#### Workshop on Quantum Materials Epitaxy

Room Ballroom A - Session QME-SaM1

#### **Oxides I**

Moderator: Bharat Jalan, University of Minnesota

#### 8:00am QME-SaM1-1 Hunting for New Oxide Superconductors using MBE, Darrell Schlom, Cornell University INVITED

Most oxide superconductors have been discovered through bulk synthesis. In this talk I will describe our efforts to use strain engineering, epitaxial stabilization, and interface engineering—all strengths of MBE—to hunt for new oxide superconductors.

\* This work was performed in collaboration with the coauthors listed in the references below.

- J.P. Ruf, H. Paik, N.J. Schreiber, H.P. Nair, L. Miao, J.K. Kawasaki, J.N. Nelson, B.D. Faeth, Y. Lee, B.H. Goodge, B. Pamuk, C.J. Fennie, L.F. Kourkoutis, D.G. Schlom, and K.M. Shen, "Strain-Stabilized Superconductivity," *Nature Communications*12 (2021) 59.
- F.V.E. Hensling, M.A. Smeaton, V. Show, K. Azizie, M.R. Barone, L.F. Kourkoutis, and D.G. Schlom, "Epitaxial Growth of the First Two Members of the Ban+1InnO2.5n+1 Ruddlesden-Popper Homologous Series," *Journal of Vacuum Science and Technology A*40 (2022) 062707.

<sup>+</sup>schlom@cornell.edu [mailto:schlom@cornell.edu]

8:30am QME-SaM1-3 Advanced Epitaxial Growth of Quantum Materials Using Thermal Laser Epitaxy, Jochen Mannhart, Max Planck Institute for Solid State Research, Germany INVITED Molecular Beam Epitaxy (MBE) and Pulsed Laser Deposition (PLD) are the primary techniques employed for the epitaxial growth of thin films and heterostructures of quantum materials [1]. Each technique has its own set of advantages and drawbacks. Thermal Laser Epitaxy (TLE) is an innovative epitaxial growth technique that aims to combine the benefits of both MBE and PLD. TLE utilizes laser-induced thermal evaporation of ultra-pure sources, facilitated by practically unlimited evaporation temperatures. Moreover, also the substrate temperatures are virtually unrestricted, as is the gas atmosphere applied throughout the entire process.

In this presentation, I will discuss the state-of-the-art in the growth of quantum materials using thermal laser epitaxy and the opportunities this advanced technique offers for the epitaxial growth of complex films and heterostructures.

H. Boschker and J. Mannhart, 'Quantum Matter Heterostructures', Annu.
Rev. Condens. Matter Phys. 8, 145 (2017)
W. Braun and J. Mannhart, 'Film Deposition by Thermal Laser Evaporation', AIP Advances 9, 085310 (2019)

The work that will be presented has been conducted together with W. Braun, H. Boschker, B. Faeth, F. Felden, F.V.E. Hensling, M. Jäger, D.-Y. Kim, L.N. Majer, and T. Smart.

#### 9:00am QME-SaM1-5 Superconductivity at Interfaces of KTaO<sub>3</sub> and its Possible Origins, Anand Bhattacharya, Argonne National Laboratory INVITED

Superconductivity in materials with broken inversion symmetry and strong spin-orbit coupling can lead to unconventional pairing states that may be of interest in quantum science and technology. In this seminar I will discuss a recently discovered superconducting electron gas formed at interfaces of a 5d transition metal oxide KTaO<sub>3</sub> (KTO) that combines these attributes intrinsically, and whose unique properties provide strong clues about the origin of its superconductivity. KTO, like its widely studied 3d cousin SrTiO<sub>3</sub> (STO), is a 'quantum paraelectric', where the onset of ferroelectricity at low temperatures is believed to be thwarted by quantum fluctuations, giving rise to a very large dielectric constant. However, unlike STO, no evidence of superconductivity has been found to date in electron-dopedKTO in the bulk. Recently, we discovered that electron gases formed interfaces of KTO are robust two-dimensional superconductors<sup>1</sup> over a wide range of carrier densities, with  $T_c$  as high as 2.2 K, about an order of magnitude higher than those found at STO interfaces. Furthermore, there is a striking dependence of  $T_c$  on the crystalline facet of KTO at which the interfacial electron gas is formed – in our samples the maximum  $T_c$  values at the KTO (111) and (110) interfaces are 2.2 K and ~ 1 K respectively, while the KTO (001) interface remains normal down to 25 mK. For the KTO (111) interface, a remarkable

non-saturating *linear* dependence of  $T_c$  on the areal carrier density ( $n_{2D}$ ) is observed, over nearly an order of magnitude of  $n_{2D}$ . The superconductivity can also be tuned by gate electric fields, which elucidates the role of the interface in mediating pairing and allows for reversible switching of superconductivity at T = 2 K. Based on these findings, we propose a mechanism<sup>2</sup> for pairing via inter-orbital interactions induced by inversionbreaking transverse optical (TO1) phonons, the same mode that softens in the quantum paraelectric phase, that explains several key aspects of superconductivity at KTO interfaces. Our results may provide insights into the pairing mechanism in other doped quantum paraelectrics, which has remained an open question for decades. Looking further, KTO interfaces are also a promising platform for exploring novel devices<sup>3</sup> for quantum science, and I will present some initial results in this direction.

References:

- 3. C. Liu et al., Science (2021). https://www.science.org/doi/abs/10.1126/science.aba5511
- 4. C. Liu et al., *Nature Communications* (2023).https://doi.org/10.1038/s41467-023-36309-2
- 5. M. Yu et al., *Nano Lett.* (2022). https://doi.org/10.1021/acs.nanolett.2c00673

#### 9:30am QME-SaM1-7 Synthesis of Electronic-Grade Quantum Heterostructures by Hybrid PLD, Chang-Beom Eom, University of Wisconsin-Madison INVITED

Modern quantum materials are inherently sensitive to point defects, and require a new synthesis route to produce epitaxial oxide thin films and interfaces clean enough to probe fundamental quantum phenomena. The recent discovery of robust superconductivity at KTaO<sub>3</sub> (111) and KTaO<sub>3</sub> (110) heterointerfaces on KaTaO<sub>3</sub> bulk single crystals offers new insights into the role of incipient ferroelectricity and strong spin-orbit coupling.Electronic grade epitaxial thin film platforms will facilitate investigation and control of the interfacial superconductivity and understanding the fundamental mechanisms of the superconductivity in KTaO<sub>3</sub>. The major challenge of research on KTaO<sub>3</sub> system is that it is difficult to grow high-quality KTaO3 epitaxial thin films due to potassium volatility. Recently, we have developed the hybrid PLD method for electronic grade KTaO<sub>3</sub> thin film growth, which successfully achieves this by taking advantage of the unique capabilities of PLD to instantly evaporate Ta<sub>2</sub>O<sub>5</sub> in a controlled manner and evaporation of  $K_2O$  to maintain sufficient overpressure of volatile species. We successfully synthesized heteroepitaxial KTaO<sub>3</sub> thin films on 111-oriented KTaO<sub>3</sub> bulk single crystal substrates with a SmScO<sub>3</sub> template by hybrid PLD, followed by a LaAlO<sub>3</sub> overlayer. Electrical transport data show a superconducting transition temperature of ~ 1.35K. We anticipate that the ability to synthesize highquality epitaxial complex oxides such as KTaO3 that contain volatile elements will provide a new platform for exploring new physics and technological applications arising from unique characteristics such as large spin-orbit coupling.

This works has been done in collaboration with Jieun Kim, Jungwoo Lee, Muqing Yu, Neil Campbell, Shun-Li Shang, Jinsol Seo, Zhipeng Wang, Sangho Oh, Zi-Kui Liu, Mark S. Rzchowski, Jeremy Levy.

This work is supported by the Gordon and Betty Moore Foundation's EPiQS Initiative, Grant GBMF9065 to C.B.E., and a Vannevar Bush Faculty Fellowship (N00014-20-1-2844).

#### Workshop on Quantum Materials Epitaxy Room Ballroom A - Session QME-SaM2

#### **Topological and Magnetic Materials I**

Moderator: Dr. Nitin Samarth, Pennsylvania State University

10:30am QME-SaM2-11 Invited Paper, Stuart Parkin, Max Planck Institute of Microstructural Physics, Germany INVITED

11:00am QME-SaM2-13 The Art and Science of Molecular Beam Epitaxy — —from Topological Materials to Interfacial Superconductivity, Cui-Zu Chang, Pennsylvania State University INVITED

In this talk, I will focus on the molecular beam epitaxy (MBE) growth of quantum materials, spanning from topological materials to interfacial superconductors. I will talk about two solid-state phenomena with zero

resistance: the quantum anomalous Hall (QAH) effect and the interface superconductivity. The QAH insulator is a material in which the interior is insulating but electrons can travel with zero resistance along onedimensional conducting edge channels. Owing to its resistance-free edge channels, the QAH insulator is an outstanding platform for energy-efficient electronics and spintronics as well as topological quantum computations. With many efforts, we were the first to realize the QAH effect in MBEgrown Cr- and V-doped topological insulator (TI) thin films. I will briefly talk about the route to the QAH effect and then focus on our recent progress on the high Chern number QAH effect and three-dimensional QAH effect in MBE-grown magnetic TI multilayers. Finally, I will talk about the interfacial superconductivity in MBE-grown TI/iron chalcogenide heterostructures. Moreover, the TI/iron chalcogenide heterostructures fulfill the two essential ingredients of topological superconductivity, i.e. topological and superconducting orders, and thus provide an alternative platform for the exploration of Majorana physics towards the scale topological quantum computations.

#### 11:30am QME-SaM2-15 Epitaxial Control of Topological Semimetals, Kirstin Alberi, National Renewable Energy Laboratory INVITED

Three dimensional topological semimetals exhibit properties that hold promise for a wide range of applications, including electronics, spintronics, photodetectors and thermoelectrics. In order to use them for these purposes, we must integrate them into device structures with control of defects, interfaces and the Fermi level. We must also learn how to manipulate the electronic structure and behavior of topological semimetals through the addition of impurities or alloying. The aim of our research is to enable these capabilities through epitaxial synthesis as well as understand how various forms of disorder (defects, impurities and interfaces) impact the resulting film properties. In this talk, I will detail our work on two relevant materials: the Dirac semimetal Cd<sub>3</sub>As<sub>2</sub> and the Weyl semimetal TaAs. We grow these films by molecular beam epitaxy using elemental sources, which allows us to control point defects and incorporate impurities. In Cd<sub>3</sub>As<sub>2</sub>, the As/Cd flux ratio can be selected to adjust the balance of native Cd vacancy and interstitial defect concentrations, permitting the free electron concentration to be varied with the 10<sup>16</sup> to 10<sup>18</sup> cm<sup>-3</sup> range. This control has allowed us to study the role of point defects on magnetotransport behavior. Likewise, the addition of Zn can be used to induce n-to-p doping and topological semimetal-semiconductor transitions. More recently, we have achieved epitaxy of single crystal-like TaAs films of arbitrary thickness directly on GaAs substrates. We map out the growth window of this material and comment on the challenges ahead for epitaxial growth of monopnictide Weyl semimetals more generally.

This work was performed by the National Renewable Energy Laboratory, operated by Alliance for Sustainable Energy, LLC, for the U.S. Department of Energy (DOE) under Contract No. DE-AC36-08GO28308. Funding was provided by the U.S. Department of Energy, Office of Science, Basic Energy Sciences, Division of Materials Sciences and Engineering, Physical Behavior of Materials Program under the Disorder in Topological Semimetals project.

#### 12:00pm QME-SaM2-17 Growth and Investigations of Topological and Quantum Phenomena in Epitaxial Semimetallic Thin Films, Chris Palmstrøm, University of California, Santa Barbara INVITED

Controlling electronic properties via band gap engineering is at the heart of modern semiconductor devices. We have extended this concept to band structure engineering of quantum materials utilizing confined thin film geometries, hetero-epitaxial interfaces and epitaxial strain to engineer the electronic structure in elemental, rare-earth monopnictide and Heusler materials. The growth and tuning of the band structure of epitaxial films have been investigated through a combination of molecular beam epitaxial growth, in-vacuo angle-resolved photoelectron spectroscopy, scanning tunneling microscopy and spectroscopy and ex-situ low temperature magnetostransport and hybrid density functional theory.

### Saturday Afternoon, September 16, 2023

Workshop on Quantum Materials Epitaxy

Room Ballroom A - Session QME-SaA1

#### 2D and Heteroepitaxial Integration

Moderator: Dr. Roman Engel-Herbert, Paul Drude Institute

2:00pm QME-SaA1-1 Invited Paper, Grace Xing, Cornell University INVITED

#### 2:30pm QME-SaA1-3 Growth of Topological Materials by Molecular Beam Epitaxy, Stephanie Law, Pennsylvania State University INVITED

Topological materials are those materials that have a topologically nontrivial band structure and include topological insulators (TIs), Dirac semimetals (DSMs), and Weyl semimetals (WSMs). TIs, in particular, have been of interest due to their ability to host two-dimensional surface states with spin-momentum locking. In order to take advantage of these states, we need to be able to grow TI thin films with controllable thickness, few to no grain boundaries, few defects, and as part of heterostructures. This has typically proven to be a challenge, since most TI materials are layered materials. The weak interaction between the film and the substrate results means that it is possible to grow TIs on a variety of substrates, but that the nucleation of the film is difficult to control. In this talk, I will describe our recent progress growing three classes of TI thin films: Bi2Se3, the prototypical TI; BiSb, a TI that can be made to be bulk insulating; and MnBi<sub>2</sub>Se<sub>4</sub>, an antiferromagnetic TI. I will describe commonalities in the growth of these materials as well as challenges unique to each material. Finally, I will give an outlook on the prospects for the growth of insulting layered TI thin films on technologically-relevant substrates.

#### 3:00pm QME-SaA1-5 Big Flat Quantum Crystals: Wafer-Scale Growth of 2D Materials by Metalorganic Chemical Vapour Deposition, James A. Gupta, University of Ottawa, Canada INVITED

Following the successful development of monolayer graphene and the subsequent Nobel Prize, the field of 2D materials has exploded into a universe of exciting new materials and applications. To date, however, most samples have been produced by mechanical exfoliation, yielding singlecrystal monolayer films with dimensions of only a few 10's of microns. Such samples have worked extremely well for proof-of-concept devices, but it is very clear that epitaxial growth is needed for large-scale device development, reproducibility and manufacturability. For the epitaxial growth of most compound semiconductors, the greatest challenges are in the requirements for the growth of many layers with precisely-controlled composition and thickness to satisfy the device requirements. In contrast, for 2D materials, the challenge is to controllably grow single-crystal monolayer films with low densities of defects and grain boundaries, and to grow heterostructures of different, weakly-bonded van der Waals materials . In this presentation I will discuss the requirements for epitaxial growth of 2D materials, particularly graphene, hexagonal boron nitride and transition metal dichalcogenides (e.g. MoS<sub>2</sub>, WS<sub>2</sub>, MoSe<sub>2</sub>, WSe<sub>2</sub>). As with any epitaxial growth process, the identification of suitable molecular sources and substrates is critical. At the University of Ottawa, we have established a 2D growth facility featuring an Aixtron 3X2" Close-Coupled Showerhead MOCVD system. I will discuss the specific materials, precursors and processes used at uO, as well as the challenges and opportunities for 2D epitaxial growth in general.

#### 3:30pm QME-SaA1-7 Advanced Heterogeneous Integration Enabled by Remote Epitaxy, Jeehwan Kim, Massachusetts Institute of Technology INVITED

For future of electronics such as bioelectronics, 3D integrated electronics, and bendable electronics, needs for flexibility and stackability of electronic products have substantially grown up. However, conventional wafer-based single-crystalline semiconductors cannot catch up with such trends because they are bound to the thick rigid wafers such that they are neither flexible nor stackable. Although polymer-based organic electronic materials are more compatible as they are mechanically complaint and less costly than inorganic counterparts, their electronic/photonic performance is substantially inferior to that of single-crystalline inorganic materials. For the past half a decade, my research group at MIT has focused on mitigating such performance-mechanical compliance dilemma by developing methods to obtain cheap, flexible, stackable, single-crystalline inorganic systems. In today's talk, I will discuss about our strategies to realize such a dream electronic system and how these strategies unlock new ways of manufacturing advanced electronic systems. I will highlight our remote epitaxy technique that can produce single-crystalline freestanding membranes including III-nitrides, III-V and complex oxide system with their

excellent semiconducting performance. In addition, I will present unprecedented artificial heterostructures enabled by stacking of those freestanding 3D material membranes, e.g., world's smallest verticallystacked full color micro-LEDs, world's best multiferroic devices, battery-less wireless e-skin, and heat dissipating system GaN power devices.

#### Workshop on Quantum Materials Epitaxy Room Ballroom A - Session QME-SaA2

#### **Poster Highlights**

Moderator: Prof. Jason Kawasaki, University of Wisconsin - Madison

4:30pm QME-SaA2-11 Poster Highlights Session - 3-Minute Oral Presentations from Poster Presenters,

Workshop on Quantum Materials Epitaxy Room Ballroom A - Session QME-SaP

#### Workshop on Quantum Materials Epitaxy Poster Session

QME-SaP-1 Optimization of Hybridized InAsSb/InGaSb Semiconductor Topological Materials, *Heather Haugan*, Air Force Research Laboratory; D. Das, L. Ram-Mohan, Worcester Polytechnic Institute; J. Corbett, K. Mahalingam, R. Bedford, K. Eyink, Air Force Research Laboratory

We are in the midst of the second quantum revolution. Research institutes and companies worldwide are working toward harnessing the power of quantum physics for technological applications. Gapless surface states on topological insulators are protected from elastic scattering on nonmagnetic impurities,1 which makes them promising candidates for low-power electronic applications. In previous years, most research efforts on demonstrating topologically protected edge states were focused on rather exotic topological materials.<sup>1,2</sup>However it was extremely difficult to generate strong enough edge currents out of these materials, to be practically useful for widespread applications, due to low emission currents.Hence, we are exploring more commonly used infrared materials such as InAsSb/InGaSb guantum wells (QWs) and superlattices (SLs). Both structures can be designed to have an inverted gap by elevating the hole state above the electron state, and a large emission current, in particular in the SL by enhancing wavefunction overlaps. These are critical components on establishing topological states to create the conducting edge and the insulating bulk state. This unique circumstance can create dissipationless transportof electrons in heterostructures, which is particularly important for a variety of sensing applications. Therefore, as an initial test, we examine two hybridized topological structures; one for 6.22 Å atomic lattice constant 82 Å InAs/34 Å In<sub>0.4</sub>Ga<sub>0.6</sub>Sb/82 Å InAs symmetric QWs on In<sub>0.32</sub>Ga<sub>0.68</sub>Sb substrate<sup>3</sup> and the other for 6.10 Å atomic lattice constant 70 Å InAs<sub>0.9</sub>Sb<sub>0.1</sub>/35 Å GaSb SLs on GaSb substrate.Both structures are tailored for the same hybridization gap of ~60 meV.By using a combination of theoretical modeling, high-resolution x-ray diffraction, and high-resolution transmission electron microscopy, we optimize the absorber designs and their molecular beam epitaxy process to achieve high-quality materials.Growth parameters in each design are carefully coordinated to mitigate the crystalline defects to produce high-quality materials.Quasiparticle interference mapping through a scanning tunneling microscope is used to investigate the band structure of a SL sample.

[1] M. Z. Hasan and C. L. Kane, "Topological insulators", Rev. Mod. Phys. 82, 3045 (2010).

[2] J. E. Moore, "The birth of topological insulators", Nature464, 194 (2010).

[3] Krishtopenko and Teppe, "Quantum spin Hall insulator with a large bandgap, Dirac fermions, and bilayer graphene analog", Sci. Adv. **4**, 7529 (2018).

QME-SaP-2 Doping the Undopable: Hybrid Molecular Beam Epitaxy Growth, n-type Doping, and Field-Effect Transistor using CaSnO<sub>3</sub>, *Fengdeng Liu*, University of Minnesota, USA; *P. Golani*, University of Minnesota; *T. Truttmann*, University of Minnesota, USA; *I. Evangelista*, University of Delaware; *M. Smeaton*, Cornell University; *D. Bugallo*, Drexel University; *J. Wen*, University of Minnesota; *A. Kamath Manjeshwar*, University of Minnesota, USA; *S. May*, Drexel University; *L. Kourkoutis*, Cornell University; *A. Janotti*, University of Delaware; *S. Koester*, University of Minnesota; *B. Jalan*, University of Minnesota, USA

The alkaline earth stannates are touted for their wide band gaps and the highest room-temperature electron mobilities among all the perovskite oxides. CaSnO<sub>3</sub> has the highest measured band gap in this family and is thus a particularly promising ultra-wide band gap semiconductor. However, discouraging results from previous theoretical studies and failed doping attempts had written off this material as "undopable". Here we redeem CaSnO<sub>3</sub> using hybrid molecular beam epitaxy (hMBE), which provides an adsorption-controlled growth for the phase-pure, epitaxial and stoichiometric CaSnO3 films. By introducing lanthanum (La) as an n-type dopant, we demonstrate the robust and predictable doping of CaSnO<sub>3</sub> with free electron concentrations, n, from  $3.3 \times 10^{19}$  cm<sup>-3</sup> to  $1.6 \times 10^{20}$  cm<sup>-3</sup>. The films exhibit a maximum room-temperature mobility of 42 cm<sup>2</sup> V<sup>-1</sup>s<sup>-1</sup> at n = $3.3 \times 10^{19}$  cm<sup>-3</sup>. Despite having a smaller radius than the host ion, La expands the lattice parameter. Using density functional calculations, this effect is attributed to the energy gain by lowering the conduction band upon volume expansion. Finally, we exploit the robust doping by fabricating the CaSnO3 -based field-effect transistors. The transistors show promise for CaSnO3's high-voltage capabilities by exhibiting low off-state leakage below

20 pA/ $\mu$ m at a drain-source voltage of 100 V and on-off ratios exceeding 10<sup>6</sup>. This work opens the door to future studies on the semiconducting properties of CaSnO<sub>3</sub> and the many devices that could benefit from CaSnO<sub>3</sub>'s exceptionally wide band gap.

#### QME-SaP-3 Solid Source Metal-Organic Molecular Beam Epitaxy for Epitaxial SrRuO<sub>3</sub> Films, Anusha Kamath Manjeshwar, S. Nair, A. Rajapitamahuni, R. James, B. Jalan, University of Minnesota

The investigation of the electrical and magnetic properties of SrRuO<sub>3</sub> and its associated Sr<sub>n+1</sub>Ru<sub>n</sub>O<sub>3n+1</sub> Ruddlesden-Popper phases requires a high degree of control over the isolation of the desired phase and its defect density. The growth of ruthenates is fundamentally challenging because ruthenium (Ru) resists scalable evaporation and oxidation. This bottleneck complicates the growth of SrRuO<sub>3</sub> films with low defect densities using inherently low-energy, ultra-high vacuum deposition techniques such as molecular beam epitaxy (MBE). Special modifications to conventional MBE such as electron-beam assisted evaporation and ozone-assisted oxidation of Ru have, so far, enabled the best defect control or the highest residual resistivity ratios (RRR =  $\rho_{300k}/\rho_{2K}$ ) in SrRuO<sub>3</sub> films among all physical vapor deposition techniques. However, these modifications are expensive and require additional interlocks to ensure safe operating conditions.

We outline a novel technique called solid source metal-organic MBE to supply a solid metal-organic precursor with pre-oxidized ruthenium with an effusion cell at T < 200 °C, a drastic decrease from the ~ 2000 °C required to produce comparable fluxes with elemental Ru. With this technique, we demonstrate the growth of phase pure, epitaxial, and stoichiometric SrRuO<sub>3</sub> films with robust ferromagnetism below 150 K on SrTiO<sub>3</sub> (001) substrates. We simplify the route to an adsorption-controlled growth window in SrRuO<sub>3</sub> films, growth conditions where the films can self-regulate their stoichiometry, which is a key ingredient for successful defect control in electron-beam and ozone-assisted MBE-grown SrRuO<sub>3</sub> films. We discuss the intricate relationship between cation stoichiometry, magnetic domains, and RRR in epitaxial SrRuO<sub>3</sub> films and outline new pathways for achieving low defect densities in SrRuO<sub>3</sub>. Using these guidelines to optimize stoichiometry and film thickness within a growth window, we achieve a RRR = 87 for a 50 nm-thick  $SrRuO_3$  film, the highest for any  $SrRuO_3$  film on  $SrTiO_3$  (001) substrates. We will also illustrate how solid source metal-organic MBE is a simple and cost-effective method to enhance the capabilities of conventional MBE for the defect-controlled growth of ruthenates.

QME-SaP-4 Growth Mechanism of SrTiO<sub>3</sub> on a Graphene-covered Substrate using Hybrid MBEO, Sooho Choo, H. Yoon, University of Minnesota, USA, Republic of Korea; B. Matthews, Pacific Northwest National Laboratory; S. Sharma, University of Minnesota, USA; S. Spurgeon, S. Chambers, Pacific Northwest National Laboratory; R. James, B. Jalan, University of Minnesota, USA

Epitaxial films grown on a substrate covered with two-dimensional (2D) materials offer many exciting possibilities: reusability of the substrate; ability to obtain a freestanding membrane; and opportunity to reduce misfit dislocations. Three growth mechanisms are argued to be responsible for epitaxial growth on 2D material-covered substrate: (1) Remote epitaxy; (2) Van der Waals epitaxy; and (3) pinhole-assisted epitaxy. It is, however, still unclear which of these three mechanisms is responsible for epitaxial growth. In this talk, we will first present the successful growth of epitaxial SrTiO<sub>3</sub> nanomembranes on SrTiO<sub>3</sub> (001) substrates covered with bilayer graphene. Titanium tetraisopropoxide (TTIP) was used as a source of titanium and oxygen. No additional oxygen was used to avoid graphene oxidation. By varying Sr/TTIP beam equivalent ratios, we reveal a wide MBE growth window for adsorption-controlled growth of stoichiometric SrTiO3 membranes [1]. Bulk-like lattice parameter of 3.905 Å was obtained for nanomembranes. By combining heteroepitaxial growth, high-resolution Xray diffraction, atomic force microscopy, transmission electron microscopy, and Raman spectroscopy, we discuss all three growth mechanisms highlighting the role of graphene thickness, pinholes, and the substrate's ionicity on epitaxial growth.

QME-SaP-5 Growing Clean Crystals from "Dirty" Precursors in MBE, Rashmi Choudhary, University of Minnesota, USA; Z. Liu, J. Cai, X. Xu, J. Chu, University of Washington; B. Jalan, University of Minnesota, USA

Ultra-high purity elemental sources have long been considered a prerequisite for obtaining low impurity concentration in compound semiconductors in the world of molecular beam epitaxy (MBE). Furthermore, to realize intrinsic properties, the material needs to be nearly free of intrinsic and extrinsic defects. For this reason, the use of ultra-high-purity elemental sources has been the historical practice in MBE, perhaps,

for the fear that impurity elements might get incorporated into the film, making it "dirty".

In this work, we challenge this conventional MBE wisdom by presenting an extension of the hybrid-MBE approach, known as solid-source metalorganic MBE, for growing superconducting Sr<sub>2</sub>RuO<sub>4</sub> films using a solid organometallic precursor, ruthenium acetylacetonate, as a source of Ru. We grew 100 nm thick (001) Sr<sub>2</sub>RuO<sub>4</sub> films on (001) LSAT substrate at 900°C substrate temperature using co-deposition of Sr, ruthenium acetylacetonate, and oxygen plasma. These films are phase-pure, singlecrystalline, fully coherent, and superconducting. The superconducting transition temperature of the film is 0.85 K. In contrast to the conventional MBE, which employs ultra-pure Ru metal evaporated at ~ 2000°C as a Ru source, along with reactive ozone to obtain Ru  $\rightarrow$  Ru<sup>4+</sup> oxidation, the use of ruthenium acetylacetonate precursor requires significantly lower temperature for Ru sublimation (less than 200°C) and eliminates the need for ozone.

This is the first-time realization of superconducting  $Sr_2RuO_4$  films using ozone-free MBE. By combining our results with the recent developments in hybrid-MBE, we argue that leveraging precursor chemistry will be necessary to realize next-generation breakthroughs in the synthesis of atomically precise quantum materials. Our results establish hybrid-MBE as a viable method for growing highest quality crystals and put this technique at the forefront of vacuum deposition technologies despite the use of a "dirty" chemical precursor.

QME-SaP-6 Growth of EuIn<sub>2</sub>As<sub>2</sub> Thin Films by MBE: Towards Investigating the Topological Properties of a Candidate Axion Insulator, *Muhsin Abdul Karim*, University of Notre Dame, Ghana; *J. Wang*, University of Notre Dame, China; *K. Yoshimura*, University of Notre Dame, Japan; *S. Bac*, University of Notre Dame, Korea (Democratic People's Republic of); *X. Liu*, University of Notre Dame, China; *B. Assaf*, University of Notre Dame

Euln<sub>2</sub>As<sub>2</sub> is a promising topological material with an antiferromagnetic ground state predicted to support axion electrodynamics[1]. To observe these predicted properties and effects, thin films of the material must be synthesized. However, thin films of Euln<sub>2</sub>As<sub>2</sub> are not easy to make due to its nature of crystallization and the fact that it competes with thermodynamically stable zincblende phases of III-V materials. In this work, we present the thin film growth of EuIn<sub>2</sub>As<sub>2</sub> on sapphire substrates by molecular beam epitaxy where we show that a high substrate temperature stabilizes the crystal structure of this compound. We have successfully grown thin films of thickness ranging between 50 - 120 nm. Magnetometry studies on the thin films agree very well with the result on bulk crystals[2]. The films exhibit the expected (ab)-plane magnetic easy axis and a Neel temperature close to 16 K. All our films are p-type and have charge carrier densities varying between  $(2.8 - 7.4)*10^{(14)}$  cm<sup>-2</sup>. We also find that the sample resistivity, ranging from  $1.8*10^{-5}$  –  $2.8*10^{-4}$   $\Omega$ m, is mainly determined by changing morphology of the films under different growth conditions. Our work provides a developed growth scheme for EuIn<sub>2</sub>As<sub>2</sub>thin films, a crucial gateway towards realizing the predicted topological properties in this candidate axion insulator material.

#### QME-SaP-7 Synthesis of Free-Standing Membranes Using a Sacrificial Layer Method Grown by Hybrid MBE, *Shivasheesh Varshney*, S. Choo, Z. Yang, J. Wen, S. Koester, B. Jalan, University of Minnesota, USA

Free-standing membranes have broad applications in the creation of symmetry-mismatched, non-equilibrium, and artificial heterostructures. We use sacrificial layer method to synthesize phase-pure epitaxial SrTiO<sub>3</sub> membranes. In this study, we will discuss the growth of strain-engineered SrTiO<sub>3</sub> films using different sacrificial layer(s) grown by hybrid MBE. We characterize the as-grown films using x-ray diffraction (XRD) and atomic force microscopy (AFM). We show exfoliation and transfer of films onto dissimilar substrates, followed by their structural characterization. Finally, we use impedance spectroscopy to characterize the dielectric properties and show a bulk-like dielectric constant of  $\approx$  300 for SrTiO<sub>3</sub> membranes transferred on Au coated Si substrate.

#### QME-SaP-8 Epitaxial Growth of Precursor Phases of Novel Cuprate Superconductors Using Oxide MBE, Jinkwon Kim, C. Kim, D. Schlom, Cornell University

Since the discovery of high- $T_c$  cuprates,<sup>[1]</sup> a plethora of research has been conducted to understand their superconducting origin. They usually have layered perovskite structures and CuO<sub>2</sub> planes are considered as crucial ingredients to host Cooper pairs and *d*-wave superconductivity. The CuO<sub>6</sub> coordination octahedron is elongated along the *c*-axis, the  $3d_{x2-y2}$  orbital dominantly contributes to the electronic structure at the Fermi level.<sup>[2]</sup> In

2019, a totally different type of cuprate superconductor was discovered: Ba<sub>2</sub>CuO<sub>4-6</sub> (bulk  $T_c \sim 73$  K for  $\delta = 0.8$ ).<sup>[3]</sup>The octahedron of Ba<sub>2</sub>CuO<sub>4-6</sub> was compressed along the *c*-axis, hence it becomes a multi-band system composed of  $3d_{x2-y2}$  and  $3d_{3z2-r2}$  orbitals. Moreover, oxygen vacancies even exist on the CuO<sub>2</sub> plane. These unique characteristics of Ba<sub>2</sub>CuO<sub>4-6</sub> strongly sugges that the Cooper pair behavior is different from previously reported cuprate superconductors. Hence the study of Ba<sub>2</sub>CuO<sub>4-6</sub> is expected to contribute to unveiling clues about the superconducting mechanism of high- $T_c$  cuprates.

Unfortunately, the synthesis of Ba2CuO4-6 poses significant challenges. Ba-Cu-O compounds energetically prefer forming the Ba<sub>2</sub>CuO<sub>3</sub> phase (a 1D CuO chain structure), hence strong oxidation is required to achieve the desired Ba2CuO3.2 phase. Li et al. achieved superconducting Ba2CuO4-6 specimens by high-pressure synthesis methods.<sup>[3]</sup> But the chemical instability and polycrystalline structure limited deeper understanding of its electronic structure and superconductivity. Epitaxial thin film growth can be an alternative approach since it provides a strong oxidation environment with a large surface-to-volume ratio, a low reaction temperature, a pseudomorphic constraint from the substrate, and the ability to reveal the electronic structure with angle-resolved photoemission spectroscopy. In this study, using oxide MBE, we grew epitaxial thin films of the precursor phase of the Ba<sub>2</sub>CuO<sub>4-δ</sub> superconductor, Ba<sub>2</sub>CuO<sub>3</sub>, and its homolog, Sr<sub>2</sub>CuO<sub>3</sub>. After growth, the Ba<sub>2</sub>CuO<sub>3</sub> and Sr<sub>2</sub>CuO<sub>3</sub> films were oxidized by various methods such as post-growth ozone exposing or topotactic oxidation. We characterized the cuprate films by reflection high-energy electron diffraction (RHEED), x-ray diffraction (XRD), and atomic force microscopy (AFM). Our work on a thin-film approach toward single-crystalline  $A_2CuO_{4-\delta}(A: Ba, Sr)$  superconductors will be presented.

#### References

[1] J. G. Bednorz et al., Z. Phys. B Condens Matter64,189 (1986).

[2] H. A. Jahn et al., Proc. R. Soc. London, Ser. A161, 220 (1937).

[3] W. M. Li et al., Proc. Natl. Acad. Sci. U.S.A. 116, 12156 (2019).

#### QME-SaP-9 MBE of Ba<sub>2</sub>BiTaO<sub>6</sub>, a Candidate *p*-type Oxide Semiconductor, Anna Park, Y. Birkhölzer, M. Barone, D. Schlom, Cornell University

Complementary metal-oxide-semiconductor (CMOS) technology is an important part of today's integrated circuit technology. CMOS replaced nchannel metal-oxide-semiconductor (NMOS) in the 1980s and with the 100 to 1000x power savings advantage it provides, enabled integrated circuits to grow from tens of thousands of NMOS transistors on a chip to tens of billions of CMOS transistors today. Today we stand at similar cross-roads for transistors made from oxide semiconductors. Only high-performance nchannel oxide transistors (and thus NMOS) exist for oxide transistors. If high performance p-channel oxide transistors could also be made, low-power CMOS would be possible in oxide systems and enable low-power transparent electronics. Although many p-type semiconducting oxides have been predicted based on their theorized electronic properties, few have been realized in experiment and those that have been achieved have much lower mobility than established n-type oxides like indium-gallium-zincoxide or In<sub>2</sub>O<sub>3</sub>. Unfortunately, the realization of *p*-type oxides is particularly difficult due to the localization of the oxygen 2p orbitals. One design criterion to realize p-type oxides is to create oxides with Sn<sup>2+</sup>, Pb<sup>2+</sup>, or Bi<sup>3+</sup> cations that have dispersive filled s-orbitals that will hybridize with oxygen 2p orbitals to delocalize the hole states and enhance mobility. We show that suboxide MBE can be used to grow Ba2BiTaO6, a candidate transparent p-type oxide.

Suboxide MBE utilizes molecular beams of suboxides, where the incoming cation precursors are already in the desired oxidation states. Many oxides crystallize with a perovskite structure and offer flexibility in accommodating a variety of cations, resulting in a wide range of properties. Double perovskites, of which Ba<sub>2</sub>BiTaO<sub>6</sub> is an example, offer additional flexibility and another degree of freedom to explore in the interplay of structure and properties. ForBa<sub>2</sub>BiTaO<sub>6</sub>, we take advantage of suboxide MBE and avoid using an electron beam to evaporate Ta by supplying a molecular beam of TaO<sub>2</sub> from a Ta<sub>2</sub>O<sub>5</sub> source. Additionally, with the volatility of Bi at our growth temperatures, we grow in an adsorption-controlled regime and fine tune the flux of Ba and Ta cations to create the desired phase.

QME-SaP-10 Flexomagnetism and Strain Induced Superconductivity in Rippled GdAuGe Heusler Membranes, *Tamalika Samanta*, Z. LaDuca, D. Du, T. Jung, S. Manzo, K. Su, M. Arnold, J. Kawasaki, University of Wisconsin - Madison

Rare earth-based Heuslers are prospective materials platforms for magnonics, topological spin texture, superconductivity, THz spintronics, etc. [1, 2]. The magneto-mechanical coupling in these materials allows for better control and manipulation of the primary order parameter and magnetic flexibility [3]. Here, we demonstrate novel flexomagnetic responses i.e., the coupling between strain gradient and magnetism, and strain-induced superconductivity, in GdAuGe Heusler membranes. The thin films of GdAuGe Heusler composition have been grown on monolayer Graphene/ Ge (111) by molecular beam epitaxy (MBE). GdAuGe films are then mechanically exfoliated to form free-standing rippled membranes.

GdAuGe shows an antiferromagnetic ordering below ~17 K, which is sustained when a homogeneous strain is applied. However, the application of strain gradient dramatically alters the magnetic ground state of GdAuGe in the rippled membranes. A phase diagram of the rippled GdAuGe membranes is shown in Fig. 1(a). Notably, a moderate strain gradient of a few tenths of a percentage transforms the ground state from antiferromagnetic to unconventional ferrimagnetic phases. These ferrimagnetic ground states in the rippled membranes offer the possibility of discovering spin reorientation and other unique magnetic phenomena; the most exciting observation is the emergence of superconductivity in GdAuGe membranes when a very large strain gradient is applied, with superconducting transitions occurring at low temperatures below ~3.5 K. Figure 1(b) shows the magnetic characterization of a superconducting GdAuGe rippled membrane.

At present, the microscopic origin of flexomagnetism and its effects on the thermodynamics of spin reorientation and phase transitions in these membranes remain unclear. Advanced spectroscopic measurements and magneto-transport experiments, combined with theoretical modeling, are planned to further investigate the phenomena in these rippled membranes. References

1. Graf, Tanja, et al. "Simple rules for the understanding of Heusler compounds." Progress in solid state chemistry 39.1 (2011): 1-50.

2. Kawasaki, Jason K. "Heusler interfaces—Opportunities beyond spintronics?." APL Materials 7.8 (2019): 080907

3. Du, Dongxue, et al. "Epitaxy, exfoliation, and strain-induced magnetism in rippled Heusler membranes." Nature Communications 12.1 (2021): 1-7

QME-SaP-11 Engineering Metal Oxidation Towards Epitaxial Growth of Complex Iridates using Molecular Beam Epitaxy, *Sreejith Nair*, *Z*. Yang, D. *Lee, S. Guo*, University of Minnesota, USA; *J. Sadowski*, Brookhaven National Laboratory; *S. Johnson*, Auburn University; *A. Saboor*, University of Delaware; *Y. Li*, *H. Zhou*, Argonne National Laboratory, USA; *R. Comes*, *W. Jin*, Auburn University; *K. Mkhoyan*, University of Minnesota, USA; *A. Janotti*, University of Delaware; *B. Jalan*, University of Minnesota, USA

The platinum group metals like Ir and Ru have captured significant interest in the condensed matter physics and materials science community due to the exotic electronic and magnetic properties that they exhibit when combined with oxygen. The oxides of these metals provide a unique platform to study and leverage the delicate interplay between electron correlations, crystal field and spin-orbit coupling energies. High quality thin films of complex platinum group metal oxides are hence, critical to realizing new phenomena such as the predicted unconventional superconductivity in Sr<sub>2</sub>IrO<sub>4</sub>. However, the platinum group metals have extremely low vapor pressures and low oxidation potentials. These factors make it challenging to synthesize their oxide thin films using an ultra-high vacuum (UHV) technique like Molecular Beam Epitaxy (MBE). Here, we have addressed these challenges using a novel solid-source metal-organic MBE approach [1,2]. We demonstrate atomically precise synthesis of binary IrO<sub>2</sub> using Ir(acac)<sub>3</sub> as the metal-organic Ir source at substrate temperatures as low as 250 °C. The use of the metal-organic precursor allows Ir supply at source temperatures less than 200 °C and enables easy oxidation due to the +3 Ir oxidation state in the precursor. Further, by combining epitaxially strained IrO2 thin film growth on different substrates, x-ray diffraction, electron microscopy, spectroscopy techniques, and DFT calculations, we demonstrate a vital role of epitaxial strain in Ir oxidation. Thus, epitaxial strain can be an additional tuning knob to engineer metal oxidation which can aid the conventional thermodynamic and kinetic driving forces [3].

However, the true test of metal oxidation in UHV occurs at high growth temperatures where oxidation becomes increasingly thermodynamically

unfavorable. Hence, in order to examine the efficacy of the solid-source metal-organic MBE approach and to realize the elusive unconventional superconducting state, we study the synthesis of Sr<sub>2</sub>IrO<sub>4</sub> thin films, which is favored at growth temperatures greater than 600-700 °C. We will present a detailed growth study, structural characterization, electrical and magneto-transport in epitaxial Sr<sub>2</sub>IrO<sub>4</sub> films, along with alternative ways to tackle the Ir oxidation challenge in UHV synthesis.

#### References:

[1] W. Nunn et al., "Solid source metal-organic molecular beam epitaxy of epitaxial RuO<sub>2</sub>", *APL Mater. 9, 091112 (2021)* 

[2] W. Nunn et al., "Novel synthesis approach for "stubborn" metals and metal oxides", *Proc. Natl. Acad. Sciences* 118, e2105713118 (2021)

[3] S. Nair et al., "Engineering Metal Oxidation using Epitaxial Strain", *Nat. Nanotechnol. (accepted) (2023)* 

QME-SaP-12 Improved Epitaxy of Unconventional Metals for Quantum Applications, *Stefania Isceri*, *M. Giparakis*, *R. Svagera*, *M. Waas*, Technische Universität Wien, Austria; *V. Butera*, *E. Kolibalova*, *O. Man*, Central European Institute of Technology, Czechia; *L. Fischer*, *H. Detz*, *W. Schrenk*, *G. Strasser*, *S. Buehler-Paschen*, *A. Andrews*, Technische Universität Wien, Austria

Strange metal thin films have attracted attention due to their promising applications in quantum devices. Strong correlations in YbRh<sub>2</sub>Si<sub>2</sub> lead to intriguing phenomena, including linear-in-temperature strange metal behavior, phase transition from Landau-Fermi liquid to antiferromagnetic at the quantum critical point, electron delocalization transition [1], unconventional superconductivity [2], and suppression of shot noise [3]. In the Weyl-Kondo semimetal Ce<sub>3</sub>Bi<sub>4</sub>Pd<sub>3</sub> films, non-trivial surface states are present [4]. In this study, we demonstrate the improvement of epitaxial YbRh<sub>2</sub>Si<sub>2</sub> films on Ge(001) and the achievement of epitaxial growth of Ce<sub>3</sub>Bi<sub>4</sub>Pd<sub>3</sub> on sapphire.

We use an MBE chamber equipped with Knudsen cells for Bi, Ce, Pd, Yb, and e-beam evaporation sources for Rh and Si. First, we investigate the conditions in terms of growth temperature, and Yb flux to obtain smooth and stoichiometric samples without any surface treatment. Then, we performed density functional theory (DFT) calculations to analyze the most favorable adsorbed atoms in the first layer of YbRh<sub>2</sub>Si<sub>2</sub> on the Ge surface. This study indicates that the adsorption of Rh on Ge (binding energy Eb=-5.4 eV) is favored over Si and Yb (E<sub>b</sub>=-4.4 eV and -2.9 eV, respectively), so strongly that Rh atoms tend to kick out the Ge atoms. On the other hand, Si atoms diffuse on the Ge surface. We analyzed the improvement of the samples' surface by soaking the substrate with 1-2 ML of Yb before the deposition of YbRh<sub>2</sub>Si<sub>2</sub> in the temperature range between 400°C and 475°C, measured by a pyrometer. The thickness of the samples spans 10 to 60 nm. The results show that with increasing Yb soaking time, a transition of the RHEED pattern from spotty to streaky, as well as the reduction of the surface roughness and defects occur.

For the second semimetal  $Ce_3Bi_4Pd_3$ , the sapphire substrates are cleaned with solvents and then annealed in the MBE machine to remove hydrocarbons. Then 50-nm-thick  $Ce_3Bi_4Pd_3$  films are grown at 60°C (heater temperature) and a 10-nm-thick Si capping layer is deposited to prevent oxidation of the samples. In the early development of this research, x-ray diffraction shows that the epitaxy of polycrystalline films of this material is possible, whilst energy dispersive x-ray spectroscopy (EDX) and inductively coupled plasma-optical emission spectroscopy (ICP-OES) techniques are used to adjust the stoichiometric composition. Future investigations are planned on diamond (lattice mismatch=0.4%).

[1]	L.	Prochaska	et	al.,	Science,	367,	285-288,		2020
[2]	D. H.	Nguyen et	al.,	Nature	Commu	nications,	12, 4	4341,	2021
[3]	L.	Chen	e	t a	l., a	arXiv:2206.	00672	,	2022

[4] S. Dzsaber, et al., PNAS 118, 202

QME-SaP-13 Growth and Angle-Resolved Photoemission of Strain-and Thickness-Tuned α-Sn Films, Aaron Engel, C. Dempsey, H. Inbar, S. Nishihaya, Y. Chang, University of California, Santa Barbara; A. Fedorov, Advanced Light Source, Lawrence Berkeley National Laboratory; M. Hashimoto, D. Lu, SLAC National Accelerator Laboratory; P. Taylor, P. Folkes, US Army Research Laboratory; C. Palmstrøm, University of California, Santa Barbara

α-Sn, the diamond structure allotrope of Sn, is a zero-gap semiconductor with band inversion. Calculations suggest that epitaxial tensile strain induces a transformation to a topological insulator (TI) phase, while epitaxial compressive strain induces a transformation to a Dirac semimetal (DSM) phase [1,2]. When this DSM phase is confined, it is suggested to form a quasi-3D TI phase [3]. There is little consensus, however, on exactly how or if these transitions occur. The α-Sn based system is expected to have less alloy disorder and anti-site defects compared to the typical (Bi,Sb)<sub>2</sub>(Se,Te)<sub>3</sub> TI system. Bulk α-Sn is also only stable at low temperatures, transformation temperature is raised above 100 °C by epitaxial stabilization of α-Sn(a=6.489 Å) on a closely lattice matched substrate like InSb (a=6.479 Å) [4]. Even at the low growth temperatures (<30 °C) necessary due to the phase transformation, incorporation of indium from the substrate as a *p*-type dopant in the epi-layer is difficult to prevent [5].

We first explore the essential role that surface preparation of the InSb(001) substrate has on both the quality and dopant density of the  $\alpha$ -Sn films. Through magnetotransport and ultraviolet photoelectron spectroscopy measurements, we find that growth on the Sb-terminated c(4x4) surface reconstruction results in higher mobility films with significantly reduced *p*-type doping. Using spin- and angle-resolved photoemission spectroscopy (ARPES), we study compressively strained  $\alpha$ -Sn films on InSb(001) at a range of film thicknesses. These measurements provide essential clarification to the band structure of  $\alpha$ -Sn: we observe the presence of a 3D TI-like phase in 13 bilayer films. Potential causes of this contradiction to the literature will be discussed.

With the previous behavior benchmarked, we then alloy the  $\alpha$ -Sn films with Ge to tune from low (-0.15%) compressive strain on InSb to multiple tensile strains (+0.5%, +0.8%, +1.3%) at the same film thicknesses. Morphology changes as a function of Ge alloying were studied with *in-situ* scanning tunneling microscopy, and strain was confirmed through XRD. Finally, the presence of topological phase transitions induced by tensile strain is studied via ARPES. Our results pave the way for a better understanding of the effect of strain and confinement on  $\alpha$ -Sn's band structure.

[1] Phys Rev B 97, 195139 (2018).

- [2] Phys Rev B 90, 125312 (2014).
- [3] Phys Rev Lett 111, 216401 (2013).
- [4] J Cryst Growth 54, 507 (1981).
- [5] Phys Rev B 105, 075109 (2022).

# QME-SaP-14 Electrostatic Gating of SrSnO<sub>3</sub> Thin Films with Improved Mobilities, *Zhifei Yang, F. Liu, T. Truttmann, B. Jalan,* University of Minnesota, USA

Ultra-wide-bandgap (UWBG) semiconducting oxides are becoming more crucial in sustainable technologies due to their promising use in applications including transparent electronics and power switching. Among them, alkaline earth stannates such as SrSnO<sub>3</sub> with the perovskite crystal structure have gained much interest in recent years. However, the roomtemperature mobility of SrSnO<sub>3</sub> thin films has been shown to be limited by defective surface scattering. By using a 4 nm undoped SrSnO<sub>3</sub> capping layer on 19 nm La-doped SrSnO<sub>3</sub> thin film, the measured room-temperature mobility has been shown to improve. In this structure, charge spill over from the doped layer to the undoped layer is expected to happen as Fermi levels equilibrate. Here, we demonstrate a reversible and electrostatic doping of SrSnO<sub>3</sub> films grown by Hybrid molecular beam epitaxy with tunable carrier densities using electric-double-layer transistor configurations with ion gels. Using modeling and a discrete two-channel model, we show that the modulation due to gating is confined within 4 nm at the top capping layer and the modulation leads to an increase of mobility in SrSnO<sub>3</sub> up to 130 cm<sup>2</sup>V<sup>-1</sup>s<sup>-1</sup> at 250 K. A detailed growth study combined with temperature-dependent Hall effect measurements and transport analysis will be presented.

QME-SaP-15 Epitaxial Growth and Transport Properties of Square-Net Rare Earth Telluride Thin Films, Adrian Llanos, J. Falson, California Institute of Technology

The square-net family of materials constitute a set of crystal structures which host a wide array of quantum phenomena including charge-density wave order, magnetism, superconductivity and topological band structures. The rare-earth ditellurides (RTe<sub>2</sub>) (Space group P4/nmm) are an especially exciting subclass of these materials whose structure consists of square-planar, conducting Te sheets interspersed with insulating R-Te corrugated layers. Due to their unique crystal structures and chemical tunability, the RTe<sub>2</sub> compounds offer opportunities to study the effects of topological phenomena in the context of broken-symmetry ground states. This research seeks to use molecular beam epitaxy to open new avenues for control of the low-temperature properties of these materials.

In this presentation, we describe our recent work on epitaxial growth of rare-earth telluride thin films LaTe<sub>2</sub> and DyTe<sub>2</sub> grown on MgO substrates. The good lattice match of DyTe<sub>2</sub> (+~1.7%) has enabled the growth of epitaxially strained films in the ultra-thin limit (<3 unit cells (uc)). Observation of RHEED oscillations along with measured surface roughness on the order of ~1uc indicates layer-by-layer growth. Out-of-plane X-ray diffraction shows intense peaks with prominent Laue fringes and rocking curve full width at half maximum of ~0.02°. Thin films of LaTe<sub>2</sub> have also been produced with comparably high structural quality yet are relaxed within 1uc due to the +7% mismatch with MgO.

Using grazing incidence X-ray diffraction, a modulated superlattice in the (hk0) plane has been observed in DyTe<sub>2</sub> and shows a V5xV5 modulation expected for Te-deficient DyTe<sub>2</sub>. Additionally, incommensurate modulations that cannot be indexed according to previously reported modulated structures have also been observed. The role of Te deficiency in producing these modulated structures and in relaxing epitaxial strain will be discussed.

Magnetotransport studies on LaTe<sub>2</sub> have revealed previously unobserved, non-saturated, negative magnetoresistance that persists to room temperature. The temperature dependence of resistivity also shows a strong dependence on growth temperature and charge density, likely resulting from variations in Te deficiency. We will discuss our current understanding of these phenomena informed by the unique features in the band structure as well as the complex defect chemistry found in these materials.

#### QME-SaP-16 Improving MBE (Bi,Sb)<sub>2</sub>(Te,Se)<sub>3</sub> Topological Materials Via Resonant and Magnetic Dopants, Patrick Taylor, Army Research Laboratory

The  $(Bi,Sb)_2(Se,Te)_3$  topological insulator (TI) system hosts robust Dirac-like topological surface states, but suffers from parasitic conduction in the bulk. This parasitic conduction channel limits the technology transition of these emergent materials into useful device technologies. In this work, we present results of the MBE growth, in-situ ARPES measurements, DFT calculations, and magneto-transport investigations of select  $(Bi,Sb)_2(Se,Te)_3$  alloys doped with Sn, a resonant dopant, [1,2] as well as Mn and Eu magnetic dopants whose function is intended to reduce bulk parasitic conduction.

The addition of tin is predicted by DFT to shift the Dirac point upwards in energy, and reduce the energy of the Fermi level out of the conduction band, and closer to the mid-gap Dirac states. Those DFT predictions were tested by in-situ ARPES measurements of MBE-grown films and are found to be consistent as shown in Figure 1.

Interestingly, as EuS is added during MBE growth as in independent flux, the carrier concentration drops and the mobility increases as the general transport behavior is consistent with significantly lower bulk conduction. Insitu ARPES measurements show the near complete absence of conduction band states. Figure 2 summarizes the magneto-transport and in-situ ARPES results from EuS doping. The significance of this work is that it highlights the potential for significant reduction in parasitic bulk conduction toward a loffe-Regel metal-insulator transition.

[1]C. Jaworski, J. Heremans, Phys. Rev. B 80, 233201 (2009)

[2]K. Kushwaha, et. al., Nature Communications, Vol. 7, No.11456 (2016)

QME-SaP-17 Growth of Cd<sub>3</sub>As<sub>2</sub> on GaAs(001), GaAs(110), and Si(001) Substrates, Anthony Rice, I. Leahy, A. Norman, K. Alberi, National Renewable Energy Laboratory

The three-dimensional Dirac semimetal Cd<sub>3</sub>As<sub>2</sub> has been shown to exhibit a variety of novel physics, providing a promising platform for their study. Thin film synthesis is enabling for scientific study as well as the realization of new devices, and growth has already been carried out on GaAs, GaSb, CdTe, SrTiO<sub>3</sub> and mica substrates. Due to its low energy (112) surface, however, the majority of thin film synthesis routes result in this orientation, while single crystals are limited by this cleave plane when performing studies requiring pristine surfaces. By expanding compatible substrate orientations, and ultimately the Cd<sub>3</sub>As<sub>2</sub> orientation, much more of the band structure may be probed via photoemission, and a broader range of device structures may be integrated with it.

Here, we present the design of II-VI buffer layers to template high quality Cd<sub>3</sub>As<sub>2</sub> in the (001) and (110) orientations on GaAs substrates. Latticematched Zn<sub>x</sub>Cd<sub>1-x</sub>Te buffers are known to reduce defects in Cd<sub>3</sub>As<sub>2</sub> epilayers grown on GaAs and improve their electron mobility [1]. We find that the addition of a ZnTe nucleation layers is critical for stabilizing Cd<sub>3</sub>As<sub>2</sub> (001) growth on GaAs (001) substrates, while Zn<sub>3</sub>As<sub>2</sub> nucleation layers are required to remove tilt in the  $Zn_xCd_{1-x}Te$  buffer when growing on GaAs (110). These films have a much different morphology due to the higher surface energy, and also a much different dependence on arsenic incorporation compared to Cd<sub>3</sub>As<sub>2</sub> (112). However, we show that the Cd<sub>3</sub>As<sub>2</sub> epilayers exhibit electron mobilities greater than 10,000 cm<sup>2</sup>/V-s. Finally, we present a methodology for growing Cd<sub>3</sub>As<sub>2</sub> (112) epilayers on GaAs (001) substrates using CdTe to switch between orientations [2], also allowing for integration with Si (001) [3]. Such schemes will allow for the design of Cd<sub>3</sub>As<sub>2</sub> orientation for specific measurement and application needs.

This work was performed by the National Renewable Energy Laboratory, operated by Alliance for Sustainable Energy, LLC, for the U.S. Department of Energy (DOE) under Contract No. DE-AC36-08GO28308. Funding was provided by the U.S. Department of Energy, Office of Science, Basic Energy Sciences, Division of Materials Sciences and Engineering, Physical Behavior of Materials Program under the Disorder in Topological Semimetals project.

[1] A. D. Rice, K. Park, E. T. Hughes, K. Mukherjee, K. Alberi. *Phys. Rev. Mat.* **3**, 121201(R) (2019)

[2] A.D. Rice, J. Nelson, A.G. Norman, P. Walker, K. Alberi,High Mobility  $Cd_3As_2(112)$  on GaAs(001) Substrates Grown via Molecular Beam Epitaxy. ACS Appl. Electron. Mater. **2022**, 4, 729

[3] A.D. Rice, K. Alberi. Crystals2023, 13, 578.

#### QME-SaP-18 Molecular Beam Epitaxy of Superconducting ZrN Thin Films on GaN Substrates, *Kevin D. Vallejo*, *D. Hurley*, *K. Gofryk*, *B. May*, Idaho National Laboratory

Group III-Nitride materials have found applications in optoelectronics and photonic devices due to the large variation in direct bandgap spanning from the infrared to the deep ultraviolet. Recent

research has pursued the integration of this well-established material system with transition-metal nitrides to create complex heterostructures with additional magnetic or superconducting functionality. ZrN is a wellknown refractory conductor with high oxidation resistance, high hardness, and has been shown to be a superconductor at temperatures <10K. The estimated lattice mismatch of ZrN with InN, GaN, and AlN is 8.5%, -1.5%, and -4.2%, respectively, suggesting strain free integration with In-based ternaries. This work focuses on the epitaxial growth of ZrN on c-plane GaN substrates via molecular beam epitaxy. An electron beam source was used to evaporate Zr, and an RF-plasma source supplied the active nitrogen. Reflection high energy electron diffraction (RHEED) and X-ray diffraction (XRD) did not reveal any crystallographic texture of ZrN deposited on fused silica at temperatures >700°C. However, growth of ZrN on cplane GaN substrates at similar temperatures was epitaxial. RHEED maintained similar hexagonal symmetry during the entirety of ZrN deposition and post-growth examination via XRD showed (111) oriented ZrN thin films. A physical properties measurement system (PPMS) was used to measure electrical transport as a function of temperature and magnetic field. Using a 4-point probe in a PPMS, initial results reveal that the epitaxial ZrN film is superconducting with a critical temperature of  $^{\sim}$  4 K and a critical field of 2 T. Because the critical temperature is lower than expected, the presence of off stoichiometry or structural disorder is suspected. These results pave the way for integration of superconductors and quantum phenomena in the III-

photonic

QME-SaP-19 Quasi Van Der Waals Epitaxy of Magnetic Topological Insulators on a Gaas (111) Substrate, Yuxing Ren, University of California at Los Angeles; L. Tai, 404 Westwood Plaza, Engineering VI, 310; K. Pan, S. Srivastava, Y. Xie, M. Goorsky, K. Wang, University of California at Los Angeles

Magnetic topological insulator could achieve quantum anomalous Hall (QAH) effect and spin-orbit torque (SOT) switching in the same structure. This is promising for its future applications in memory or switching with its robust surface properties bv topological protection. Considering the van der Waals nature of the epitaxial layer s.ithasvervweakvanderWaals bonding with the substrate. This gives rise to a novel auasi Van der Waals epitaxial growth mode attheinterfaceofGaAs(111) substratesand the epitaxiallayers, which has the advantages of both good crystallinity from substrate confinement, and a less influence from defects and roughness on the substrate surfaces. This is very crucial for achieving the quantization regime.

Inthisworkwehavedone hetero-epitaxy ofCr:(BixSb1-x)2Te3andother topological insulatorsonGaAs(111)substrates magnetic bv MBE(MolecularBeamEpitaxy). Unlike the pure Van der Waals epitaxy which has more freedom at the interfaces epitaxial layer and substrates, we found out that inthisquasi Van der Waals growthmode, strain exist and relaxesquickly within the1<sup>st</sup>epitaxial layer. While the surface defects quickly get screened within the 1<sup>st</sup> layer, the surface confinement also gives the epitaxial layer a uniform in-plane orientation which is important for achieving structure. а single crystalline Growthmechanismandtheinfluenceonitstransportproperties are also discussed.

#### QME-SaP-20 Transforming Rotating RHEED Data for Post-Growth Characterization Using Automated Machine Learning, Chris Price, Atomic Data Sciences

Reflection high-energy electron diffraction (RHEED) data is a characterization technique used to monitor the real-time surface structure and material morphology during epitaxial growth. Analyzing RHEED data over the growth duration can reveal a wealth of information about the relationship between the structure of the growing material and the growth procedure. In practice, this information is difficult to access and frequently neglected because transforming RHEED data into physically interpretable information is challenging and time-consuming. This is especially true if the growth stage is rotating, an important step to synthesize materials at device-relevant length scales with uniform growth across the substrate. Existing strategies to deconvolute the rotational and intensity oscillation frequencies must be calibrated to the materials system and measurement conditions and are brittle to structural changes during growth. We present an automated, material system-agnostic, and parameter-free approach to analyze rotating RHEED data. Using an entire unlabeled RHEED video as input, we extract the rotational frequency as a function of time in the video and create a complementary dataset averaged over the rotational period. The averaged data is used as input to a series of unsupervised dimensional reduction and clustering algorithms to identify transition points in the growth independent of rotation. Within each growth segment identified between these transition points, we extract original RHEED patterns at high symmetry scattering angles and quantify them using image segmentation models. Transitions are validated using small, labeled datasets of expertidentified growth transitions. The metrics automatically extracted at these angles are compared with the equivalent angles throughout the growth to label and quantify the evolution of the materials system. We align these quantified pattern metrics with in-situ environmental metrology data, such as quenching temperature, to build correlations between synthesized material structure and process variables. Fusing domain knowledge with machine learning, we reduce the time and effort barriers to accessing all the physical information collected with RHEED, producing physically interpretable datasets on materials structure over the course of a rotating growth.

### Sunday Morning, September 17, 2023

#### Workshop on Quantum Materials Epitaxy

Room Ballroom A - Session QME-SuM1

#### Oxides II

Moderator: Julia Mundy, Harvard University

#### 8:30am QME-SuM1-1 Design of Quantum Oxide Heterostructures, Nini Pryds, Technical University of Denmark INVITED

The wide range of fascinating properties observed in complex oxide continue to attract great interest such as ferro-, piezo- and pyroelectricity. Several strategies have been employed to break the lattice symmetry and expand the range of functionalities by design. Here, I will show and discuss a wide class of quantum materials, including strongly correlated oxides in the form of thin films and freestanding membranes. Using the freestanding oxide membranes, it is possible to create atomically thin stacks of oxide membranes with an extensive range of interfacial propeties some of which I will discuss during my talk. Ultimately, goal is to understand, control, and exploit the physical properties of quantum oxide heterostructures and their interfaces for next generation of electronic, information and energy.

#### 9:00am QME-SuM1-3 Interfacial Phenomena in 4d and 5d Transition Metal Oxides Grown by Metal-organic MBE, Ryan Comes, Auburn University INVITED

Complex oxides comprised of transition metal cations exhibit a host of intriguing properties for new technologies that can be tuned by the choice of ions from the 3d, 4d, and 5d blocks of the periodic table. Perovskite oxides with the chemical formula ABO3 have some of the richest behavior, where they can exhibit ferroelectricity, ferromagnetism, or superconductivity depending on the choice of B-site metal ion. This combination of properties in a single class of materials offers rich opportunities for engineering of unusual behavior through the design of multi-layer thin films that incorporate epitaxial strain and interfacial electronic band offsets. Using hybrid metal-organic molecular beam epitaxy (MBE), we are able to control these materials down to the atomic level so that interfaces between two different materials can be tuned to produce novel quantum phenomena. In this talk, we will show how novel behavior can be tuned and studied using in situ techniques to understand the film growth process and resulting functional properties. We have employed MBE and in situ X-ray photoelectron spectroscopy (XPS) to explore 4d and 5d oxide films that exhibit strong spin-orbit coupling and interfacial charge transfer. We have demonstrated the growth of hard-to-grow materials including SrNbO<sub>3</sub>, SrIrO<sub>3</sub>, and SrHfO<sub>3</sub> using metal-organic precursors and examined how interfacial phenomena can be tuned via charge transfer into materials such as BaSnO<sub>3</sub> and SrCoO<sub>3</sub>. Ongoing work focuses on the use of these materials to produce novel oxide heterostructures for topological phases and high electron mobility 2D electron gases.

9:30am QME-SuM1-5 Investigating the Electronic Structure of Coupled Electric Fields at the Surface and Buried Interface of an Epitaxial Complex Oxide/Group IV Semiconductor Heterostructure, Scott Chambers, Pacific Northwest National Laboratory; J. Ngai, University of Texas at Arlington; P. Sushko, Pacific Northwest National Laboratory; E. Ramirez, University of Texas at Arlington; T. Lee, D. Biswas, Diamond Light Source, UK INVITED We have probed the relationship between electron trapping at the surface and electron transfer across the interface of MBE-grown SrTiO3 and unintentionally doped Si(001). The latter, driven by shallow O donors in the near-surface region, gives rise to a 2D hole gas on the semiconductor side of the interface if enough charge transfer occurs to reach inversion. The former results in surface depletion within the top ~2 nm of the film as a consequence of charge trapping at the surface. By varying the composition of the film surface, we have found that charge transfer from Si to STO and thus hole gas formation in Si can be controlled. That is, the initiating step in charge transfer across the interface is charge trapping at the surface and first-principles modeling points to extra oxygen at the STO surface as being the electron trap that initiates the process. Surface compositional changes that prevent the trapping of extra oxygen at the surface quench charge transfer across the interface. As a result, the two electric fields constitute a coupled state that can be manipulated by means of surface composition engineering. Resonant soft x-ray photoemission near the Ti  $L_3$ -to- $e_g$ excitation yields valuable information on the electronic properties of gap states associated with trapped electrons at the surface and itinerant electrons in the subsurface region of the films. This measurement nicely complements hard x-ray photoemission that has been used to probe the coupled electric fields.

#### Workshop on Quantum Materials Epitaxy Room Ballroom A - Session QME-SuM2

#### **Topological and Magnetic Materials II**

Moderator: Prof. Chris Palmstrøm, University of California, Santa Barbara

# 10:30am QME-SuM2-9 Molecular Beam Epitaxy of Topological Semimetal Heterostructures, *Nitin Samarth*, Penn State University INVITED The landscape of topological quantum materials has expanded greatly with the discovery of topological Dirac states in both the bulk and surface of certain semimetals. This talk provides an overview of the synthesis by molecular beam epitaxy (MBE) of topological semimetal thin films (Cd<sub>3</sub>As<sub>2</sub>[1], ZrTe<sub>2</sub>[2], TaAs [3], NbAs [4] and their characterization using x-ray diffraction, angle resolved photoemission spectroscopy, and quantum transport. The potential application of these films for spintronics is studied by measuring spin to charge interconversion after interfacing them with conventional metallic ferromagnets (permalloy) or two dimensional

This work was supported by SMART/nCORE, a Semiconductor Research Corporation program, sponsored by NIST, the Institute for Quantum Matter (DOE EFRC grant DE-SC0019331) and the Penn State Two-Dimensional Crystal Consortium-Materials Innovation Platform (2DCC-MIP) under NSF DMR 2039351.

1. W. Yanez et al., Phys. Rev. Applied 16, 054031 (2021).

2. Y. Ou et al., Nat. Commun. 13, 2972 (2022).

ferromagnets (CrTe<sub>2</sub>).

3. R. Xiao et al., Phys. Rev. B 106, L201101 (2022).

4. W. Yanez et al., Phys. Rev. Applied 18, 054004 (2022).

#### 11:00am QME-SuM2-11 Controlling Magnetism in Layered Quantum Materials Through Designer Defects, Matthew Brahlek, Oak Ridge National Laboratory INVITED

Understanding how functional phenomena can be modified in epitaxial thin films is crucial for designing and manipulating properties. In this talk I will discuss several interesting examples where novel routes to control magnetic properties arose from understanding why defects form and ultimately how to control their formation. I will discuss the large electronic and magnetic response that is induced in the layered magnetic topological insulator MnBi<sub>2</sub>Te<sub>4</sub> by controlling the propagation of surface oxidation as well as native defects imparted during synthesis. I will also discuss how ferromagnetism can be externally turned on with a high level of continuous control through the application of low energy helium implantation in the ultra-high conductivity, non-magnetic layered oxide PdCoO<sub>2</sub>. These two examples highlight how a detailed understanding of synthesis by molecular beam epitaxy is critical to understanding and designing properties which is critical to driving new applications.

#### 11:30am QME-SuM2-13 Epitaxy of Rare Earth Compounds on Atomically Flat Surfaces, Joseph Falson, Caltech INVITED

In this presentation I will discuss the epitaxy of oxides and chalcogenides on atomically flat crystalline surfaces generated by high temperature laser annealing. In the case of rare-earth tellurides, we can induce large amounts of epitaxial strain and relieve this by tuning the thickness of films one monolayer at a time. Furthermore, I will discuss the role of laser heating in stabilizing off-stoichiometric oxide films. Finally, I will discuss ongoing efforts to reduce the residual impurity concentrations in ZnO-based heterostructures, where we expect laser heating to play a key role.

12:00pm QME-SuM2-15 Reactive Force Field Simulations as Versatile Tool to Explore the Growth Kinetics in Molecular Beam Epitaxy of Quantum Materials at the Atomic Scale, Roman Engel-Herbert, Paul Drude Institute, Germany INVITED

**Novel Materials** 

**Room Ballroom A - Session NM-MoM1** 

Oxide Semiconductors Moderator: Bharat Jalan, University of Minnesota

7:45am NM-MoM1-1 Welcome and Sponsor Thank Yous,

8:00am NM-MoM1-2 Art Gossard MBE Innovator Awardee Talk: Setting a New Quality Standard for Both Holes and Electrons in GaAs Ultra-High Mobility Quantum Wells, *Loren N. Pfeiffer*, Princeton University INVITED

8:30am NM-MoM1-4 Silicon-doped *B*-Ga<sub>2</sub>O<sub>3</sub> Films Grown at 1 µm/h by Suboxide Molecular-Beam Epitaxy, *Kathy Azizie*, *F. Hensling*, *C. Gorsak*, Cornell University; *Y. Kim*, Air Force Research Laboratory; *N. Pieczulewski*, Cornell University; *D. Dryden*, Air Force Research Laboratory; *M. Senevirathna*, *S. Coye*, Clark Atlanta University; *S. Shang*, Penn State University; *J. Steele*, *P. Vogt*, *N. Parker*, *Y. Birkhölzer*, Cornell University; *Z. Liu*, Penn State University; *M. Williams*, Clark Atlanta University; *K. Chabak*, Air Force Research Laboratory; *D. Muller*, Cornell University; *A. Neal*, *S. Mou*, Air Force Research Laboratory; *D. Schlom*, Cornell University

We report the use of suboxide molecular-beam epitaxy (S-MBE) to grow  $\beta$ - $Ga_2O_3$  at a growth rate of ~1  $\mu$ m/h with control of the silicon doping concentration from 5x10<sup>16</sup> to 10<sup>19</sup> cm<sup>-3</sup>. In S-MBE, pre-oxidized gallium in the form of a molecular beam that is 99.98% Ga<sub>2</sub>O, i.e., gallium suboxide, is supplied. Directly supplying Ga<sub>2</sub>O to the growth surface bypasses the ratelimiting first step of the two-step reaction mechanism involved in the growth of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> by conventional MBE. As a result, a growth rate of ~1  $\mu$ m/h is readily achieved at a relatively low growth temperature ( $T_{sub} \approx 525$ °C), resulting in films with high structural perfection and smooth surfaces (rms roughness of < 2 nm on ~1  $\mu$ m thick films). Silicon-containing oxide sources (SiO and SiO<sub>2</sub>) producing an SiO suboxide molecular beam are used to dope the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> layers. Temperature-dependent Hall effect measurements on a 1 µm thick film with a mobile carrier concentration of 2.7x10<sup>17</sup> cm<sup>-3</sup> reveal a room-temperature mobility of 124 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> that increases to 627 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> at 76 K; the silicon dopants are found to exhibit an activation energy of 27 meV. We also demonstrate working MESFETs made from these silicon-doped  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> films grown by S-MBE at growth rates of ~1 um/h.

# 8:45am NM-MoM1-5 Improving Si Dopant Control in n-type *8*-Gallium Oxide, *Brenton Noesges*, *Y. Kim, A. Neal, S. Mou, T. Asel*, Air Force Research Laboratory, Materials and Manufacturing Directorate, USA

Ultra-wide band gap materials such as  $\beta$ -gallium oxide ( $\beta$ -Ga<sub>2</sub>O<sub>3</sub>) are promising for high power electronic devices since breakdown voltage scales with band gap. Within this material class,  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> is unique since  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> can be grown from the melt and demonstrates n-type conductivity with carrier concentration controllable between 1016 -1020 cm-3 with low donor activation energies.<sup>1</sup> However, oxidation of dopant material affects source vapor pressures which can impact dopant profile uniformity in  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> grown via PAMBE.<sup>2,3</sup> This work is focused on optimizing uniform dopant profiles in the low Si doping regime (<10<sup>18</sup> cm<sup>-3</sup>) of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> films since Si doping concentrations show a gradient, increasing toward the surface of the thin films. SIMS and C-V measurements both show Si concentrations of ~3x10<sup>17</sup> cm<sup>-3</sup>. We also examine another aspect of Si incorporation in  $\beta$ - $Ga_2O_3$  that needs consideration when using traditional effusion cells as a Si dopant source. Previous work in PAMBE 8-Ga<sub>2</sub>O<sub>3</sub> growth demonstrated the importance of oxygen plasma power and Si cell temperature on the background amount of Si present in B-Ga<sub>2</sub>O<sub>3</sub> films.<sup>4</sup> In this work, we continued to explore sources of unintentional Si accumulation during the PAMBE growth process from sources like the quartz plasma bulb and Si effusion cell. The presence of Si at the interface between  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> substrate and film provides a parasitic conduction channel which is problematic for device performance. Attempts have been made to remove Si at 8-Ga<sub>2</sub>O<sub>3</sub> interfaces via etching. Our results indicate that removing interfacial Si may not be as simple since Si can re-accumulate during PAMBE, thus limiting the effectiveness of pre-growth surface treatments. Exposing a clean  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> surface to a quartz plasma bulb alone did not produce Si accumulation at the surface. On the other hand, a growth interrupt exposing a fresh  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> surface to a hot Si cell with shutter closed for several minutes produced Si accumulation equal to that of the interfacial Si between substrate and film. These results demonstrate that removing Si prior to loading into PAMBE may be inadequate to remove interfacial Si since Si can be re-introduced from the Si dopant cell during pre-deposition stages. These results point

toward important challenges and potential solutions when growing Sidoped  ${\cal B}\text{-}Ga_2O_3$  thin films.

1 Neal, A. T. et al., Appl. Phys. Lett. 113, 062101 (2018).

2 Kalarickal, N.K., et al., Appl. Phys. Lett. 115, 152106 (2019).

3 McCandless, J.P., et al. Appl. Phys. Lett. 121, 072108 (2022).

4 Asel, T. J., et al., J. Vac. Sci. Technol. A38, 043403 (2020).

9:00am NM-MoM1-6 The Effect of Gallium Beam Flux on Electron Transoport in 8-Ga<sub>2</sub>O<sub>3</sub> Grown via Plasma Assisted Molecular Beam Epitaxy, *Thaddeus Asel*, *B. Noesges, Y. Kim, A. Neal, S. Mou,* Air Force Research Laboratory, Materials and Manufacturing Directorate

β-Ga<sub>2</sub>O<sub>3</sub> has been of interest due to its large bandgap and high critical electric field, making it an excellent candidate for power electronic and RF applications [1]. The community has made significant improvements in the growth of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> via several techniques including molecular beam epitaxy (MBE). However, there has not been a study of the relationship between the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> growth parameters and their effects on the defects present in the crystal and their impact on the electronic transport properties. Utilizing temperature dependent Hall Effect measurements and a self-consistent fitting of the temperature dependent carrier density and mobility data, we are able to quantify the concentration defect states, including compensating acceptors, deep level impurities, and unintentional donors, present in the epitaxial films. The Ga beam flux can control the oxygen to Ga ratio present in the chamber as our oxygen is held constant for each growth run. This allows for growths that occur in the "oxygen rich" regime where the amount of Ga supplied limits the growth rate and the "gallium rich" regime where the amount of O supplied limits the growth rate. In both regimes the formation of the volatile suboxide Ga<sub>2</sub>O occurs, but in the gallium rich regime, the desorption of Ga2O causes a lower growth rate than that seen in the oxygen rich regime, due to the excess of Ga presence preventing the second reaction step in Ga<sub>2</sub>O<sub>3</sub> growth. Two preliminary samples were grown in the gallium rich (Ga Beam Flux =  $1 \times 10^{-7}$  Torr) and oxygen rich (Ga Beam Flux =  $6 \times 10^{-8}$  Torr) regimes. The samples had different doping densities of  $4.25 \times 10^{17}$  cm<sup>-3</sup> in the gallium rich sample and  $2.60 \times 10^{17} \text{ cm}^{-3}$  in the oxygen rich sample, this discrepancy is due to an inaccuracy in the Si doping source. The acceptor concentration in the gallium rich sample was calculated to be 7.26  $\times$  10  $^{16}\,cm^{\text{-3}}$  and 4.50  $\times$  10  $^{15}$ cm<sup>-3</sup> in the oxygen rich sample, a factor of 16 difference based on the Ga beam. This is likely due to Ga vacancies that can form during desorption of Ga<sub>2</sub>O during growth. These results indicate that there is significant impact on the electron transport properties of  $\beta$ -Ga-2-O3 with a change in the Ga to O ratio during growth, and that optimization of growth parameters is needed to optimize the electronic properties of MBE grown 8-Ga-2-O3.

9:15am NM-MoM1-7 Growth of α-(Al<sub>x</sub>Ga<sub>1-x</sub>)<sub>2</sub>O<sub>3</sub> by Suboxide Molecular-Beam Epitaxy, Jacob Steele, K. Azizie, N. Pieczulewski, J. McCandless, Cornell University; I. Matara Kankanamge, M. D. Williams, Clark Atlanta University; H. Xing, D. Jena, D. Muller, Cornell University; T. Onuma, Kogakuin University, Japan; D. Schlom, Cornell University (USA) and Leibniz-Institut für Kristallzüchtung (Germany)

Ga<sub>2</sub>O<sub>3</sub> has attracted significant interest due to its ultra-wide bandgap, high electron mobility, and large breakdown field. These properties exceed the current benchmarks set by materials such as SiC and GaN, making Ga2O3 optimal for next-generation power devices. Still, it has been proposed that the properties of Ga<sub>2</sub>O<sub>3</sub>can be extended further by alloying with Al to form (Al<sub>x</sub>Ga<sub>1-x</sub>)<sub>2</sub>O<sub>3</sub> which can raise the bandgap to 8.6 eV. This goal presents a challenge for the most researched phase,  $\beta$ , as  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>thermodynamically prefers a monoclinic structure and  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> is stable in the corundum structure. This structural mismatch limits the compositional range and the range of attainable bandgaps. In contrast,  $\alpha$ -Ga<sub>2</sub>O<sub>3</sub> occupies the corundum structure and has been shown to alloy over the full compositional range, enabling bandgaps from 5.3 - 8.6 eV. One method of growing α-(Al<sub>x</sub>Ga<sub>1-x</sub>)<sub>2</sub>O<sub>3</sub> is molecular-beam epitaxy (MBE). MBE is a powerful and highly controllable growth technique for  $\alpha$ -(Al<sub>x</sub>Ga<sub>1-x</sub>)<sub>2</sub>O<sub>3</sub> thin films with drawbacks being slow growth rates of a few hundred nm/h and narrow adsorption-controlled growth windows. One method to increase the growth rate is the technique of suboxide MBE, which allows growth of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>thin films at rates exceeding 1 µm/h with large adsorption-controlled growth regimes.

We show that suboxide MBE can be used for the epitaxial growth of high quality  $\alpha$ -(Al<sub>x</sub>Ga<sub>1-x</sub>)<sub>2</sub>O<sub>3</sub> thin films on A plane sapphire substrates over the full range of x at greater than 1 µm/h. For our study, gallium suboxide, Ga<sub>2</sub>O,

and elemental Al are the MBE sources. The oxidant is 80% distilled ozone which is held at constant pressure (5 x 10-6 Torr) while the Ga<sub>2</sub>O and Al fluxes are varied to control composition. We measure the composition of our films with XRD and confirm that we cover the full range of 0 < x < 1 with vacuum ultraviolet transmittance measurements showing that the bandgaps of our films shift from  $\alpha$ -Ga<sub>2</sub>O<sub>3</sub>to  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>. We show that the film composition can be controlled directly by the relative ratios of the Ga<sub>2</sub>O and Al fluxes. Our films have high structural quality as revealed by the full width at half maximum (FWHM) of rocking curves of the  $\alpha$ -(Al<sub>x</sub>Ga<sub>1-x</sub>)<sub>2</sub>O<sub>3</sub> films ranging from 11 - 15 arcseconds; these FWHMs are identical to the underlying sapphire substrates. The surfaces of the films are also smooth with RMS roughnesses measured by atomic force microscopy ranging from 0.3 - 1.1 nm on  $\alpha$ -(Al<sub>x</sub>Ga<sub>1-x</sub>)<sub>2</sub>O<sub>3</sub> films with thicknesses in the 17.8 - 47.8 nm range. We also show our progress with growing  $\alpha$ -(Al<sub>x</sub>Ga<sub>1-x</sub>)<sub>2</sub>O<sub>3</sub> films over 100 nm thick and with doping our films.

# 9:30am NM-MoM1-8 Electrostatic Gating of SrSnO<sub>3</sub> Thin Films with Improved Mobilities, *Zhifei Yang, F. Liu, T. Truttmann, B. Jalan,* University of Minnesota, USA

Ultra-wide-bandgap (UWBG) semiconducting oxides are becoming more crucial in sustainable technologies due to their promising use in applications including transparent electronics and power switching. Among them, alkaline earth stannates such as SrSnO3 with the perovskite crystal structure have gained much interest in recent years. However, the roomtemperature mobility of SrSnO3 thin films has been shown to be limited by defective surface scattering. By using a 4 nm undoped SrSnO<sub>3</sub> capping layer on 19 nm La-doped SrSnO3 thin film, the measured room temperature mobility has been shown to improve. In this structure, charge spill over from the doped layer to the undoped layer is expected to happen as Fermi levels equilibrate. Here, we demonstrate a reversible and electrostatic doping of SrSnO<sub>3</sub> thin films grown by Hybrid molecular beam epitaxy with tunable carrier densities using electric-double-layer transistor (EDLT) configuration with ion gels. Using modeling and a discrete two-channel model, we show that the modulation due to gating is confined within 4 nm at the top capping layer and the modulation leads to an increase of mobility in SrSnO<sub>3</sub> up to 130  $cm^2V^{-1}s^{-1}$  at 250 K. A detailed growth study combined with temperature-dependent Hall effect measurements and transport analysis will be presented.

#### Novel Materials Room Ballroom A - Session NM-MoM2

#### Nitrides

**Moderator:** Dr. Lutz Geelhaar, Paul-Drude-Institut für Festkörperelektronik Leibniz-Institut im Forschungsverbund Berlin

10:15am NM-MoM2-11 Molecular Beam Epitaxy of Binary and Ternary Manganese and Chromium Nitrides, K. Vallejo, K. Gofryk, Idaho National Laboratory; S. Gutierrez-Ojeda, Universidad Nacional Autónoma de México; G. Cocoletzi, Benemérita Universidad Autónoma de Puebla, Mexico; Brelon May, Idaho National Laboratory

Transition metal nitrides have exceptional properties and are used in a wide variety of electrochemical, structural, photochemical, and plasmonic applications. Among these compounds Mn- and Cr- nitrides have shown exceptional potential for magnetic sensing and spintronics. The Mn<sub>x</sub>N<sub>y</sub> system is complex with several different metastable phases both predicted and experimentally realized. Cr<sub>x</sub>N<sub>y</sub> has two primary phases, cubic (CrN) and hexagonal (Cr<sub>2</sub>N), which exhibit desirable mechanical, thermal, wear, anticorrosion, thermoelectric properties. Recent studies have provided valuable insights into the growth and formation of phases of both materials using various vapor deposition techniques. However, there are conflicting reports on the electrical and magnetic properties of Cr<sub>x</sub>N<sub>y</sub> which could be attributed to impurities, nitrogen vacancies, substrate effects, and strain. This controversy calls for a more detailed study and preparation of high-quality monocrystalline CrN to investigate he intrinsic physical properties. This study uses molecular beam epitaxy to synthesize epitaxial thin films of different Mn-N and Cr-N phases. The electrical and magnetic properties of these films are investigated with the rocksalt MnN and CrN both showing metallic behavior, with the latter showing a magnetic transition ~280K. However, when combining these materials at similar growth conditions, instead of maintaining the rocksalt structure, a new ternary cubic phase of Mn<sub>x</sub>Cr<sub>y</sub>N is obtained which shows narrow-gap semiconducting behavior. This work presents an avenue for the epitaxial integration of metallic, magnetic, and semiconductor materials.

10:30am NM-MoM2-12 Achieving Atomically Ordered GaN/AlN Quantum Heterostructures: The Role of Surface Polarity, *Yuanpeng Wu*, *P. Zhou*, *Y. Xiao, K. Sun, D. Wang, P. Wang, Z. Mi*, University of Michigan, Ann Arbor

A central goal of modern material physics and nanotechnology is the control of materials and their interfaces to atomic scales. However, for interfaces between polar layers, this goal is thwarted by the atomic substitution process among cations with different ionicities. In traditional semiconductor heterostructures, such as InAs/GaAs, Si/Ge, AlGaN/GaN and ABO<sub>3</sub> perovskites, diffusive interfaces have been widely observed, which deteriorates the performance of electronic and optoelectronic devices. Interfacial diffusion also prohibits achieving atomically ordered quantum heterostructures for applications such as quantum light sources and sensors. The studies on the origin of interfacial diffusion are often compounded by factors such as various synthesis parameters, available epitaxial substrates, strain distribution and surface reconstruction while a vital solution for achieving a perfect heterointerface remains elusive.

In this work, we discovered a strong dependence of interfacial diffusion on surface polarity in GaN/AIN quantum heterostructures. Atomically ordered quantum interface can be readily synthesized on the semipolar plane instead of the conventional c-plane of GaN/AIN heterostructures. The underlying mechanism of this dependence is explored through firstprinciples density functional theory calculations and it is found that the chemical bonding configurations at the semipolar plane can effectively eliminate the cation substitution process, which leads to an atomic sharp interface. The near-perfect interface quality ensures extreme quantum confinement and superior optical properties including record-high internal quantum efficiency of ~75% in the deep ultraviolet wavelength regime. We developed a scalable and robust fabrication method and demonstrated that electroluminescence energies of interdiffusion-free GaN are free from the quantum-confined Stark effect. In addition, we demonstrated a unique strategy of controlling surface polarities through different strain relaxation mechanisms in a core-shell nanostructure platform. This work provides, for the first time, a viable path for the synthesis of interdiffusion-free polar quantum heterostructures, which is paramount for high-performance devices across various material platforms.

10:45am NM-MoM2-13 Epitaxial Cubic Boron Nitride Grown by Ion Beam-Assisted Molecular-Beam Epitaxy on Diamond, *David Storm*, *S. Maximenko, A. Lang, N. Nepal, T. Feygelson, B. Pate, D. Meyer,* US Naval Research Laboratory

Cubic boron nitride (c-BN) shares several properties with diamond, including high mechanical hardness; high thermal conductivity, second only to diamond; and an ultra-wide band gap ( $E_g \sim 6.2$  eV, indirect). In addition, c-BN can be doped both *n*- and *p*-type, and the lattice mismatch between c-BN and diamond is only ~1.3%. These similarities suggest the potential for novel electronic devices based on c-BN/diamond heterostructures for high temperature and high power applications. However, the growth of device-quality layers of c-BN is challenging: boron nitride occurs in multiple phases; the desired cubic phase is metastable at pressures and temperatures typical of vapor-phase growth; and the absence of large-area bulk c-BN crystals necessitates heteroepitaxial growth on non-native substrates.

Single crystal epitaxial cubic boron nitride films were grown on (100) oriented IIa diamond substrates by ion beam-assisted molecular-beam epitaxy (MBE) in a custom MBE system equipped with an Ar ion source, a N<sub>2</sub> plasma source, and an electron beam evaporator for supplying elemental boron. The films are fully cubic, as indicated by Fourier transform infrared spectroscopy and corroborated by x-ray photoelectron spectroscopy. Transmission electron microscopy reveals an epitaxial c-BN film with the presence of isolated misfit dislocations but no indication of h-BN. The interface between the c-BN layer and the diamond substrate is structurally abrupt, and no interlayer between the c-BN film and diamond substrate is seen. It was found that trace amounts of impurities, such has Mg, Be, and Si, facilitate the growth of c-BN on diamond by ion-assisted MBE.

11:00am NM-MoM2-14 Optical Properties of ScN Layers Grown on Sapphire Using Plasma-Assisted Molecular Beam Epitaxy, *Duc V. Dinh*, Paul-Drude-Institut für Festkörperelektronik, Leibniz-Institut im Forschungsverbund Berlin e.V., Germany; *F. Peiris*, Kenyon College; *J. Lähnemann, O. Brandt*, Paul-Drude-Institut für Festkörperelektronik, Leibniz-Institut im Forschungsverbund Berlin e.V., Germany

Since the report of rock-salt scandium nitride (ScN) on sapphire 50 years ago [1], ScN has been mostly used in combination with other materials. In particular, ScN has been combined with wurtzite AIN to form (AI,Sc)N for surface acoustic wave [2] and high-electron mobility transistor applications [3]. However, very recent studies have shown that ScN itself can also be used for electronic [4] and infrared opto-electronic applications [5].

Here, we report on the optical properties of (3-250)-nm-thick ScN(111) layers grown on sapphire Al2O3(0001) substrates using plasma-assisted molecular beam epitaxy. The optical properties of the layers were investigated by three different techniques, namely, spectroscopic ellipsometry, confocal Raman and photoluminescence spectroscopies. The optical constants of the layers are investigated by variable-angle spectroscopic ellipsometry covering a spectral range from far infrared to far ultraviolet (0.045-8.5 eV). Refractive indices (n, k) of the layers are determined by fitting ellipsometry data using a parametric semiconductor model, taking into account the effects of surface roughness. Fits of ellipsometry data return the energies of four oscillators representing the band-to-band transitions. These correspond to the high-symmetry points in the band structure of ScN including 2.03 eV at the X point and three transitions (3.89, 5.33, and 6.95 eV) at the  $\Gamma$  point. These three oscillators are associated with direct transitions that occur between the degenerate heavy and light hole-bands and the first, second and third conduction bands at the  $\Gamma$  point, respectively. These energy transition values and refractive indices are consistent with theoretical studies previously reported for ScN [6]. Despite the rocksalt structure of ScN, Raman spectra of all the layers reveal several first-order phonon modes with an LO(L) mode at 675-680 cm-1. The appearance of these first-order modes is attributed to defects and impurities in the layers. Room-temperature photoluminescence measurements of the layers are dominated by a band with a peak energy decreasing from 2.3 to 2.2 eV, attributed to a reduction of the oxygen concentration in the thicker lavers.

#### References

[1]	N.	Scl	ar,	J.	Appl	. Pl	nys.	35,	1534	(1964)
[2]	M.	Akiya	ima	et	al,	Adv.	Mater.	, 21,	593	(2009).
[3]	D.V.	Dinh	et	al,	Appl.	Phys.	Lett.	122,	152103	(2023).
[4]	N.L.	Adams	ki e	et al,	Appl.	Phys.	Lett.,	115,	232103	(2019).
[5]	K.C.	Mau	ırya	et	al,	Nano	Lett.	22,	5182	(2022).
[6]	R.	Deng	et	al,	Phys.	Rev.	Lett.,	91,	045104	(2015).

11:15am NM-MoM2-15 Epitaxial Growth of High ScN Fraction ScAIN on NbN and SiC, Matthew Hardy, A. Lang, E. Jin, N. Nepal, B. Downey, V. Gokhale, S. Katzer, V. Wheeler, U.S. Naval Research Laboratory

ScAlN thin films have attracted significant attention due to their factor of five increase in piezoresponse over AlN for Sc<sub>0.43</sub>Al<sub>0.57</sub>N. Integration of metallic epitaxial NbN with ScAlN using molecular beam epitaxy (MBE) enables a pathway towards a highly conductive lower electrode while preserving high crystal quality even in relatively thin ScAlN films suitable for use at or above X-band frequencies. Maintaining phase-pure and high crystal quality Sc<sub>x</sub>Al<sub>1-x</sub>N at high *x* is critical to improve resonator bandwidth and reduce insertion loss.

In this work, we show the importance of layer nucleation—both an AIN interlayer, and the initial ScAIN layer—to the final crystal quality of MBE-grown ScAIN films on SiC and NbN/SiC. With the inclusion of a 5-nm AIN interlayer, the  $Sc_{0.32}Al_{0.68}N$  XRD FWHM decreases from 2.0° to 1.13°. An AIN interlayer is also critical to growth of ScAIN on NbN thin films. A two-step AIN growth process can effectively encapsulate the NbN layer while

providing a smooth surface on which to nucleate ScAIN growth, and is critical to maintaining high crystal quality for ScAIN grown on NbN.

The ScAIN initiation steps also have a strong impact on the final quality of the film. Instead of a two-step ScAIN growth we previously demonstrated, initiation using a linear composition grade from Sc0.32Al0.68N to Sc0.40Al0.60N over 100 nm leads to further improvements in the RHEED pattern, including a narrowing of the spots early in the growth, as well as elimination of remaining ring-like character in the final RHEED pattern after an additional 40 nm of growth, and a XRD FWHM as low as 1.22° for ScAlN films grown on SiC. Transmission electron micrographs show near elimination of cubic grains that otherwise form in the initial layers of the  $Sc_{0.40}AI_{0.60}N$ . The graded sample has the same average ScN fraction and thickness as the twostep sample. The grade thickness can be reduced to 25 nm (with the remaining 125 nm  $Sc_{0.40}AI_{0.60}N$ ) without degrading the XRD FWHM or RHEED pattern, increasing the average ScN fraction from 0.373 to 0.393. Finally, a 500-nm-total-thickness sample (100 nm  $Sc_{0.32}AI_{0.68}N \rightarrow$  $Sc_{0.40}AI_{0.60}N$ , 400 nm  $Sc_{0.40}AI_{0.60}N$ ) was grown to show the impact of defect annihilation in thicker films, resulting in a reduction of XRD FWHM to 0.89°. Employing the same 25-nm grade followed by 125-nm of Sc<sub>0.40</sub>Al<sub>0.60</sub>N grown on AIN/NbN/SiC results in a FWHM of 1.97°. The improved layer initiation shows that more gradual changes in surface energy and strain reduces the nucleation of undesirable cubic grains, and may point to a general strategy for elimination of anomalous grains in high ScN fraction ScAIN.

## 11:30am NM-MoM2-16 High Efficiency Micrometer Scale Green and Red Light Emitting Diodes, *Yixin Xiao*, *R. Maddaka*, *Y. Wu*, *Y. Malholtra*, *Y. Guo*, *S. Yang*, *J. Liu*, *K. Sun*, *A. Pandey*, *J. Min*, *Z. Mi*, University of Michigan

High efficiency light emitting diodes (LEDs) with characteristic length scales on the order of microns or less, also known as µLEDs, have been under intense investigations for their immense promise in various display and communications scenarios. Among the many material systems investigated for µLEDs, the III-nitride family possesses many desirable material properties such as comparatively low surface recombination velocities and excellent wavelength tunability. To date, however, it has remained a challenge to achieve efficient green and red emitting µLEDs, largely due to the enhanced surface recombination and poor p-type doping related to the top-down etching. Moreover, it has remained difficult to achieve the high levels of indium incorporation required for a red emitting indium gallium nitride (InGaN) active region. Here we demonstrate that the efficiency bottleneck of µLEDs can be fundamentally addressed by utilizing bottom-up III-nitride nanostructures. We report on the demonstration of micrometer scale green and red LEDs with an external quantum efficiency of 25% and 8%, respectively, which are the highest values ever reported to the best of our knowledge. We employ selective area plasma-assisted molecular beam epitaxy as the material synthesis platform. Due to efficient strain relaxation, such bottom-up nanostructures are largely free of dislocations. By exploiting the large exciton binding energy and oscillator strength of quantum-confined InGaN nanostructures, we show that the external quantum efficiency of a green-emitting micrometer scale LED can be dramatically improved from ~4% to >25%. The dramatically improved efficiency is attributed to the utilization of semipolar planes in strainrelaxed nanostructures to minimize polarization and quantum-confined Stark effect and the formation of nanoscale quantum-confinement to enhance electron-hole wavefunction overlap. We have further developed a new approach that included an InGaN/GaN short period superlattice together with an InGaN quantum dot active region to achieve high efficiency red emission. A maximum quantum efficiency of >7% was measured. Our studies offer a viable path to achieve high efficiency micrometer scale LEDs for a broad range of applications including mobile displays, virtual/augmented reality, biomedical sensing, and high-speed optical interconnects, that were difficult for conventional quantum well based LEDs.

11:45am NM-MoM2-17 AlN/AlGaN Short Period Superlattices With Sub 2 nm Layers Grown by MME, Alexander Chaney, Azimuth Corporation; C. Bowers, K. Mahalingam, UES INC; S. Mou, K. Averett, T. Asel, Air Force Research Laboratory, Materials and Manufacturing Directorate, USA

In this work we present a method for creating AIN/AI<sub>x</sub>Ga<sub>1-x</sub>N short period superlattices (SPSL) with individual layer thicknesses down to 3-4 ML. By taking advantage of the oscillatory nature of metal modulated epitaxy (MME), the formation of alternating AIN and AI<sub>x</sub>Ga<sub>1-x</sub>N layers can be achieved through the introduction of a constant Ga overpressure during the MME growth of AIN. However a typical MME process results in SPSLs with periods between 5 and 6 nm. In order to reduce the thickness of the AIN and AI<sub>x</sub>Ga<sub>1-x</sub>N layers, the shutter timings are reduced in order to limit the

total amount of Al deposited on the surface. Initial results with this shortened shutter timing showed Oth order SL peak with Pendellosung fringes in an XRD coupled scan. Such behavior is typically indicative of a high quality SPSL. However, TEM imaging showed a strong intermixing of between the AIN and Al<sub>x</sub>Ga<sub>1-x</sub>N layers, leading to a structure that more resembles a random Al<sub>x</sub>Ga<sub>1-x</sub>N alloy than a well ordered SPSL. In order to prevent intermixing at the interface layers, 3 altered growth processes were investigated: alternating Al and Ga shutters, inclusion of a N plasma only exposure step and reduction of the Ga flux. The goal with each of these changes was to limit the interaction of the Ga and Al during the final stages of the AIN layer formation. TEM examination of each growth's layer structures showed a significant reduction in layer intermixing for all samples. Alternating the Al and Ga shutter resulted in a SPSL with high order in its layer structure, with AlN and  $Al_xGa_{1-x}N$  thicknesses of 4 ML and 3 ML respectively. Introduction of the plasma only step resulted similarly improved interfaces however the thicknesses of AIN and Al<sub>x</sub>Ga<sub>1-x</sub>N layers had changed to be 3 ML and 4 ML respectively. Finally, reducing the Ga flux led to a layer structure almost identical to what was obtained using alternating shutters. XRD coupled scans each sample showed Pendellosung fringes centered on a Oth order peak. Fitting of the peaks enabled determination of the average Al composition for the SPSL, which ranged from a low of 84% for alternating shutters to a high of 90% for lower Ga flux. Because these 2 samples showed the same layer thicknesses in TEM, reducing the Ga flux resulted in higher Al content Al<sub>x</sub>Ga<sub>1-x</sub>N layers. A coupled scan of the N pause sample showed a broad peak to the left of the main alloy peak. The cause of this is a section of growth where the layered structure had non uniformity which can be seen in TEM indicating further optimization is needed. Based on these results, MME shows promise as method for forming digital alloys.

**MBE-Grown Devices** 

#### **Room Ballroom A - Session GD-MoA1**

#### **Photonic Devices**

**Moderator:** Prof. Dr. Minjoo Larry Lee, University of Illinois Urbana-Champaign

1:30pm GD-MoA1-1 Separate Absorption, Charge, and Multiplication Avalanche Photodiodes With InGaAs/GaAsSb Type-II Superlattices Grown by Molecular Beam Epitaxy, *Hyemin Jung, S. Lee,* The Ohio State University; *X. Jin, Y. Liu,* University of Sheffield, UK; *T. Ronningen,* The ohio state university; *C. Grein,* University of Illinois at Chicago; *J. David,* University of Sheffield, UK; *S. Krishna,* The Ohio State University

Avalanche photodiodes (APDs) can be used in remote sensing applications. such as atmospheric greenhouse gas monitoring, free-space optical communications, and medical diagnostics, including the extended shortwavelength infrared (eSWIR) range between 1.3  $\mu$ m and 2.5  $\mu$ m. APDs have internal multiplication characteristics that allow them to detect weak signals. However, traditional p-i-n structure APDs, commonly used for longer infrared wavelengths  $(1.55 - 3.4 \,\mu\text{m})$ , can be performance-limited by the high dark currents due to band-to-band tunneling when subject to high electric fields. A separate absorption, charge, and multiplication (SACM) structure can be employed to improve their performance. The SACM structure decouples the absorption region from the multiplication regions, maintaining the absorber region below the tunneling threshold while keeping a high electric field only in the multiplier. In this work, we designed, grew, and fabricated an eSWIR SACM APD. The epitaxial layers grown by molecular beam epitaxy consist of a 5 nm-In0.53Ga0.47As/5 nm-GaAs0.51Sb0.49(InGaAs/GaAsSb) type-II superlattice (SL) absorber and 1000 nm-thick Al0.85Ga0.15AsSb (AlGaAsSb) multiplier on a InP substrate. The development of the SL involved the growth of InGaAs with a growth rate of 1 µm/hr and a V/III beam equivalent pressure (BEP) ratio of ~10, along with GaAsSb grown at a rate of 0.47  $\mu$ m/hr and a V/III BEP ratio of ~20. Both were grown at a temperature of 470 °C, as measured by the Bandit system. Meanwhile, AlGaAsSb was grown at a rate of 0.6 µm/hr and a temperature of 500 °C. The X-ray diffraction (XRD) result of our devices is presented in Figure 1, which reveals a mismatch of -440 arcsec and -260 arcsec for AlGaAsSb and SL, respectively. The period of the SL was calculated from a distance between the satellite peaks, determined to be 9.75 nm. These devices have a cut-off wavelength of 2.4 µm, a gain of up to 60 at room temperature, and a resulting quantum efficiency of 600% at 2  $\mu$ m. as shown in Figure 3. Furthermore, the excess noise factor of our AlGaAsSb-based SACM APD stays below 2.2, up to a gain of 30. This value is considerably lower than competing technologies, with 2 times lower than InAlAs-based SACM APD and 1.2 times lower than InAlAsSb SACM APD on GaSb at the gain of 10. These results demonstrate the potential of AlGaAsSb-based SACM APDs with InGaAs/GaAsSb SLs for various eSWIR detection applications and provide insight into the design and fabrication of SACM APDs with good noise and sensitivity characteristics, making them suitable for a range of remote sensing applications in the eSWIR range.

1:45pm GD-MoA1-2 Growth of MWIR ICLEDs on Silicon using Molecular Beam Epitaxy, Mega Frost, T. Rotter, F. Ince, G. Balakrishnan, University of New Mexico; M. McCartney, D. Smith, Arizona State University; C. Canedy, W. Bewley, S. Tomasulo, C. Kim, U.S. Naval Research Laboratory; M. Kim, Jacobs Corporation; I. Vurgaftman, J. Meyer, U.S. Naval Research Laboratory Previously, interband cascade light-emitting diodes (ICLEDs) grown on GaSb substrates have been demonstrated as useful emitters in the mid-wave infrared (MWIR) region of 3 - 5 µm for room-temperature (RT) continuous wave (CW) operation [1,2]. Transferring this technology to growth on Silicon substrates would be advantageous for applications in chemical sensing and IR scene projectors (IRSPs), providing improved manufacturability through direct integration onto these circuits. This presentation will discuss the comparison of high-performance ICLEDs grown at NRL on GaSb/Si buffers that were grown at UNM and on lattice-matched GaSb substrates, including L-I characteristics, cross-section transmission electron microscopy (XTEM) and x-ray reciprocal space mapping (RSM).

The growth of GaSb/Si involves GaSb buffer layers which were grown on Silicon (001) with a 4° offcut towards (111). The native oxide was removed using a dilute HF solution to obtain a hydrogen-passivated surface. To

achieve III-V nucleation on Silicon, a ~10 nm thick AlSb layer was grown at a substrate temperature of 500°C followed by a 1  $\mu$ m buffer layer and an antimony cap to prevent oxidation. The GaSb/Si wafers were then transferred to NRL where an additional 2-3  $\mu$ m GaSb buffer and the ungrouped active ICLED stages were grown. This same 22-stage structure was grown on a GaSb substrate as a control sample. Accounting for differences in architecture, the ICLED structures grown on Silicon show efficiencies that are 75% of those measured in ICLEDs grown on GaSb. At 100 mA, 200- $\mu$ m-diameter mesas produce 184  $\mu$ W CW at 25°C and 140  $\mu$ W at

Threading dislocations were observed in GaSb buffer grown on Si from the XTEM images, showing a higher density near the Silicon substrate but reduced near the ICLED. Individual dislocations which reached the active ICLED layers exhibited a multiplying effect throughout the structure. Another growth artifact seen in these images was a slow-varying oscillation in the ICLED layers. Our presentation will provide a detailed explanation for both mechanisms and a comparison of the ICLEDs grown on Silicon to those grown on GaSb. Possible strategies for improving the epitaxial quality and device performance will also be discussed.

[1] C. S. Kim et al., Opt. Engr. 57, 011002 (2018).

[2] N. Schäfer et al., Opt. Engr. 58, 117106 (2019).

\* Author for correspondence: mdfrost@unm.edu [mailto:mdfrost@unm.edu]

2:00pm GD-MoA1-3 Monlithic Integration of InAs Quantum Dot Lasers with Silicon Photonic Waveguides, *Alec Skipper, K. Feng,* University of California Santa Barbara; *G. Leake, J. Herman,* SUNY POLY, Albany; *C. Shang, R. Koscica,* University of California Santa Barbara; *D. Harame,* SUNY POLY, Albany; *J. Bowers,* University of California Santa Barbara

Modern silicon photonic platforms promise to drastically improve bandwidth, data rate, and power consumption in data centers while data usage is rising rapidly every year. Silicon's indirect band gap makes it impractical for use in the lasers necessary for transmitting data, requiring integration with other materials such as III-V semiconductors. Monolithic integration by growth of III-V semiconductors on silicon is a promising pathway for cost-efficient production due to the large wafer sizes and ease of packaging and testing compared to hybrid approaches utilizing separate chips [1]. However, on-chip coupling between silicon photonic waveguides and directly grown III-V lasers shows extremely high insertion losses of 7.35 dB [2], significantly reducing the chip's overall performance.

After etching a window in a patterned silicon photonics wafer, a laser stack can be grown in the opening by molecular beam epitaxy to create a chip where silicon nitride-based waveguides are aligned with InAs quantum dot active regions. Careful calibration of etch and growth rates allows good control of the vertical alignment of the active region and waveguides, but typical MBE growth conditions result in a large horizontal gap between them. While high-performance lasers have been demonstrated on 300mm silicon photonics wafers [3], this gap makes coupling challenging and needs to be addressed for monolithic integration to be competitive with hybrid and heterogeneous silicon photonic implementations.

To improve the coupling between III-V lasers and silicon photonic waveguides, growth conditions must be tailored to minimize the gap between the laser's active region and the waveguide. Polycrystalline III-V material formed on the silicon dioxide sidewall during the growth of the GaAs buffer causes the crystalline laser stack material to facet and grow away from the sidewall. By increasing the growth temperature, decreasing the growth rate, and decreasing the arsenic overpressure during the growth of the GaAs buffer, the formation of polycrystalline material is suppressed and the horizontal gap between the active region and the waveguide can be reduced from ~5  $\mu$ m to ~1  $\mu$ m. Active-passive coupling tests are in-progress and will be reported at the conference.

 $\left[1\right]$  Z. Zhou et al., "Prospects and applications of on-chip lasers," eLight 2023.

[2] W.-Q. Wei et al., "Monolithic integration of embedded III-V lasers on Soi," Light: Science & Applications 2023.

[3] C. Shang et al., "Electrically pumped quantum-dot lasers grown on 300 mm patterned si photonic wafers," Light: Science & Applications 2022.

2:15pm GD-MoA1-4 Reducing Threading Dislocation Density of Pocket-Grown InAs Quantum Dot Lasers on Patterned SiO<sub>2</sub>/Si, *Rosalyn Koscica*, *C. Shang, K. Feng, J. Bowers*, University of California Santa Barbara

The push for scalable silicon photonics drives interest in monolithic integration of InAs quantum dot lasers on the same chip as passive waveguides. One integration method embeds silicon nitride waveguides in SiO<sub>2</sub> across a Si (001) wafer. The SiO<sub>2</sub> is patterned to form rectangular "pockets" where the III-V laser stack is grown. Pocket lasers were recently developed on 300 mm Si (001) wafers, but they have limited optical performance and thermal tolerance compared to devices grown on unpatterned "blanket" Si (001) substrate [1]. Identifying the source of this discrepancy is essential to produce blanket-level device performance within the monolithically integrated platform. One fundamental aspect is the material quality of the pocket-grown III-V epi.

Threading dislocations (TD) generated from the mismatch in III-V/Si lattice and coefficient of thermal expansion diminish laser performance and reliability. In blanket III-V on Si growth, thermal cyclic annealing (TCA) and insertion of dislocation filter layers (DFL) lowers TD density (TDD) by over two orders of magnitude compared to untreated growths [2]. In fully pocket-grown III-V epi on Si, it is impractical to apply TCA due to temperature control and uniformity challenges across a wafer surface containing mixed SiO<sub>2</sub>, polycrystalline III-V, and epitaxial III-V regions. Pocket-grown lasers lacking TCA have TDD on the order of  $10^7 \text{ cm}^2$ , suggesting TDD as a major contribution to limited performance compared to blanket-grown lasers.

Here, an annealed GaAs interlayer is inserted between the Si substrate and the patterned SiO<sub>2</sub> to reduce the TDD of in-pocket devices. A planar GaAs buffer is grown on a blanket GaP/Si (001) wafer and exposed to TCA before SiO<sub>2</sub> deposition. After SiO<sub>2</sub> and waveguide patterning, DFLs and the QD laser are grown inside the pockets. The modified structure retains a patterned SiO<sub>2</sub> device configuration while successfully reducing TD presence in pocket-grown material to mid- $10^6$  cm<sup>-2</sup>. Lasers with continuous-wave (CW) power of 15 mW and CW lasing up to 55 °C are demonstrated and compared to equivalent lasers grown on native GaAs substrates, showing the effect of TDs. The reduced TDD platform helps separate the known performance impact of TDD presence from subtler pocket-specific thermal or geometric effects that may also differentiate the characteristics of pocket and blanket lasers. Thus, these reduced TDD pocket-grown devices lead one step closer to realizing blanket-quality lasers for monolithic integration.

[1] C. Shang, K. Feng, E. T. Hughes, et al. Light Sci Appl 11, 299 (2022).

[2] C. Shang, Y. Wan, J. Selvidge, et al. ACS Photonics 8, 2555-2566 (2021).

2:30pm GD-MoA1-5 MBE Growth of Near-Infrared Heterojunction Phototransistors and Visible LEDs for Night Vision Applications, David Montealegre, University of Illinois at Urbana Champaign; Y. Song, Yale University; S. Lee, University of Washington; B. Kim, M. Kim, University of Illinois at Urbana Champaign; F. Xia, Yale University; M. Li, A. Majumdar, University of Washington; M. Lee, University of Illinois at Urbana Champaign

Night-vision goggles (NVGs) based on GaAs photocathodes and vacuumbased charge multipliers are highly sensitive with a long battery life exceeding 20 hours [1]. However, the weight and length of the bulk lenses, high-voltage transformer, and fiber bundle inverter lead to neck strain and fatigue for users. Furthermore, the response of the GaAs photocathodes cuts off at 870 nm and cannot fully use the night glow spectrum extending to 1.7 µm. The DARPA ENVISION program aims to create night vision systems that combine thin and lightweight meta-lenses with a solid-state NIR upconverter to reduce the neck torque from current NVGs. In this work, describe MBE growth of InP/InGaAs npn heterojunction we phototransistors (HPTs) and AlInGaP/GaAs visible LEDs, enabling us to demonstrate upconversion of 1.55  $\mu m$  light to 625 nm visible light. Rapid thermal annealing (RTA) of the LED greatly reduced the forward current needed for visible electroluminescence and promises to increase the upconversion sensitivity by ~6x. The high sensitivity and low power consumption of the devices demonstrated here are promising for nextgeneration night vision systems.

The HPT was grown on n-InP (001) and consisted of an n-InP:Si emitter (1.5  $\mu$ m, n = 1E18 cm<sup>-3</sup>), p-InGaAs:Be base (1  $\mu$ m, p=5E16 cm<sup>-3</sup>) and n-InGaAs:Si collector (500 nm, n = 1E18 cm<sup>-3</sup>). HPTs showed responsivity of ~700-1000 A/W over a wide wavelength range of 1.25-1.65  $\mu$ m and were sensitive to incident power densities as low as 12.2 nW/cm<sup>2</sup>; high responsivity is critical, as the HPT acts as a current source to drive the LED. The LED was grown on n-GaAs (001) and consisted of a single, lattice-matched 4 nm In<sub>0.48</sub>Ga<sub>0.52</sub>P

QW with 50 nm Al<sub>0.53</sub>In<sub>0.47</sub>P barriers emitting at 625 nm. Rapid thermal annealing at 975°C greatly improved the external quantum efficiency of the LED, enabling a strong reduction in the forward current needed to see light with the unaided eye. An HPT and LED (each on their own separate wafers) were configured in series such that the n-type emitter of the HPT was shorted to the n-type cathode of the LED, leading to clear upconversion for incident 1.55 µm laser power density as low as ~288 µW/cm<sup>2</sup>; in the nearterm, we anticipate reducing the required incident power by ~6x. In addition to improving the efficiency of the discrete components, future work will aim to realize imagers where each pixel consists of a series-connected HPT and LED, integrated through metal-metal wafer bonding.

#### 2:45pm GD-MoA1-6 Optically-Addressed Monolithically-Integrated Multiband Photodetectors Using Type-II Superlattice Materials, *Zheng Ju*, *X. Qi*, *A. McMinn*, *Y. Zhang*, Arizona State University

Multiband photodetectors and FPAs have been developed for various commercial and defense applications such as resources survey, chemical sensing, target seeking, and eye-safe imaging for autonomous automobiles. When implementing multiband photodetectors into an FPA with more than two bands, additional terminals for each pixel greatly complicate the FPA layout and device processing, decrease the fill factor, and increase the ROIC complexity.[1] It is therefore highly desirable to minimize the number of terminals so that the FPA can be integrated with off-the-shelf single-band ROICs. This talk reports the demonstration of multiband monolithically integrated optically-addressed photodetectors using GaSb and InAs/InAsSb type-II superlattices (T2SLs) to cover SWIR, MWIR, and LWIR detection range. The operating principle of the optical-addressing design is to use multiple optical biases on a stack of photodiodes (PDs) connected in series to switch detection bands. The detecting PD is the current-limiting device and determines the spectral response.

Several test structures have been grown on GaSb substrates for this study. The epitaxial growth of these samples starts with a GaSb buffer grown at 500 °C, followed by InAs/InAsSb superlattices overgrowth at 410°C. RHEED patterns for the buffer growths on GaSb(100) show steaky 3×1 surface reconstruction, while growing InAs/InAsSb superlattice show the alternative transitions between a streaky 2×4 and a streaky 3×1. XRD results show that the MBE-grown InAs/InAsSb T2SLs as MWIR and LWIR photodetectors are perfectly strain-balanced onto GaSb, and PL measurements show that the cutoff wavelengths are 6 and 10  $\mu m$  for MWIR and LWIR photodetectors, respectively. The SIMS measurements confirm that the Be (p-type dopant) and Te (n- type dopant) doping levels reach 1019 atom/cm^(-3) in both GaSb and barrier layers. Device fabrications are processed by sequentially applying top contact deposition, mesa etching, bottom contact deposition and annealing. The metal combinations of Ti/Pt/Au and Ge/Ni/Au are used for p-contacts and n-contacts, respectively. A mixture of HF, H2O2 and DI water is used for wet etching. Additionally, device performance such as dark current and spectral responsivity have been characterized and compared with the-state-of-art multiband photodetectors. More details will reported at the conference. he

[1] E. H.	Steenbergen.	Appl.	Phvs.	Lett.	97.	161111-161114	(2010).

+Author	for	correspondence:	zhengju@asu.edu
---------	-----	-----------------	-----------------

Novel Materials

**Room Ballroom A - Session NM-MoA2** 

#### **Strong Spin-Orbit Oxides**

Moderator: Prof. Ryan Comes, Auburn University

3:30pm NM-MoA2-9 Atomic Scale Modeling of the Hybrid Molecular Beam Epitaxy Growth Process Using Reactive Force Field Simulations, B. Yalcin, D. Yilmaz, A. van Duin, Pennsylvania State University; Roman Engel-Herbert, Pennsylvania State University, USA, Paul-Drude-Institute for Solid State Electronics, Berlin, Germany

The hybrid molecular beam epitaxy approach for the growth of oxides – i.e. the co-supply of atomic and molecular species in their elemental and metalorganic form – has allowed accessing a self-regulated growth window for complex oxides thin films with a range of chemistries, including titanates, vanadates, stannates, and ruthenates. While it is widely accepted that the favorable growth kinetics is enabled by the volatility of the metalorganic molecules supplied, their thermal decomposition and surface

reaction kinetics is complex and far from understood. For example, while it is generally assumed that the thermal decomposition of the widely employed metalorganic molecule titanium(IV)-isopropoxide (TTIP) used for the growth of titanates by hybrid MBE takes place by dissociating C-O bonds via the  $\theta$ -hydride elimination process, alternative reaction pathways, in particular in proximity of solid surface with different chemistries, such as SrO and TiO<sub>2</sub> terminated growth fronts, might be relevant as well.

In this talk, we present reactive force field molecular dynamics (ReaxFF-MD) and metadynamics simulations to shed light on the reaction kinetics of TTIP at the atomic scale. While the initial organic ligand pyrolysis step was spontaneous and occurred exclusively by breaking a C-O bond, albeit not always via  $\beta$ -hydride elimination in subsequent thermal decomposition stages, C-O bond dissociation events typically occurred with incomplete hydronation of the remaining Ti-containting fragment. In addition, Ti-O bond dissociation events were observed for Ti-containing molecule fragments at subsequent stages of the thermal decomposition process, challenging the simplistic picture of  $\theta$ -hydride elimination. The complete reaction scheme for the thermal decomposition of TTIP will be presented along with the reaction barriers and thermodynamic driving force for the different bond dissociation events and the role of the formal titanium oxidation state on the decomposition kinetics will be discussed, which offer a simple explanation to understand the different decomposition behavior of TTIP when interacting with the different terminations of SrTiO<sub>3</sub> surfaces. The computational approach provides a predictive and computationally inexpensive framework to identify chemical reaction pathways relevant to MBE film growth processes at the atomic scale under realistic, while experimentally relevant conditions. It can be easily expanded to different metalorganic precursor molecules, therefore allowing to develop a computationally informed engineering strategy to design MO molecules for hybrid MBE beyond the currently established chemistries.

3:45pm NM-MoA2-10 Engineering Metal Oxidation Towards Epitaxial Growth of Complex Iridates using Molecular Beam Epitaxy, *Sreejith Nair, Z. Yang, D. Lee, S. Guo,* University of Minnesota, USA; *J. Sadowski,* Brookhaven National Laboratory; *S. Johnson,* Auburn University; *A. Saboor,* University of Delaware; *Y. Li, H. Zhou,* Argonne National Laboratory; *R. Comes, W. Jin,* Auburn University; *K. Mkhoyan,* University of Minnesota, USA; *A. Janotti,* University of Delaware; *B. Jalan,* University of Minnesota, USA

The platinum group metals like Ir and Ru have captured significant interest in the condensed matter physics and materials science community due to the exotic electronic and magnetic properties that they exhibit when combined with oxygen. The oxides of these metals provide a unique platform to study and leverage the delicate interplay between electron correlations, crystal field and spin-orbit coupling energies. High quality thin films of complex platinum group metal oxides are hence, critical to realizing new phenomena such as the predicted unconventional superconductivity in Sr<sub>2</sub>IrO<sub>4</sub>. However, the platinum group metals have extremely low vapor pressures and low oxidation potentials. These factors make it challenging to synthesize their oxide thin films using an ultra-high vacuum (UHV) technique like Molecular Beam Epitaxy (MBE). Here, we have addressed these challenges using a novel solid-source metal-organic MBE approach [1,2]. We demonstrate atomically precise synthesis of binary IrO2 using Ir(acac)<sub>3</sub> as the metal-organic Ir source at substrate temperatures as low as 250 °C. The use of the metal-organic precursor allows Ir supply at source temperatures less than 200 °C and enables easy oxidation due to the +3 Ir oxidation state in the precursor. Further, by combining epitaxially strained IrO2 thin film growth on different substrates, x-ray diffraction, electron microscopy, spectroscopy techniques, and DFT calculations, we demonstrate a vital role of epitaxial strain in Ir oxidation. Thus, epitaxial strain can be an additional tuning knob to engineer metal oxidation which can aid the conventional thermodynamic and kinetic driving forces [3].

However, the true test of metal oxidation in UHV occurs at high growth temperatures where oxidation becomes increasingly thermodynamically unfavorable. Hence, in order to examine the efficacy of the solid-source metal-organic MBE approach and to realize the elusive unconventional superconducting state, we study the synthesis of Sr<sub>2</sub>IrO<sub>4</sub> thin films, which is favored at growth temperatures greater than 600-700 °C. We will present a detailed growth study, structural characterization, electrical and magneto-transport in epitaxial Sr<sub>2</sub>IrO<sub>4</sub> films, along with alternative ways to tackle the Ir oxidation challenge in UHV synthesis.

#### References:

[1] W. Nunn et al., "Solid source metal-organic molecular beam epitaxy of epitaxial RuO<sub>2</sub>", *APL Mater. 9, 091112 (2021)* 

[2] W. Nunn et al., "Novel synthesis approach for "stubborn" metals and metal oxides", *Proc. Natl. Acad. Sciences* 118, e2105713118 (2021)

[3] S. Nair et al., "Engineering Metal Oxidation using Epitaxial Strain"*Nat. Nanotechnol. (accepted) (2023).* 

# 4:00pm NM-MoA2-11 Solid Source Metal-Organic Molecular Beam Epitaxy for Epitaxial SrRuO<sub>3</sub> Films, *Anusha Kamath Manjeshwar, S. Nair, A. Rajapitamahuni, R. James, B. Jalan,* University of Minnesota

The investigation of the electrical and magnetic properties of SrRuO<sub>3</sub> and its associated Sr<sub>n+1</sub>Ru<sub>n</sub>O<sub>3n+1</sub> Ruddlesden-Popper phases requires a high degree of control over the isolation of the desired phase and its defect density. The growth of ruthenates is fundamentally challenging because ruthenium (Ru) resists scalable evaporation and oxidation. This bottleneck complicates the growth of SrRuO<sub>3</sub> films with low defect densities using inherently low-energy, ultra-high vacuum deposition techniques such as molecular beam epitaxy (MBE). Special modifications to conventional MBE such as electron-beam assisted evaporation and ozone-assisted oxidation of Ru have, so far, enabled the best defect control or the highest residual resistivity ratios (RRR =  $\rho_{300k}/\rho_{2k}$ ) in SrRuO<sub>3</sub> films among all physical vapor deposition techniques. However, these modifications are expensive and require additional interlocks to ensure safe operating conditions.

We outline a novel technique called solid source metal-organic MBE to supply a solid metal-organic precursor with pre-oxidized ruthenium with an effusion cell at T < 200 °C, a drastic decrease from the ~ 2000 °C required to produce comparable fluxes with elemental Ru. With this technique, we demonstrate the growth of phase pure, epitaxial, and stoichiometric SrRuO<sub>3</sub> films with robust ferromagnetism below 150 K on SrTiO<sub>3</sub> (001) substrates. We simplify the route to an adsorption-controlled growth window in SrRuO<sub>3</sub> films, growth conditions where the films can self-regulate their stoichiometry, which is a key ingredient for successful defect control in electron-beam and ozone-assisted MBE-grown SrRuO<sub>3</sub> films. We discuss the intricate relationship between cation stoichiometry, magnetic domains, and RRR in epitaxial SrRuO<sub>3</sub> films and outline new pathways for achieving low defect densities in SrRuO<sub>3</sub>. Using these guidelines to optimize stoichiometry and film thickness within a growth window, we achieve a RRR = 87 for a 50 nm-thick SrRuO<sub>3</sub> film, the highest for any SrRuO<sub>3</sub> film on SrTiO<sub>3</sub> (001) substrates. We will also illustrate how solid source metal-organic MBE is a simple and cost-effective method to enhance the capabilities of conventional MBE for the defect-controlled growth of ruthenates.

4:15pm NM-MoA2-12 Growth of Ruddlesden-Popper Ruthenates via Thermal Laser Epitaxy, Brendan D. Faeth, F. Hensling, V. Harbola, L. Majer, Max Planck Institute for Solid State Research, Germany; H. Boschker, Epiray GMBH, Germany; W. Braun, J. Mannhart, Max Planck Institute for Solid State Research, Germany

Thermal laser epitaxy (TLE) is a novel technique for thin film deposition which employs continuous wave lasers to simultaneously heat both the substrate and elemental sources. This laser heating approach allows for evaporation or sublimation of nearly all elements from the periodic table, ultrahigh substrate temperatures exceeding 2000 C, and broad compatibility with process gases at a wide range of pressures from UHV up to 1 Torr, among other benefits. As a result, TLE dramatically expands the parameter space available for thin film synthesis compared to existing epitaxy techniques. However, to date it has been experimentally challenging to achieve simultaneous control of multiple laser based elemental sources with the flux stability and systematic fidelity necessary for the growth of ternary or multernary systems of interest such as complex oxides.

In order to establish the capabilities of TLE for the growth of such complex oxides, we demonstrate here the successful epitaxial synthesis of several Ruddlesden-Popper phases of the Sr-Ru-O ternary oxide system via TLE. Near instant thermalization of both source elements and substrates from laser heating allows the process of thermodynamic phase control to be achieved rapidly during film deposition without the need for physical shuttering of sources.Additionally, we find that the "n=1" phase Sr<sub>2</sub>RuO<sub>4</sub> can be reliably synthesized at substrate temperatures in excess of 1200 C and in a background environment of pure molecular oxygen, within an adsorption-controlled growth window that is inaccessible to conventional MBE systems.We show that films grown under these conditions demonstrate high structural, electronic, and chemical quality comparable to that of MBE-grown films.A detailed accounting of the experimental approach, growth thermodynamics, and film characterization will be discussed.

This work not only demonstrates the feasibility of TLE for the synthesis of high-quality complex oxide thin films, but also suggests new routes to

achieving thin film growth in other materials systems that remain as-yet inaccessible to conventional epitaxy techniques.

4:30pm NM-MoA2-13 Growing Clean Crystals from "Dirty" Precursors in MBE, *Rashmi Choudhary*, University of Minnesota, USA; *Z. Liu, J. Cai, X. Xu, J. Chu*, University of Washington; *B. Jalan*, University of Minnesota, USA

Ultra-high purity elemental sources have long been considered a prerequisite for obtaining low impurity concentration in compound semiconductors in the world of molecular beam epitaxy (MBE). Furthermore, to realize intrinsic properties, the material needs to be nearly free of intrinsic and extrinsic defects. For this reason, the use of ultra-high-purity elemental sources has been the historical practice in MBE, perhaps, for the fear that impurity elements might get incorporated into the film, making it "dirty".

In this work, we challenge this conventional MBE wisdom by presenting an extension of the hybrid-MBE approach, known as solid-source metalorganic MBE, for growing superconducting Sr<sub>2</sub>RuO<sub>4</sub> films using a solid organometallic precursor, ruthenium acetylacetonate, as a source of Ru. We grew 100 nm thick (001) Sr<sub>2</sub>RuO<sub>4</sub> films on (001) LSAT substrate at 900°C substrate temperature using co-deposition of Sr, ruthenium acetylacetonate, and oxygen plasma. These films are phase-pure, singlecrystalline, fully coherent, and superconducting. The superconducting transition temperature of the film is 0.85 K. In contrast to the conventional MBE, which employs ultra-pure Ru metal evaporated at ~ 2000°C as a Ru source, along with reactive ozone to obtain Ru  $\rightarrow$  Ru<sup>4+</sup> oxidation, the use of ruthenium acetylacetonate precursor requires significantly lower temperature for Ru sublimation (less than 200°C) and eliminates the need for ozone.

This is the first-time realization of superconducting  $Sr_2RuO_4$  films using ozone-free MBE. By combining our results with the recent developments in hybrid-MBE, we argue that leveraging precursor chemistry will be necessary to realize next-generation breakthroughs in the synthesis of atomically precise quantum materials. Our results establish hybrid-MBE as a viable method for growing highest quality crystals and put this technique at the forefront of vacuum deposition technologies despite the use of a "dirty" chemical precursor.

4:45pm NM-MoA2-14 MBE Growth of BaCo<sub>2</sub>Ru<sub>4</sub>O<sub>11</sub>: A Metallic Ferromagnetic Hexaferrite, *Yilin Evan Li*, Department of Materials Science and Engineering, Cornell University; *M. Brützam*, Leibniz-Institut für Kristallzüchtung, Germany; *R. Cava*, Department of Chemistry, Princeton University; *C. Guguschev*, Leibniz-Institut für Kristallzüchtung, Germany; *D. Schlom*, Department of Materials Science and Engineering, Cornell University

Multiferroics with coupled magnetic and electric orders, although rare, hold potential for low-energy-consumption materials for logic and memory capable of electric-field control of magnetism. Barium hexaferrite (BaFe<sub>12</sub>O<sub>19</sub>) is the leading seller in today's multi-billion-dollar magnetic ferrite market and is predicted to gain electric polarization order at room temperature in addition to its robust ferrimagnetism under in-plane, biaxial, compressive strain [1]. Epitaxial strain engineering has achieved significant success in modifying materials' electric and magnetic ground states. For instance, the tensile strain exerted by a DyScO3 substrate transforms EuTiO<sub>3</sub>, an antiferromagnetic paraelectric material, into a ferroelectric ferromagnet [2]. The recent realization of single-crystal substrates of Sr1.03Ga10.81Mg0.58Zr0.58O19(SGMZ) [3], an insulator that is isostructural to BaFe12O19, enables straining BaFe12O19 as SGMZ has a ~1.1% smaller in-plane lattice constant. In addition to strain, to induce the ferroelectric state, an epitaxial bottom electrode is needed to control the electric state for this hexaferrite multiferroic candidate.  $BaCo_2Ru_4O_{11}$  is a metallic ferromagnetic oxide [4], belongs to the same hexaferrite family as BaFe12O19, and has small (~0.3%) in-plane lattice mismatch to the SGMZ substrate. Consequently, a coherent  $BaCo_2Ru_4O_{11}$  epitaxial thin film on the SGMZ substrate would be ideal for straining BaFe12O19and serving as the bottom electrode of a metal-insulator-metal structure to test for ferroelectricity in this predicted strain-induced multiferroic.

Epitaxial BaCo<sub>2</sub>Ru<sub>4</sub>O<sub>11</sub>thin films have not previously been reported, perhaps due to the lack of hexagonal ferrite substrates and the synthesis challenges associated with the growth of complex ruthenate epitaxial thin films. Films of BaCo<sub>2</sub>Ru<sub>4</sub>O<sub>11</sub> were grown by MBE on (0001) SGMZ substrates in an adsorption-controlled regime, similar to the way that films of SrRuO<sub>3</sub> and CaRuO<sub>3</sub> films with high resistivity ratios have been grown by MBE [5]. With matching substrates and adsorption-controlled growth, we have grown epitaxial BaCo<sub>2</sub>Ru<sub>4</sub>O<sub>11</sub>thin films with resistivity ratios comparable to BaCo<sub>2</sub>Ru<sub>4</sub>O<sub>11</sub> single crystals [4]. Wang, P. S., Xiang, H.J. *Physical Review X4*, 011035 (2014).
Lee, J. H. et al. *Nature*476, 114 (2011).

[3] Guguschev, C. et al. Crystal Growth & Design 22 (4), 2557-2568 (2022).

[4] Foo, M. L. et al. Journal of Solid State Chemistry 179, 563-572 (2006).

[5] Nair, H. P. et al. APL Materials6, 046101 (2018).

5:00pm NM-MoA2-15 Hybrid Molecular Beam Epitaxy of SrIrO<sub>3</sub> Films and Heterostructures, *Gaurab Rimal*, *T. Tasnim*, *J. Ahammad*, Auburn University; *G. Sterbinsky*, Argonne National Lab; *J. Sadowski*, Brookhaven National Laboratory; *M. Boebinger*, Oak Ridge National Laboratory; *R. Comes*, Auburn University

The iridiates form a fascinating class of oxides. The high spin-orbit coupling of Ir has been predicted to lead to novel physics such as topological phases and superconductivity in ternary iridate films and heterostructures. For investigating these effects, high quality materials, including epitaxial films, are needed. The low vapor pressure of Ir, however, precludes the use of effusion cells for MBE growth using elemental iridium. Recent progress in solid-source hybrid MBE, in which metalorganic precursors are used as a cation source, has evolved for high-quality MBE growth of films containing similar low vapor-pressure elements. In this study, we use MBE with a metalorganic iridium acetylacetonate source to grow the high-quality films of SrIrO<sub>3</sub> and heterostructures of SrIrO<sub>3</sub> with SrCoO<sub>3</sub>. Specifically, we discuss the role of different substrates on film properties via temperaturedependent transport, X-ray absorption and scanning transmission electron microscopy. We examine interfacial phenomena in SrIrO<sub>3</sub>/SrCoO<sub>3</sub> heterostructures for charge transfer and novel topological magnetic behavior.

MBE-Grown Devices

#### Room Hall of Ideas E-J - Session GD-MoP

#### **MBE-Grown Devices Poster Session**

#### GD-MoP-2 Parity-Time Symmetry Single-Mode Double-Microdisk InGaAs Quantum Dot Lasers, K. Lin, C. Xu, Tsong-Sheng Lay, National Chung Hsing University, Taiwan

We successfully demonstrate the parity-time symmetry (PT-symmetry) single-mode lasing operation of laterally coupled double-microdisk lasers. The microdisk lasers of disk diameter = 2.85 µm are fabricated by usingMBE-grown InGaAs quantum dots as the gain medium. The gain materials of dots-in-a-well (DWELL) structures are grown on (001) n<sup>+</sup>-GaAs substrate by molecular beam epitaxy. The wafer structure consists of a 1 µm-thick Al<sub>0.5</sub>Ga<sub>0.5</sub>As sacrifice layer, and an active layercomprised of a stack of six InGaAs DWELLs.In spite of the lasing output of multiple whisperinggallery modes (WGMs) from the single microdisk lasers, the laterally coupled double-microdisk lasers achieve single WGM lasing under gain-loss contrast pumping condition, literally pumping only one disk for the doublemicrodisk. We change the air gap distance (d) for the coupled doublemicrodisk structures to change the coupling strength (k) between the microdisks. Under single selective pumping (gain-loss contrast) at room temperature, the laterally coupled double microdisk lasers of d = 150 nm, and 200nm show single lasing mode at WGM m = 1,21 ( $\lambda$ = 1199nm). We also fabricate the double-microdisk lasers by coating the microdisks with  $HfO_2$  to change the coupling strength k. Under single selective pumping, the HfO<sub>2</sub>-coated double microdisk lasers show a single lasing mode at WGM m = 1,20 (λ= 1277nm).

GD-MoP-3 Impact Ionization Coefficients of AlxGa1-xAsSb (x=0 – 1) Lattice Matched to InP Substrates, *Seunghyun Lee*, The Ohio State University; *X*. *Jin*, University of Sheffield, UK; *H*. *Jung*, The Ohio State University; *J*. *David*, University of Sheffield, UK; *S*. *Krishna*, The Ohio State University

Impact ionization is a crucial process in the physics of semiconductors that influences the operation and performance of various semiconductor devices. It is utilized in avalanche photodiodes (APDs) to increase the signal-to-noise ratio, but it can also lead to avalanche breakdown in electronic devices. To ensure reliable device operation, it is vital to determine the impact ionization coefficients of electrons and holes ( $\alpha$  and  $\beta$ ), respectively. In this study, we present the  $\alpha$  and  $\beta$  for a range of Al<sub>x</sub>Ga<sub>1</sub>-xASSb compositions, covering x from 0 to 1, as determined through measurements of avalanche multiplication. Additionally, we explore the relationship between the impact ionization coefficients and the bandgap (E<sub>g</sub>) change ( $\Gamma$  and X points) along with the indirect-to-direct transition. This is because the impact ionization process is influenced by the material's band structure and the E<sub>g</sub>.

Four PIN Al<sub>x</sub>Ga<sub>1-x</sub>AsSb APDs with x of 0, 0.5, 0.65 and 0.85 were grown on InP using the RIBER Compact 21DZ molecular beam epitaxy, and were fabricated for electrical characterizations. The measured photocurrent spectra of the four APDs are presented in Fig. 1 (a), which illustrates that the cut-off tail moves toward lower energy as the x gradually decreases. To investigate the behavior of the E<sub>g</sub> with various Al compositions, the  $E_{g,T}$ , and  $E_{g,x}$  were extracted, as shown in Fig. 1 (b), and compared with the theoretical change in Eg proposed by Adachi. The discrepancy of Eg between the theory and experiment may come from the alloy disorder that can induce lower E<sub>g</sub> than expected in the theoretical calculation. The result suggests that the cross-over should happen around x=0.5 which is similar value predicted by Adachi.

The  $\alpha$  and  $\beta$  for Al<sub>x</sub>Ga<sub>1-x</sub>AsSb with x=0, 0.85, and 1 were plotted as a function of inverse electric field as shown in Fig. 2 (a). Fig. 2 (b) illustrates the  $\alpha$  and  $\beta$  for Al<sub>x</sub>Ga<sub>1-x</sub>AsSb with x=0, 0.85, and 1 as a function of x at 290 kV/cm. The  $\alpha$  remains fairly constant until x=0.85, where it jumps up at x=0, while the  $\beta$  gradually increases as the x decreases from 1 to 0. This suggests that the  $\alpha$  can change abruptly at a critical x point, and a similar point may exist for the rate of change in the  $\beta$ , as seen in other material systems such as AlGaInP and AlGaAs on GaAs. To gain more insight, we will explore the behavior of Eg and  $\alpha$  and  $\beta$  for additional x=0, 0.2, 0.4, 0.45, 0.50, 0.55, 0.65, 0.75, 0.85, and 1 in Al<sub>x</sub>Ga<sub>1-x</sub>AsSb. Knowing these coefficients and Eg parameters will allow engineers and scientists to design and optimize the performance of optoelectronic and electronic devices.

GD-MoP-4 Superconducting Germanium for Scalable Qubit Architectures, Patrick Strohbeen, A. Brook, J. Shabani, Center for Quantum Information Physics, New York University

As superconducting qubit platforms mature and algorithms demand everincreasing qubit numbers, it is becoming increasingly clear that platform scalability is an issue that must be addressed[1]. Current state-of-the-art transmon qubits simply take up too large of a footprint to reach the number of qubits required for the exciting proposed applications[1]. One proposed solution is to merge the large shunting capacitance with the nonlinear Josephson inductance into a singular circuit element[2]. This "mergemon" design is similar in concept to the original superconducting qubits in this design philosophy, however the junction area is significantly larger to accommodate the large shunt capacitance desired[2]. By increasing the area of the junction however, the contribution of the superconductor-insulator interface to the overall loss of the qubit becomes much more impactful. Thus, highlighting the need for new materials discovery and development to tackle these scalability challenges in new qubit designs.

In this talk, I will discuss our work in the Shabani lab on the development of one such superconducting material system: superconducting germanium thin films. We have previously shown attainable superconductivity via MBE growth[3] and will now discuss our work in the context of developing new qubits. The growth of these covalent superconductors by MBE is highly enticing for applications in such mergemon architectures due to the natural homoepitaxial growth relationship with lower microwave loss substrates. Growth of these superconducting films and characterization of the superconducting phase in context of qubit applications will be discussed.

[1] Y-.P. Shim and C. Tahan, Nat. Commun. 5, 4225 (2014).

[2] R. Zhao et. al, Phys. Rev. Applied 14, 064006 (2020).

[3] P. J. Strohbeen et. al, In Preparation (2023).

GD-MoP-5 Epitaxial Growth of High-Quality Aluminum Thin Films via Mbe for the Experimental Realization of Majorana Bound States, A. Elbaroudy, B. Khromets, E. Bergeron, T. Blaikie, Y. Shi, A. Tam, S. Sadeghi, F. Sfigakis, Zbig Wasilewski, J. Baugh, University of Waterloo, Canada

In-situ epitaxial AI on InAs\InGaAs shallow guantum well (QW) has become a promising material platform for condensed matter systems hosting Majorana Zero Modes (MZMs) and, ultimately, topological quantum computing. This is due to the high mobility, large Landé g-factor, and strong spin-orbit interactions (SOI) of the two-dimensional electron gas (2DEG) in InAs quantum wells as well as the relatively high critical value of the inplane magnetic field for very thin films of aluminum. It has been shown that with a sufficiently transparent Al/InGaAs interface, a proximity-induced superconducting gap in InAs approaches that of aluminum. Other benefits of AI are its large coherence length at low temperatures and its presence in III-V MBE systems; growing Al in situ produces an ultra-clean aluminum layer and a low-defect metal-semiconductor interface. However, growing a thin (-10 nm) continuous Al layer in standard MBE systems is challenging due to the high surface mobility of aluminum in a UHV environment, even at room temperature, and its tendency for 3D nucleation. In this work, we report a study of epitaxial Al thin film growth on InGaAs surface inside a standard Veeco GEN10 MBE reactor. We investigated the effect of Al deposition rate and substrate temperature on the quality of Al layers grown. Reflection High-Energy Electron Diffraction (RHEED) was performed simultaneously at four azimuths, and Band Edge Thermometry (BET) was used to monitor the substrate heating by the radiation from the Al source. Atomic Force Microscopy (AFM) and Scanning Electron Microscopy (SEM) were used to analyze the morphology of the Al films. X-Ray Reflectivity (XRR) and critical magnetic field measurements were performed to verify the thickness and superconductivity of the continuous smooth Al layers, respectively. The results give new insights into the kinetics of Al deposition and show that with sufficiently high Al flux, at close to room substrate temperatures, 2D growth can be achieved within the first few monolayers of Al deposition. This eliminates the need for complex LN2 cooling of the substrate and paves the way for the development of high-quality superconductor-semiconductor interfaces in standard MBE systems.

**GD-MoP-6 Submicron-Scale Light Emitting Diode with Efficient and Robust Red Emission**, *Yixin Xiao*, *R. Maddakka*, University of Michigan, Ann Arbor Light emitting diodes (LEDs) with characteristic length scales on the order of microns or less, also known as  $\mu$ LEDs, have been under intense investigations for their immense promise in various display and communications scenarios. Among the many material systems investigated

for µLEDs, the III-nitride family possesses many desirable material properties such as comparatively low surface recombination velocities and excellent wavelength tunability. To date, however, it has remained a challenge to achieve efficient red III-nitride µLED that is robust under different operational conditions, largely due to the material synthesis difficulties in the high levels of indium incorporation in the indium gallium nitride (InGaN) active region that are required for red emission. Here, we demonstrate, for the first time, a device synthesis strategy that enables robust and efficient red-emitting  $\mu\text{LED}$  with device dimensions near the submicron scale. We employ selective area plasma-assisted molecular beam epitaxy as the material synthesis platform and a combination of short-period superlattice, thick n-type GaN interlayer, and a relatively thick single-segment InGaN active region as the device structure to achieve nearsubmicron-scale µLEDs that emits at red wavelengths (>625 nm) over two orders of magnitude of current injection levels with 3% external quantum efficiency

GD-MoP-7 Impact of Built-in Electric Field Direction on Performance of GaN-Based Laser Diodes, *Henryk Turski*, Institute of High Pressure Physics PAS, Poland; *L. van Deurzen*, Cornell University; *M. Hajdel*, *M. Chlipala*, *M. Zak, G. Muziol, C. Skierbiszewski*, Institute of High Pressure Physics PAS, Poland

Nitride devices are mainly obtained along [0001] direction. That is why the internal polarization-induced electric fields in violet to green nitride light emitting diode (LED) and laser diode (LD) structures point in a direction opposite to what is desired for efficient flow of electrons and holes. This arrangement persists because of the need to have p-type layers on top of the structure to activate it and the lack of efficient structures grown along [000-1] direction.

To go around these problems one can use plasma-assisted molecular beam epitaxy (PAMBE), instead of metalorganic vapor phase epitaxy, to grow buried p-type layers and bottom tunnel junction (TJ) to invert current flow direction, with respect to the built-in polarization [1]. We also shown that p-type-down construction can be used to realize true-blue laser diodes, but obtained devices suffered from relatively high operating voltage [2].

In the present work, we report on optimization of the Ge-doping in PAMBE [3] for the growth of low resistance and high crystal quality TJs. The use of InGaN instead of GaN led to significant enhancement of Ge incorporation enabling the improvement in the operating voltage, bringing it to the similar level as for standard laser diodes without tunnel junction. This recent improvement in electrical performance of bottom TJ laser diodes opens the possibility to present the advantages of this constructions. Thanks to suppressed current overflow past the active region and the placement of p-type layers in close contact with the substrate bottom TJ laser diodes are expected to profit from lower optical losses and more efficient heat dissipation, respectively. Continues-wave operation of the cyan bottom TJ laser diodes will be presented. Comparison between p-up and bottom TJ devices will be discussed. Other device constructions built based on bottom тι design will he shown.

Acknowledgements: This work was supported partially by the Homing POIR.04.04.00-00-5D5B/18-00 project of the Foundation for Polish Science co-financed by the European Union under the European Regional Development Fund, the National Science Centre, Poland no. 2021/43/D/ST3/03266 and the European Horizon 2020 project VISSION (Grant ID:101070622).

#### Reference

[1] H. Turski, S. Bharadwaj, H. Xing, and D. Jena, Journal of Applied Physics **125**, 203104 (2019).

[2] H. Turski *et al.*, ECS Journal of Solid State Science and Technology **9**, 015018 (2019).

[3] H. Turski et al., Materials 15 (2022).

GD-MoP-8 Demonstration of a 4.32 μm Cutoff InAsSbBi nBn Photodetector, a Lattice-matched Random Alloy III-V Solution for Midwave Infrared Sensing, P. Webster, J. Logan, L. Helms, P. Grant, C. Hains, R. Carrasco, A. Newell, Air Force Research Laboratory, Space Vehicles Directorate; *Marko S. Milosavljevic, S. Johnson*, Arizona State University; G. Balakrishnan, University of New Mexico; D. Maestas, C. Morath, Air Force Research Laboratory, Space Vehicles Directorate

InAsSbBi *nBn* photodetectors are demonstrated that are lattice-matched to the underlying GaSb substrate with a 4.32  $\mu$ m wavelength cutoff at 150 K; *Monday Evening, September 18, 2023* 

that is 0.3  $\mu$ m longer than that of lattice-matched InAsSb at this temperature reflecting a 0.5% Bi mole fraction in the InAsSbBi active region. A low growth temperature was utilized to facilitate the incorporation of Bi, resulting in a minority carrier lifetime on the order of 24 ns in the InAsSbBi active region. Nevertheless, the detectors exhibits a quantum efficiency of 17% at 3.3  $\mu$ m wavelength with a dark current density of 50  $\mu$ A/cm<sup>2</sup> at 150 K and -0.4 V bias, and the strong photoresponse turn-on characteristic of a random alloy at 4.32  $\mu$ m wavelength and 150 K. A shot noise-equivalent irradiance analysis indicates that these detectors' dark-current-limited noise-equivalent irradiance of 10<sup>12</sup> cm<sup>-2</sup>s<sup>-1</sup> is 2 orders of magnitude greater than the Rule 07 expectation for this cutoff, and dark-current-limited shot noise-equivalent irradiance performance transitions to photon-limited at 1.7×10<sup>15</sup> photons/cm<sup>2</sup>s. [1]

P.T. Webster, J.V. Logan, L. Helms, P.C. Grant, C. Hains, R. A. Carrasco, A. T. Newell, M.S. Milosavljevic, S.R. Johnson, G. Balakrishnan, D. Maestas, C. P. Morath, Appl. Phys. Lett. 123, TBD (2023). DOI: https://doi.org/10.1063/5.0161051

GD-MoP-11 Zeeman Field-Induced Two-Dimensional Weyl Semimetal Phase in Cadmium Arsenide Thin Films, *Binghao Guo, W. Miao, V. Huang, A. Lygo,* University of California, Santa Barbara; *X. Dai,* University of Science and Technology, Hong Kong, China; *S. Stemmer,* University of California, Santa Barbara

We report a topological phase transition in MBE-grown, quantum-confined cadmium arsenide (Cd<sub>3</sub>As<sub>2</sub>) thin films under an in-plane Zeeman field when the Fermi level is tuned into the topological gap via an electric field. Symmetry considerations in this case predict the appearance of a two-dimensional Weyl semimetal (2D WSM), with a pair of Weyl nodes of opposite chirality at charge neutrality that are protected by space-time inversion ( $C_2T$ ) symmetry. We show that the 2D WSM phase displays unique transport signatures, including saturated resistivities on the order of  $h/e^2$  that persist over a range of in-plane magnetic fields. Moreover, applying a small out-of-plane magnetic field, while keeping the in-plane field within the stability range of the 2D WSM phase, gives rise to a well-developed odd integer quantum Hall effect, characteristic of degenerate, massive Weyl fermions. A minimal four-band k p model of Cd<sub>3</sub>As<sub>2</sub>, which incorporates first-principles effective g factors, qualitatively explains our findings.

Reference: B. Guo ... S. Stemmer, Phys. Rev. Lett. 131, 046601 (2023)

#### GD-MOP-12 MBE Grown InAs/GaAs Quantum Dot Columns as a Buffer Layer for Spatial and Spectral Homogeneity, *Nazifa Tasnim Arony*, *L. McCabe*, *J. Zide*, University of Delaware

MBE grown InAs quantum dots (QDs) on GaAs substrates have been widely studied<sup>1</sup> because of their wide range of applications in complex quantum devices including quantum sensors and quantum computers, since the QDs can serve as a basis for potential qubits<sup>2</sup>. Advanced quantum devices require spatial, spectral, compositional and structural homogeneity and scalability in the grown quantum dots that must have definitive locations in an array on the substrate. Non-templated self-assembled MBE grown QDs are not a feasible option in this regard due to their spatial and spectral randomness. There is ongoing research being conducted on effective methods to overcome the current challenges of producing defect-free, homogeneous and scalable QD platforms<sup>3</sup>. Recent work<sup>4</sup> has been done by our group on low-density site-controlled MBE grown InAs QDs on GaAs platform using nano-fabricated arrays of nano pits. However, achieving spectral homogeneity and thus, scalability is still a challenge because of the impurities introduced in the regrowth surface from the nanofabrication steps. Hence, we are exploring the domain of quantum dot columns (QDCs) as a buffer layer for the top QD-arrays of interest. In this process, the spatial homogeneity can be maintained by the templated QDs in the bottom layers while burying defects underneath the QDCs and potentially, scalable platforms for devices can be achieved.

**Novel Materials** 

#### Room Hall of Ideas E-J - Session NM-MoP

#### **Novel Materials Poster Session**

NM-MoP-1 A Study of the Effect of Substrate Misorientation on the Strain Relaxation of InSb Grown on GaAs (001), *Trevor Blaikie*, *M. Tam, Y. Shi*, University of Waterloo, Canada; *A. Rahemtulla*, *N. Appathurai*, *B. Moreno*, Canadian Light Source, Inc., Canada; *Z. Wasilewski*, University of Waterloo, Canada

High quality growths of InSb crystals are vital for advancing the fabrication of subwavelength plasmonic nanostructures for Terahertz (THz) applications. InSb is uniquely suited to applications with THz plasmonics because it is the only semiconductor that intrinsically supports the excitation of surface plasmons at THz frequencies.

GaAs (001) is chosen as the substrate because of its low cost and availability. Naturally, the high lattice mismatch between InSb and GaAs of 14.6% leads to high dislocation densities. The effects of substrate misorientation were studied by using substrates with two different offcuts. Sample A was grown with 0° misorientation from the (001) planes, while sample B has a 2° misorientation towards the [010] crystallographic direction.

A synchrotron X-ray beamline, a standard diffractometer, and a scanning electron microscope were used to characterize the two samples of InSb grown by molecular beam epitaxy on GaAs substrates. X-ray diffraction (XRD) and electron channeling contrast imaging (ECCI) were used to, independently, estimate threading dislocation density (TDD) in both samples.

TDD estimates from XRD and ECCI are nearly matched and show that there are significant differences in TDD between the two samples. The TDD of sample B was 63-74% of the TDD in sample A. This reduction in TDD is linked to the substrate misorientation.

ECCI also revealed that a high density of microtwin defects were created during the growth of sample A. From XRD, three-dimensional reciprocal space maps (3D RSMs) were created for both samples. The 3D RSMs for sample A revealed that these microtwins significantly broaden the full width at half maximum (FWHM) of the 004 InSb Bragg peak, but only if the direction of the X-ray beam is parallel to the microtwin boundary lines. Evidence of such microtwin defects was not present in the ECCI or XRD of sample B.

Additionally, a novel method is proposed to visualize the 3D RSMs, allowing for the effects of strain and tilt caused by defects to be independently studied. Measurements from the standard diffractometer revealed that the FWHM of the Bragg peak is anisotropic for both samples. This effect could not be explained by the occurrence of microtwins alone. It is proposed that the anisotropic FWHM is a result of two different types of dislocations,  $\alpha$  and  $\beta$ , that form in {111} glide planes. Glide velocities and nucleation energies are not equal in perpendicular directions. This leads to higher densities of  $\beta$  dislocations compared to the density of  $\alpha$  dislocations.

NM-MoP-2 2DEG Transport at the Interface of SrNbO<sub>3</sub>/BaSnO<sub>3</sub>, Brian Opatosky, S. Thapa, T. Tasnim, G. Rimal, P. Gemperline, Auburn University; S. Mahatara, New Mexico State University; H. Paik, University of Oklahoma; R. Vukelich, M. Giri, D. Hilton, Baylor University; B. Kiefer, New Mexico State University; R. Comes, Auburn University

Following confirmation of a high carrier concentrated 2D electron gas (2DEG) at the interface of SrNbO<sub>3</sub>/BaSnO<sub>3</sub> (SNO/BSO) via interfacial Nb 4d to Sn 5s injection, we investigate the transport properties of this 2DEG. As there can be transport contributions from the BSO 2DEG and the depleted SNO layer, we measure the carrier mobility via both temperature-dependent electronic transport and magnetic THz spectroscopy to decouple the contributions of the conducting pathways. In order to stabilize the material for measurement, we cap the SrNbO<sub>3</sub> film with a layer of SrHfO<sub>3</sub> (SHO), which provides an inert interface in terms of charge transfer. In establishing the transport properties of SNO/BSO, we provide a framework for future SNO interfacial studies.

NM-MoP-3 SrIrO<sub>3</sub> Films and Heterostructures Grown by Hybrid Molecular Beam Epitaxy, *Tanzila Tasnim*, Auburn University, Bangladesh; *G. Rimal, B. Opatosky*, Auburn University; *G. Sterbinsky*, Argonne National Laboratory; *M. Boebinger*, Oak Ridge Natinal Laboratory; *R. Comes*, Auburn University The 5d iridium-based transition metal oxides have sparked considerable recently due to their ability to host unusual and exotic quantum states, originating from strong spin-orbit coupling, electron correlations, and octahedral rotations. We utilized hybrid molecular beam epitaxy to grow semi-metallic SrIrO<sub>3</sub> films and heterostructures on different substrates such as SrTiO<sub>3</sub>, Nb-doped SrTiO<sub>3</sub>, and LSAT. The iridium was supplied through a metalorganic precursor, iridium acetylacetonate [Ir(acac)<sub>3</sub>]. The growth of the films was closely monitored using Reflected High Energy Electron Diffraction while the stoichiometry was characterized using in-situ X-ray Photoelectron Spectroscopy (XPS). To confirm the ideal growth window for the material, we used Rutherford Backscattering for comparison with XPS results. High-resolution X-ray Diffraction was used to determine the thickness of the films, lattice parameters, and in-plane coherence to the substrate. Scanning transmission electron microscopy studies were performed to investigate the strain-induced distortions and interfacial phenomena in the films. Ongoing work focuses on the synthesis of multilayer films with SrNbO<sub>3</sub> donor layers within SrIrO<sub>3</sub> films for interfacial charge transfer to produce novel electronic phases in the material.

NM-MoP-4 Characterization of MBE Grown Fe0.75Co0.25 in Composite Multiferroics, Katherine Robinson, Ohio State University; M. Newburger, M. Page, Air Force Research Laboratory; R. Kawakami, Ohio State University Composite multiferroics contain both ferromagnetic and ferroelectric layers and are promising candidates for future magnonics applications. These materials have generated much interest recently because they present the opportunity to efficiently control magnon generation and propagation via electrical methods. The ferromagnet Fe0.75Co0.25 has many attractive properties, such as a low growth temperature and metallic behavior, making it easier to detect magnetic properties of the material electrically. Fe0.75Co0.25 has low ferromagnetic damping, allowing for more straightforward study of magnon propagation, as well as a relatively high magnetoelastic constant.<sup>1-3</sup> This work studies the growth and properties of epitaxial Fe0.75C00.25 on ferroelectric materials by Molecular Beam Epitaxy and using Ferromagnetic Resonance (FMR), Brillouin Light Scattering (BLS), and Magneto-Optical Kerr Effect (MOKE). FMR and MOKE are utilized to determine the magnetic properties including damping parameters and coercivity while BLS illuminates the magnon dynamics and interactions. Leveraging the magnetoelastic nature of Fe0.75Co0.25, the multiferroic coupling is investigated by applying a voltage to the ferroelectric substrate, causing a strain unto the magnetic film, and altering the magnetic properties

#### References

1. Edwards, E. R. J., Nembach, H. T. & Shaw, J. M. Co<sub>25</sub>Fe<sub>75</sub> Thin Films with Ultralow Total Damping of Ferromagnetic Resonance. *Phys. Rev. Appl.* **11**, 054036 (2019).

2. Lee, A. J. et al. Metallic ferromagnetic films with magnetic damping under  $1.4 \times 10^{-3}$ . *Nat Commun* **8**, 234 (2017).

3. Schwienbacher, D. et al. Magnetoelasticity of Co<sub>25</sub>Fe<sub>75</sub> thin films. J Appl Phys **126**, https://doi.org/10.1063/1.5116314 (2019).

NM-MoP-5 Multicolor Micrometer Scale Light Emitting Diodes Monolithically Grown on the Same Chip, *Yifu Guo*, *Y. Xiao*, *Y. Malholtra*, *Y. Wu*, *S. Yang*, *J. Liu*, *A. Pandey*, *Z. Mi*, University of Michigan

Micro LEDs have emerged as a strong contender for next generation display devices due to their high efficiency, fast response, high brightness, and extended lifetime. For practical applications, it is highly desired that full color LEDs can be monolithically integrated on the same chip, which, however, has remained extremely challenging to achieve via the conventional quantum well based approach. In recent years, N-polar indium gallium nitride (InGaN) based light emitting diodes on the (sub)micron scale, also known as µLEDs, that are synthesized via selective area plasma assisted molecular beam epitaxy, have achieved record levels of efficiency at the (sub)micrometer device scale, with 25% external quantum efficiency (EQE) for green emission and 8% EQE for red. Such advances are enabled by selective area plasma assisted molecular beam epitaxy, in which, unlike thin film epitaxial growths, local kinetics can be controlled by substrate mask patterning. Moreover, the selective area openings on the substrate mask naturally lead to the formation of a photonic crystal. Here, we demonstrate the effect of pattern opening diameters on the InGaN photoluminescence (PL) wavelength. We show that, for a multiple-quantum-disk structure designed for green emission, given a certain photonic crystal lattice constant, the PL peak wavelength can vary over nearly 100 nm as the opening diameter varies over 60 nm, thereby enabling the achievement of multi-color emission for LED structures grown on a single chip in a single epitaxial step. More importantly, we have demonstrated strong coherent emission over a wide wavelength range for such nanowire photonic crystal LED structures. Their emission wavelengths can be precisely controlled and tuned by varying the

design and processing parameters. Such nanowire photonic crystal devices not only enable a wide range of wavelength tuning but also lead to high efficiency and highly directional emission which is desired for future neareye display applications. By further optimizing the design and epitaxial process, the realization of full-color emission for such unique N-polar IIInitride photonic nanostructures can be potentially realized. Work is currently in progress to demonstrate high efficiency micrometer scale green and red LEDs that can exhibit strong coherent emission.

#### NM-MoP-6 Bismuth Surfactant Enhancement of Surface Morphology and Film Quality of Low-Temperature Grown Gasb, Pan Menasuta, K. Grossklaus, J. McElearney, T. Vandervelde, Tufts University

Epitaxial growth of GaSb is critical for emerging mid-IR optoelectronics including thermal imaging, optical communications, LEDs, and thermophotovoltaic (TPV) cells [1-3]. Lower GaSb growth temperatures may be favorable for several reasons, ranging from compatibility with other layers that require low-temperature growth to lowered bulk mobility to prevent defects [4]. However, the surface of GaSb may degrade during growth at lower temperatures, leading to surface defects and device performance degradation. As the temperature decreases, the growth front transition from layer-by-layer to Stranski–Krastanov (SK) and eventually to the rough 3D-islanding regime. Furthermore, systematic characterization of homoepitaxial GaSb surfaces has not been done at temperatures beyond the range of 350°C to 450°C, not to mention in the presence of a surfactant [4-5].

We investigate the surface morphologies of two series of homoepitaxial GaSb(100) thin films grown on GaSb(100) substrates by MBE in a Veeco GENxplor system. The first series was grown at temperatures ranging from  $290^\circ\text{C}$  to  $490^\circ\text{C}$  and serves as the control. The second series was grown using the same growth parameters, with Bi used as a surfactant during the growth. We compared the two series to examine the impacts of Bi over the range of growth temperatures. AFM is used to characterize the surface morphology. The surface feature is investigated using SEM. Raman spectroscopy and ellipsometry are used to examine the films' properties. HRXRD is performed to analyze the film properties and any Bi incorporation. We found that the morphological evolution of the GaSb series grown without Bi is consistent with the standard surface nucleation theory, and we identified the 2D-3D transition temperature to be near 290°C. In contrast, the presence of a Bi surfactant during growth was found to significantly alter surface morphology and prevent undesired 3D islands at low temperatures. We observe a preference for hillocks over step morphology at high growth temperatures, anti-step bunching effects at intermediate temperatures, and the evolution from step-meandering to mound morphologies at low temperatures. This morphological divergence from the first series indicates that Bi significantly increases in the 2D Erlich-Schwöebel (ES) potential barrier of the atomic terraces, inducing an uphill adatom flux that can smoothen the surface. Our findings demonstrate that Bi surfactant can improve the surface morphology and film structure of low-temperature grown GaSb. Bi surfactant may also improve other homoepitaxial III-V systems grown in non-ideal conditions.

NM-MoP-7 Study the Temperature Effect on the Stability and Performance of III-Nitride HEMT Based Magnetic Fields Sensors , Satish Shetty, Institute for Nanoscience and Engineering, University of Arkansas; A. Kuchuk, 1Institute for Nanoscience and Engineering, University of Arkansas; H. Mantooth, Department of Electrical Engineering, University of Arkansas; G. Salamo, Institute for Nanoscience and Engineering, University of Arkansas

We investigated the reliability of Al<sub>0.34</sub>Ga<sub>0.66</sub>N/GaN micro-Hall-effect sensors under industry-relevant environmental conditions. The 2DEG formation heterostructure was grown on a GaN/sapphire template by molecular beam epitaxy. The performance and stability of Hall sensor was correlated by monitoring the Hall sensitivity, sheet density of two-dimensional electron gas, and contact resistance while the device was subjected to 200 °C thermal aging for a long-time duration of 2800 hours under atmospheric conditions. The stability and performance of Al<sub>0.34</sub>Ga<sub>0.66</sub>N/GaN micro-Hall sensors was evaluated by correlating electrical results with the microstructural evolution of the Al<sub>0.34</sub>Ga<sub>0.66</sub>N/GaN Hall sensor heterostructure. Overall, we have found that the design Al<sub>0.34</sub>Ga<sub>0.66</sub>N/GaN Hall-effect sensors structure has yielded a stable response for a prolonged 2800 hours of thermal aging at 200 °C. The output result of Hall device was evaluated in terms of Hall sensitivity and ohmic contacts, data shows very stable performance without any significant degradation. However, at the early stage of thermal aging we notice a small change in performance but after

subsequent aging sequence the performance was further stabilized and provided stable output Hall sensitivity for 2800 hours of thermal aging at 200  $^{\circ}$ C.

#### NM-MoP-8 Optimization of Heteroepitaxial ZnGeN<sub>2</sub>/GaN Quantum Wells for Green LEDs, *M. Miller*, Colorado School of Mines; *A. Rice*, National Renewable Energy Laboratory; *D. Diercks*, Colorado School of Mines; *A. Tamboli*, *Brooks Tellekamp*, National Renewable Energy Laboratory

Newly theorized hybrid II-IV-N<sub>2</sub>/III-N heterostructures, based on current commercialized (In,Ga)N light-emitting diodes (LEDs), are predicted to significantly advance the design space of highly efficient optoelectronics in the visible spectrum, specifically in the green to amber regions where LED efficiencies are orders of magnitude lower than other colors. Yet, there are few epitaxial studies of II-IV-N2 materials. ZnGeN2, a ternary analogue of the wide bandgap material GaN, is explored as a potential green-to- amber emitter which can be integrated into existing GaN LED heterostructures due to structural similarity. Cation-ordered ZnGeN<sub>2</sub> has a theoretical band gap of 3.4 eV, which is expected to be reduced with cation disorder. ZnGeN<sub>2</sub> is wurtzite when disordered, and is structurally and electronically similar to GaN, possessing a lattice mismatch of ~0.8%. Past work by this group has demonstrated epitaxial growth of ZnGeN2 on GaN and AlN via molecular beam epitaxy (MBE) [1,2]. Here we present the first growth of well-defined quantum wells of ZnGeN<sub>2</sub> within GaN by nitrogen plasma-assisted MBE, including successful five-layer multiple quantum well (MQW) structures.

Detailed structural and elemental analysis of the heterostructures was performed, including X-ray diffraction (XRD), scanning transmission electron microscopy (STEM), energy dispersive X-ray spectroscopy (STEM-EDS), and atom probe tomography (APT). These methods demonstrate high-quality and abrupt interfaces in the heterostructures, even after multiple repeating heterointerfaces. Through changes in growth methodology, we also demonstrate methods to improve unintentional incorporations, including associated improvements in structural quality. We include reports of a full LED stack growth, including n- and p-type GaN for carrier injection, an InGaN/GaN short-period superlattice, the ZnGeN<sub>2</sub>/GaN active region, and an AlGaN electron blocking layer. Together, this data demonstrates both the promise of heteroepitaxially integrated hybrid ternary/binary nitride systems along with the challenges associated with growing such systems, including an outlook on methods to improve the materials and devices.

#### References

[1] M. B. Tellekamp et al. Heteroepitaxial integration of  $ZnGeN_2$  on GaN buffers using molecular beam epitaxy. Crys. Growth Des. 2020; 20, 3, 1868–1875.

[2] M. B. Tellekamp et al. Heteroepitaxial ZnGeN<sub>2</sub> on AlN: Growth, Structure, and Optical Properties. Crys. Growth Des. 2022; 22, 2, 1270–1275.

#### NM-MoP-9 Machine Learning Analysis and Predictions of PAMBE III– Nitride Growth, Andrew Messecar, S. Durbin, R. Makin, Western Michigan University

There is considerable interest in applying machine learning techniques to optimize the synthesis of crystalline materials. Already, Bayesian optimization has been employed to optimize the molecular beam epitaxy (MBE) synthesis of SrRuO<sub>3</sub> and TiN thin films. Also, dimensionality reduction techniques and clustering algorithms have been applied to identify significant features in reflection high–energy electron diffraction (RHEED) patterns for a broad range of material systems, and convolutional neural networks have been shown to be useful in the classification of RHEED spot patterns for arsenide materials. Here, we explore how supervised machine learning techniques can be utilized to understand the relationships between the plasma–assisted molecular beam epitaxy (PAMBE) growth parameter space and the quality of GaN and InN thin film samples.

Data from over 100 PAMBE growth runs of GaN and InN (each) have been organized into material–specific data sets, including substrate temperature, metal source effusion cell temperature, initial N<sub>2</sub> pressure, and RF power. These variables were selected, as they are the direct system parameters a machine learning model would control. Each run took place in a Perkin– Elmer 430 MBE system equipped with an Oxford Applied Research HD–25 RF plasma source. RHEED was used as the primary quality metric, with crystallinity represented for the initial study by a binary numerical value (1 for monocrystalline and 0 for polycrystalline). The values of the growth variables were then mapped to this crystallinity label and other structural properties using supervised learning algorithms to perform both inference and prediction.

P-values corresponding to the growth parameters in each data set were measured with respect to the crystallinity; decision tree algorithms were fit to the same data for additional inference. Results from these two separate analyses were found to agree when deciding the most statistically significant synthesis variables: initial N<sub>2</sub> pressure and substrate temperature for GaN, and indium effusion cell temperature and initial N<sub>2</sub> pressure for InN. Supervised learning algorithms were subsequently trained on the synthesis data and used to predict the probability of growing monocrystalline and other metrics including the Bragg–Williams order parameter across a broad range of synthesis parameter values. The resulting machine learning–predicted growth maps agreed with conventional experimental wisdom while also providing new insight on the processing space for these materials.

This work was supported in part by the National Science Foundation (grant number DMR–2003581).

NM-MoP-10 Tuning the Emission Wavelength by Varying the Sb Composition in InGaAs/GaAsSb W-quantum Wells Grown on GaAs(001) Substrates, ... Zon, S. Voranthamrong, Department of Electrical Engineering, National Chung Hsing University, Taichung, Taiwan; C. Cheng, Department of Physics, National Central University, Chung-Li, Taiwan; Z. Lee, T. Lo, C. Liu, Department of Electrical Engineering, National Chung Hsing University, Taichung, Taiwan; C. Chiang, L. Hung, M. Hsu, Epileds Co., Ltd., Tainan, Taiwan; W. Liu, Department of Electrical Engineering, Yuan Ze University, Chung-Li, Taiwan; J. Chyi, Department of Electrical Engineering, National Central University, Chung-Li, Taiwan; Charles W. Tu, Department of Electrical Engineering, National Chung Hsing University, Taichung, Taiwan

Current vertical-cavity surface-emitting lasers (VCSELs) on cell phones for facial recognition are based on 940 nm VCSELs consisting of GaAs/AlAs distributed Bragg reflectors (DBRs) grown on GaAs(001) substrates. It is desirable to have longer-wavelength VCSELs, however, because the screen of a smart phone is transparent at longer wavelength (1380 nm) and because of eye safety. The maximum permissible exposure to the retina is higher for wavelength longer than 940 nm.

Long-wavelength lasers beyond 1300 nm is commonly fabricated on InP substrates, but InP-based VCSELs present many practical challenges. Thus, there is a great deal of interest in long-wavelength VCSELs, especially at 1550 nm, on GaAs substrates. Several approaches have been developed, including dilute nitrides, quantum dots, and antimonides. Here we explore strain-compensated GaAsP/InGaAs/GaAsSb W-quantum wells (W-QWs).

In this study, we investigate the effect of the Sb composition in GaAsSb on the photoluminescence (PL) wavelength. The tensile-strained GaAsP layer is inserted to compensate the compressive strain from the InGaAs/GaAsSb/InGaAs W-QWs. The samples are grown on GaAs(001) substrates by solid-source molecular beam epitaxy (MBE) with valved cracker cells for group-V materials.

Because of technical issues, our Sb flux is limited. We, therefore, vary the Sb composition in the range of 6-20% by controlling the growth temperature of GaAsSb, while the other parameters (thickness and composition) are kept constant for the In<sub>0.3</sub>Ga<sub>0.7</sub>As and GaAs<sub>0.66</sub>P<sub>0.34</sub> layers. All samples are grown at 520°C, except during the growth of GaAsSb. The higher Sb composition is realized at lower growth temperature of GaAsSb.

X-ray rocking curve (XRC) measurements and simulations are performed to investigate the material composition and layer thickness. Low-temperature photoluminescence (PL) spectra are consistent with the XRC results. At the lowest Sb composition of 6%, the PL intensity is the strongest, and room-temperature PL is realized at ~1100 nm. By increasing the Sb composition in the GaAsSb layer, low-temperature (20 K) PL emits at longer wavelength up to ~1400 nm at 20% Sb while the PL intensity is the weakest. The XRC is also degraded.

In summary, this study shows that the composition of the GaAsSb layer, which can be controlled by its growth temperature, is an important factor to tune the PL wavelength. When the Sb composition is higher, the lattice mismatch with GaAs is larger, resulting in larger strain. To maintain the sample quality then requires adjusting the layer thickness of the GaAsP strain-compensation layer. This work is in progress.

# NM-MoP-11 Strong Correlation in Two-Dimensional 1T- NbSe<sub>2</sub>, Joy Hsu, R. Birchmier, M. Altvater, V. Madhavan, University of Illinois at Urbana-Champaign

Two-dimensional 1T-phase NbSe<sub>2</sub>, a strongly correlated system, has drawn enormous attention since it was predicted to be a candidate to host quantum spin liquid.<sup>[1]</sup> However, the insulating mechanism of 1T-NbSe<sub>2</sub> is still unclear, and there is ongoing debate regarding whether the gap is

dominated by Mott physics or charge transfer within each charge density wave (CDW).<sup>[2,3]</sup> More experimental studies need to be conducted to determine the potential of 1T-NbSe<sub>2</sub> to support a quantum spin liquid.

In this work, monolayer and bilayer 1T-NbSe<sub>2</sub> were grown with molecular beam epitaxy method and investigated with scanning tunneling microscopy (STM). During the growth, the film was monitored by *in situ* reflection highenergy electron diffraction, and a quenching treatment was applied to ensure retaining of the 1T-phase. The sample was further transferred to 4K-STM *via* a vacuum suitcase to avoid contamination. At low temperature, 1T-NbSe<sub>2</sub> experienced a CDW transition and displayed ordered triangular superlattice with start of David motifs, which were clearly shown by our 4K-STM. The density of states of monolayer and bilayer 1T-NbSe<sub>2</sub> was measured with scanning tunneling spectroscopy, and the gap character was discussed. Our measurements reveal that the gap feature is very sensitive to local perturbations, including CDW domains, defects, and interlayer coupling. In summary, we achieved controlled growth of monolayer and bilayer 1T-NbSe<sub>2</sub> and shed light on the delicate modulation of correlation-driven insulating states.

#### Acknowledgment

The work was supported by the National Science Foundation through grant DMR-200378, with partial support from the Gordon and Betty Moore Foundation through EPiQS grant 9465.

#### References

[1] G. Misguich, C. Lhuillier, B. Bernu, C. Waldtmann, *Phys. Rev. B* **1999**, *60*, 1064.

[2] M. Liu, J. Leveillee, S. Lu, J. Yu, H. Kim, C. Tian, Y. Shi, K. Lai, C. Zhang, F. Giustino, *Sci. Adv.* **2021**, *7*, eabi6339.

[3] Y. Nakata, K. Sugawara, R. Shimizu, Y. Okada, P. Han, T. Hitosugi, K. Ueno, T. Sato, T. Takahashi, *NPG Asia Mater.* **2016**, *8*, e321.

#### NM-MoP-12 Growth of Cobalt-containing Compounds for Back-End-of-Line Interconnects, Yansong Li, G. Zhou, C. Hinkle, University of Notre Dame

The resistivity of conventional metal interconnects increases rapidly with decreasing size, which greatly limits the performance of devices and causes high energy consumption. Electron scattering at surfaces and grain boundaries are found to be the main causes for this size effect. To solve this size effect issue, we synthesized two types of cobalt-containing materials, topological semimetal CoSi and delafossite metal PtCoO2, which could be promising candidates to replace currently used conventional metals. CoSi, a topological semimetal with multifold fermions, possesses unique topologically protected surface states that are expected to decrease resistivity at scaled dimensions where surface transport dominates. Here we demonstrate the growth of CoSi thin films and single-crystal CoSi nanowires by MBE. Multiple characterization techniques including RHEED, HRXRD, Raman microscopy, and TEM are utilized for optimizing growth conditions and realization of single-phase CoSi growth. Another candidate PtCoO2, a delafossite metal with an anisotropic 2D fermi surface and layered structure, is expected to have a very low in-plane resistivity even at ultra-downscaled dimensions. We achieved highly conductive PtCoO2 thin films by the technique combining shutter-controlled MBE growth and postgrowth annealing. Apart from the characterization techniques mentioned above, XPS and XRF are also utilized to detail chemical information and optimize the stoichiometry. We will show resistivity vs. dimension data for both CoSi and PtCoO2 and provide an outlook for using these materials as scaled interconnects.

NM-MoP-13 Development of Al<sub>x</sub>Ga<sub>1-x</sub>As<sub>1-x</sub>Bi<sub>Y</sub> for the Next Generation of APDs, *Matthew Carr*, *N. Bailey*, University of Sheffield, UK; *M. Sharpe*, *J. England*, University of Surrey, UK; *R. Richards*, *J. David*, University of Sheffield, UK

Alloying Bismuth into GaAs has been shown to produce a marked alternation of the valence band structure. The increased spin orbit splitting energy in Ga<sub>1-x</sub>AsBi<sub>x</sub> has been shown to dramatically reduce the excess noise of GaAs APDs<sup>[1]</sup>. Higher performing III-V APDs could yet be achieved by Bi alloying with Al<sub>0.8</sub>Ga<sub>0.2</sub>As potentially promising a new family of ultralow-noise, photodetectors. Isolating the effect that Al may have on the incorporation of Bi will be of benefit opening up other potential material systems. This could include telecommunication APDs based on the inclusion of Bi into InAlAs, lattice matched to InP.

This study aims to investigate the synthesis and growth optimisation of Al<sub>x</sub>Ga<sub>(1-x)</sub>As<sub>(1-y)</sub>Bi<sub>y</sub> with a viewto understand how adding Bi to an Al containing alloy affects its material properties. We present a series of

 $AI_xGa_{(1-x)}As_{(1-y)}Bi_{\gamma}$  structures, grown in an Omicron MBE STM reactor. Crystallographic and optical material quality was assessed using X-ray diffraction, photoluminescence, Rutherford backscattering and time of flightmeasurements.

Samples of Al<sub>x</sub>Ga<sub>(1-x)</sub>As<sub>(1-y)</sub>Bi<sub>y</sub> with between 0-80% Al and up to 6.2% Bi were synthesised successfully. The incorporation efficiency of Bi was unaffected by the group III substitution of Ga for Al. The inclusion of 2.5% of Al into the ternary GaAs<sub>0.975</sub>Bi<sub>0.025</sub> showed an acute reduction in the optical quality with thePL intensity reduced by a factor of 36, with further degradation at increased Al concentrations up to 30% with loss of optical activity. Improvements to optical quality and wafer homogeneity were observed with annealing for 30s at temperatures between 400-600°C under N<sub>2</sub>. Beyond 600°C optical quality decreased by a factor of 0.5. The bandgap reduction caused by Bi incorporation is strikingly similar to GaAs. There is a strong relation between Bi incorporation and the key growth parameters of temperature and Bi flux that is also akin to those observed in GaAs[2]. Growth temperature variation by 60 °C alone altered Bi content in the between 0.8 – 6.2%.

The study has been successful in the synthesis of Al<sub>x</sub>Ga<sub>(1-y)</sub>As<sub>(1-y)</sub>Bi<sub>y</sub>. However further work remains in the optimization of the epitaxial growth. Optical quality remains limited by the increase in non-radiative recombination centres with alloying of Al. We attribute this increase in part due to the reduced bond stability between Al and Bi. It is however promising that the incorporation of Bi into the group V lattice site showed no sensitivity to the Al content. This reveals a non-trivial relationship between the Bi incorporation into Al containing III-V alloys.

NM-MoP-15 Epitaxial Growth of *a*-plane Mn<sub>3</sub>Sn on *c*-plane Al<sub>2</sub>O<sub>3</sub> using Molecular Beam Epitaxy, Sneha Upadhyay, T. Erickson, Ohio University; J. Moreno, Universidad Autonoma de Puebla, Mexico; H. Hall, Ohio

University; K. Sun, University of Michigan, Ann Arbor; G. Cocoletzi, Universidad Autonoma de Puebla, Instituto de fisica, Mexico; N. Takeuchi, Centro de Nanociencias y Nanotecnología, Universidad Nacional Autonoma de México; A. Smith, Ohio University

Noncollinear antiferromagnetic Weyl semimetal Mn<sub>3</sub>Sn has become fascinating in the current times because it is one of the rare antiferromagnets that exhibits large anomalous Hall and Nernst effects<sup>1</sup>. For future device applications, it is necessary to grow high-quality crystalline films, which has been particularly challenging to achieve. Higo et al. reported a large perpendicular switching in an Mn<sub>3</sub>Sn (0110) film grown on a MgO substrate with a W buffer layer by MBE<sup>2</sup>. Gao et al. reported the growth of Mn<sub>3</sub>Sn (0001) on Al<sub>2</sub>O<sub>3</sub>(0001) with a Pt buffer layer, while Mn<sub>3</sub>Sn (1120) was grown on R-plane  $Al_2O_3$  and MgO (110) substrates using PLD<sup>3</sup>. In this work, we grew Mn<sub>3</sub>Sn (1120) directly on Al<sub>2</sub>O<sub>3</sub> (0001) without a buffer layer in our molecular beam epitaxy chamber. Compared to our previous single-step deposition at high temperature, which resulted in a crystalline but rough and discontiguous film, here the growth was carried with a twostep deposition method at room temperature. This method results in a smooth, epitaxial Mn<sub>3</sub>Sn (1120) film having a thickness of ~220 nm. The growth is monitored in-situ using reflection high energy electron diffraction (RHEED) and measured ex-situ using X-ray diffraction, Rutherford backscattering, and cross-sectional STEM. We observe that the RHEED patterns are streaky, and the XRD shows a predominant single crystalline (1120) orientation. Additional results pertaining to the growth and structure, as well as empirical models, will be discussed.

#### Acknowledgment:

The authors acknowledge support from the U.S. Department of Energy, Office of Basic Energy Sciences, Division of Materials Sciences and Engineering under Award No. DE-FG02-06ER46317. The authors would like to thank Dr. Eric Stinaff and his students for back-coating the sapphire (0001) substrates.

<sup>1</sup> S. S. Zhang *et al.*, "Many-body resonance in a correlated topological Kagome Antiferromagnet," *Physical Review Letters* **125**, 046401 (2020).

<sup>2</sup> T. Higo *et al.*, "Perpendicular full switching of chiral antiferromagnetic order by current", *Nature***607**, 474 (2022).

<sup>3</sup> D. Gao *et al.*, "Epitaxial growth of high quality Mn<sub>3</sub>Sn thin films by pulsed laser deposition", *Applied Physics Letters* **121**, 242403 (2022).

NM-MoP-16 Surfactant Effect of Mn on AlN MBE Growth, Jesús Fernando Fabian Jocobi, R. Trejo Hernández, A. Martínez López, Nanoscience and Nanotechnology Program, Center for Research and Advanced Studies of the National Polytechnic Institute (CINVESTAV), Mexico; Y. Casallas Moreno, CONACYT-Interdisciplinary Professional Unit in Engineering and Advanced Technologies, National Polytechnic Institute, Mexico; I. Koudriavysev, Electrical Engineering Department, Solid State Electronic Section, Center for Research and Advanced Studies of the National Polytechnic Institute (CINVESTAV), Mexico; D. Olguin Melo, Center for Research and Advanced Studies of the National Polytechnic Institute Querétaro Unit, Mexico; S. Gallardo Hernández, M. López López, Physics Department, Center for Research and Advanced Studies of the National Polytechnic Institute (CINVESTAV), Mexico

Diluted magnetic III-N semiconductors (DMSs) have attracted significant attention due to their attractive applications for spintronic devices. The introduction of Mn atoms has been used to induce a ferromagnetic behavior in III-nitride materials [1], such as AIN. The presence of doping atoms on the surface during the MBE growth process can significantly influence the properties of the films [2]. In this study, we investigated the surfactant effects of Mn during the MBE growth of AIN.

The heterostructures were grown on Si (111) substrates employing a 200 nm-thick AlN buffer layer grown at 850 °C. After the buffer growth, clear streak (1X1) reflection high-energy electron diffraction (RHEED) patterns were observed (Fig. 1). Subsequently, three layers of AlN were grown with increasing doping levels of Mn (BEP<sub>Mn</sub>=1.9, 3.9 and 5X10<sup>-9</sup> Torr, respectively). A set of samples were prepared by varying the growth temperature from 790 to 830 °C.

During the growth of AlN:Mn layers at 790  $^{\circ}$ C, the streak (1X1) RHEED patterns were conserved, and the RMS surface roughness as evaluated by AFM was in the order of nanometers (Fig. 2). Employing secondary ion mass spectrometry (SIMS), we observed that the Mn concentration (Fig. 3), for the AlN layer grown at BEP<sub>Mn</sub>= 5X10<sup>-9</sup> Torr was in the order of 1X10<sup>19</sup> atoms/cm<sup>3</sup>. On the other hand, we observed a complete distinct behavior for the growth temperature of 830 °C. No significant Mn incorporation was observed by SIMS in the films, regardless of the Mn flux used. However, for this growth temperature, the appearance of a 3X RHEED reconstruction was observed in the AlN:Mn growth (Fig. 1). Furthermore, the surface of the AlN:Mn film showed a very flat morphology with a RMS roughness of 0.3 nm.

The absence of Mn incorporation in AlN layers at 830 °C, coupled with the observed 3X surface reconstruction and a very flat surface morphology, suggest a surfactant behavior of Mn in AlN grown at these conditions. These findings contribute to the fundamental understanding of surfactant effects in III-nitride growth on Si substrates and may have implications for the optimization of AlN-based optoelectronic devices.

[1] R. Frazier et al., "Indication of hysteresis in AlMnN," *Appl. Phys. Lett.*, vol. 83, no. 9, pp. 1758–1760, 2003, doi: 10.1063/1.1604465.

2] T. F. Kuech, "Surfactants in semiconductor epitaxy," *AIP Conf. Proc.*, vol. 916, pp. 288–306, 2007, doi: 10.1063/1.2751920.

NM-MoP-17 Growth and Scattering Mechanisms of Metamorphic In<sup>--0.81</sup>Ga0.19As Quantum Wells, Jason Dong, University of California at Santa Barbara; Y. Gul, University College London, UK; A. Engel, C. Dempsey, S. Chatterjee, University of California at Santa Barbara; M. Pepper, University College London, UK; C. Palmstrøm, University of California at Santa Barbara

 $\label{eq:link} In_xGa_{1-x}As/In_xAl_{1-x}As quantum wells with high In content have potential advantages over the GaAs/AlGaAs structures for spintronics and topological quantum computing applications. In comparison to GaAs/AlGaAs quantum wells, In_xGa_{1-x}As/In_xAl_{1-x}As quantum wells possess a lower electron effective mass, higher g-factor, and higher Rashba spin-orbit coupling. Due to a lack of latticed matched substrates, In_xGa_{1-x}As/In_xAl_{1-x}As quantum wells are grown on lattice mismatched substrates such as GaAs and InP with a metamorphic buffer layers. However, the growth of high mobility In_xGa_{1-x}As/In_xAl_{1-x}As quantum wells is hampered by enhanced interface roughness scattering from the metamorphic buffer layers and alloy scattering within the well [1].$ 

In this work, we report the growth of modulation doped In\_0.81Ga\_0.19AS/In\_0.81Al\_0.19AS quantum wells grown on semi-insulating InP (001) substrates. The quantum wells are characterized utilizing low temperature magnetotransport, which is performed using gated Hall bars and the van der Pauw geometry structures. Quantum wells with electron mobilities in excess of 380,000 cm<sup>2</sup>/Vs have been grown. The electron

mobility of the In<sub>0.81</sub>Ga<sub>0.19</sub>As quantum wells is comparable to that of the current state of the art In<sub>0.75</sub>Ga<sub>0.25</sub>As quantum wells. The role of growth parameters on electron mobility is discussed. The low temperature electron mobility and carrier density of the quantum wells is modeled to extract the dominant scattering mechanisms that limit the mobility. The influence of an InGaAs digital alloy on the electron mobility and alloy scattering of the quantum well is investigated.

[1] Chen, C. et al. Journal of Crystal Growth 425, 70-75 (2015).

#### NM-MoP-19 Light-enhanced Gating Effect at Conducting Interface of Laser MBE Grown EuO-KTO<sub>3</sub>, *Manish Dumen*, *S. Chakraverty*, Institute of Nano Science and Technology, India

Light illumination and electrostatic gating field are two widely used stimuli for controlling electronic processes in low-dimension systems. KTaO<sub>3</sub>(KTO)based conducting interfaces have gained tremendous interest because its spin-orbit coupling strength is one order of magnitude higher than STO, which makes it a promising candidate for spintronics and optoelectronic devices. In this talk, I will present the combined effect of light illumination and electrostatic gate on the conducting EuO-KTO interface. An unusual illumination enhanced gating effect is observed for this metallic system at room temperature. This enormous change in conductance is observed even with visible light of very low power intensity of 0.5 mW along with the back gate. This unusual effect offers a new perspective for tuning the photoelectrical properties at the oxide interfaces, which can be helpful for designing advanced photoelectric devices with high performance and multifunctionality

#### NM-MoP-20 4.3 µm InAs/AISb Quantum Cascade Detector Strain-Balanced to a GaSb Substrate, *Stefania Isceri*, *M. Giparakis*, *W. Schrenk*, *B. Schwarz*, *G. Strasser*, *A. Andrews*, Technische Universität Wien, Austria

Quantum cascade detectors (QCD) are high-speed, low-noise, photovoltaic detectors based on intersubband (ISB) transitions operating in the mid-infrared range at room temperature [1]. The active region of a QCD is composed of multiple periods of superlattice (SL) like structures. Each period includes an optical transition quantum well (QW) and an extraction cascade composed of thinner QWs. Previously, InAs/AISb on InAs substrates was used for QCDs operating at 2.7  $\mu$ m, because InAs offers a low effective electron mass of 0.023 m0, which increases the optical transition strength and improves responsivity [2]. In this study, we present the development of molecular beam epitaxy (MBE) techniques to produce high-quality InAs/AISb layers for a QCD detecting at 4.3  $\mu$ m on GaSb substrates. The advantages are that wavelengths longer than 1.7  $\mu$ m (0.74 eV band gap) are not absorbed by the substrate and it enables subsequent waveguides and light coupling.

Before the superlattice, we tuned the temperature and the Sb flux to remove the native oxide and grow a GaSb buffer layer, which improves the surface roughness, as observed by the root mean square (RMS) surface roughness of 0.27 nm measured with atomic force microscopy (AFM).

We then optimized the growth temperature for the InAs/AlSb heterostructures. Due to the As-for-Sb exchange, the strain-compensated InAs/AlSb SLs growth is challenging. The bond strength of As is stronger than for Sb and excess As on the surface during growth preferentially forms AlAs, instead of AlSb, leading to growth defects and lattice mismatch. We adjusted the As flux, shutter sequences, and "soak" times in order to have sharp interfaces, as determined by high-resolution x-ray diffraction (HR-XRD) and AFM. The devices are Te-doped, since Si and Sn are amphoteric in GaSb and AlSb. As the dopant source, we use the volatile compound GaTe instead of the element itself. The current active region design results in the InAs to AlSb thickness ratio of 2.4:1. This is not strain balanced. To overcome this problem, we include InSb interlayers for strain balancing.

The grown QCD with contact layers was processed into  $150 \times 150 \ \mu m$  mesas with the  $45^{\circ}$  wedge-facet substrate illuminated geometry and then optically characterized with a Fourier transform infrared (FTIR) spectrometer and a Globar source. The spectrum shows a strong intersubband absorption at the designed wavelength of 4.3  $\mu m$ . Device performance and comparisons will be presented.

- F. Giorgetta et al., IEEE Journal of Quantum Electronics 45(8), 1039 (2009)
- 7. M. Giparakis, et al., Appl. Phys. Lett. 120, 071104 (2022)

NM-MoP-21 Growth and Surface Investigation of Antiferromagnetic D0<sub>19</sub>-Mn<sub>3</sub>Ga Thin Films on GaN (0001), Ashok Shrestha, A. Abbas, D. Ingram, A. Smith, Ohio University

In recent years,  $Mn_3Ga$  has garnered significant attention due to its exotic physical properties and potential applications in spintronic devices [1,2]. One of the most intriguing, yet less explored, phases is the hexagonal antiferromagnetic phase of  $Mn_3Ga$  ( $D0_{19}$ ), which exhibits anomalous Hall effect and topological Hall effect in distinct temperature ranges [2]. In this presentation, we will delve into the growth and surface studies of a thin film of  $D0_{19}$ - $Mn_3Ga$  on a Ga polar- GaN (0001) substrate.

The experiments are carried out in an ultra-high vacuum chamber equipped with a molecular beam epitaxy (MBE) system and a room-temperature scanning tunneling microscope (STM). Initially, the GaN epilayer is deposited on a GaN (0001) substrate at 700 °C under gallium-rich conditions, followed by the growth of  $\mathsf{D0}_{19}\text{-}\mathsf{Mn}_3\mathsf{Ga}$  at 250  $^o\mathsf{C}$  under manganese-rich conditions. Reflection high-energy electron diffraction (RHEED) is used during growth to monitor the sample, and the in-plane lattice constant is evaluated. Both RHEED and STM confirm that the grown sample exhibits epitaxial growth. Furthermore, STM measurements show atomic resolution images with multiple flat terraces and steps. The ex-situ-X-ray diffraction (XRD) clearly shows the Mn<sub>3</sub>Ga 0002 peak, and the calculated *d*-spacing matched well with the step heights measured by STM. These measurements are consistent with the theoretically reported *c*-value of D019-Mn3Ga. The concentration of manganese and gallium in the sample is confirmed to be 3.2:1.0 by Rutherford backscattering (RBS). Various insitu and ex-situ measurements confirm the D019-Mn3Ga growth. Further work is planned to refine the sample stoichiometry and investigate the noncollinear antiferromagnetism.

This work is supported by the U.S. Department of Energy, Office of Basic Energy Sciences, Division of Materials Sciences and Engineering under Award No. DE-FG02-06ER46317.

#### References:

[1] L. Song, B. Ding, H. Li, S. Lv, Y. Yao, D. Zhao, and J. He, Appl. Phys. Lett. **119**, 152405 (2021).

[2] Z. H. Liu, Y. J. Zhang, G. D. Liu, B. Ding, E. K. Liu, H. Mehdi Jafri, Z. P. Hou,
W. H. Wang, X. Q. Ma, and G. H. Wu, Scientific Reports 7, 515 (2017).

#### NM-MoP-22 Guided Anisotropic Oxygen Transport in Vacancy Ordered Oxides, Jeffrey Dhas, Y. Du, Pacific Northwest National Laboratory

Understanding the transport processes of ions under external stimuli is critical as they govern the operation and failure mechanisms within energyconversion systems and microelectronic devices. The atomically precise fabrication of materials through methods such as molecular beam epitaxy or pulsed laser deposition enables the reliable study of novel functional states, which can be probed to characterize relevant fundamental processes at play. Using in situ transmission electron microscopy, we show that oxygen migration in vacancy ordered, semiconducting SrFeO<sub>2.5</sub> epitaxial thin films can be guided to proceed in two different types of diffusion pathways. Depending on the pathway which the oxygen ions undertake, different polymorphs of SrFeO<sub>2.75</sub> can be achieved, which give rise to different ground electronic properties before reaching a metallic, fully oxidized SrFeO3 phase. Utilization of oxygen tracer exchange experiments and timeof-flight secondary ion mass spectrometry helps probe the characteristics of oxygen ion transport in the system via determination of the oxygen depth profile. Additionally, ab initio calculations are implemented to reveal the diffusion steps and reaction intermediates. Ultimately, the underlying principles of controlling oxygen diffusion pathways and reaction intermediates which we demonstrate can be beneficial to advancing the design of structurally ordered oxides and novel devices for tailored applications.

#### NM-MoP-23 Impact of Unintentional Sb in the Tensile Electron Well of Type-II InAs/InAsSb Superlattices Grown on GaSb by Molecular Beam Epitaxy, *Marko Milosavljevic*, Arizona State University; *P. Webster*, Air Force Research Laboratory; *S. Johnson*, Arizona State University

High-performance materials that cover the mid-wave (3 to 5  $\mu$ m) and long-wave (8 to 14  $\mu$ m) infrared atmospheric transmission windows are essential for detection applications such as thermal sensing, gas detection, and infrared spectroscopy. Strain-balanced type-II InAs/InAsSb superlattices provide a high-quality material system with design flexibility in both the mid-wave and long-wave infrared regions that offer long lifetimes, robust absorption, and the ability to grow thick pseudomorphic layers on commercially available GaSb substrates. Despite many advantages,

InAs/InAsSb superlattice performance is hindered by the incorporation of unintentional Sb into the tensile InAs layer.

In this work, the impact of unintentional Sb in the tensile InAs electron well of type-II InAs/InAsSb superlattices is investigated. Several coherently strained mid and long wave superlattices are grown on (100) GaSb substrates by molecular beam epitaxy and examined using X-ray diffraction and temperature-dependent photoluminescence. The zero-order diffraction angle provides average strain and hence the average Sb mole fraction in a superlattice period. Analysis of higher order diffraction angles provides period thickness, which along with the individual layer growth times and the average strain, provides the tensile InAs and compressive InAsSb layer thicknesses. Analysis of the photoluminescence measurements provides the ground-state transition energy of the superlattice, which along with simulations of the ground state energies of the electrons and heavy-holes using a Kronig-Penney model, specify the distribution of Sb among the compressive hole well and the tensile electron well, which contains 1.8% (1.2%) unintentional Sb in the mid (long) wave superlattices.

A model of the Sb mole fraction profile in the compressive and tensile layers is developed and fit to the measured average Sb mole fractions of the compressive and tensile layers. The best-fit parameters provide the saturation and depletion rates of surface Sb and the Sb mole fraction. When the Sb shutter is opened, the compressive Sb mole fraction rapidly saturate at 41% in less than 1 s (1 monolayer); when the Sb shutter is closed, the tensile Sb mole fraction decays to a background of 0.6% in less than 3 s. Dilute amounts of Sb in the tensile electron well reduces the tensile strain, requiring a thicker well to achieve a strain balance. Analysis of the electron and heavy hole wavefunctions show that this increases the electron confinement, reducing the wavefunction overlap, and thus the optical absorption performance of the superlattice.

### NM-MoP-24 Local Droplet Etching and Filling Behavior of Nanoholes in In<sub>0.52</sub>AI<sub>0.48</sub>As Layers, *Dennis Deutsch*, *V. Zolatanosha*, *C. Buchholz*, *K. Jöns*, *D. Reuter*, Paderborn University, Germany

Semiconductor quantum dots fabricated via filling of local droplet etched nanoholes in Al<sub>x</sub>Ga<sub>1-x</sub>As with GaAs are excellent candidates for on-demand sources of entangled photon pairs due to their low exciton fine structure splitting. However, photon emission in the GaAs/Al<sub>x</sub>Ga<sub>1-x</sub>As system is limited to wavelengths below 800 nm and long-haul quantum communication via the global fiber network requires sources emitting photons in the optical Cband, i. e., ca. 1550 nm. One way to tackle this challenge, is to transfer the approach of local droplet etching and re-filling to the  $In_{0.53}Ga_{0.48}As/In_{0.52}Al_{0.48}As\-system$  lattice matched to InP. In this study we report on the influence of various growth parameters, as etching temperature, metal species and residual As pressure on the shape, areal density and size of the nanoholes, as these properties play an important role for the later quantum dot's emission characteristics. We present detailed statistical analysis of the nanohole morphology and densities that were obtained by analyzing measurements performed via atomic force microscopy and scanning electron microscopy. The areal density decreases strongly with increasing etching temperature (see Fig. 1) and the hole depth and diameter increase with increasing etching temperature. With increasing etching temperature, the nanoholes also become more and more elongated along the [011]-direction. Overgrowth of the nanoholes with Ino 52Alo 48As under As2-atmosphere preserves the holes (see Fig. 2) and we observed that a moderate overgrowth with 50 nm In<sub>0.52</sub>Al<sub>0.48</sub>As notably improved the number of symmetric nanoholes for samples etched at 410°C and 435°C. We found that filling the nanoholes with In0.53Ga0.48As is possible either under As<sub>2</sub>- or under As<sub>4</sub>-atmosphere but it works significantly better under As<sub>4</sub>-atmosphere. We also observed that the shape of the etched holes strongly depends on the metal species used for etching. Under the same etching conditions, the holes etched with pure Al tend to be significantly more elongated than those etched with In, as can be clearly seen in Fig. 3. Photoluminescence measurements on overgrown filled holes show that the emission wavelength shifts with the filling level of the nanoholes and QD emission in the optical C-band can be achieved when filling holes generated at etching temperatures above 435°C (see Fig. 4).

NM-MoP-25 Heteroepitaxial growth of (111)-oriented SrTiO<sub>3</sub> on ScAlN/GaN, E. Jin, A. Lang, B. Downey, V. Gokhale, Matthew Hardy, N. Nepal, S. Katzer, V. Wheeler, Naval Research Laboratory

Ultra-wide bandgap (UWBG) semiconductor materials have been highly studied in recent years, owing to their attractive materials properties for high power and RF electronics. In particular, ScAIN has been an UWBG material that not only possesses a large bandgap, but also exhibits very

high piezoelectric and spontaneous polarization coefficients, making it an appealing material for telecommunications and non-volatile memory applications. High quality epitaxial ScAIN films demonstrated by molecular beam epitaxy (MBE) have enabled high power density GaN field effect transistors utilizing ScAIN as a barrier layer. Heterogeneous integration of epitaxial oxides with ScAIN could realize novel hybrid electronics that can couple the added functionalities observed in oxides with this emergent semiconductor platform. For example, high-permittivity oxides such as SrTiO<sub>3</sub> (STO) could be used to greatly improve electric field management in RF high-electron-mobility transistors (HEMTs).

Integration of epitaxial STO with ScAIN comes with several challenges, including the lattice and crystal structure mismatch between a cubic and wurtzite material. In our previous work, we demonstrated that (111)-oriented STO films can be grown on AlGaN/GaN HEMT structures via a thin rutile TiO<sub>2</sub> buffer layer that mitigates the strain between the two different materials. We leverage that approach in this work to demonstrate the growth of STO on ScAIN/GaN HEMT structures via RF-plasma-assisted oxide MBE.

The preparation of the ScAlN surface prior to STO growth can also greatly impact both the crystal quality of the STO film and the channel electrical properties of the ScAlN/GAN heterostructure. To study the effects of surface pre-treatment prior to STO growth, we prepare the ScAlN surface with a series of different chemical cleans, including piranha acid, UV ozone and hydrofluoric acid, and a sulfuric-phosphoric acid mixture. We show that the a sulfuric-phosphoric solution results in the best combination of STO crystallinity (measured with x-ray diffraction) and ScAlN/GaN channel electrical properties (measured with Hall effect measurements). We also perform scanning transmission electron microscopy imaging to compare the impacts of the chemical cleans on the microstructure and find a significantly rougher oxide-nitride interface for the piranha-cleaned sample.

This work presents some of the growth and process optimization that is required to obtain high crystal quality epitaxial STO/ScAIN/GaN heterostructures, and can pave the way for subsequent perovskite oxide-UWBG semiconductor integration for the development of functional oxide-nitride electronics.

NM-MoP-26 Strain-Mediated Sn Incorporation and Segregation in Compositionally Graded Ge<sub>1:x</sub>Sn<sub>x</sub> Epilayers Grown by MBE at Different Temperature, Nirosh M Eldose, H. Stanchu, S. Das, S. Shetty, C. Li, Y. I Mazur, S. Yu, G. J. Salamo, University of Arkansas

Group IV alloys of Ge and Sn are extensively studied for various electronic and optoelectronic applications on a Si platform. Ge\_1-xSn\_x with  $\alpha\text{-Sn}$ concentrations as low as 6% [1] allows for a transition from an indirect bandgap to a direct optical. Higher Sn content makes possible mid and even long-range infrared optical emission and detection [2]. At the same time, due to the low solid solubility of Sn in Ge (~1%), as well as the large lattice mismatch of  $\alpha$ -Sn with Ge (~14%), the realization of high-quality Sn-rich Ge1-xSnx structures has proved challenging. In this study, we demonstrate enhanced Sn content using molecular beam epitaxy (MBE) growth of compositionally graded Ge1-xSnx on Ge (001). High-quality GeSn alloys with Sn composition reaching 6% at constant temperature. The maximal fraction of Sn was further increased to 9.0% when the growth temperature was continuously lowered while increasing the Sn flux. The analysis of surface droplets and SIMS (secondary ion mass spectrometry) profiles of elemental composition give evidence of Sn rejection during the growth, potentially associated with a critical energy of elastic strain. The intentional reduction of the coherent strain by decreasing the Sn flux near the sample surface has shown to trap a higher fraction of Sn in the Ge<sub>1-x</sub>Sn<sub>x</sub> layer and lower surface segregation. Supporting data (Fig.2) shows an approach for XRD spectra simulation was developed for strain and composition characterization.

 $\left[1\right]$  S. Wirths, D. Buca, S. Mantl, Prog. Cryst. Growth Charact. Mater. 2016, 62 (1), 1–39.

[2] J. Bass, H. Tran, W. Du, R. Soref, S.-Q. Yu, Opt. Exp. 2021, 29 (19),30844-30856.

NM-MoP-27 Growth and Characterization of GaAs (111) on 4H-SiC for Infrared Sensor, Subhashis Das, N. M Eldose, H. Stanchu, F. Maia de Oliveira, C. Li, M. Benamara, Y. I. Mazur, G. Salamo, University of Arkansas Epitaxial growth of III-V semiconductors on 4H-SiC would potentially allow the integration of optical sensors on SiC based power devices. We report on the growth of high-quality crystalline GaAs layer on the SiC hexagonal substrate by molecular beam epitaxy (MBE). For fabrication on SiC, a 5 nm AlAs nucleation layer was grown at 700 °C followed by a 60 nm GaAs layer buffer grown at 600 °C. We will discuss the surface morphology, structural quality, and the optical properties of the MBE grown samples. The  $\omega$ -2 $\theta$ scan result (fig.1. (a)) corroborates the crystalline growth of GaAs (111) on 4H-SiC. The structural quality is further illustrated by the cross-sectional TEM image in fig. 1(b). It consists of a high-quality GaAs layer and a highly defected interface region between GaAs and the 4H-SiC substrate. This defect region is attributed to the lattice and crystal structure mismatch between substrate and film. Fig. 1(c) shows the temperature dependent photoluminescence properties of the grown structure. Good free-exciton (FE) emission has been observed at room temperature (300 K) and lower temperature (77 K). Excitingly, the optical results were comparable with the same structure grown on a GaAs substrate. Overall, these observations exhibit potential to achieve an optical emitter for sensors integrated on SiC based power device platform.

#### NM-MoP-28 Growth and Conductivity Control of AlN by Plasma Assisted MBE, Neeraj Nepal, M. Hardy, B. Downey, A. Lang, D. Katzer, E. Jin, D. Storm, V. Gokhale, T. Growden, D. Meyer, V. Wheeler, U.S. Naval Research Laboratory

Aluminum nitride (AIN) is an ultra-wide direct bandgap semiconductor of interest due to its bandgap of ~6.2 eV, large critical electric field breakdown ( >15 MV/cm), high saturation velocity (~2x10<sup>7</sup> cm/s) and high thermal conductivity. Compared to GaN, it provides higher Baliga's figure-of-merit for power devices and higher Johnson's figure-or-merit for RF devices. Realizing the full potential of this material in electronic device applications requires the ability to tailor the electrical conductivity in active AIN layers through impurity dopings.

Due to AlN's large bandgap, impurity doping is challenging. To-date there are only a few reports on achieving impurity doping of AlN by molecular beam epitaxy (MBE) [1], ion implantation, [2] and metal organic chemical vapor deposition (MOCVD) [3]. Recently, MOCVD was used to grow metal semiconductor field effect transistor structure with n-type AlN channel [4]. Still, there is limited understanding of how to control and implement repeatable impurity doping in AlN-based devices.

In this paper, we report the plasma-assisted MBE growth of ~500 nm thick Si doped AIN films grown on AIN/sapphire templates using a metal modulated epitaxy (MME) approach Specifically, the parameters of growth temperature (760-1060°C), growth rate (3.7-11.1 nm/min), and Si flux (1E17-5E19 cm<sup>-3</sup>) were investigated and correlated with the resulting sheet resistance. All films were nucleated using an optimum in-situ cleaning Alabsorption and desorption technique monitoring the evolution of the growth surface with reflection high-energy electron diffraction. This was followed by a ~20 nm unintentionally doped AIN layer and ~500 nm Si doped AIN layers. Hall measurements show that sheet resistance increases with increasing growth rate, while a minimum resistance is attained at a mid-range thermocouple temperature of 860 °C (~688 °C real temperature). Additional results correlating XRD, AFM, and electrical measurements for the full parameter space will be discussed and related to potential defects limiting the conductivity in these films. Si-doping in AIN/sapphire templates will be compared with that on bulk substrates to determine the impact of threading dislocations on conductivity.

References:

- 8. H. Ahmad et al., J. Appl. Phys. 131, 175701 (2022) and references therein.
- 9. P. Bagheri et al., Appl. Phys. Lett. 122, 142108 (2023) and references therein.
- 10. Y. Taniyasu, Appl. Phys. Lett. 85, 4672 (2004).
- 11. M. Hiroki et al., IEEE Electron Dev. Lett. 43, 350 (2022).

NM-MoP-29 Molecular Beam Epitaxy Grown Group-IV Alloys for Infrared Photodetector and Quantum Transport Applications, *Tyler McCarthy*, Arizona State University; *R. Basnet*, University of Arkansas; *Z. Ju*, *X. Qi*, *A. McMinn*, Arizona State University; *J. Hu*, *S. Yu*, University of Arkansas; *Y. Zhang*, Arizona State University

Group-IV alloys are an emerging material system for potential applications in quantum transport and infrared photodetectors while remaining CMOS compatible. By utilization of strain, magnetic fields, and light illumination, the zero-gap, diamond-cubic phase of Sn,  $\alpha$ -Sn, is predicted to be a topological insulator, Dirac semimetal, or Weyl semimetal[1]. Focusing on the unexplored alloys with other Group-IV elements, Ge or Si, offers a novel tool to navigate the exciting boundaries of these topological phases. Additionally, SiGeSn is a model material system to demonstrate the momentum(k)-space charge separation (k-SCS) idea[2].Photodetectors with SiGeSn compositions near the indirect-to-direct bandgap transition have broad wavelength range of 2 to 22  $\mu$ m covering multiple IR spectrum bands.

Both Sn-rich and Ge-rich SiGeSn samples were grown at Arizona State University by molecular beam epitaxy in a VG-V80 chamber equipped with elemental effusion cells of In, Sb, Cd, Te, Sn and Ge, and a Si sublimation source. Complete sample details investigated using quantum PPMS for quantum and magneto transport measurements and RHEED, XRD, SEM, AFM, XPS, FTIR, and TEM methods for optical and structural characterization to be presented at the conference.

For thin film  $\alpha$ -Sn(Ge) samples, INSb substrates were chosen for lattice match conditions. The thermal oxide desorption was done under excess Sb flux at a pyrometer temperature of 480 C after which temperature was lowered to 390 C for Sb-rich InSb buffer growth. To separate from the conducting InSb substrate while maintaining lattice match conditions, a semi-insulating Cd-rich CdTe buffer was grown at 280 C. Samples were cooled overnight via contact with LN<sub>2</sub> shroud and thin films of  $\alpha$ -Sn and dilute Ge-containing SnGe alloys were grown. Due to heating by the thermal radiation from the Sn and Ge effusion cells during growth, there is a temperature creep on the sample surface. Therefore, to maintain the substrate at a temperature below the  $\alpha$ - to  $\beta$ -Sn phase transition, a short-pulse modulated technique, shutter cycles open for 2 seconds and shut for 10 seconds, was employed to grow the pure  $\alpha$ -Sn samples but not for the SnGe films.

Ge-rich SiGeSn alloys with thermalization barrier between  $0.4k_BT$  and  $3k_BT$  were grown on Ge and GeSn virtual substrates. Ge substrate surfaces were cleaned using HF and HCl solutions prior to UHV outgas at 550 °C, GeSn virtual substrates used HF and H<sub>2</sub>O<sub>2</sub>. A Ge buffer was grown at a substrate temperature of 500 °C before cooling down to 200 °C for SiGeSn growth. The Ge cell was held constant while Sn and Si fluxes were altered to obtain designed composition.

NM-MoP-30 Transport of Rare-Earth Nitrides Deposited via Molecular Beam Epitaxy, *Kevin Vallejo*, Z. Hua, Y. Zhang, K. Gofryk, B. May, Idaho National Laboratory

Rare-earth nitrides have a variety of attractive physical properties including magnetic, semiconducting, and superconducting behaviors. These heavy elements have high spin orbit coupling, and their compounds could enable potential spintronic devices. However, the physical properties of these materials is intrinsically linked to crystalline quality. Thus, a systematic investigation of these properties requires high quality samples with minimal defects and tunable dopant density. Molecular beam epitaxy is an ideal tool for such synthesis and this work explored the effects of temperature, metal flux, and nitrogen plasma power on the synthesis of cerium, neodymium, and samarium nitrides on several substrates (silicon, yttria-stabilized zirconia, and fused silica) and orientations. The team performed structural characterization of these materials using atomic force microscopy, x-ray diffraction, and transmission electron microscopy. The thermal and electrical transport characteristics were identified using non-destructive, laser-based metrology techniques and resistivity measurements as a function of temperature and magnetic field. These results serve as a platform for understanding the growth conditions of elements with complex oxidation states, low vapor pressures, and large atomic masses, paving the way for the high-quality synthesis of other lanthanoid and actinoid compounds.

NM-MoP-31 High Al-Content AlGaN Grown on TaC Virtual Substrates with Metallic Conductivity, *M. Brooks Tellekamp*, *D. Roberts*, National Renewable Energy Laboratory; *M. Miller*, Colorado School of Mines; *A. Rice*, National Renewable Energy Laboratory; *J. Hachtel*, Oak Ridge National Laboratory; *N. Haegel*, National Renewable Energy Laboratory

The lack of lattice matched substrates for AlGaN is the primary limitation to achieving high-performance power electronics, high-frequency electronics, and deep UV LEDs. This substrate limitation affects both material quality, through the formation of misfit-induced threading dislocations and strain-induced phase separation, and limitations to device geometry due to resistive or insulating electrical behavior. Dislocations and phase separation prevent AlGaN from reaching its full potential, and in the case of semiconducting substrates the primary loss mechanism in a vertically conductive device is resistive loss in the substrate itself. Thus, AlGaN alloys coulddrive disruptive technology iflong-standing substrate issues can be solved [1]. For Al<sub>x</sub>Ga<sub>1-x</sub>N there are competing effects of increasing alloy scattering, increased bandgap with increasing Al fraction, and decreasing dopant activation such that ideal compositions for power devices fall in the range 0.3 < x < 0.85 [2]. For these compositions pseudomorphic growth on GaN and AlN is very difficult or impossible.

Recently we have reported the design of virtual substrates for Al<sub>x</sub>Ga<sub>1-x</sub>Nepitaxy consisting of (111) TaC<sub>x</sub> grown on sapphire substrates via RF sputtering [3]. The crystallinity is subsequently improved by face-to-face annealing. These substrates offer several opportunities to improve power electronic devices through lattice and thermal conductivity matching, high electrical conductivity, high stability, and epitaxial liftoff.

In this talk we will discuss the growth of AlGaN on TaC templates by molecular beam epitaxy (MBE). Annealed TaC substrates show streaky-smooth reflection high-energy electron diffraction (RHEED) patterns and 6-fold rotational symmetry. The epilayers consist of Al<sub>x</sub>Ga<sub>1-x</sub>N in the range 0.7 < x < 1. Using RHEED, X-ray diffraction, atomic force microscopy, and scanning transmission electron microscopy (STEM) we investigate the impact of nucleating conditions on the structure of the film and interface. During metal-rich growth we observe incommensurate RHEED features associated with laterally contracted bilayers of metal which are not observed in nitrogen-rich growth. For Al<sub>0.7</sub>Ga<sub>0.3</sub>N we observe relaxed growth on TaC and strained growth on co-loaded AIN templates, and corresponding to this relaxed growth only the film on TaC exhibits a step-terrace structure in AFM observed as spiral hillocks. The impact of TaC defects on the AlGaN epilayer will be discussed, informed by aberration-corrected STEM.

[1] Kaplar et al., ECS J. Solid State Sci. Technol., 6 (2), p. Q3061, 2016.

[2] Coltrin et al., J. Appl. Phys., 121, p. 055706, 2017.

[3] Roberts et al., arXiv, 2208.11769, 2022.

NM-MoP-32 Grafted AlGaAs/GeSn p-i-n Heterojunction for GeSn MIR Electrically Pumped Laser Application, Yang Liu, J. Zhou, D. Vincent, J. Gong, S. Haessly, Y. Li, Q. Zhang, University of Wisconsin - Madison; S. Yu, University of Arkansas; Z. Ma, University of Wisconsin - Madison

In recent years, there has been significant progress in the development of germanium-tin (GeSn) lasers, which are promising candidates for applications in on-chip photonics, The recent advances in the growth of GeSn alloys have enabled the realization of high-performance GeSn lasers with improved efficiency, power output, and wavelength tunability. The electrically pumped GeSn laser diode is of much interest, as it presents the capacity of heterogeneous integration with the existing Si CMOS platform. However, the electrically pumped GeSn laser diode stops lasing at 90 K[1], due to increased free carrier absorption loss and competing non-radiative recombination at higher temperature. To get a higher operating temperature, introducing carrier confinement with heterostructures is desired[2]. However, the current epitaxy lattice-matched heterostructures, such as SiGeSn/GeSn and Ge/GeSn, shows insufficient electrical confinement to electrically driven GeSn laser at room temperature due to small band offset.

Here, we introduce a semiconductor grafting technology to form an AlGaAs/GeSn heterostructure to provide a viable approach to creating a larger band offset using the AlGaAs confinement layer[3],regardless of their respective lattice constant. In this grafting strategy, an ultrathin oxide (UO) layer is first deposited on the GeSn substrate, serving as a quantum tunneling layer and a double passivation layer. The formation of heterojunction is followed by transferring a single crystalline AlGaAs layer onto the passivated GeSn and finished by a thermal process to chemically bond them together. The introduction of the UO layer exhibits significantly suppressed interfacial density of states, which rivals the one obtained from *Monday Evening, September 18, 2023* 

epitaxy

The grafted AlGaAs/GeSn heterojunction confines the electrons in the active GeSn layer due to the 0.324eV band offset between AlGaAs and GeSn(Figure 1a). It shows the well-passivated interfaces reflected from the uniform diode ideality factor IF~1.5(Figure 3a) in all of the 341 devices, which are similar to the IF obtained from the MBE growth[4]. And I-V measurement also reveals the benefits from a larger band offset with an On/Off ratio of around 4 orders(Figure 3a). Most of the capacity-voltage sweeping measurements are consistent when the frequency changes from 10kHz to 200kHz(Figure 3b). The formation of the high-quality AlGaAs/GeSn diode indicates the feasibility of semiconductor grafting. The preliminary diode performance has also manifested a great potential for room-temperature electrically pumped GeSn laser by employing AlGaAs/GeSn heterojunction with better electrical confinement.

NM-MoP-34 Molecular Beam Epitaxy of Kagome-Structured Antiferromagnetic FeSn Grown on LaAlO<sub>3</sub> (111), *Tyler Erickson*, S. Upadhyay, H. Hall, D. Ingram, S. Kaya, A. Smith, Ohio University

Iron and tin can be alloyed to form different structures of alternating stackings of Kagome Fe<sub>3</sub>Sn and honeycomb Sn<sub>2</sub> (stanene) layers. This alternating sequence results in either Fe<sub>3</sub>Sn<sub>2</sub> or FeSn depending on whether there are Fe<sub>3</sub>Sn bilayers or Fe<sub>3</sub>Sn monolayers separating the stanene layers [1,2]. Fe<sub>3</sub>Sn<sub>2</sub> and FeSn provide interesting avenues for spintronics with flat bands arising from geometrical frustration leading to novel topological phases [3]. Fe<sub>3</sub>Sn<sub>2</sub> and FeSn have both been grown using molecular beam epitaxy on various substrates. [2, 4, 5] In this study, we grow FeSn by MBE following the method described by Hong et al. [2] Namely, we grew our FeSn on LaAlO<sub>3</sub> substrates at a temperature of 500 °C. The choice of using LaAlO<sub>3</sub> is based on the relatively good lattice match with difference of only 1%. Four samples have been grown with Fe:Sn flux ratios of 0.8:1, 1:1, 1.2:1, and 1.5:1. We compare the results of the 4 samples by means of RHEED, XRD, RBS, and AFM. In all cases, smooth streaky RHEED patterns are observed, and from the streak spacing we calculate the in-plane lattice constants which are then complemented by the lattice constants calculated from the XRD spectra. For the case of the 1:1 flux ratio, using RHEED we find an a = 5.290 Å as compared to the expected a\_for FeSn = 5.297 Å [2] and using XRD we find c = 4.56 Å as compared to the expected c for FeSn = 4.481 Å [2]. In this presentation, we will discuss the lattice parameters as functions of the incident flux ratios as well as the phases and phase purity of the resultant samples. We will also present results for the surface smoothness as a function of flux ratios as measured by the AFM images, and we will also address the resultant film stoichiometry as a function of incident flux ratios.

This work is supported by the U.S. Department of Energy, Office of Basic Energy Sciences, Division of Materials Sciences and Engineering under Award No. DE-FG02-06ER46317.

#### References:

27

[1] L A Fenner et. al., J. Phys. Condens. Matter 21 452202 (2009).

[2] Deshun Hong et. al., AIP Advances 10, 105017 (2020).

[3] Yaofeng Xie et. al. Communications Physics 4:240 (2021).

[4] Shuyu Cheng et. al., APL Mater. 10, 061112 (2022).

[5] Igor Lyalin et. al., Nano Lett. 21, 16, 6975–6982 (2021).

NM-MoP-35 Tuning Interface Sharpness and Superconductivity at Oxide Heterostructures, Y. Eren Suyolcu, Max Planck Institute for Solid State Research, Germany; G. Kim, Max Planck Institute for Solid State Research; Y. Wu, G. Logvenov, P. van Aken, Max Planck Institute for Solid State Research, Germany

The structural adaptability of transition-metal oxides allows for designing different heterostructures emerging unique physical properties at interfaces<sup>1</sup>. High-temperature interface superconductivity takes place at the interface between overdoped (metallic) and undoped (insulating) La<sub>2</sub>CuO<sub>4</sub> layers grown by oxide molecular beam epitaxy (MBE)<sup>2</sup>. In addition to homo-epitaxial systems<sup>3,4</sup>, multilayers of La<sub>2</sub>CuO<sub>4</sub> combined with La<sub>2-x</sub>Sr<sub>x</sub>NiO<sub>4</sub><sup>5</sup>, LaNiO<sub>3</sub><sup>6</sup>, and LaSrMnO<sub>3</sub><sup>7</sup> layers revealed the impact of the interface sharpness on the occurrence of superconducting<sup>5</sup>, thermoelectric<sup>6</sup>, and magnetic<sup>7</sup> properties, respectively. In this work, we designed new *cuprate-manganite* interfaces using oxide MBE<sup>7,8</sup> and focused on the interface sharpness and superconducting properties compared to *cuprate-cuprate* interfaces.

We probed the interfaces using scanning transmission electron microscopy (STEM) techniques, including high-angle annular dark-field (HAADF),

annular bright-field (ABF) imaging, and electron energy-loss spectroscopy (EELS).

Our findings demonstrate that hetero-epitaxial contacts with manganite layers can realize sharper Sr-doped La<sub>2</sub>CuO<sub>4</sub> interfaces. The dopant distribution in La<sub>2</sub>CuO<sub>4</sub> is affected by the elemental intermixing at the first atomic monolayer of the interfacial LaMnO<sub>3</sub> contact, and different superconducting behavior (*e.g.*, interface vs filamentary) can be customized with the interfacial design<sup>8</sup>. With such a design, we create interface superconductivity confined down to one monolayer thickness but with a cost of filamentary behavior due to local intermixing. We also demonstrate that structurally sharp interfaces can be chemically rough, and the chemical intermixing dominates the physical properties.<sup>8</sup>

#### References:

- 12. Y. E. Suyolcu et al., J. Supercond. Nov. Magn. **33**, 107–120 (2020).
- 13. A. Gozar et al., Nature 455, 782–785 (2008).
- 14. Y. E. Suyolcu et al., Adv. Mater. Interfaces 4, 1700737 (2017).
- 15. F. Baiutti et al., Nanoscale **10**, 8712–8720 (2018).
- 16. P. Kaya et al., ACS Appl. Mater. Interfaces 10, 22786–22792 (2018).
- 17. Y.-M. Wu, Y. E. Suyolcu et al., ACS Nano 15, 16228–16235 (2021).
- 18. G. Kim et al., Phys. Rev. Mater. 3, 084420 (2019).
- 19. Y. E. Suyolcu et al., submitted, 2023.

# NM-MoP-37 Molecular Beam Epitaxial Growth of GalnAs, GaNAs and GalnNAs Nanowires over 2-inch Si(111) Substrate Showing Emission at Near Infrared Regime, *Keisuke Minehisa*, H. Hashimoto, K. Nakama, F. Ishikawa, Hokkaido University, Japan

Semiconductor nanowires are the materials with one-dimensional structures and are expected to be applied to next-generation optical and electronic devices. Besides, III-V compound semiconductor GaAs has high electron mobility and photoelectric conversion efficiency, and has been used for lasers, solar cells, and transistors. Monolithic structures of GaAs nanowires grown heteroepitaxially on Si substrates are thus promising for future device applications. Among them, dilute nitride GaNAs or GaInNAs are materials of interest since the introduction of few % of N into host matrix Ga(In)As provides efficient tunability of band gap and lattice constant, working at the near infrared wavelengths of solar spectrum. In this study, we report the molecular beam epitaxial growth and the characteristics of GaAs-related GaNAs and GaInNAs core-multishell nanowires on 2-inch Si(111) substrates.

We fabricated GaAs-related core-multishell nanowires samples having optically active GaInNAs, GaNAs, or GaInNAs shells, respectively, on 2-inch n-type Si(111) substrates by constituent Ga-induced vapor liquid solid growth using a plasma-assisted molecular beam epitaxy. We prepared several samples with different shell layers. GaInAs shell contains 20% In. GaNAs shell have its nitrogen concentration 1%. The concentration of In and nitrogen was 20% and 1%, respectively, for GaInNAs. After the nanowire growth, the substrate wafer was observed to be black, resulting from efficient light absorption. GaInAs, GaNAs, and GaInNAs nanowires showed PL peak at 1000, 1050, and 1100 nm, respectively at room temperature. The intensity of the GaInNAs was comparable with GaInAs and the peak width was smaller than that of GaInAs, considered to be induced by the mediation of strain deformation by the introduction of nitrogen. The results is promising for the realization of high quality GaInNAs material operating at near infrared regime.

#### NM-MoP-38 Tunable Superconductivity in Hybrid Interface FeTe<sub>1-</sub> **\*Sex/Bi**<sub>2</sub>Te<sub>3</sub> Grown by Molecular Beam Epitaxy, *An-Hsi* (*Jane*) *Chen*, Oak Ridge National Laboratory, USA; *Q. Lu, R. Moore, M. Brahlek*, Oak Ridge National Laboratory

Hybrid interfaces of topological insulators and s-wave superconductors are great candidates for realizing Majorana bound states which have been projected to have paradigm-changing possibilities in quantum computing. The epitaxial FeTe<sub>1\*</sub>Se<sub>x</sub>/Bi<sub>2</sub>Te<sub>3</sub> platform possess the necessary parameters for topological states, high transition temperatures, and a high level of tunability available through doping and interfacial engineering. Recently, monolayer of superconducting FeTe<sub>1\*</sub>Se<sub>x</sub> (x=0.25) grown on the Bi<sub>2</sub>Te<sub>3</sub> was reported to exhibit emergent topological interfacial Dirac states at the Fermi energy. Pushing to lower Se levels reduces disorder which is critical for interrogating Majorana bound states, yet pure FeTe is not superconducting. Here we systematically interrogate how modifications to the molecular beam epitaxy growth of Bi<sub>2</sub>Te<sub>3</sub> and the FeTe<sub>1\*</sub>Se<sub>x</sub> can enable tailoring both superconductivity and topological properties at low Se

doping levels. Low temperature transport measurement, angle resolved photoemission spectroscopy and X-ray diffraction are combined to unravel the roles of band structure, crystallinity, and superconductivity which can be tailored as a function of growth conditions. This study will reveal the complex relation of strain and charge at FeTe<sub>1-x</sub>Se<sub>x</sub>/Bi<sub>2</sub>Te<sub>3</sub> interface which will hopefully create a robust platform for Majorana bounds states and advancing quantum devices.

This material was based on work supported by the U.S. DOE, Office of Science, Basic Energy Sciences, Materials Science and Engineering Division, and U.S. Department of Energy, Office of Science, National Quantum Information Sciences Research Centers, Quantum Science Center.

NM-MoP-39 Van Der Waals Epitaxy of 2D Ferromagnetic Fe<sub>5-x</sub>GeTe<sub>2</sub> Films with Curie Temperature Above Room Temperature on Graphene, Joao Marcelo J. Lopes, H. Lv, A. Kassa, A. da Silva, J. Herfort, M. Hanke, A. Trampert, R. Engel-Herbert, M. Ramsteiner, Paul-Drude-Institut für Festkörperelektronik, Leibniz-Institut im Forschungsverbund Berlin, Germany

Van der Waals (vdW) heterostructures combining layered ferromagnets and other two-dimensional (2D) crystals such as graphene and transition metal dichalcogenides are promising building blocks for the realization of ultracompact devices with integrated magnetic, electronic and optical functionalities. Their implementation in various technologies depends strongly on the development of a bottom-up, scalable synthesis approach allowing to realize highly uniform heterostructures with well-defined interfaces between different 2D layered materials. It also requires that each material component of the heterostructure remains functional, which ideally includes ferromagnetic order above room temperature for 2D ferromagnets. In this contribution, we will present our recent results on van der Waals (vdW) epitaxy of the 2D itinerant ferromagnetic metal Fe5-xGeTe2 (FGT,  $x \sim 0$ ) on single crystalline epitaxial graphene using molecular beam epitaxy. For the growth of FGT films (with thickness ranging from 10 to 15 nm), elemental Fe, Ge, and Te were co-supplied from conventional effusion cells, and a growth temperature of 300 °C was employed. As a substrate, epitaxial graphene on 4H-SiC(0001), synthesized via SiC surface was employed. Morphological graphitization, and structural characterization using methods such as atomic force microscopy, synchrotron-based grazing incidence X-ray diffraction, and scanning transmission electron microscopy (STEM) revealed that epitaxial FGT films exhibiting very good surface morphology, high crystalline quality, and a sharp interface to graphene could be realized. Interestingly, stacking faults related to the presence of single FGT layers with thicknesses exceeding those expected for the Fe<sub>5</sub>GeTe<sub>2</sub> phase could be identified by STEM. We expect these to be novel FGT metastable phases with Fe composition higher than 5 and potentially enhanced magnetic properties. Temperaturedependent magneto-transport measurements and superconducting quantum interference device (SQUID) magnetometry were employed to assess the magnetic properties of the samples. Ferromagnetic order with a predominant out-of-plane magnetization was shown to persist above 350 K. Furthermore, magneto-transport also revealed that the epitaxial graphene continues to exhibit a high electronic quality. These results represent an important advance beyond non-scalable flake exfoliation and stacking methods, thus marking a crucial step toward the implementation of ferromagnetic 2D materials in practical applications.

NM-MoP-40 Molecular Beam Epitaxy of MnBi<sub>2</sub>Te<sub>4</sub> and Bi<sub>2</sub>Te<sub>3</sub>/ MnBi<sub>2</sub>Te<sub>4</sub> Heterostructures, *Hyunsue Kim*, University of Texas at Austin; *M. Liu*, Harvard University; *L. Frammolino*, *Y. Li*, *F. Zhang*, University of Texas at Austin; *W. Lee*, University of Chicago; *X. Li*, *A. MacDonald*, *C. Shih*, University of Texas at Austin

Intrinsic Magnetic Topological Insulator (MTI) has been widely recognized as an excellent platform to study topological surface state critical for understanding exotic quantum phenomena, including the Quantum Anomalous Hall effect and Axion insulator states. Using molecular beam epitaxy (MBE), we gain control of high-quality MnBi2Te4 thin films on Si (111) and epitaxial graphene substrates, and Bi2Te3/ MnBi2Te4 heterostructure. By combining several *in-situ* characterization techniques, we obtain critical insights toward atomical control of MBE growth of MnBi2Te4 and Bi2Te3/ MnBi2Te4 heterostructures. In specific, we extract the free energy landscape for the epitaxial relationship as a function of the in-plane angular distribution. Furthermore, with the optimized layer-bylayer growth, we map out the chemical potential and Dirac point of the thin film grown. Lastly, we observe Mn out-diffusion behavior across the interface on Bi2Te3/ MnBi2Te4 heterostructure with an abrupt Bi2Te3/ MnBi2Te4 heterostructure with an abrupt Bi2Te3/

diffusion behavior across the interface. These scientific insights secure the foundation for understanding growth dynamics and pave the way for the future applications of MBE for magnetic topological insulators and their heterostructure for emerging topological quantum materials.

NM-MoP-41 Effect of Spin-Orbit Field on the Magnetization Reversal in a Crystalline (Ga,Mn)(As,P) Ferromagnetic Layer, Seongjin Park, K. Lee, S. Lee, Korea University, Republic of Korea; X. Liu, Unversity of Notre Dame; M. Dobrowolska, J. Furdyna, University of Notre Dame

Effect of current induced spin-orbit field (SOF) on the magnetization reversal have investigated in a crystalline (Ga,Mn)(As,P) ferromagnetic layer with perpendicular anisotropy. To study the dependence of SOFs on current direction, two types of Hall devices along the <110> and the <100> crystallographic directions, in which the Rashba-type and the Dresselhaustype SOFs are collinear and orthogonal to each other have been fabricated. The current scan experiments clearly show magnetization switching in all devices regardless of current direction, which varies in 4 different crystal directions of the film. However, magnetization switching chirality in current scan hysteresis depends on the crystal direction of current flow. The effect of SOF was further studied external field scan experiments, in which Hall resistance hysteresis shows clear difference between current polarity (i.e., positive and negative currents) with increasing magnitude current. The observed SOT switching chirality in current scan hysteresis and the current polarity dependent shift of in field scan hysteresis are consistently explained with the Rashba-type and the Dresselhaus-type spin-orbit fields induced by tensile strain in the (Ga,Mn)(As,P) film. Furthermore, the differences of magnetization switching field between opposite current polarities show clear dependence on the direction of Hall devices (i.e., <110> and <100>). We have systematically measured crystalline dependences of magnetization switching process by varying magnitude of current and external field strength. From the magnitudes of hysteresis shifts between two opposite current polarities measured for the <110> and <100> Hall devices, we are able to quantify magnitudes of the Rashba-type and the Dresselhaus-type spin-orbit fields.

#### NM-MoP-42 Unraveling the Role of Dopant Clustering in Magnetic Impurity Doped Monolayers of Transition Metal Dichalcogenides, *Rehan Younas, G. Zhou, C. Hinkle,* University of Notre Dame

Efforts to achieve above room temperature ferromagnetism in monolayers of transition metal dichalcogenides (TMDs) through substitutional doping with magnetic impurities are actively being pursued for energy-efficient logic and memory devices. However, the current limitations stem from phase separation and multi-layered growth at heavy doping levels, restricting the doping in monolayers to levels well below the threshold established by density functional theory (DFT) for above room temperature Curie temperature. On the other hand, room temperature magnetism has been frequently observed at significantly lower doping levels (0.1-1%), but this magnetism arises from a combination of substitutional dopants, point defects, contaminants, interstitials, or edge states. As a result, the origin of purely substitutional doping-induced ferromagnetism remains a subject of debate.

Toward this end, this study employs molecular beam epitaxy (MBE) to achieve up to 30% substitutional doping of vanadium (V) and iron (Fe) in a monolayer of tungsten diselenide, surpassing the doping requirements (>15%) indicated by DFT for room temperature ferromagnetism. Magnetometry measurements, however, reveal the absence of ferromagnetism down to a temperature of 4 K in these phase-pure films, with only the phase-separated films exhibiting any room temperature ferromagnetic behavior at Fe doping levels exceeding 30%. Structural characterization utilizing plan-view transmission electron microscopy reveals significant dopant clustering, even at modest doping levels (~5%), which serves as the primary factor responsible for the absence of ferromagnetism in phase-pure films. Remarkably, these observations align with DFT calculations, which predict a low formation energy for dopant clustering, leading to a weakened exchange interaction that subsequently suppresses ferromagnetism. The insights gained from this exploratory study offer a promising pathway to attain high doping densities in monolayer TMDs while emphasizing the influence of dopant clustering on the magnetic properties of the films.

#### NM-MoP-43 Atomic Layer Molecular Beam Epitaxy Growth of Kagome Ferrimagnet RMn<sub>6</sub>Sn<sub>6</sub> (R = Rare Earth) Thin Films, *Shuyu Cheng*, *W. Zhou*, *R. Kawakami*, Ohio State University

Materials with quasi-2D Kagome layers are an ideal platform for studying physics at the junction of non-trivial band topology and magnetism. In recent years, Kagome-structured ternary compounds  $RMn_6Sn_6$  (R = rare

earth) have drawn much attention due to their highly tunable physical properties. With different rare earth elements R, the magnetic anisotropy of RMn<sub>6</sub>Sn<sub>6</sub> varies from within the Kagome plane (e.g. Gd) to perpendicular direction (e.g. Tb) [1, 2]. Especially for TbMn<sub>6</sub>Sn<sub>6</sub>, a large anomalous Hall conductance arises from gapped Dirac cones that are close to the Fermi level [1]. In this work, we synthesized (0001)-oriented thin films of ErMn<sub>6</sub>Sn<sub>6</sub> and TbMn<sub>6</sub>Sn<sub>6</sub> using atomic layer molecular beam epitaxy (AL-MBE). The structure of the sample was characterized by RHEED, AFM, and XRD. The magnetic properties were measured with SQUID, and the transport properties were measured with PPMS. We show that ErMn<sub>6</sub>Sn<sub>6</sub> thin films exhibit easy-plane anisotropy up to room temperature, while TbMn<sub>6</sub>Sn<sub>6</sub> exhibits uniaxial anisotropy at low temperatures. In general, the AL-MBE growth recipe can be applied to other materials in the RMn<sub>6</sub>Sn<sub>6</sub> family. This work establishes RMn<sub>6</sub>Sn<sub>6</sub> thin films as a highly tunable system for fundamental research and potential applications in the future.

#### References

[1]. Yin, et al. "Quantum-limit Chern topological magnetism in  $TbMn_6Sn_6$ ." Nature 583.7817 (2020): 533-536.

[2]. Ma et al. "Rare Earth Engineering in  $RMn_6Sn_6$  (R= Gd- Tm, Lu) Topological Kagome Magnets." Physical review letters 126.24 (2021): 246602.

#### NM-MoP-44 Investigating Phase Transformations and Stability of Pt-Te Van Der Waals Materials Through Pt Vapor Exposure and Post-Growth Annealing, *Kinga Lasek*, Purdue University; University of South Florida

In this research, we investigate the growth and transformation of ultrathin Pt-telluride van der Waals (vdW) compounds by vacuum annealing and Pt-vapor exposure. We find that molecular beam epitaxy readily grown PtTe2 thin films can be converted into Pt3Te4- and furthermore Pt<sub>2</sub>Te<sub>2</sub>-bilayers through vacuum-induced Te-loss. Using scanning tunneling microscopy, x-ray, and angle resolved photoemission spectroscopy, we find that Pt<sub>3</sub>Te<sub>4</sub> remains thermally stable up to 350C while achieving Pt<sub>2</sub>Te<sub>2</sub> requires a higher annealing temperature of 400C. Interestingly, bilayer Pt<sub>2</sub>Te<sub>2</sub> can be re-tellurized by exposure to Te-vapor. This causes the topmost Pt<sub>2</sub>Te<sub>3</sub>.

Additionally, we introduce a novel method to transform monolayer  $PtTe_2$  into  $Pt_2Te_2$ , using vapor-deposited Pt atoms. This innovative process allows for well-defined metal-semiconductor junctions by nucleating the  $Pt_2Te_2$  phase within  $PtTe_2$ . These compositional phase transformations hold significant potential for efficient in-plane metal contacts, particularly in materials with substantial spin-orbit coupling like  $PtTe_2$ . The comprehensive understanding of these processes enables the controlled synthesis of all known Pt-telluride vdW compounds in their ultrathin form by precisely managing Te removal or Pt addition.

Furthermore, we investigate the chemical stability of these materials through exposure to oxygen and air. Remarkably, even after extended air exposure, only the surface Te layer is modified by oxygen chemical bonds, leading to a 3-eV shift to the higher binding energy of the Te-3d core levels. However, these oxygen species can be effectively removed through vacuum annealing at 280 C, restoring the pristine state of Pt-telluride samples. This demonstrates the excellent air stability of these materials.

#### NM-MoP-45 Layer-Dependent Optical Properties of MBE-Grown ZrTe<sub>2</sub> Determined by in-Situ Spectroscopic Ellipsometry, E. Houser, Frank Peiris, Kenyon College; A. Richardella, M. Stanley, N. Samarth, Pennsylvania State University

Two-dimensional transition metal dichalcogenides (TMDCs) are an interesting platform to interrogate fundamental physics questions as well as to advance the development of optoelectronic technologies. Both these endeavors are heavily dependent on having high-quality TMDCs, realized only after employing extensive growth optimization procedures. In this work, we investigated the growth and the optical properties of ZrTe<sub>2</sub>, a candidate topological Dirac semimetal. During the growth of 12 unit cells (u.c.) ofZrTe<sub>2</sub> on a sapphire substrate, we obtained in-situ spectroscopic ellipsometry after the deposition of each u.c. After the deposition of the ZrTe<sub>2</sub>, we deposited a Te capping layer to protect the TMDC film. X-ray reflectivity measurements performed subsequently indicated that the total thickness of ZrTe<sub>2</sub> and the thickness of Te to be 5.95 nm and 19 nm, respectively.

A standard inversion technique was used to model the ellipsometry spectra by specifying a three layer model (i.e., sapphire substrate,  $ZrTe_2$  layer and the Te capping layer) to fit the final ellipsometry spectra. The thicknesses obtained from X-ray reflectivity allowed us to obtain the precise dielectric function of the final  $ZrTe_2$  layer (i.e., 12 u.c.). Subsequently, we fit the

remaining ellipsometry spectra obtained for 11 u.c. through 1 u.c.  $ZrTe_2$  and obtain their dielectric functions. Clearly, the dielectric functions show a noticeable change with the thickness of the  $ZrTe_2$  layers, where the absorption depicted by the imaginary part of the dielectric function increases with the thickness of  $ZrTe_2$ . Additionally, the layer-dependent dielectric functions were analyzed by incorporating a Drude oscillator to account for the free electrons and two Kramers-Kronig-consistent oscillators to represent the band-to-band transitions. Interestingly, we find that the Drude contribution reduces as the thickness of  $ZrTe_2$  gets smaller, suggesting that its metallic character diminishes as the thickness reduces. Further analysis of the optical conductivity verifies this observation.

The work at Kenyon is funded by DMR-2004812 and the work at The Pennsylvania State University Two-Dimensional Crystal Consortium – Materials Innovation Platform (2DCC-MIP) is supported by NSF cooperative agreement DMR-1539916 and DMR-2039351.

NM-MoP-46 Bi Heteroantisites at Ga(As,Bi)/(Al,Ga)As Interface: Role of the Surface Reconstruction?, *Esperanza Luna*, *A. da Silva*, *K. Biermann*, Paul-Drude-Institut für Festkörperelektronik, Leibniz-Institut im Forschungsverbund Berlin e.V., Germany; *J. Puustinen*, *J. Hilska*, *M. Guina*, Optoelectronics Research Centre, Tampere University, Finland; *P. Laukkanen*, *M. Punkkinen*, University of Turku, Finland; *A. Trampert*, Paul-Drude-Institut für Festkörperelektronik, Leibniz-Institut im Forschungsverbund Berlin e.V., Germany

Innovative growth strategies, including the use of (Al,Ga)As barriers, have been proposed to improve the performance of optoelectronic devices based on Ga(As,Bi) quantum wells (QWs). It is argued that the presence of Al might suppress the well-known Bi surface segregation but the exact role of Al is unclear, as well as its impact on the Ga(As,Bi)/(Al,Ga)As interface properties.

We investigate the interfaces of GaAs<sub>0.96</sub>Bi<sub>0.04</sub>/Al<sub>0.15</sub>Ga<sub>0.85</sub>As QW structures using a combination of (scanning) transmission electron microscopy (S)TEM techniques. The samples were grown by solid source MBE on GaAs(001). The Ga(As,Bi) QWs, with nominal thickness of 7 nm, were grown at 370 °C, while the substrate temperature T<sub>s</sub> was raised to 580 °C for the barriers growth. There were growth interruptions (GI) before and after the QW to adjust T<sub>s</sub> and the V/III ratio. In addition to As-flux during the GI, Bi-flux was supplied just before the QW growth at the Ga(As,Bi)-on-(Al,Ga)As interface.

Our TEM investigations reveal that the layers grow pseudomorphically on the GaAs substrate. Whereas high-angle annular dark-field (HAADF) micrographs with Z-contrast show the expected sequence of layers with their expected thickness and compositions, diffraction-based chemicallysensitive g002 dark-field TEM images reveal the striking presence of "dark lines" at both Ga(As,Bi)-on-(Al,Ga)As and (Al,Ga)As-on-Ga(As,Bi) interfaces, precisely at the GI positions, delimiting the interfaces. The line at the Ga(As,Bi)-on-(Al,Ga)As interface is ~2 nm thick and remarkably welldefined. Formation of quaternary (Al,Ga)(As,Bi) at the interface may cause the features, but theoretical estimations of the  $g_{002}$  diffracted intensity  $I_{002}$ for (Al,Ga)(As,Bi) result in a much brighter contrast than observed experimentally. In the calculation Bi and Al are incorporated substitutionally at V- and III-element positions, respectively. Interestingly, Bi incorporation at III-element position, i.e., the presence of Bi antisites,  $Bi_{Ga}$ , has a remarkable impact decreasing  $I_{002}$  and 1%  $Bi_{Ga}$  would explain the observed contrast, EDX and HAADF-STEM reveal Ga depletion and Bi accumulation at the Ga(As,Bi)-on-(Al,Ga)As interface, consistent with the presence of BiGa at this location. Furthermore, CuPt<sub>B</sub> atomic ordering is detected at the 7-nm thick Ga(As.Bi) QW but not at the GI positions before and after the QW. suggesting QW growth on (2x1) reconstruction. With support of densityfunctional-theory calculations, we discuss the role of the surface reconstruction and/or the impact of Al on  ${\rm Bi}_{{\rm Ga}}$  formation, a largely anticipated defect in Ga(As,Bi) yet challenging to detect.

NM-MoP-47 Substrate Preparation Methods for the MBE Growth of Van Der Waals Materials, *Ryan Trice*, *M. Yu*, *A. Richardella*, *M. Hilse*, *S. Law*, Penn State University

The growth of van der Waals thin films by MBE has exploded in recent years, including Bi<sub>2</sub>Se<sub>3</sub>, a popular prototypical 3D topological insulator. Despite the interest in these materials, the growth of high-quality Bi<sub>2</sub>Se<sub>3</sub> films by MBE with low carrier density and high mobility remains *Monday Evening, September 18, 2023* 

challenging, in part due to a lack of understanding of the influence of the substrate. In this study, we investigate how the preparation of c-plane sapphire substrates influences film quality. Sapphire was chosen as the substrate of investigation due to its widespread use in van der Waals epitaxy. Although  $Bi_2Se_3$  was used as the material of interest, these results are likely applicable to growth of any van der Waals material on c-plane sapphire.

The Bi<sub>2</sub>Se<sub>3</sub> thin films were grown using MBE in a DCA Instruments R450 reactor. Bismuthand selenium were supplied using thermal evaporation from standard Knudsen effusion cells. All films showed streaky reflection high energy electron diffraction patterns and the expected x-ray diffraction patterns, indicative of good film growths. Further characterization was done with atomic force microscopy and room-temperature Hall effect measurements.

We explored three significant substrate preparation methods. The first was an ultra-high vacuum anneal of the substrate at 800°C for 10 minutes. This gave a 9.4% increase in mobility without noticeable change to the surface of the substrate. Second, we found that the previous use of Nano-strip<sup>®</sup>, a stabilized sulfuric acid and hydrogen peroxide mix, reduced the mobility of the film by 5-12%. It was previously thought that this reagent's ability to eliminate positive and negative resists, remove organic materials, and create an atomically smooth surface would be beneficial to the growth of thin films. However, the use of Nano-strip® likely resulted in a sulfurterminated surface, as characterized by XPS. This sulfur-terminated surface was detrimental to good Bi<sub>2</sub>Se<sub>3</sub> film growth. Third, we found that annealing sapphire at temperatures which formed a terrace-step morphology had approximately a 40% improvement in mobility of the film. Changes in the anneal temperature showed slight changes in the sapphire step heights following previous literature. Use of UV-light to clean the substrate surface showed mixed results with improvement of mobility and carrier density on less terraced surfaces but worse carrier density and mobility of the highly terraced surfaces. AFM characterization of the films showed no considerable changes in RMS roughness values. Further studies can focus on optimizing these step heights to better match Bi<sub>2</sub>Se<sub>3</sub> growth conditions.

NM-MoP-49 Comparison of the Optoelectronic Properties of InGaAs and GaAsSb Absorbers on InP for 1.55 μm Avalanche Photodiodes, Nathan Gajowski, The Ohio State University; P. Webster, Air Force Research Lab; S. Lee, The Ohio State University; P. Grant, Air Force Research Lab; S. Krishna, The Ohio State University

The development of short-wave infrared Avalanche Photodiodes (APDs) operating at the 1.55 µm wavelength are critical for advancement of remote sensing and optical communication. APDs achieve internal gain through the impact ionization process, which yields a sensitive, high-speed detector that suppresses the system's circuit noise. The 1.55 µm wavelength is notable in optical communication for its low loss in optical fiber and can also be used in eye-safe LiDAR systems which, along with high atmospheric transmission and low solar background at this wavelength. enable detection at longer distances than conventional systems [1]. Separate Absorption, Charge, and Multiplication (SACM) APDs are specifically well suited to both applications due to the highly tunable device design. By separating the absorption and multiplication regions of the device, each can be optimized individually, resulting in devices with lower dark currents, lower excess noise factors, and higher gains. The InP substrate is well situated for SACM APD applications at 1.55 µm due to the availability of lattice-matched quaternary multipliers that exhibit extremely low excess noise as well as two lattice-matched bulk absorbers for this wavelength; In<sub>0.47</sub>Ga<sub>0.53</sub>As and GaAs<sub>0.50</sub>Sb<sub>0.50</sub> [2].

In this work, lattice-matched In<sub>0.47</sub>Ga<sub>0.53</sub>As and GaAs<sub>0.50</sub>Sb<sub>0.50</sub> alloys are grown on InP substrates by molecular beam epitaxy to compare their optoelectronic properties as a function of doping and evaluate their performance as the absorber volume in SACM APD applications. The band gap and Urbach energy are measured as a function of temperature using steady-state photoluminescence and evaluated using an Einstein single oscillator model to extract the frozen in disorder, average phonon energy, and electron-phonon coupling parameters. The minority carrier lifetime of each material is extracted from time-resolved photoluminescence to assess how doping modifies the minority carrier lifetime, providing insight into the optimal design of an effective absorber in an SACM APD.

NM-MoP-50 Transmission Electron Microscopy Studies of the Formation of In<sub>2</sub>Se<sub>3</sub> Layers via Selenium Passivation of InP(111)B Substrates, Kaushini Wickramasinghe, C. Forrester, City College of New York, City University of New York; M. McCartney, D. Smith, Arizona State University; M. Tamargo, City College of New York, City University of New York

Three-dimensional topological insulators (3d-TIs) are a new class of materials with their non-trivial topology giving rise to exotic metallic surface states protected by time reversal symmetry and an insulating bulk. However, exploiting the surface channels is often hindered by the presence of crystal defects, such as antisites, vacancies and twin domains. In particular, twinning is shown to be highly deleterious for terahertz device applications. Twinning reduces helicity dependent topological photocurrent, thus eliminating twinning can provide a path to chip-scale polarimeters, among other devices. In the past, it has been challenging to fully suppress the twin domains. In our previous study, we have demonstrated that the growth of fully twin-free Bi<sub>2</sub>Se<sub>3</sub> and other 3D TIs on smooth non-vicinal InP(111)B substrates is feasible by incorporating a newly developed selenium (Se) passivation technique during the oxide removal process of the substrate<sup>1</sup>. This technique allows the formation of several quintuple layers of untwined In<sub>2</sub>Se<sub>3</sub> on the InP surface that serve as the platform for the growth of twin-free Bi<sub>2</sub>Se<sub>3</sub>.

In this study, we investigate the structural details of the In<sub>2</sub>Se<sub>3</sub> and Bi<sub>2</sub>Se<sub>3</sub> layers formed by this novel technique using high resolution transmission electron microscopy (HR-TEM) and scanning transmission electron microscopy (STEM). The data show that well-ordered In<sub>2</sub>Se<sub>3</sub> van der Walls layers form over the InP (111)B surface. The interface between the zinc blende InP lattice and the rhombohedral In<sub>2</sub>Se<sub>3</sub> layers is abrupt and flat, and largely free of imperfections and defects. Similarly abrupt interfaces are evident at the Bi<sub>2</sub>Se<sub>3</sub>/In<sub>2</sub>Se<sub>3</sub> interface. Additionally, STEM bright field (BF) and dark field (DF) images show clear evidence of significant Se diffusion into the substrate beyond the In<sub>2</sub>Se<sub>3</sub>/InP interface. The presence of this excess Se does not alter the crystal structure of the InP, which remains zinc blende. This observation suggests that during the In<sub>2</sub>Se<sub>3</sub> formation process, the In atoms remain fixed in their lattice sites while Se diffuses into the substrate. When sufficient Se is present at the appropriate temperature, the lattice transforms into the rhombohedral In<sub>2</sub>Se<sub>3</sub> lattice, maintaining its registry with the substrate and precluding the formation of twins. Once this twin-free In<sub>2</sub>Se<sub>3</sub> layer is formed, it serves as a perfect template for twin-free Bi<sub>2</sub>Se<sub>3</sub> layer or other 3D TI formation. This novel approach for forming a high quality twin free 2-dimensional crystal on a 3-dimensional zinc blende crystal lattice may have more general applications to other technologically important substrates.

1. Wickramasinghe et al. Crystals, 13(4),677 (2023)

#### Science and Technology of MBE Room Hall of Ideas E-J - Session ST-MoP Science and Technology of MBE Poster Session

ST-MoP-1 Shadow Mask Molecular Beam Epitaxy, Shagorika Mukherjee, S. Sitaram, X. Wang, University of Delaware; S. Law, Penn State University Shadow mask molecular beam epitaxy (SMMBE) is a form of selected area epitaxy (SAE) in which one can pattern vacuum-deposited films using a mechanical mask. If the mask is not a contamination source itself and is near the substrate, epitaxial layers may be deposited on the substrate sequentially through the patterned mask. In this way, a patterned film can be produced without etching. A unique feature of SMMBE is the shadowing effect that arises near the mask edges, which causes the elemental fluxes to change as a function of position. The shadowing effect occurs when the mask edges are not perfectly vertical. This results in a gradient of film thickness and/or composition. The gradient can be controlled by changing the mask thickness and/or the angle of the mask edges. In this presentation, we demonstrate the potential of SMMBE to produce in-plane gradient permittivity materials (GPMs) by taking advantage of the shadowing effect.

GPMs are materials in which the permittivity changes horizontally in the plane of the sample. There are a variety of applications of GPMs, we are focused on the excitation of surface plasmon polaritons (SPPs) which can confine light and enhance its electric field. In a GPM, light of different wavelengths is confined at different positions on the sample. If successful, this type of material could be used to develop an on-chip spectrometer. Since we are interested in working in the infrared, our GPMs consist of heavily silicon-doped InAs, which is known to be a good infrared plasmonic material. By varying the silicon doping density in the plane, we can control the Si:InAs permittivity in the lateral direction. We propose to create such a material using SMMBE with reusable silicon shadow masks. The shadow masks are each 200  $\mu$ m thick and have a dimension of 1 cmx1 cm. Each mask has an aperture at its center. The aperture has slanted inner walls, which makes its dimension 0.5 cmx0.5 cm at the top and 0.528 cmx0.528 cm at the substrate side. To date, we have studied the effect of several growth parameters in controlling the in-plane permittivity in the GPMs. In comparison to non-SMMBE growth recipe, we have found that increased quantities of Bi surfactant (2.5x higher) and As (1.5x higher) in the growth recipe of the GPM can significantly minimize poor growth regions around the edges of GPM where the permittivity gradients are. Characterization by s-SNOM (scattering-type scanning near-field optical microscopy) using mid-IR light source illustrates that the GPMs we have made confine different wavelengths of light at different in-plane positions in the GPMs, setting the stage for the creation of an on-chip spectrometer.

# ST-MoP-2 Strain Relaxation of Graded InGaN and AlGaN, Reem Alhelais, F. Mais de Oliveira, H. Stanchu, M. Sarollahi, Y. Mazur, N. Al-Hosainy, M. Ware, University of Arkansas

III-nitride semiconductor materials have proved to be ideal materials for high-power, high-frequency, and high-temperature applications because of their tunable direct band gaps, high breakdown voltage, high absorption coefficient, resistance to defects, lattice match, and polarization characteristics. These materials form a continuous alloy system with direct bandgaps from 6.2 eV (AIN) through 3.4 eV (GaN) to 0.7 eV (InN). The compositionally graded Group III-nitride alloy enables access to a large range of energies through varying the bandgap. This change in bandgap is achieved by varying the indium and aluminum composition during growth, which yields excellent compatibility for various optoelectronic applications.

The growth and quality of mismatched heteroepitaxial III-Nitrides layers are generally influenced by strain relaxation mechanisms that release the accumulated strain energy. Plastic relaxation is generally started by the formation of misfit dislocations above the critical thickness. This has been well studied and is generally understood for heteroepitaxial films with a fixed composition. Graded composition films have been investigated recently for potential incorporation into semiconductor devices, however, the issue of plastic relaxation for graded III-Nitride semiconductors has not been thoroughly investigated.

Graded InGaN and AlGaN can be grown pseudomorphically strained to its substrate until some critical amount of strain energy is built up. This can happen either as a result of reaching a maximum composition or a maximum growth thickness. These two parameters are not independent in terms of their contribution to the buildup of strain energy, and the goal of this study is to determine both the range over which these alloy layers can be grown without relaxation and the mechanisms by which they exhibit relaxation.

In the present work, we have grown both graded InGaN and AlGaN layers with 30 % of In and Al composition of increasing thicknesses for 15 min, 30 min, and 60 min on GaN substrates. We investigated their properties through X-ray diffraction reciprocal space mapping (RSM). With increasing the thickness of these graded layers, the InGaN or AlGaN signature in the RSM shifts from a fully strained position. Atomic force microscopy will be also used to characterize the sample surface of interest, including dislocation density, while transmission electron microscopy will be used to understand the nature of relaxing defects that is formed in these layers. How these introduced dislocations impact the electrical and optical properties will be demonstrated through photoluminescence and Raman spectroscopy.

#### ST-MoP-3 Improved N-Type Doping Activation in AlGaAs:Si Through Digital Alloy Growth, *Xizheng Fang*, B. Kim, D. Montealegre, Y. Wang, A. Birge, M. Lee, University of Illinois at Urbana Champaign

Despite its maturity, MBE growth of Si-doped Al<sub>x</sub>Ga<sub>1-x</sub>As (n-Al<sub>x</sub>Ga<sub>1-x</sub>As:Si) with moderately high electron concentration ( $n_0 > 1e17$  cm<sup>-3</sup>) remains challenging due to the formation of DX centers and other traps, particularly near the direct to indirect band transition at x=0.45.<sup>1</sup> Al<sub>x</sub>Ga<sub>1-x</sub>As surfaces are also prone to roughening during growth<sup>2</sup>, which can negatively impact subsequent growth of active regions. In this work, we show that n-Al<sub>0.4</sub>Ga<sub>0.6</sub>As:Si grown as a 3 monolayer (ML) GaAs/2 ML AlAs digital alloy at 610°C exhibits ~34x higher activation than in bulk samples, enabling straightforward doping up to  $n_0 = 2.5e17$  cm<sup>-3</sup> with smooth surface morphology.

All samples were grown on semi-insulating GaAs (001) and consisted of a 200 nm GaAs buffer, a 100 nm undoped  $Al_{0.4}Ga_{0.6}As$  layer to prevent charge transfer into the GaAs buffer, and a 500-nm-thick  $n-Al_{0.4}Ga_{0.6}As$ :Si layer

followed by a 9 nm n<sup>+</sup>-GaAs cap to facilitate ohmic contact formation. In all cases, the growth rate was held at 0.65  $\mu$ m/hr with V/III of 30 and a target Si concentration [Si] = 1e19 cm<sup>-3</sup>.

We started with the growth of a bulk n-Al<sub>0.4</sub>Ga<sub>0.6</sub>As:Si alloy control sample, which gave a root mean square (RMS) roughness of 2.1 nm and  $n_0 = 7.3e15$ cm<sup>-3</sup> (activation = 0.073%) according to Hall effect measurements. In an attempt to increase the activation, we reduced [Si] to 5e18 cm<sup>-3</sup> and added 10x Si delta doping spikes<sup>3</sup> at 50 nm intervals to bring the integrated Si concentration to 1e19 cm<sup>-3</sup>. However, the activation only improved by 40%, while the roughness increased by 2x. Next we grew a 6 ML GaAs/4 ML AlAs digital alloy (6/4 DA), with the rationale that Si atoms residing within layers of pure GaAs or AlAs and away from the GaAs/AlAs interfaces may be less prone to DX-center formation. Hall effect measurements showed that activation increased by 2.6x over the control, while the RMS roughness decreased to 1.0 nm; the position of satellite peaks in 004  $\omega/2\theta$  x-ray scans agreed with the 10 ML periodicity. Encouraged by the improvement in surface morphology and activation, we next grew 3/2 DAs at 500°C and 610°C, attaining 14x and 34x improvements in activation, respectively, while maintaining roughness similar to the 6/4 DA; the peak n<sub>0</sub> achieved in this work of 2.5e17 cm<sup>-3</sup> is sufficient for use as the n-cladding layer in GaAs/AlGaAs laser diodes.<sup>4</sup> All samples exhibited photoluminescence at the expected wavelength (~640 nm) and similar reflectance spectra to bulk n-Al<sub>0.4</sub>Ga<sub>0.6</sub>As:Si, showing that the DAs mimic the optical properties of random alloys. In future, we will further explore the effects of periodicity and composition on noto elucidate the mechanism for n-doping enhancement in DAs.

#### ST-MoP-4 Metal Contact Etch Process Optimization of Metal Contact Etching in 3D Integration Devices, Sung Gyu Pyo, CAU, Republic of Korea

In this paper, we would like to report the metal contact etch, which is different from the existing device contact process, on the film stack side and the supercontact etching characteristics accordingly. General metal contact etch etch is organically related to physical profile and electrical properties, so evaluating only one item does not have much meaning, but 3D Integration. The physical profile characteristics of metal contact etch etch as upercontact were examined. As a result of the 1st step etch evaluation, the etch target in the wafer left area was approximately 2365A, and the bottom surface was found to be good with a bottom rounded profile.

After the 1st step etch for liner TEOS and nitride removal, the stopping margin was evaluated using 1) metal contact etch etch conditions where the target was reduced by about 22 sec, 2) CMOS image sensor metal contact etch baseline conditions to which an ILD reduction scheme was applied to improve optical characteristics, and 3) the selectivity was improved by increasing the C5F8/O2 ratio and the etch target was reduced. As a result, all three conditions were punch-through of BLC nitride has occurred.

In the 1st and 2nd steps, after proceeding with etch to the appropriate target, in the 3rd step, a good stopping margin was secured as a result of evaluating the 3-step etch recipe that over-etched using high selectivity.

It was confirmed that the stopping margin according to the over etch target split and process window change was also good, and the CD bias also secured good results.

### **Tuesday Morning, September 19, 2023**

**Novel Materials** 

**Room Ballroom A - Session NM-TuM1** 

**Novel Materials, Optoelectronics** 

Moderator: Prof. Jason Kawasaki, University of Wisconsin - Madison

7:45am NM-TuM1-1 Welcome and Sponsor Thank Yous,

#### 8:00am NM-TuM1-2 NAMBE Young Investigator Awardee Talk: Dislocation Dynamics in InGaSb Graded Buffers on GaSb Grown by MBE, Stephanie Tomasulo, M. Twigg, S. Maximenko, I. Vurgaftman, J. Nolde, U.S. Naval Research Laboratory INVITED

Conventional III-V materials possess a large range of bandgap energies (Eg) enabling their use in a wide variety of applications. However, researchers are typically constrained to the few lattice-constants available in commercial binary substrates, which greatly restricts access to the lowest-Eg (0.05-0.35 eV) options. Graded buffers can and have been used to bridge the gap between an available substrate and a bulk-like material with the desired Eg while mitigating excessively high threading dislocation densities (TDD). However, the dislocation dynamics in III-Sb materials is still poorly understood. Here, we explore the effects of substrate temperature  $(T_{sub})$  on surface morphology and dislocation dynamics in In<sub>x</sub>Ga<sub>1-x</sub>Sb graded buffers. We grew two identical samples at  $T_{sub}\!=\!455$  and 495 °C (pyro). These had 8  $\times$ 365 nm steps, increasing x by ~0.03 per step and capped with 1  $\mu$ m of In<sub>0.28</sub>Ga<sub>0.72</sub>Sb (grading rate of=0.58%/µm). Nomarski microscopy shows the lower T<sub>sub</sub>=455 °C results in a rough surface, while T<sub>sub</sub>=495 °C has a smooth, cross-hatched appearance as expected for a graded buffer. We then measured x-ray reciprocal space maps (RSMs) with the x-ray beam incident along both [110] and [1-10] to assess potential asymmetric strain relaxation. Such relaxation has been observed in other III-V graded buffers and explained by different dislocation formation energies/glide velocities along each direction, which results from the core structure of the dislocation being terminated with either a group-III (beta core) or a group-V (alpha core) element [1]. Thus, asymmetries in RSMs can be linked to dislocation glide behavior. We previously observed asymmetry in InAsSb graded buffer RSMs, finding minimal tilt when measured along [110] and significant tilt when measuring along [1-10], suggesting a preference for beta dislocation cores [2]. Interestingly, InGaSb graded buffers reveal the opposite asymmetry, i.e. measuring along [110] results in tilt while measuring along [1-10] shows little tilt, indicating alpha dislocation cores. Furthermore, in InGaSb, the tilt direction changes from positive (more glide on (111)) for T<sub>sub</sub>=455 °C to negative (more glide on (-1-11)) for 495 °C revealing the significant role  $T_{\mbox{\scriptsize sub}}$  plays in dislocation dynamics. Additional samples will be explored, varying the V/III and grading rates, to build an understanding of how to control dislocation dynamics and TDD in InGaSb graded buffers. Etch pit density and transmission electron microscopy will be used to quantify TDD and supplement the understanding of dislocation dynamics gained from RSMs. [1] France et al. JAP 107, 103530 (2010). [2] Tomasulo et al., NAMBE 2019.

# 8:30am NM-TuM1-4 Emergent Ferromagnetism in Altermagnetic Candidate MnTe Films Grown on InP (111), Matthew Brahlek, Oak Ridge National Laboratory

To push into new generations of spintronic devices requires understanding new magnetic phenomena and also how to control both known and emerging material platforms as high-quality epitaxial thin films. Specifically, altermagnets are a new phase that is predicted to exhibit a strong spin splitting of the band structure, which can form the basis for new spintronic applications. MnTe, a candidate altermagnet, is a room-temperature antiferromagnet (TN  $\approx$  310 K) semiconductor (energy gap of order 1 eV) with a NiAs structure. Here, we present results on the molecular beam epitaxy growth and properties of MnTe/InP(111). Using polarized neutron reflectivity and magnetotransport, we find that there is emergent ferromagnetic behavior likely driven by a combination of charge transfer and strain. The ferromagnetic component is likely a slight canting of the bulk-like A-type antiferromagnetic state, as seen by neutron diffraction. This high level of tunability of MnTe opens the door to tailoring interlayer magnetic interactions in this layered material system. Together these results provide a potential mechanism of tuning antiferromagnetic ordering for applications in high-speed, next-generation spintronics.

8:45am NM-TuM1-5 Growth Parameters Impact on Electronic and Optical Properties of ErAs:InGaAlBiAs Materials, *Wilder Acuna*, *W. Wu*, *J. Bork*, *M. Jungfleisch*, *L. Gundlach*, *J. Zide*, University of Delaware

We present our work on the growth of ErAs:InGaAlBiAs through a digital alloy approach where a superlattice of thin layers of the quaternary InGaBiAs and InAlBiAs, behave as a guinary and allows tuning the bandgap with the gallium and aluminum composition just with layer thickness variation. We target InGaAlBiAs for 1550 nm optical excitation for photoconductive switches (PCS) that can generate and detect terahertz (THz) pulses. Semiconductors implemented in PCS need high dark resistance, subpicosecond carrier lifetime, and a high carrier mobility is desirable. Incorporating erbium above the solubility limit creates ErAs nanoparticles, decreasing the material's carrier lifetime. At the same time, ErAs nanoparticles have a strong pinning effect of the effective Fermi level on the material. In this narrow bandgap semiconductor, the Fermi level is close to the conduction band, i.e., there is a high electron concentration, which decreases the dark resistivity. Other authors have found that the size of ErAs nanoparticles affects the electron concentration in InGaAs, where bigger nanoparticles cause a lower electron concentration. Er can be codeposited, forming nanoparticles at the same time the matrix is grown, which saves growth time; however, while the usual 490°C growth temperature could give enough adatom mobility to create sizeable nanoparticles, our structure of interest is a dilute bismuthide that requires lower temperature growth (~280°C). At this low temperature, we use interrupt growth and migration-enhanced epitaxy, which allows us to have lower carrier concentration. All these samples are characterized by different techniques to determine relevant properties. Dark resistance is obtained from Hall effect and Van der Pauw measurements, carrier lifetime from optical pump THz probe spectroscopy, optical band gap from spectrophotometry, material quality from high-resolution x-ray diffraction, and composition through Rutherford backscattering spectrometry.

#### 9:00am NM-TuM1-6 Heteroepitaxial Growth of Site-determined Quantum Emitters in 2D GaSe Films, *Mingyu Yu*, University of Delaware; *S. Law*, Pennsylvania State University

GaSe is an attractive van der Waals (vdW) material due to its intriguing bandgap behavior and is an ideal choice for quantum photonic technology. We will grow GaSe films on patterned substrates by molecular beam epitaxy (MBE) to obtain wafer-scale films with site-controlled localized quantum emitters (QEs). A proper substrate is critical as the substrate pattern is used to funnel electrons and holes to a local minimum to form emission. Unlike traditional epitaxy, vdW materials may grow on substrates with large differences in lattice constant/structure as the weak interlayer bonds make them less affected by substrate. However, it also means less control over the crystal structure and surface morphology of the deposited film. To achieve the goal—GaSe films with strain-localized QEs—we first need flat atomically-thin GaSe films with minimal twin boundaries. Our previous work[1] proved that Al<sub>2</sub>O<sub>3</sub> was not suitable for the growth of GaSe films due to the poor wettability of Ga, so we turned to GaAs(111)B, which has the same cation as GaSe and a relatively small lattice mismatch (6%).

Typically, vdW materials cannot grow heteroepitaxially on 3D substrates unless the dangling bonds are terminated. We developed a processannealing in Se at 680°C-to deoxidize GaAs(111)B in an As-free MBE while generating a Se-terminated surface smooth enough for subsequent growth (Fig. S1). Preliminary work has resulted in GaSe films with somewhat rough surfaces (Fig. S2a, 2d). Additional annealing of the substrate prior to growth can notably promote film coalescence and reduce roughness (Fig. S2b, 2e), probably because it facilitates a fully Se-terminated GaAs surface. Fig. S2e shows a surface comprised of twinned triangular domains, and the reflection high energy electron diffraction (Fig. S2g) confirms the coexistence of two GaSe orientations. In addition, the 2-step method (lowtemperature nucleation followed by high-temperature growth) facilitates film coalescence (Fig. S2c, 2f), since high substrate temperature can enhance adatom mobility. Moreover, fresh GaAs(111)B wafers lead to more directional nucleation of GaSe (Fig. S3), implying that additional oxidation due to air exposure is detrimental to the growth of high-quality GaSe. We also found that the quality of vdW films had an opposite dependence on growth conditions (temperature and rate) in 2D/2D and 2D/3D growth modes. The next challenge is to achieve uniform coverage and unidirectional nucleation. The effects of growth parameters, substrate pretreatment, and uncracked/cracked Se will be examined to better understand the growth mechanics and morphology evolution of GaSe.

### **Tuesday Morning, September 19, 2023**

9:15am NM-TuM1-7 Evaluating (001) and (111)A InAs Quantum Emitters at Telecommunication Wavelengths Grown by Droplet Epitaxy, Margaret Stevens, US Naval Research Laboratory; W. McKenzie, G. Baumgartner, Laboratory for Telecommunication Sciences; J. Grim, A. Bracker, US Naval Research Laboratory

Droplet epitaxy is a versatile growth method that can produce nanostructures with emission in the telecommunication wavelength ranges. The MBE growth conditions have a significant impact on the morphology of the nanostructures, and as a result, the emission of the guantum dots. Due to the low surface diffusion of In on InAlAs surfaces, conventional growth conditions in MBE lead to rings of small quantum dots on (001) surfaces that limit the emission to shorter wavelengths. Previously, we have shown that we can manipulate the crystallization stage of droplet epitaxy to create new morphologies on (001) surfaces and shift the resulting emission to 1300-1550 nm at 4 K [1]. By utilizing a two-stage As flux exposure, including a low flux exposure immediately after metal droplet deposition followed by a high flux exposure with a temperature ramp [2], we can create clusters of larger quantum dots at densities ~  $1/\mu m^2$  surrounded by a higher density of smaller quantum dots. Further understanding of why the different morphologies form, and how we can better control quantum dot density, is needed.

In this work, we have expanded on the two-stage As flux exposure to further manipulate the morphology and density of InAs QDs grown on InAlAs/InP (001). We grew samples with different temperature ramps and different low-flux conditions to change the density and size of the different QD features that form. Atomic force microscopy (AFM) is used to study the quantum dot morphology and density and photoluminescence spectroscopy (PL) is used to study the resulting emission. Additionally, we grew InAs QDs via droplet epitaxy on InAlAs/InP (111)A surfaces to compare quantum dot size, density, and resulting emission to samples grown by the two-stage flux technique on (001) substrates. Finally, we incorporate our telecommunication range emitting quantum dots in diode structures to tune the QD charge state and study resulting emission properties.

[1]	Μ	۱.	Α.	Steve	ns	et	al.,	JVST	Α,	41,	032703	2023
[2]	S.	V.	Bala	akirev	et	al.,	Appl.	Surf.	Sci.,	578,	152023,	2022

9:30am NM-TuM1-8 Growth of InSb Quantum Well on InAs Using AlInSb Buffer Layer Assisted by Interfacial Misfit Dislocation Arrays, Fatih Furkan Ince, A. Newell, T. Rotter, G. Balakrishnan, University of New Mexico; M. McCartney, D. Smith, Arizona State University

Mid-wave infrared (MWIR) detectors are widely used in various fields, such as medical devices, remote sensing, and spectroscopy. InSb-based infrared focal plane arrays (FPAs) have emerged as a popular choice for their affordability, scalability, and temporal stability, as well as their spatial uniformity. However, achieving fully relaxed tunable absorber to cover MWIR and LWIR is challenging due to lack of binary substrates. Thus, type-II superlattices and metamorphic buffers are employed to cover the MWIR spectrum and extend into the long-wave infrared (LWIR) region [1], [2]. In this study, we propose using interfacial misfit dislocations to grow fully relaxed InSb on an InAs substrate, combined with direct growth of Al<sub>x</sub>In<sub>1-x</sub>Sb buffer layers on InAs, to develop tunable InAsSb absorbers for MWIR and LWIR applications. This approach is expected to lead to the production of high-performance MWIR and LWIR detectors, potentially opening new avenues for future applications.

We will discuss using interfacial misfit dislocations to form instantly relaxed buffer layers on InAs substrates, and analyze the directly grown AlInSb and InSb epilayers using HR-XRD  $\omega$ -20 scans and reciprocal space mapping. TEM analysis showed misfit dislocation arrays at the AlInSb/InAs interface. We investigated InSb quantum wells grown with different AlxIn1-xSb barrier layers using PL and present a detailed analysis of the InSb quantum wells based on PL and TRPL results. These findings provide a better understanding of the properties of these materials and their potential for MWIR and LWIR applications.

[1] W. L. Sarney, S. P. Svensson, Y. Xu, D. Donetsky, and G. Belenky, "Bulk InAsSb with 0.1 eV bandgap on GaAs," J. Appl. Phys., vol. 122, no. 2, p. 025705, Jul. 2017.

[2] A. Rogalski, "Next decade in infrared detectors," in Electro-Optical and Infrared Systems: Technology and Applications XIV, Warsaw, Poland, 2017. \*Author for correspondence: incef@unm.edu

# 9:45am NM-TuM1-9 Molecular Beam Epitaxy of Kagome Antiferromagnetic Mn<sub>3</sub>GaN grown on MgO(001), Ali Abbas, A. Shrestha, A. Smith, Ohio University

There have been very few studies of antiperovskite structure Mn<sub>3</sub>GaN in general and it was seen in MBE growth mainly as a second-phase precipitate when growing MnGaN [4]. However, this material is very interesting since it can support the Kagome antiferromagnetic spin structure [3]. And so, we grow thin films of Mn<sub>3</sub>GaN on cubic MgO(001) substrates using rf N-Plasma MBE. In our work, Mn<sub>3</sub>GaN is deposited at 250 ±10°C with a Mn:Ga:N flux ratio of 3:1:1. We keep the Ga:N ratio fixed using an RF plasma nitrogen source. The sample surface is continuously monitored throughout the growth using reflection high energy electron diffraction. During the growth, the RHEED pattern was observed to be highly streaky, indicating an atomically smooth surface. In addition, we observed half-order fractional streaks (2x pattern) in the [100] direction. The calculated in-plane lattice constant based on RHEED is 3.89 ±0.06 Å. This value is very close to the lattice constant a of Mn<sub>3</sub>GaN according to theory (3.898 Å) [1] and with the in-plane experimental value for sample growth by sputtering (3.896 Å); and in that work, the authors also observed a 2x pattern [2]. We also measure the out-of-plane lattice constant using Xray diffraction. For the major 002 peak, the value calculated is 3.84 ±0.06 Å which also agrees well with the theoretical value (3.898 Å) [1] and with the experimental reported c value [2] (3.881 Å). Since we did not observe significant second-phase peaks, the phase purity of the sample is quite high, and Rutherford backscattering confirms a stoichiometry of 3:1:1.

This work is supported by the U.S. Department of Energy, Office of Basic Energy Sciences, Division of Materials Sciences and Engineering under Award No. DE–FG02–06ER46317.

References:

[1] E. F. Bertaut, D. Fruchart, J. P. Bouchad, and R. Fruchart, (1968). Diffraction Neutronique de  $Mn_3GaN$ . *Solid State Commun.***6**, 251–256.

[2]T. Hajiri, K. Matsuura, K. Sonoda, E. Tanaka, K. Ueda, & H. Asano, (2021). Spin–Orbit–Torque Switching of Noncollinear Antiferromagnetic Antiperovskite Manganese Nitride Mn 3 Ga N. *Physical Review Applied*, **16(2)**, 024003.

[3]T. Nan, C.X. Quintela, J. Irwin, G. Gurung, D.F. Shao, J. Gibbons, N. Campbell, K. Song, S.Y. Choi, L. Guo, and R.D. Johnson, (2020). Controlling spin current polarization through non–collinear antiferromagnetism. *Nature communications*, **11(1)**, p.4671.

[4] KH. Kim, KJ. Lee, HS. Kang, FC. Yu, JA. Kim, DJ. Kim, KH. Baik, SH. Yoo, CG. Kim, YS. Kim, (2004).Molecular beam epitaxial growth of gan and gamnn using a single precursor. physica status solidi (b), 241(7):1458–1461

#### **Novel Materials**

#### Room Ballroom A - Session NM-TuM2

#### **Thin Film Membranes**

Moderator: Dr. Roman Engel-Herbert, Paul Drude Institute

10:30am NM-TuM2-12 Doping the Undopable: Hybrid Molecular Beam Epitaxy Growth, n-type Doping, and Field-Effect Transistor using CaSnO<sub>3</sub>, Fengdeng Liu, P. Golani, T. Truttmann, University of Minnesota, USA; I. Evangelista, University of Delaware; M. Smeaton, Cornell University; D. Bugallo, Drexel University; J. Wen, A. Kamath Manjeshwar, University of Minnesota; S. May, Drexel University; L. Kourkoutis, Cornell University; A. Janotti, University of Delaware; S. Koester, University of Minnesota; B. Jalan, University of Minnesota, USA

ABSTRACT: The alkaline earth stannates are touted for their wide band gaps and the highest room-temperature electron mobilities among all the perovskite oxides. CaSnO<sub>3</sub> has the highest measured band gap in this family and is thus a particularly promising ultra-wide band gap semiconductor. However, discouraging results from previous theoretical studies and failed doping attempts had written off this material as "undopable". Here we redeem CaSnO<sub>3</sub> using hybrid molecular beam epitaxy (hMBE), which provides an adsorption-controlled growth for the phase-pure, epitaxial and stoichiometric CaSnO<sub>3</sub> films. By introducing lanthanum (La) as an n-type dopant, we demonstrate the robust and predictable doping of CaSnO<sub>3</sub> with free electron concentrations, *n*, from  $3.3 \times 10^{19}$  cm<sup>-3</sup> to  $1.6 \times 10^{20}$  cm<sup>-3</sup>. The films exhibit a maximum room-temperature mobility of 42 cm<sup>2</sup> V<sup>-1</sup>s<sup>-1</sup> at *n* =  $3.3 \times 10^{19}$  cm<sup>-3</sup>. Despite having a smaller radius than the host ion, La expands the lattice parameter. Using density functional calculations, this

### **Tuesday Morning, September 19, 2023**

effect is attributed to the energy gain by lowering the conduction band upon volume expansion. Finally, we exploit the robust doping by fabricating the CaSnO<sub>3</sub> -based field-effect transistors. The transistors show promise for CaSnO<sub>3</sub>'s high-voltage capabilities by exhibiting low off-state leakage below 20 pA/µm at a drain-source voltage of 100 V and on-off ratios exceeding 10<sup>6</sup>. This work opens the door to future studies on the semiconducting properties of CaSnO<sub>3</sub> and the many devices that could benefit from CaSnO<sub>3</sub>'s exceptionally wide band gap.

10:45am NM-TuM2-13 Controlling the Balance between Remote, Pinhole, and van der Waals Epitaxy of Heusler Films on Graphene/Sapphire, *Taehwan Jung*, Z. LaDuca, D. Du, S. Manzo, K. Su, X. Zheng, V. Saraswat, University of Wisconsin - Madison; J. McChesney, Argonne National Lab; M. Arnold, J. Kawasaki, University of Wisconsin - Madison

Remote epitaxy is promising for the synthesis of lattice-mismatched materials, exfoliation of membranes, and reuse of expensive substrates. However, clear experimental evidence of a remote mechanism remains elusive. Alternative mechanisms such as pinhole-seeded epitaxy or van der Waals epitaxy can often explain the resulting films. Here, we show that growth of the Heusler compound GdPtSb on clean graphene/sapphire produces a 30° rotated (R30) superstructure that cannot be explained by pinhole epitaxy [1]. With decreasing temperature, the fraction of this R30 domain increases, compared to the direct epitaxial R0 domain, which can be explained by a competition between remote versus pinhole epitaxy. Careful graphene/substrate annealing and consideration of the relative lattice mismatches are required to obtain epitaxy to the underlying substrate across a series of other Heusler films, including LaPtSb and GdAuGe. The R30 superstructure provides a possible experimental fingerprint of remote epitaxy, since it is inconsistent with the leading alternative mechanisms.

This work was supported by the Air Force Office of Scientific Research FA9550-21-0127  $\,$ 

[1] D. Du, et. al. Nano Lett. 2022, 22, 21, 8647-8653

11:00am NM-TuM2-14 Synthesis of Flexomagnetic GdAuGe Membranes via Van Der Waals Epitaxy on Graphene Terminated Germanium, Zachary LaDuca, S. Manzo, T. Jung, T. Samanta, K. Su, M. Arnold, J. Kawasaki, University of Wisconsin - Madison

The ability to synthesize freestanding membranes of crystalline materials is critical for advancing the understanding of strain and strain gradient effects on materials properties. A common approach for the synthesis of freestanding membranes is epitaxial growth on graphene terminated substrates, where direct bonding between the film and substrate is prevented, allowing for exfoliation of the film from the substrate. However, challenges with wetting on the low surface energy graphene frequently prevents the growth of smooth epitaxial films. Here we demonstrate high quality growth of the flexomagnetic Heusler compound GdAuGe on graphene terminated germanium substrates using a cold-seeded growth approach. Scanning electron microscopy, atomic force microscopy, and xray diffraction experiments illustrate the tradeoffs resulting from growth at high and low temperatures and confirm that multistep growth approach comprised of a few-nanometer, amorphous deposition at room temperature followed by an anneal and subsequent growth at elevated temperature result in smooth, highly ordered films. This improved morphology and crystallinity enhances the ability to control the strain state in rippled GdAuGe membranes.

# 11:15am NM-TuM2-15 Flexomagnetism and Strain Induced Superconductivity in Rippled GdAuGe Heusler Membranes, Tamalika Samanta, Z. LaDuca, D. Du, T. Jung, S. Manzo, K. Su, M. Arnold, J. Kawasaki, University of Wisconsin - Madison

Rare earth-based Heuslers are prospective materials platforms for magnonics, topological spin texture, superconductivity, THz spintronics, etc. [1, 2]. The magneto-mechanical coupling in these materials allows for better control and manipulation of the primary order parameter and magnetic flexibility [3]. Here, we demonstrate novel flexomagnetic responses i.e., the coupling between strain gradient and magnetism, and strain-induced superconductivity, in GdAuGe Heusler membranes. The thin films of GdAuGe Heusler composition have been grown on monolayer

Graphene/ Ge (111) by molecular beam epitaxy (MBE). GdAuGe films are then mechanically exfoliated to form free-standing rippled membranes.

GdAuGe shows an antiferromagnetic ordering below ~17 K, which is sustained when a homogeneous strain is applied. However, the application of strain gradient dramatically alters the magnetic ground state of GdAuGe in the rippled membranes. A phase diagram of the rippled GdAuGe membranes is shown in Fig. 1(a). Notably, a moderate strain gradient of a few tenths of a percentage transforms the ground state from antiferromagnetic to unconventional ferrimagnetic phases. These ferrimagnetic ground states in the rippled membranes offer the possibility of discovering spin reorientation and other unique magnetic phenomena; the most exciting observation is the emergence of superconductivity in GdAuGe membranes when a very large strain gradient is applied, with superconducting transitions occurring at low temperatures below ~3.5 K. Figure 1(b) shows the magnetic characterization of a superconducting GdAuGe rippled membrane.

At present, the microscopic origin of flexomagnetism and its effects on the thermodynamics of spin reorientation and phase transitions in these membranes remain unclear. Advanced spectroscopic measurements and magneto-transport experiments, combined with theoretical modeling, are planned to further investigate the phenomena in these rippled membranes.

References

1. Graf, Tanja, et al. "Simple rules for the understanding of Heusler compounds." Progress in solid state chemistry 39.1 (2011): 1-50.

2. Kawasaki, Jason K. "Heusler interfaces—Opportunities beyond spintronics?." APL Materials 7.8 (2019): 080907

3. Du, Dongxue, et al. "Epitaxy, exfoliation, and strain-induced magnetism in rippled Heusler membranes." Nature Communications 12.1 (2021): 1-7

11:30am NM-TuM2-16 Growth Mechanism of SrTiO<sub>3</sub> on a Graphenecovered Substrate Using Hybrid MBE, Sooho Choo, University of Minnesota, Republic of Korea; *H. Yoon*, University of Minnesota, USA, Republic of Korea; *B. Matthews*, Pacific Northwest National Laboratory; *S. Sharma*, University of Minnesota, USA; *S. Spurgeon*, *S. Chambers*, Pacific Northwest National Laboratory; *R. James*, *B. Jalan*, University of Minnesota, USA

Epitaxial films grown on a substrate covered with two-dimensional (2D) materials offer many exciting possibilities: reusability of the substrate; ability to obtain a freestanding membrane; and opportunity to reduce misfit dislocations. Three growth mechanisms are argued to be responsible for epitaxial growth on 2D material-covered substrate: (1) Remote epitaxy; (2) Van der Waals epitaxy; and (3) pinhole-assisted epitaxy. It is, however, still unclear which of these three mechanisms is responsible for epitaxial growth. In this talk, we will first present the successful growth of epitaxial SrTiO<sub>3</sub> nanomembranes on SrTiO<sub>3</sub> (001) substrates covered with bilayer graphene. Titanium tetraisopropoxide (TTIP) was used as a source of titanium and oxygen. No additional oxygen was used to avoid graphene oxidation. By varying Sr/TTIP beam equivalent ratios, we reveal a wide MBE growth window for adsorption-controlled growth of stoichiometric SrTiO<sub>3</sub> membranes [1]. Bulk-like lattice parameter of 3.905 Å was obtained for nanomembranes. By combining heteroepitaxial growth, high-resolution Xray diffraction, atomic force microscopy, transmission electron microscopy, and Raman spectroscopy, we discuss all three growth mechanisms highlighting the role of graphene thickness, pinholes, and the substrate's ionicity on epitaxial growth.

## 11:45am NM-TuM2-17 Synthesis of Free-Standing Membranes Using a Sacrificial Layer Method Grown by Hybrid MBE, Shivasheesh Varshney, S. Choo, Z. Yang, J. Wen, S. Koester, B. Jalan, University of Minnesota, USA

Free-standing membranes have broad applications in the creation of symmetry-mismatched, non-equilibrium, and artificial heterostructures. We use sacrificial layer method to synthesize phase-pure epitaxial SrTiO<sub>3</sub> membranes. In this study, we will discuss the growth of strain-engineered SrTiO<sub>3</sub> films using different sacrificial layer(s) grown by hybrid MBE. We characterize the as-grown films using x-ray diffraction (XRD) and atomic force microscopy (AFM). We show exfoliation and transfer of films onto dissimilar substrates, followed by their structural characterization. Finally, we use impedance spectroscopy to characterize the dielectric properties and show a bulk-like dielectric constant of  $\approx$ 300 for SrTiO<sub>3</sub> membranes transferred on Au coated Si substrate.

### **Tuesday Afternoon, September 19, 2023**

#### **MBE-Grown Devices**

**Room Ballroom A - Session GD-TuA1** 

#### Solar Cell and Quantum Computing

 ${\rm {\it Moderator:}}$  Prof. Dr. Minjoo Larry Lee, University of Illinois Urbana-Champaign

#### 1:30pm GD-TuA1-1 InSb-Based Dilute-Bismide Alloys Towards Long-Wave Infrared Sensing, *Corey White*, *M. Bergthold*, *T. Leonard*, *A. Ricks*, *D. Wasserman*, *S. Bank*, The University of Texas at Austin

It is well-established that bismuth incorporation into III-V alloys produces a significant reduction in bandgap energy<sup>1, 2</sup> that can be advantageously leveraged for optoelectronic devices operating at extended wavelengths. For accessing the long-wave infrared, InSb is a particularly promising candidate host alloy due to both its narrow bandgap energy and the relatively similar ideal growth conditions for InSb and III-Bi materials. Here we report progress towards achieving wavelength extension from InSbBi with photoluminescence (PL) from InSbBi with the highest substitutional bismuth concentrations yet reported as well as methods for post-growth bismuth droplet removal.

Epitaxial InSbBi films were grown by solid-source MBE on InSb substrates. Low substrate temperatures, Sb/In flux ratios near stoichiometry, and fast growth rates were employed to encourage significant bismuth incorporation. From these films, X-ray diffraction and Rutherford backscattering spectrometry measurements were used to confirm the bismuth concentration in the films and extrapolate the InBi lattice parameter, which was found to be ~6.63 Å. This is in line with previous reports of the InBi lattice parameter<sup>3-5</sup> confirming that the incorporation of bismuth into InSb causes the lattice to expand. PL measurements from the InSbBi films demonstrated significant wavelength extension beyond that of InSb with emission out to ~6.5  $\mu$ m at 83 K. Atomic force microscopy surface morphology measurements revealed that the samples exhibited droplet formation.

As is typically observed for dilute-bismide alloys,<sup>3, 6</sup> it is increasingly challenging to achieve droplet-free InSbBi surfaces at elevated bismuth concentrations. We investigated post-processing techniques for droplet removal as an alternative to attempting to completely mitigating droplet formation during growth.  $HCl/H_2O_2$  digital wet etching, physical polishing, and ion milling were performed on a film with large bismuth droplets resulting in improvements in roughness of 2.2×, 3.7×, and 4.0×, respectively. Studies on the effects of these techniques on InSbBi optical quality are underway and will be reported at the conference.

<sup>1</sup>S. Francoeur et al., APL 82 (2003).

<sup>2</sup>S. Tixier et al., APL 82 (2003).

<sup>3</sup>M. Rajpalke et al., APL **105** (2014).

<sup>4</sup>A. Shalindar et al., JAP **120** (2016).

<sup>5</sup>B. Joukoff and A. Jean-Louis, JCG&D **12** (1972).

<sup>6</sup>A. Ptak et al., JCG&D **338** (2012).

This work was performed at the UT Austin MRC, a member of the NNCI (NSF No. ECCS-1542159) and supported by Lockheed Martin, NSF (ECCS-1933836), and an NSF GRF (RCW). RBS measurements were performed at Rutgers LSM.

1:45pm GD-TuA1-2 Abrupt Te Doping of MBE-Grown GalnP for Solar Cell Applications, *Brian Li*, *Y. Sun*, *R. Hool*, *M. Lee*, University of Illinois, Urbana-Champaign

For III-V multi-junction solar cells (MJSC), 1.9 eV Ga<sub>0.51</sub>-In<sub>0.49</sub>P (hereafter GaInP) is a primary material for the high-bandgap top cell [1]. In recent years, GaInP solar cells grown by molecular beam epitaxy (MBE) have achieved high efficiencies after post-growth rapid thermal annealing (RTA) [2] similar to cells grown by metalorganic chemical vapor deposition (MOCVD). Another avenue to improve MBE-grown GaInP cells is the use of group VI dopants (Se, Te) instead of group IV dopants such as Si. For example, MOCVD-grown AlGaAs and AlGaInP cells reported superior internal quantum efficiency (IQE) by changing the dopant from Si to Se [3], [4], an improvement that correlates with the shallower defect energy levels of Se [5], [6]. MBE-grown phosphide cells may also be improved by the use of a group VI dopant, but such studies have not been reported. In this work, we show improved IQE of GaInP solar cells with Te doping of the n-type emitter over Si, which was modeled to show improved carrier lifetime in n-GaInP:Te.

A well reported issue for MOCVD GalnP:Te is undesirable Te surface segregation [7], so we first optimized MBE growth conditions to minimize segregation. GalnP samples were grown at different target concentrations of [Te] =  $5.7*10^{17}$  to  $1.7*10^{18}$  cm<sup>-3</sup>, and doping profiles from secondary ion mass spectrometry (SIMS) indeed showed high Te surface segregation at the nominal substrate growth temperature  $T_{sub}$  = 460 °C. For example, the doping profile for target [Te] =  $1.7*10^{18}$  cm<sup>-3</sup> ranged by orders of magnitude ( $5*10^{16}$  cm<sup>-3</sup> at the start of growth to  $7*10^{19}$  cm<sup>-3</sup> at the surface). Next, reducing  $T_{sub}$  from 460 °C to 420 °C greatly suppressed surface segregation for all samples. Abrupt doping profiles were obtained for target [Te] =  $5.7*10^{17}$  cm<sup>-3</sup> solely by this reduced  $T_{sub}$ , while [Te] =  $1.7*10^{18}$  cm<sup>-3</sup> also required a Te "pre-dose" or deposition prior to GalnP growth to achieve an abrupt doping profile. The cause of this difference in pre-dose requirement as a function of doping will be the subject of future work.

Lastly, n-on-p front junction GaInP cells were grown with both GaInP:Te and GaInP:Si n-type emitters (n > 1\*10<sup>18</sup> cm<sup>-3</sup>). For as-grown cells, the IQE of the GaInP:Te cell was significantly higher than that of GaInP:Si, with the short-circuit current density derived from IQE increasing from 13.2 mA/cm<sup>2</sup> to 14.1 mA/cm<sup>2</sup>. Modeling of the IQE indicates a ~4x higher carrier lifetime of GaInP:Te than GaInP:Si. This work shows the promise of GaInP:Te for improved optical material quality, and future work will explore GaInP:Te for rear-junction cells with n = 1–5\*10<sup>17</sup> cm<sup>-3</sup>, as well as the effect of RTA on GaInP:Te.

2:00pm GD-TuA1-3 MBE Growth of Metamorphic 1 eV InGaAs Solar Cells with Low Threading Dislocation Density, Adrian Birge, M. Kim, B. Kim, D. Montealegre, M. Lee, University of Illinois at Urbana-Champaign, USA

Growth of metamorphic (MM) In<sub>x</sub>Ga<sub>1-x</sub>As enables optoelectronic devices, such as near-infrared photovoltaics, detectors, and lasers with bandgap energies (E<sub>g</sub>) ranging from 0.8-1.3 eV, to be integrated on GaAs substrates. Most previously reported MM In<sub>x</sub>Ga<sub>1-x</sub>As structures with x=0.3 (E<sub>g</sub>=1.0 eV) have used metal-organic chemical vapor deposition (MOCVD) at substrate temperatures (T<sub>sub</sub>) over 700°C to facilitate high dislocation glide velocity and thus low threading dislocation density (TDD)<sup>1</sup>, while thermodynamically suppressing phase separation. MBE growth of MM In<sub>x</sub>Ga<sub>1-x</sub>As cannot be done at such high T<sub>sub</sub> due to excessive In desorption. Thus, the MBE growth window must be chosen carefully to balance the need for adequate dislocation glide velocity while kinetically suppressing phase separation. In this work, we describe a two-step method for fully relaxed In<sub>0.3</sub>Ga<sub>0.7</sub>As with a TDD = ~7x10<sup>5</sup> cm<sup>-2</sup>, comparable to the lowest reported values for this material. On this platform we fabricated the first MM 1.0 eV In<sub>0.3</sub>Ga<sub>0.7</sub>As solar cells grown via MBE to our knowledge.

In<sub>x</sub>Ga<sub>1-x</sub>As (x=0-0.3) buffers showed strong phase separation in electron channeling contrast imaging (ECCI) when all growth was carried out at T<sub>sub</sub>=500°C. Therefore, we adopted a two-step approach where In<sub>x</sub>Ga<sub>1-x</sub>As (x=0-0.18) was grown at T<sub>sub</sub>=500°C to facilitate dislocation glide<sup>2</sup>, followed by 460°C for x=0.18-0.3 to prevent PS<sup>3</sup>; an In<sub>0.35</sub>Ga<sub>0.65</sub>As overshoot layer was also included to facilitate strain relaxation. ECCI yielded a TDD < 1x10<sup>6</sup> cm<sup>-3</sup> without PS, while reciprocal space mapping showed 97.9% relaxation in the In<sub>0.3</sub>Ga<sub>0.7</sub>As cap.

Using our two-step  $In_xGa_{1*x}As$  graded buffer as a virtual substrate, a 1 eV InGaAs solar cell was grown and devices were fabricated both as-grown (AG) and with rapid thermal annealing (RTA)<sup>4</sup>. Internal quantum efficiency (IQE) measurements showed improved minority electron diffusion length in the base of the RTA cell, resulting in a 1.15x increase in short-circuit current density (J<sub>SC</sub>). Lighted current-voltage (LIV) curves showed a strong improvement in the open-circuit voltage (V<sub>oC</sub>) after RTA, with a bandgap-V<sub>oC</sub> offset (W<sub>oC</sub>) of 0.46 V, ~0.1 V higher than the best cells in the literature grown by MOCVD at NREL<sup>5</sup>. The fill factor (FF) and cell efficiency ( $\eta$ ) of our RTA'd cells were hampered by high contact resistance, which will be addressed in future runs. Nonetheless, the J<sub>SC</sub> (29.96 mA/cm<sup>2</sup>) and V<sub>oC</sub> (0.538 V) values of our RTA'd cells compare favorably with the best latticematched 1.0 eV solar cells (e.q. InGaAsN on GaAs, InGaAsP on InP) grown by MBE<sup>6</sup>, presenting a promising path for MBE-grown MM 1.0 eV In<sub>0.3</sub>Ga<sub>0.7</sub>As solar cells.

2:15pm GD-TuA1-4 Strained Superlattice InAlGaAs/AlGaAs Spin-Polarized Photocathodes Implemented with Random and Digital Alloying, *Aaron Engel*, University of California, Santa Barbara; *M. Stutzman*, Thomas Jefferson National Accelerator Facility; *J. Dong*, University of California, Santa Barbara; *C. Palmstrøm*, University of California, Santa Barbara

Spin-polarized electron beams are a crucial tool in particle physics experiments and specialized characterization techniques. Their production involves exciting a III-V semiconductor (the photocathode) with nearbandgap circularly polarized light. Spin-polarized photoelectrons are
## Tuesday Afternoon, September 19, 2023

produced through optical selection rules. These photoelectrons are then extracted after functionalizing the surface to create a negative electron affinity condition. Increasing the electron spin polarization (ESP) and the quantum efficiency (QE) of the photocathodes enhances overall experimental efficiency, extends the photocathode lifetime, and enables the production of high bunch charge beams. The current state-of-the-art spin-polarized photocathodes consist of GaAs/GaAs<sub>0.67</sub>P<sub>0.33</sub> strained superlattice quantum wells grown on metamorphic GaAs<sub>0.67</sub>P<sub>0.33</sub> virtual substrates on GaAs(001), which consistently produce QE over 1% and ESP around 85%.

While the GaAs/GaAsP system is mature, it still faces many limitations. Growth of the relaxed virtual substrate results in a higher density of threading dislocations than there would be in a fully pseudomorphic system, thereby limiting QE, ESP, and photocathode lifetime. In addition, since the As/P ratio in the virtual substrate controls the lattice constants, band offsets, and band gaps, it is difficult--if not impossible--to design a stack that optimizes each of these parameters simultaneously.

Here we demonstrate fully pseudomorphic spin-polarized photocathodes based on the InAlGaAs/AlGaAs system grown by molecular beam epitaxy. By switching to this quaternary well/ternary barrier system, we may more independently optimize the strain, superlattice bandgap, and band offsets. Furthermore, control of Group III alloy composition should be more precise, reproducible, and uniform than control of Group V alloy composition. The fully pseudomorphic stack simplifies growth and reduces the dislocation density in the active region. Unfortunately, the transition from a binary/ternary system to a quaternary/ternary system will increase the random alloy disorder in the photocathode. This random alloy disorder can then cause significant reductions in ESP. To solve this, we investigate using a digital alloy for the well and/or barrier in the superlattice. Our first test structure using a random alloy photocathode design yielded polarization over 80%, but a QE of only about 0.3%. We will report on further optimization of the growth conditions of both random and digital alloy photocathodes. In addition, we will discuss the changes to the QE and ESP of the photocathodes between the two alloy schemes.

2:30pm GD-TuA1-5 Molecular Beam Epitaxy Growth and Characterization of InAs Quantum Wells Grown on Metamorphic III-V Buffer Layers, *I. Levy, Patrick Strohbeen, W. Strickland, M. Hatefipour, J. Issokson, L. Baker,* New York University; *M. Mikalsen,* new york University; *S. Farzaneh, J. Shabani,* New York University

Heterostructures of a 2-dimensional electron gas (2DEG) semiconductor and a superconductor are prime candidates for various applications including quantum computing and topological superconducting circuits [1,2]. The 2DEG layer needs to be in close proximity to the superconductor, within the structure, and to form an Ohmic contact. Heterostructure of containing a quantum well (QW) of InAs (InGaAs/InAs/InGaAs) that have a narrow bandgap with the Fermi level close to the conduction band as a 2DEG semiconductor and thin epitaxial Al layer grown on it as a superconductor. This heterostructure makes a prime candidate due to the high mobility in the 2DEG and the compatibility between the QW and the epitaxial Al.

In this work, we present our study of the growth of these heterostructures (InAs QW/AI) grown by molecular beam epitaxy [3]. We show the effect of modification of the growth conditions on the heterostructures and analyze the disorder in the samples using different techniques. Various impurities including background impurity, surface impurities, alloy scattering and surface roughness can be moderated as a function of the growth conditions. We find samples that have high indium content in their graded buffer layer show a decrease in surface roughness with growth temperature. In addition, anisotropy along rapid diffusion directions during the growth results in anisotropy in the resulting carrier mobilities parallel with these directions. We also analyze the typical cross hatch pattern of the surface on the InAs heterostructures as a function of growth parameters and relate the strain state anisotropy (or lack thereof) to the observed electrical characteristics. Lastly, we show that growth of a thin capping layer of 1 - 6 nm of In<sub>0.81</sub>Al<sub>0.19</sub>As modifies the surface scattering on the quantum well, the implications of which will be discussed.

[1] H. Kroemer, Physica E 20, 196 (2004)

[2] J. A. del Alamo, Nature 479, 317 (2011)

[3] W.M. Strickland et al. Appl. Phys. Lett. 121, 092104 (2022)

2:45pm GD-TuA1-6 Cryogenic Growth of Superconducting Thin Films on GaAs, Si, and Sapphire Substrates, *Teun van Schijndel, A. Engel,* University of California Santa Barbara; *J. Dong,* University of California at Santa Barbara; *A. McFadden,* NIST-Boulder; *C. Palmstrøm,* University of California Santa Barbara

Superconducting thin films are crucial in various fields of quantum information technology, including superconducting qubits and topological quantum computing. The vast majority of either of these qubit technologies use aluminum as the superconducting component. Al is generally grown at low temperatures to achieve smooth thin films. This allows for easy integration of Al-based devices with material systems such as sapphire, Si, Ge, or III-V materials due to minimal interfacial reactions.

While aluminum is the most common, other superconductors show promising results as well. In particular, Ta- and Nb-based superconducting qubits on sapphire show low loss and long coherence times.<sup>1,2</sup> Due to its resilience to high temperatures, low-loss sapphire can withstand the growth of Ta and Nb at elevated temperatures necessary for the realization of desirable superconducting properties. These superconductors could be interesting for topological qubits as well, but integration of these refractory metals on high-purity semiconducting substrates with high spin-orbit coupling remains a challenge due to interfacial compound formation at higher temperatures. In fact, this can lead to additional roughness and consequently degradation of superconducting properties such as T<sub>c</sub>.

In this work, we explore the MBE growth of superconducting thin films at ultralow temperatures below 10K (LT). In particular, we focus on the growth of refractory metals such as Ta, Nb, and V. Through a comparative study of thin film growth at ultralow temperature and room temperature (RT), significant changes in the properties of the superconducting thin films are observed. Initial experiments on SiO2/Si surfaces indicate a notable improvement in film smoothness with lower temperature growth measured by Atomic Force Microscopy. Furthermore, four-probe resistance measurements demonstrate a significant increase in the superconducting Tc for all superconductors studied. Most dramatically, we found that 20 nm Ta grown at RT did not exhibit a superconducting transition above 2K, while LT growth of Ta resulted in a superconducting film with a Tc of 3.95K. We will further investigate this behavior on GaAs(001), Si(001), and Al<sub>2</sub>O<sub>3</sub>(0001) surfaces and extend our study to superconductors such as Pb, Sn, and Al. Our work demonstrates the growth of high-quality superconducting thin films with ultralow temperature growth, which enables the exploration of different substrate and superconductor combinations for use in quantum information technology.

References

1)	Nat	Commun	12,		1779	(2021).
2)	npj	Quantum	Inf	8,	3	(2022).

### Novel Materials Room Ballroom A - Session NM-TuA2

#### **Novel Oxides and Superconductors**

Moderator: Dr. Matthew J. Brahlek, Oak Ridge National Laboratory

### 3:30pm NM-TuA2-9 TaO<sub>2</sub> - The New Kid on the 5*d* Block, *Yorick Birkhölzer*, A. Park, M. Barone, D. Schlom, Cornell University

For the realization of the next generation of fast, energy-efficient nanoelectronics, there is a great need for new materials whose electrical and optical conductivities can be sensitively tuned between high (on) and low (off) states by altering a thermodynamic control parameter, e.g., strain or temperature. Unfortunately, most materials are either metallic or insulating and their conductivities cannot be changed substantially. Materials exhibiting a metal-insulator transition (MIT) at or above room temperature are quite rare, limiting their applicability in devices. The archetypical compound displaying an MIT is VO<sub>2</sub> with a transition temperature of 65 °C. Shortly after the discovery of the MIT in  $3d^1$  VO<sub>2</sub>[1], a similar effect was discovered in  $4d^1$  NbO<sub>2</sub> albeit at a much higher temperature of 807 °C[2]. Thus far, the  $5d^1$  analog TaO<sub>2</sub> has remained elusive.

## Tuesday Afternoon, September 19, 2023

Here, we show the growth of epitaxial thin films of phase-pure, rutile-like  $5d^{1}$  TaO<sub>2</sub> using suboxide MBE. This recently developed flavor of MBE makes use of the TaO<sub>2</sub> molecular vapor emanating from a Ta<sub>2</sub>O<sub>5</sub> charge in an effusion cell heated to temperatures around 1700 °C[3]. This approach avoids the notoriously unstable electron-beam evaporation of Ta metal and need for subsequent oxidation using a background gas or plasma. The latter is particularly challenging to control in the quest for TaO<sub>2</sub> as the stable bulk phase of tantalum oxide is the  $5d^{0}$  compound Ta<sub>2</sub>O<sub>5</sub>, a band insulator without an MIT, similar to the case of  $3d^{0}$  V<sub>2</sub>O<sub>5</sub> and  $4d^{0}$  Nb<sub>2</sub>O<sub>5</sub>. In the suboxide MBE approach, the Ta<sup>4+</sup> is delivered to the substrate from a preoxidized molecular beam of TaO<sub>2</sub>.

Unlike VO<sub>2</sub> that can be formed at back-end-of-line-compatible temperatures below 400 °C[4], we find that exceptionally high substrate temperatures above 1000 °C are needed to crystallize TaO<sub>2</sub> by suboxide MBE. Such high temperatures exceed the range of typical MBE systems but are attainable at the PARADIM Thin Film Facility, an NSF-supported national user facility[5], thanks to a recently installed CO<sub>2</sub> laser-based substrate heater.

Ongoing efforts entail the detailed investigation of the effect of epitaxial strain on the structural and spectroscopic properties of  $TaO_2$  thin films on various substrates. To this end, we are employing an ensemble of X-ray, optical, and electrical transport techniques, searching for signs of a structural and electronic phase transition in this candidate  $5d^1$  MIT compound.

(1959)[1] F.J.. Phvs. Rev. Lett.3. 34 Morin, Janninck, R.; Whitmore, D., J. Phys. Chem. Solids27 (1966) [2] [3] Schwaigert, T. et al, J. Vac. Sci. Technol. A41, 022703 (2023) Paik, H. et al., Appl. Phys. Lett. 107, 163101 (2015) [4] [5] www.PARADIM.org

3:45pm NM-TuA2-10 Growth of KTaO<sub>3</sub>, KNbO<sub>3</sub> and KNb<sub>x</sub>Ta<sub>1-x</sub>O<sub>3</sub> solid solutions by Suboxide Molecular-Beam Epitaxy, *Tobias Schwaigert*, Cornell University; *S. Hazra*, Penn State University; *S. Salmani-Razaie*, Cornell University; *T. Kuznetsova*, Penn State University; *S. Ganschow*, Leibniz-Institut für Kristallzüchtung, Germany; *H. Paik*, Oklahoma State University; *D. Muller*, Cornell University; *R. Engel-Herbet*, Paul Drude Institute, Germany; *V. Gopalan*, Penn State University; *D. Schlom*, Cornell University; *K. Ahadi*, North Carolina State University

Strain-engineering is a powerful means to tune the polar, structural, and electronic instabilities of ferroelectrics. KTaO3 is an incipient ferroelectric, with a very large spin-orbit coupling, in which highly anisotropic superconductivity emerges near a polar instability in electron doped samples[1, 2]. Growth of high-quality epitaxial films provides an opportunity to use epitaxial strain to finely tune electronic and polar instabilities in KTaO<sub>3</sub>. KNbO<sub>3</sub> is a well-known ferroelectric with multiple structural transitions[3]. Using a molecular beam of the suboxides TaO<sub>2</sub> and NbO<sub>2</sub> emanating from effusion cells containing  $Ta_2O_5$  or  $Nb_2O_5$  in combination with a molecular beam of potassium emanating from an indium-potassium intermetallic in an oxidant (~10% O<sub>3</sub> + 90% O<sub>2</sub>) background pressure of 1x10<sup>-6</sup> Torr, KTaO<sub>3</sub>, KNbO<sub>3</sub> and KNb<sub>x</sub>Ta<sub>1-x</sub>O<sub>3</sub> films are grown under conditions of excess potassium in an absorption-controlled regime. Biaxial strains ranging from -0.1 % to -2.1 % are imposed on the commensurately strained KTaO<sub>3</sub> films by growing them upon SmScO<sub>3</sub>, GdScO<sub>3</sub>, TbScO<sub>3</sub>, DyScO<sub>3</sub> and SrTiO<sub>3</sub> substrates, all with the perovskite structure. Reciprocal space mapping shows the epitaxial thin films are coherently strained to the underlying perovskite substrates provided the films are sufficiently thin. Cross-sectional scanning transmission electron microscopy does not show any extended defects and confirms that the films have an atomically abrupt interface with the substrate. X-ray diffraction rocking curves (full width at half maximum < 30 arc sec on all of the above substrates) are the narrowest reported to date for KTaO<sub>3</sub>, KNbO<sub>3</sub> and KNb<sub>x</sub>Ta<sub>1-x</sub>O<sub>3</sub>

films grown by any technique. Laue fringes confirm that the films are smooth with a well-defined thickness. Atomic force microscopy reveals atomic steps at the surface of the grown films. SIMS measurements confirm that the films are free of indium contamination.

#### References

[1] Ueno K et al. Discovery of superconductivity in  $KTaO_3$  by electrostatic carrier doping. Nat. Nanotechnol. 2011; 6:408

[2] Bruno FY et al. Band structure and Spin-orbital Texture of the (111)- KTaO\_3 2D Electron Gas. Adv. Electron. Mater. 2019; 5:1800860

[3] Hewat, A. W. Cubic-tetragonal-orthorhombic-rhombohedral ferroelectric transitions in perovskite potassium niobate: neutron powder

profile refinement of the structures. J. Phys. C: Solid State Phys. 1973; 6.16:2559.

4:00pm NM-TuA2-11 Studying Electronic Structures of Pure SrCoOx Thin Films, Ordered Phases, and Heterostructures, Jibril Ahammad, G. Rimal, Auburn University; J. Sadowski, Brookhaven National Laboratory; G. Sterbinsky, Argonne National Laboratory; M. Boebinger, Oak Ridge National Laboratory; R. Comes, Auburn University

SrCoO<sub>x</sub> (SCO) exhibits contrasting crystalline, electronic, and magnetic states with varying oxygen content. While in brownmillerite (BM) phase, SrCoO<sub>2.5</sub> is anantiferromagnetic insulator with ordered oxygen vacanices, SrCoO<sub>3</sub> is a ferromagnetic metal with the cubic perovskite (P) structure. Although SCO has been a relatively well-studied system, its growth using molecular beam epitaxy (MBE) has remained very limited. P-phase SCO films show high sensitivity to cation stoichiometry and oxygen chemical potential, with secondary phases present depending on growth conditions. In this work, we grew SrCoO<sub>x</sub> using MBE- via both shuttered and codeposition methods- in oxygen plasma, with  $2.5 \le x \le 2.87$  as suggested by X-ray diffraction (XRD). X-ray absorption study (XAS) of Co L and O K edges confirms Co3+ or higher cobalt oxidization states. In-situ reflection highenergy electron diffraction (RHEED) indicates excellent quality and highly epitaxial growth of our films. Stoichiometry ratios between Co and Sr were determined using in-vacuo X-ray photoelectron spectroscopy (XPS), followed by Rutherford backscattering (RBS). We grew our films on three different substrates: LaAlO<sub>3</sub> (LAO), La<sub>0.3</sub>Sr<sub>0.7</sub>Al<sub>0.65</sub>Ta<sub>0.35</sub>O<sub>3</sub> (LSAT), and SrTiO<sub>3</sub> (STO) which have lattice mismatches of - 1.0%, 1.0% and 2.0% with P-SCO respectively. Therefore, our study also provides a scope to explore straininduced oxygen vacancies in the SCO films and their impacts on phase stability. Measurements of these phases were performed using temperature dependent transport property measurements and Scanning transmission electron microscopy (STEM). An evolution of electronic structure of 3d SCO films when coupled with 5d Ir/ Ta based systems in superlattice/double-perovskite structures can provide a further scope to study charge transfer in metastable oxide perovskites.

4:15pm NM-TuA2-12 Synthesizing Metastable Ruddlesden-Popper Titanates by Mbe in Pursuit of Next-Generation Millimeter-Wave Tunable Dielectrics, Matthew Barone, Cornell University; Z. Tian, University of California at Berkeley; M. Papac, National Institute of Standards and Technology, Boulder; B. Goodge, E. Fleck, G. Olsen, Cornell University; K. Lee, Hongik University, Republic of Korea; L. Kourkoutis, Cornell University; N. Orloff, National Institute of Standards and Technology, Boulder; L. Martin, University of California at Berkeley; D. Schlom, Cornell University

While paraelectric (Ba,Sr)TiO<sub>3</sub> films were once used as tunable dielectrics in radio frequency (RF) circuits, dielectric loss above 10 GHz renders (Ba,Sr)TiO<sub>3</sub> incompatible with the high frequency future of RF electronics.<sup>1</sup>The related Ruddlesden-Popper titanates— $(ATiO_3)_{a}AO$  with A =(Ba,Sr)-have demonstrated low loss up to 100 GHz, but these experiments have used interdigitated capacitors compatible with the in-plane dielectric tunability of these phases,<sup>2,3</sup> rather than commercially preferable<sup>1</sup> metalinsulator-metal (MIM) capacitors requiring out-of-plane dielectric tunability. To achieve out-of-plane tunability in a Ruddlesden-Popper film, first-principles calculations indicate the concentration of barium and the series member, n, should both be maximized,<sup>3,4</sup> but synthesizing such films is extremely challenging. Here, we refine existing synthesis techniques to grow a high-*n* Ruddlesden-Popper (n = 20), containing the highest concentration of barium ever accomplished in a Ruddlesden-Popper (A = Ba<sub>0.6</sub>Sr<sub>0.4</sub>) as shown in Fig. 1(b).<sup>5</sup> With a firm grasp on the synthesis, we have demonstrated epitaxially strained heterostructures of (ATiO<sub>3</sub>)<sub>n</sub>AO dielectric layers with metallic SrRuO3 electrodes (Fig. 1(b)). Measurements confirm that such Ruddlesden-Popper films are, in fact, ferroelectric (Fig. 1(c)) and that the dielectric constant is highly tunable at room temperature (Fig. 1(d)). To assess their relevance to the future of tunable dielectrics for GHz electronics, it remains to evaluate the dielectric loss of these new phases at frequencies greater than 10 GHz.

### References

<sup>1</sup> G. Subramanyam, M.W. Cole, N.X. Sun, T.S. Kalkur, N.M. Sbrockey, G.S. Tompa, X. Guo, C. Chen, S.P. Alpay, G.A. Rossetti, K. Dayal, L.Q. Chen, and D.G. Schlom, J. Appl. Phys. **114**, 191301 (2013).

<sup>2</sup> C.-H. Lee, N.D. Orloff, T. Birol, Y. Zhu, V. Goian, E. Rocas, R. Haislmaier, E. Vlahos, J.A. Mundy, L.F. Kourkoutis, Y. Nie, M.D. Biegalski, J. Zhang, M. Bernhagen, N.A. Benedek, Y. Kim, J.D. Brock, R. Uecker, X.X. Xi, V. Gopalan, D. Nuzhnyy, S. Kamba, D.A. Muller, I. Takeuchi, J.C. Booth, C.J. Fennie, and D.G. Schlom, Nature **502**, 532 (2013).

## **Tuesday Afternoon, September 19, 2023**

<sup>3</sup> N.M. Dawley, E.J. Marksz, A.M. Hagerstrom, G.H. Olsen, M.E. Holtz, V. Goian, C. Kadlec, J. Zhang, X. Lu, J.A. Drisko, R. Uecker, S. Ganschow, C.J. Long, J.C. Booth, S. Kamba, C.J. Fennie, D.A. Muller, N.D. Orloff, and D.G. Schlom, Nat. Mater. **19**, 176 (2020).

<sup>4</sup> K. Lee, W. Lee, M. Jeong, Y. Kim, E. Kim, H. Mun, J. Lee, C. Lee, K. Cho, D.G. Schlom, C.J. Fennie, N.M. Dawley, G.H. Olsen, and Z. Wang, U.S. Patent No. 11,133,179 B2 (2021).

 $^5$  M.R. Barone, M. Jeong, N. Parker, J. Sun, D.A. Tenne, K. Lee, and D.G. Schlom, APL Mater.  ${\bf 10},$  91106 (2022).

## 4:30pm NM-TuA2-13 MBE Synthesis and UV Raman Characterization of hexagonal ScFeO<sub>3</sub> films, *Nicholas Parker, D. Schlom,* Cornell University; *D. Tenne,* Boise State University; *M. Barone,* Cornell University

Hexagonal ScFeO<sub>3</sub> films synthesized by molecular beam epitaxy on (0001) oriented Al<sub>2</sub>O<sub>3</sub> were studied by variable temperature ultraviolet Raman spectroscopy. Films were grown in a monolayer controlled method by shuttering of the MBE fluxes as well as by a co-deposition method so that properties of the resulting films could be compared. In addition, thin films were characterized by X-ray diffraction as well as atomic force microscopy. ScFeO<sub>3</sub> in the hexagonal phase is theorized to be multiferroic at near room temperature while being isostructural to YMnO3 and other rare-earth ferrites. This multiferroicity is a combination of ferroelectricity and antiferromagnetic behavior often seen in the family of rare-earth ferrites, but at significantly higher temperatures in hexagonal ScFeO<sub>3</sub>. The Raman spectra of hexagonal ScFeO3 at room temperature are indicative of the polar hexagonal P63cm structure. Furthermore, the x-ray diffraction spectra of the films are correlated with hexagonal ScFeO<sub>3</sub>. Atomic force microscopy provided detailed images of the surface layers of the films allowing for roughness estimates in the range of 1.5-10 nm depending on thickness and growth procedure. The temperature evolution over the range of 10-1450 K of the Raman spectra for hexagonal ScFeO<sub>3</sub> to a non-polar phase, and fitting the Raman intensities as a function of temperature provides a transition temperature of 950 K ± 50 K.

### 4:45pm NM-TuA2-14 Growth and Characterization of Complex Nickelates with High Sr Concentration, B. Paudel, L. Wang, Z. Yang, M. Bowden, Pacific Northwest National Laboratory; J. Liu, Shanghai Institute of Microsystem and Information technology, China; K. Koirala, T. Kaspar, P. Sushko, S. Chambers, Yingge Du, Pacific Northwest National Laboratory

Complex nickelates (*R*/A)NiO<sub>3</sub> (where *R* denotes lanthanide and *A* denotes alkaline earth metals) are of great interests owing to their diverse structures and functionalities. Dynamic tunability of Ni valence states in nickelate thin films (from 1+ to 4+) has offered emergent properties such as superconductivity, enhanced electrocatalytic activity, quantum confinement effect, and metal insulator transitions. Despite many efforts, stabilization of the quadrivalent (Ni<sup>4+</sup>)state through Sr doping in rare earth nickelates has been proven difficult. In this talk, I will present our effort in growing perovskite La<sub>1-x</sub>Sr<sub>x</sub>NiO<sub>3</sub> in thin film and superlattice forms by oxygen-plasma assisted molecular beam epitaxy. We show that it is difficult to achieve high-quality single-phase epitaxy at higher Sr concentration (x>0.5) as bulk SrNiO<sub>3</sub> has a hexagonal phase. Phase segregation (SrNiO<sub>x</sub> to SrNi<sub>2</sub>O<sub>3</sub> + Sr<sub>2</sub>NiO<sub>3</sub>) and phase transition (perovskite to hexagonal) are revealed for the end member at different growth and strain conditions by XPS, XRD, and STEM.

## 5:00pm NM-TuA2-15 Sharpening the Superconducting Transition of Bi<sub>2</sub>Sr<sub>2</sub>Ca<sub>n-1</sub>Cu<sub>n</sub>O<sub>2n+4</sub> Films with n=1-3 Grown by MBE, Y. Eren Suyolcu, Y. Li, D. Schlom, Cornell University

In bulk, the Bi<sub>2</sub>Sr<sub>2</sub>Ca<sub>n-1</sub>Cu<sub>n</sub>O<sub>2n+4</sub> (i.e., Bi-cuprate) family provides the highest superconducting transition temperature—up to 105 K [1]—without highly toxic constituents like thallium or mercury. When prepared as epitaxial films, however, the superconducting transitions of this family are broad and depressed for as-grown epitaxial films, with the highest zero-resistance  $T_c$  reported to date of 97 K [2]. In this work, we use ozone-assisted MBE to grow Bi<sub>2</sub>Sr<sub>2</sub>Ca<sub>n-1</sub>Cu<sub>n</sub>O<sub>2n+4</sub> (*n*=1–3) single-layers and bi-layers on (100) SrTiO<sub>3</sub> substrates, where weuse a non-superconducting Bi<sub>2</sub>Sr<sub>2</sub>CuO<sub>6</sub> (Bi-2201, *n*=1) buffer layer to reduce disorder caused by the underlying substrate. We demonstrate that combining adsorption-controlled *co-shuttered* growth and the initial precise flux calibration provides the best structural and superconducting properties for Bi-cuprate films formed at growth rates of 0.1 µm/hr or higher.

*In-situ* reflection high-energy electron diffraction exhibits incommensurate structural modulation and also proves the absence of any secondary phases. The structural quality of the samples is confirmed using a combination of four-circle x-ray diffraction, atomic force microscopy, and

transmission electron microscopy. The superconducting properties are studied by resistivity vs temperature measurements. Bi-2201/Bi-2212 bilayers exhibit the smoothest reported surfaces with subnanometer root-mean-square (rms) roughness of ~0.4 nm and the sharpest superconducting transition width ( $\Delta T_c$ ) ~10 K, similar to Bi-2212 single crystals. While the  $T_{c(R=0)}$  for the as-grown bilayer is low, i.e., ~50 K, this is a matter of oxygen content and can be modified via a post-growth process. This presentation focuses on achieving the high-structural and surficial quality and the sharp  $\Delta T_c$  of the as-grown samples. We conclude that combining the *co-shuttered* growth and the initial precise flux calibration provides the best structural and superconducting properties [3].

References

[1]	H. M	laeda	et	al.,	Jpn.	J.	Appl.	Phys.	27	(1988),	L209.
[2]	К.	Endo	)	et	al.,		Nature.	35	5	(1992),	327.
[3]	Y.	E.		Suy	olcu	e	et	al.,	in	prepa	ration.

5:15pm NM-TuA2-16 High-Temperature Superconductor FeSe Films Enabled Through Temperature and Flux Ratio Control, Maria Hilse, H. Yi, C. Chang, N. Samarth, The Pennsylvania State University; R. Engel-Herbert, Paul-Drude-Institut für Festkörperelektronik, Germany

FeSe, a bulk superconductor with a T<sub>c</sub> of 9K has attracted a high level of attention since a skyrocketing boost in Tc was reported for a single unit cell (UC) layer of FeSe grown on SrTiO<sub>3</sub>(001) by molecular beam epitaxy (MBE) to as high as 100K. FeSe-SrTiO<sub>3</sub> heterostructures have since been fabricated by many groups but the record T<sub>c</sub> proved difficult to reproduce and thus the mechanism behind it remains concealed. After extensive work in the past, the field appears to agree on certain key "ingredients" in the sample preparation that are believed essential for the boost in T<sub>c</sub>. Those are; 1. an ultra-clean substrate surface of a double TiO2 termination realized by a chemical and thermal *ex-situ* and/or thermal *in-situ* substrate preparation; 2. ultra-thin - one UC thickness - limit of FeSe; 3. a high number of Se vacancies in the FeSe film ensured through post-growth annealing steps in ultra-high vacuum (UHV) for several hours; 4. followed by a capping layer growth protecting FeSe against oxidation during ex-situ characterization. We present our findings on FeSe thin film growth by MBE and present a roadmap for high-T<sub>c</sub> - 222% higher than the reported bulk value in ex-situ transport measurements - circumventing above mentioned steps 1, 2, and 3 by simple in-situ Se/Fe flux ratio and temperature control during FeSe growth. FeSe films of 20-UC-thickness grown at varying temperatures and Se/Fe flux ratios and the structural and morphological properties of the obtained uncapped FeSe films were analyzed. The morphology of the films showed a sensitive dependence on the growth temperature and flux ratio spanning from perfectly smooth and continuous films with atomic terraces at 450 °C growth temperature and a low flux ratio of 2.5 to exclusively disconnected island growth of large height but smooth top surfaces at lower temperatures and/or higher flux ratios. Surprisingly, the tetragonal P4/nmm crystal structure of FeSe was maintained for all investigated films and the *in-situ* observed diffraction pattern in reflection high energy diffraction also maintained the streaky pattern characteristic for smooth FeSe films even for the samples with the most pronounced island growth resulting in a root mean square atomic force microscopy roughness of more than 18nm. Smaller flux ratios than 2.5 resulted in mixed - FeSe/elemental Fe - phase samples. FeSe films grown under optimized conditions at 450°C and a flux ratio of 2.5 (but without any post-growth UHV anneal) and capped with the commonly used FeTe (300°C) and elemental Te (room temperature) layers yielded superconducting onset temperatures of about 30K and a Tc of 20K.

Science and Technology of MBE

### **Room Ballroom A - Session ST-WeM1**

### Advancement in MBE Growth Approaches

Moderator: Dr. Darrell Schlom, Cornell University

8:00am ST-WeM1-1 Thermal Laser Epitaxy - the Universal Epitaxy Tool(?), Wolfgang Braun, D. Kim, F. Hensling, T. Smart, L. Majer, B. Faeth, S. Smink, D. Dereh, Max Planck Institute for Solid State Research, Germany; H. Boschker, Epiray GmbH, Germany; J. Mannhart, Max Planck Institut for Solid State Research, Germany

Thermal laser epitaxy (TLE) uses quasi-CW lasers to heat the substrate and the sources in a deposition geometry similar to MBE. The high energy densities of the laser beams enable the evaporation or sublimation of any element in the periodic table in any combination and substrate temperatures exceeding the melting point of sapphire at 2040 °C. This leads to a dramatic expansion of the parameter space available to MBE-type growth modes such as, e.g., the adsorption-controlled growth of oxides.

In addition, TLE offers new opportunities in terms of the available gas pressures and environments during growth. A TLE growth chamber is conceptually simple, besides the laser entrance windows it basically contains only mechanical mounts for substrates and sources. This means that background gases at any pressure and with any reactivity are possible, as long as these parts do not get damaged, and the laser beam is not absorbed in the gas phase. This is the case for most reactive gases such as oxygen, ozone, plasma-excited nitrogen or ammonia.

As an example, we study the evaporation of Al in a molecular oxygen atmosphere. At high laser power densities, the flux of the source is independent of the oxygen pressure up to  $10^{-1}$  hPa. At lower power densities, the source flux depends on the oxygen pressure and the area of exposed metallic surface on the source. This is consistent with volatile suboxide molecules proportional to the oxygen flux being generated on the exposed surface. For the current maximum laser power, the source finally gets passivated by a stable oxide at pressures above  $10^{-1}$  hPa, and no longer produces a stable flux. The deposition rate on the substrate starts dropping at  $10^{-2}$  hPa for the given working distance of 80 mm, as the mean free path drops below this distance and the transport transitions from the ballistic to the diffusive regime. Irrespective of the reactive flux generation and deposition, the more than five orders of magnitude dynamical range of the deposition flux and high maximum growth rates are maintained.

This implies that TLE is able to cover the entire range of growth conditions traditionally employed in MBE, GS-MBE, MOMBE, MOCVD, CBE and CVD. TLE thereby enables the evaporation or sublimation of practically any combination of elements, together with reactive gas sources, at low and also very high growth rates. CVD growth modes, e.g. for thick buffer layers, may be combined with kinetically limited growth, e.g. for active layers at low temperatures, in a single growth to produce devices with very dissimilar layers and growth conditions in a single growth run.

Results of TLE with nitrogen and ammonia will be presented and discussed.

## 8:15am ST-WeM1-2 Routes Towards Making BaZrSe<sub>3</sub> Thin Films in the Perovskite Structure by MBE, *Ida Sadeghi, K. Ye, J. Van Sambeek, R. Jaramillo,* MIT

Chalcogenide (sulfide and selenide) perovskite semiconductors are anticipated to have favorable structural, optical and electronic characteristics for solar energy conversion. The most studied compound is BaZrS<sub>3</sub>, with a band gap of 1.9 eV. Alloying on the anion or cation sites has been explored to lower the band gap into a range suitable for single-junction solar cells. The pure selenide perovskite BaZrSe<sub>3</sub> is predicted to have band gap 0.5 eV lower than the sulfide.<sup>1</sup> However, BaZrSe<sub>3</sub> may form in different polymorphs, theory predicts that the needle-like (non-perovskite) phase with band gap below 1 eV is the most stable, and solid-state synthesis attempts have resulted in a semi-metallic hexagonal ordered defect phase.<sup>1,2,3,4</sup>

We previously reported the first epitaxial synthesis of chalcogenide perovskite thin films by gas-source MBE: BaZrS<sub>3</sub> film on LaAlO<sub>3</sub> substrate.<sup>5</sup> The films grow on a self-assembled interface layer that relieves the epitaxial strain. Here we demonstrate alloying BaZrS<sub>3</sub> with Se, including the first report of a pure selenide perovskite, and we confirm the tunability of the direct band gap of this alloy system.<sup>6</sup> We demonstrate two processing routes to make BaZrS<sub>(3-y)</sub>Se<sub>y</sub> alloys in the perovskite structure. (i) We deposit BaZrS<sub>(3-y)</sub>Se<sub>y</sub> on a BaZrS<sub>3</sub> template layer on LaAlO<sub>3</sub>. The template stabilizes

the perovskite structure for BaZrS<sub>(3-y)</sub>Se<sub>y</sub>, which otherwise does not grow as a perovskite directly on LaAlO<sub>3</sub>. Scanning transmission electron microscopy analysis shows that there is anion intermixing between the sulfide template and the alloy during growth, such that the final film has nearly uniform composition. (ii) We deposit BaZrS<sub>3</sub> on LaAlO<sub>3</sub>, followed by post-growth selenization. The RHEED pattern during selenization remains unchanged, indicating that the crystal structure is static even as the sulfur and selenium anions exchange. Both processing routes can produce selenide perovskite films with band gap of 1.4 eV.

This work sets the stage for developing chalcogenide perovskites as a family of semiconductor alloys with properties that can be tuned with composition in high-quality epitaxial thin films, as has been long-established for other semiconductor materials. The band gap of high-selenium-content BaZrS<sub>(3-y)</sub>Se<sub>y</sub> suggests applications in thin-film solar cells.

- 1. Sun, Y.-Y., et al. Nano Lett. 15, 581 (2015).
- 2. Tranchitella, L. J. et al. J. Am. Chem. Soc. 120, 7639 (1998).
- 3. Jess, A. et al. Chem. Mater.34, 6894 (2022).
- 4. Aslanov, L.A. Russ. J. Inorg. Chem.9, 1090 (1964).
- 5. Sadeghi, I. et al. Adv. Funct. Mater. 31, 2105563 (2021).
- 6. Sadeghi, I. et al. arXiv:2211.10787.

8:30am ST-WeM1-3 Adsorption Controlled Homoepitaxial Growth of *c*-Plane Sapphire by Thermal Laser Epitaxy, *Felix Hensling, L. Majer, S. Smink, J. Mannhart, W. Braun,* Max Planck Institute for Solid State Research, Germany

The interest in sapphire as an electronic materialis due to itslow cost, superior properties over silicon, high quality wafer availability, and the possible integration with silicon.Indeed, its rapidly growing market share suggests sapphire to be the substrate materialof the future.[1] A cornerstone for further establishing sapphire in a wide range of (electronic) applications is the ability to grow high quality homoepitaxial sapphire films. Key applications that can benefit from this are, *e.g.* diodes and high-power electronics based on nitrides [1] and (ultra) wide band gap semiconductors, respectively [2,3]. However, epitaxial films of c-plane oriented sapphire, one of the most common cuts of sapphire, have so far been out of reach due to a preferred formation of the  $Al_2O_3\gamma$ -phase.[3]

I will present how thermal laser epitaxy (TLE) overcomes this issue thanks to the availability of a parameter space that far exceeds the reach of other deposition methods.[4] The substrate laser heating system allows to easily heat sapphire substrates even beyond their melting point. The high accessible temperatures enable a precise and smooth sapphire substrate preparation – the first step for successful homoepitaxy.[5] I further present how the crystal quality and surface smoothness of homoepitaxial sapphire increase with increasing substrate temperature in case the growth is performed in the adsorption-controlled mode. Even at the respective high temperatures growth rates far exceeding 1  $\mu$ m/h can be realized. Films were investigated by scanning transmission electron microscopy, atomic force microscopy, and x-ray diffraction. At a growth temperature of 1600 °C, the films were found to be practically undistinguishable from the underlying substrate.

#### References

[1] M.S. Akselrod, F.J. Bruni, "Modern trends in crystal growth and new applications of sapphire," J. Cryst. Growth, 360, pp. 134-145, 2012.

[2] H. Okumura, "Sn and Si doping of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> (10-10) layers grown by plasma-assisted molecular beam epitaxy", JJAP, 61, 125505, 2022.

[3] R. Jinno *et al.*, "Crystal orientation dictated epitaxy of ultrawide-bandgap 5.4- to 8.6-eV  $\alpha$ -(AlGa)<sub>2</sub>O<sub>3</sub> on m-plane sapphire", *Sci. Adv.*, 7, eabd5891, 2021

[4] W. Braun and J. Mannhart, "Film deposition by thermal laser evaporation," *AIP Advances*, 9, 085310, 2019.

[5] W. Braun *et al.*, "*In situ* thermal preparation of oxide surfaces", *APL Mater.*, 8, 071112, 2020.

8:45am ST-WeM1-4 Prediction of MBE-grown Oxide Film Composition Using Neural Networks and Big Data Analytics, Patrick Gemperline, R. Paudel, S. Thapa, S. Provence, S. Battles, R. Markland, Auburn University; R. Vasudevan, Oak Ridge National Laboratory; R. Comes, Auburn University

Reflected high energy electron diffraction (RHEED) is a highly common form of real time analysis used in growth systems such as molecular beam

epitaxy (MBE) and pulsed laser deposition (PLD). Traditional RHEED analysis focuses only on the intensity and shape of the diffraction pattern for a few still images taken during growth or on the intensity of a single diffraction peak in real time. While this information can be quite insightful, there is far more information that can be gleaned from RHEED, which is often qualitative and learned through many repeated trials by the film grower. In order to obtain greater insight from RHEED videos, component analysis (PCA) and k-means clustering were applied to the recordings of the RHEED taken during the MBE growth of epitaxial thin film perovskite oxides, including SrTiO<sub>3</sub>, LaFeO<sub>3</sub> and SrHfO<sub>3</sub>. To further enhance the utilization of RHEED, a generative adversarial neural network was trained to predict X-ray photoelectron spectroscopy data from RHEED images, thus allowing for real-time prediction of film stoichiometry. These methods yield more quantitative results from the RHEED with minimal time requirements and open the door for future development of real-time computer control of film growth for optimal growth conditions.

### 9:00am ST-WeM1-5 Autonomous Synthesis in the MBE Using Real-Time Artificial Intelligence, *Tiffany Kaspar*, L. Wang, J. Christudasjustus, M. Sassi, B. Helfrecht, J. Pope, A. Harilal, S. Akers, S. Spurgeon, Pacific Northwest National Laboratory

Materials are the key components of nearly all advanced technologies, including quantum information systems, microelectronics, catalysis, and energy conversion and storage. Modern synthesis methods enable the fabrication of an ever-expanding array of novel, non-equilibrium, and/or metastable materials and composites that may possess unique and desirable functionality. Thin film deposition by molecular beam epitaxy (MBE) can produce atomically precise (or nearly so) materials with a wide range of functional electronic, magnetic, ferroelectric/multiferroic, optical, and/or ion-conducting properties. The current state of the art in precision design of functional materials is to manually explore the "growth phase space" of the deposition techniqueto optimize the film properties of interest. Limitations of time and resources often result in incomplete exploration of the growth phase space and resulting properties. Faced with this lack of complete information, materials design and synthesis decisions are made based in part on intuition and luck, slowing both materials optimization and materials discovery. This current synthesis paradigm can be disrupted by employing artificial intelligence (AI)-accelerated analysis of in situ and ex situ data streams that will enable targeted synthesis of novel materials with desired structure, chemical stability, and functional properties. Here we present a preliminary implementation of such an AIcontrolled MBE.We are integrating the control of key synthesis parameters (temperatures, gas flow rates, shutters) with Al-guided computer control.Guidance will be based on near-real-time analysis of reflection high energy electron diffraction (RHEED) patterns using sparse data analytics, with low-latency feedback to the control software.As an initial demonstration, we will control the morphology and phase purity of epitaxial anatase TiO<sub>2</sub> thin films.

## 9:15am ST-WeM1-6 The Role of Optical Excitation on Misfit Dislocations in Epitaxial ZnS on GaP, *Alexandra Fonseca Montenegro*, *M. Baan*, *A. Blackston*, *R. Myers*, *T. Grassman*, The Ohio State University

Light decreases dislocation mobility in bulk ZnS, as was recently shown through photo-indentation measurements. Here we investigate the impact of above band gap optical excitation on ZnS epilayers grown by molecular beam epitaxy (MBE) on (001) GaP. For example, optical excitation during MBE could potentially increase the critical thickness for misfit dislocation (MD) nucleation at the ZnS/GaP heterovalent interface. GaP (001) homoepitaxial buffer layers are prepared on GaP wafers, As-capped, and transferred to the chalcogenide system where ZnS is grown by compound source MBE at 150C on the As-desorbed GaP surface. Using electron channeling contrast imaging (ECCI), the MD ensembles are measured at various film thicknesses to determine the onset of MD formation. Although HRXRD shows negligible strain-relaxation in films up to 50nm, ECCI reveals that the MD nucleation processes begins at ZnS thicknesses of between 15 and 20nm, far lower than previously reported critical thicknesses. Additionally, high densities of dipole-like features appear at large densities, which appear to be dislocation loops nucleated at the surface. Image analysis is used to quantify the MD density, length, as well as the density of dipole features as a function of film thickness. We will discuss changes in the MD content with and without above band gap excitation. Additionally, post-growth strain biasing is used to increase the MD content taking advantage of the thermal mismatch via temperature cycling. Photoluminescence spectra are acquired revealing the emergence of a subband gap peak at 3.1 eV, which increases in intensity within ZnS epilayers upon repeated temperature cycles. We will discuss changes in the MD

content and PL spectra for samples thermally cycled with and without optical excitation.

### Science and Technology of MBE Room Ballroom A - Session ST-WeM2

### **Fundamentals of MBE Growth**

Moderator: Dr. Stephanie Law, University of Delaware

# 10:00am ST-WeM2-9 Doping and Surfactant Behavior of Antimony and Gallium in Molecular Beam Epitaxy Grown Germanium-Tin, Amanda Lemire, K. Grossklaus, C. Jamison, V. Vazquez, P. Hennessey, T. Vandervelde, Tufts University

The bandgap of germanium-tin (Ge1-xSnx) alloys can vary from ~0.8 eV into the far-IR as the Sn composition increases, and transition from indirect to direct bandgaps between 6 and 25 at% Sn depending on strain. Consequently they are being developed for a range of thermophotovoltaic (TPV), thermal imaging, and photonic data transmission applications. However, at useful Sn contents these alloys are metastable and require low epitaxial growth temperatures. Therefore they can only be grown to limited thickness, otherwise films may experience Sn segregation and defect formation. Growing below ideal temperatures for Ge can also induce defects, reducing electrical conductivity. Growth and thickness limitations complicate PV cell design and optimization. One mitigation strategy for non-optimal growth conditions is to deposit and maintain on the growth surface a thin coverage of an additional atomic species, which acts as a surfactant to modify adatom interactions. Surfactants change the surface energy of the growth front, which controls the mobility of adatoms and affects the rate of positional exchange between buried and surface layers. A surfactant that preferentially moves to the growth surface could increase the incorporation of Sn into deeper layers and reduce defects formed due to growth at low temperatures. Additionally, the filling of partial growth layers by surfactant atoms may reduce islanding and so improve surface roughness.

In this work, antimony was applied as an *n*-type dopant and surfactant during the growth of GeSn alloys to potentially improve the crystallinity of Ge and GeSn films grown by molecular beam epitaxy. Previous work has shown Sb suppresses roughness in SiGe film growth. Gallium was applied as a p-type dopant and surfactant, providing a comparison between dopant types and differing surfactant atomic sizes. Pairs of GeSn samples were deposited without and with the codeposition of Sb or Ga. The growth temperature, film thickness, and Sn content were varied to investigate the impact of each surfactant on Sn incorporation and critical thickness. Amount of Sb or Ga incorporation into the growing film depends on growth conditions, so this study also provides data on the retention of Sb and Ga as dopants during low-temperature GeSn growth. Sn content and film quality were characterized by X-ray diffraction, photoluminescence and Raman spectroscopy, spectroscopic ellipsometry, and optical microscopy. Electrical properties were examined by Hall effect measurement. Results will be discussed in terms of how surfactants have modified the kinetics of the growth process and considered with design of a GeSn based photodiodes and TPV cells in mind.

### 10:15am ST-WeM2-10 SiO<sub>2</sub> Surface Planarization for Selective Area Regrowth of High Aspect Ratio Microstructures, Ashlee Garcia, A. Skipper, M. Bergthold, S. Bank, University of Texas at Austin

A molecular beam epitaxy (MBE) approach to selective area epitaxy (SAE) of III-V semiconductors has the potential to advance optoelectronic structures through seamless integration of metals, dielectrics, and high-quality crystalline semiconductors. While SAE by metal organic chemical vapor deposition has been widely successful due to its high material deposition selectivity, an all-MBE method could enable further advances through its high layer precision and access to non-equilibrium growth conditions<sup>1,2</sup>.

SAE is difficult to achieve with conventional MBE due to III-V nucleation on the amorphous mask necessitating high growth temperatures and low growth rates to mitigate polycrystal formation<sup>2-4</sup>. While an all-MBE approach has enabled the embedding of features up to 300nm tall, applications requiring high aspect ratio microstructures such as mid-/longwave infrared high-contrast photonics<sup>5-7</sup> and aspect ratio trapping of dislocations for metamorphic growth<sup>8</sup>, are more challenging to access due to increased surface roughness of micron-scale features<sup>7</sup>. Increased roughness and defects on the mask surface lowers the barrier for nucleation further restricting the selective growth regime<sup>8</sup>.

In this study, we explore the use of hydrogen silsesquioxane (HSQ) surface planarization to restore the surfaces of micron-scale films for fabrication of high aspect ratio SiO<sub>2</sub> features with surfaces comparable to that of thin films<sup>9,10</sup>. Integrating surface planarization with a 100nm layer of HSQ in the fabrication of 1.5µm tall features demonstrated a significant decrease in film roughness from a root-mean-square roughness of 3.95nm to 0.75nm. Furthermore, selectivity studies on cured HSQ films have shown selectivity equivalent to that of SiO<sub>2</sub> deposited by plasma-enhanced chemical vapor deposition. Experiments are underway to utilize the technique to achieve selective growth of high aspect ratio microstructures.

[1] D.J. Ironside et al., J. Cryst. Growth (2019). [2] A.M. Skipper et al., 2019 MRS EMC. [3] F.E. Allegretti et al., J. Cryst. Growth (1995). [4] S.C. Lee et al. J. of Appl. Phys. (2002). [5] Jun Wang et al. 2017 Laser Phys. Lett. 14 125801. [6] J.Z. Li et al. Appl. Phys. Lett. 91 (2) (2007). [7] M. R. Amirzada et al. Appl Nanosci 6, 215–222 (2016). [8] M. Ohring, The Material Science of Thin Films, Academic Press (1992). [9] F. Salmassi et al, Applied Optics, Vol. 45, No. 11 (2006). [10] C.-C. Yang and W.-C. Chen, J. Mater. Chem., 2002,12, 1138-1141.

This research was performed at the Texas Nanofabrication Facility (NSF NNCI-1542159) and was supported by Lockheed Martin and NSF via the UT CDCM MRSEC (DMR-1720595, CCF-1838435, DMR-1839175 and ECCS-1926187).

10:30am ST-WeM2-11 InP Lateral Epitaxial Overgrowth by Solid-Source Molecular Beam Epitaxy, *Yiteng Wang*, *R. Hool, W. North, S. Pandey, E. Raftery, K. Choquette, M. Lee,* University of Illinois at Urbana-Champaign

Lateral epitaxial overgrowth (LEO) of InP on buried dielectrics (typically SiO2 and SiN) has been investigated for optical telecommunication devices [1],[2]. LEO on dielectric patterns usually requires coalescence from multiple crystal fronts, which can introduce structural defects including threading dislocations (TDs) and stacking faults (SFs) [3]. In this work, we demonstrate selective InP nucleation and coalescence on patterned InP substrates with designed feature sizes of 100-1110 nm by solid-source molecular beam epitaxy. Structural defects formed due to lateral growth and coalescence for LEO on all features, including triangles, polygons, and stripes. We found that LEO on features < 300 nm with sides aligned away from [0  $\cdot 1$  1] and [0 1 1] directions gave the lowest defect density.

The dielectric patterns investigated here included arrays of 40 nm thick SiN isosceles-right triangles with side lengths of 465-1110 nm. The sides of the triangles were aligned along [0 -1 1] and [0 1 1], and the hypotenuses were aligned along [0 1 0] crystallographic directions. 20 and 350 nm layers of Be-doped InP were grown at a substrate temperature (Tsub) of 500°C and a growth rate of 0.2  $\mu$ m/hr to maintain selectivity [4]. Thicker samples of 1300 nm p-InP were initiated using the same conditions for the first 500 nm, followed by a second growth step at more typical InP MBE growth conditions of 440°C, 1  $\mu$ m/hr.

InP growths of 20-350 nm at  $T_{sub} = 500^{\circ}$ C show selective growth was maintained, as no polycrystalline nucleation was found on dielectrics by secondary electron imaging (SEI). Next, after 1300 nm growth, the InP surface was fully coalesced for all feature sizes, with small voids enclosed above the dielectrics. Electron channeling contrast imaging (ECCI) reveals that LEO on 465 and 1110 nm triangles possess high threading dislocation densities TDD > 10<sup>7</sup> cm<sup>-2</sup>. Further study of InP LEO on SiO2 lines (120 nm thick/230 nm wide) oriented along the three low-index in-plane directions shows high defect density along [0 1 1] and [0 -1 1] and lower defect density along [0 1 0].

To reduce TDD, we performed overgrowth on an array of 40 nm thick SiN polygons with side lengths of 266 nm and most of the sides oriented away from [0 1 1] and [0 -1 1] directions. Taking advantage of the side orientation and reduced feature sizes < 300 nm, InP LEO on polygon patterns shows an order of magnitude reduction in TDD ( $5.0 \times 10^6$  cm<sup>-2</sup>) and fewer SFs compared to the 465 nm and 1110 nm triangles. Future work will focus on better understanding the effect of dielectric thickness, size, and orientation on the formation of extended defects during InP LEO.

10:45am ST-WeM2-12 Effect of Molecular Beam Epitaxy (Mbe) Growth Conditions on the Structural and Magnetic Properties of High Curie Temperature (MnSb<sub>2</sub>Te<sub>4</sub>)<sub>x</sub>(Sb<sub>2</sub>Te<sub>3</sub>)<sub>1-x</sub> Magnetic Topological Insulators, *Candice Forrester*, The City college of New York, Lehman College; *C. Testelin*, Institute des NanoSciences de Paris, France; *K. Wickramasinghe*, The City College of New York; *I. Levy*, New York University; *X. Ding*, *L. Krusin-Elbaum*, The City College of New York; *G. Lopez*, Lehman College; *M. Tamargo*, The City College of New York

Recently it has been shown that structural disorder in 3D Topological Insulators (TIs) has considerable effects on the properties of the materials. The addition of magnetic ions like Mn breaks time reversal symmetry and opens a gap in the Dirac point.<sup>1</sup> This addition also changes the crystal structure from the typical quintuple layer (QLs) structure of non-magnetic TIs to a septuple layer (SLs) structure.<sup>2</sup> Furthermore, addition of a Mn flux during MBE growth results in self-assembled structures of mixed QLs and SLs.<sup>2</sup>

Previously we reported the MBE growth of self-assembled structures of  $(MnSb_2Te_4)_x(Sb_2Te_3)_{1\times}$  magnetic topological materials, and showed that their Curie temperature  $(T_c)$  is dependent on the composition x (or %SL). Samples with 0.7 < x < 0.85 exhibit very high T<sub>c</sub> values. Additionally, it was observed that as the Mn beam equivalent pressure (BEP) ratio used was increased, there was a corresponding increase in T<sub>c</sub>.Decreasing the growth rate further increased T<sub>c</sub> to >100 K, the highest values reported for this material system. An understanding of how the changes in growth conditions lead to the T<sub>c</sub> enhancement is not well-established.

Here we investigate the structural properties of the materials as they relate to the growth conditions, specifically growth rate (GR). Samples grown at slow GRs (0.4 – 0.6 nm/min) were compared to samples grown at fast GRs (0.8 – 1.0 nm/min). Samples were investigated by X-ray diffraction, Energy Dispersion X-Ray Spectroscopy (EDS), Hall effect and scanning transmission electron microscopy (STEM).

We found that for the same Mn BEP ratio, low GR yields similar composition x as fast GRs. On the other hand, EDS showed that for x > 0.7 there was increased intermixing between Sb and Mn in both the fast and the slow GR samples. However, the samples grown with slow GR showed much greater Mn and Sb intermixing, as well as Mn and Te intermixing, suggesting increased Mn incorporation at the slow GR. Cross sectional EDS studies reveal a high Mn content in the QLs, consistent with  $(Sb_{1-y}Mn_y)_2Te_3$  alloy formation. Hall effect measurements show that GR does not significantly affect the electrical doping in  $(MnSb_2Te_4)_x(Sb_2Te_3)_{1-x}$  supporting the proposal that a super-exchange magnetic mechanism is likely at play. Other techniques, such as magnetic force microscopy (MFM) are being explored to better understand the magnetic mechanisms leading to high T<sub>c</sub> values. Our results provide insight as to how to achieve on-demand magnetic TIs with enhanced properties.

<sup>1</sup>Y. Tokura et al, *Nature Reviews Physics***1**, 126 (2019)

<sup>2</sup> J.A. Hagmann et al, New Journal of Physics 19, 085002 (2017)

<sup>3</sup>I. Levy et al, Crystal Growth & Design **22**, 3007 (2022)

11:00am ST-WeM2-13 Evolution of Antiferromagnetic Spin Texture in MBE-Grown Epitaxial Multiferroic BiFeO3, Maya Ramesh, 105 Worth Street; P. Meisenheimer, University of California, Berkeley; S. Zhou, Brown University; P. Stevenson, Northeastern University; L. Caretta, Brown University; R. Ramesh, Rice University; D. Schlom, Cornell University Bismuth ferrite (BiFeO3) is a lead-free magnetoelectric multiferroic showing antiferromagnetic order and a large spontaneous polarization at room temperature. This antiferromagnetic order in BiFeO3 is complex, where, as a consequence of the Dzyaloshinskii-Moriya Interaction (DMI), a small canting of the antiferromagnetic order forms a chiral spin cycloid in bulk samples. Understanding the interplay between the ferroelectric polarization and the spin cycloid, as well as its electric field manipulation, is of significant interest for antiferromagnetic spintronics and next generation computation. There is still much to learn about BiFeO3's intrinsic antiferromagnetic structure in thin films, where epitaxial strain imposed by an underlying substrate can perturb and even destroy the spin cycloid. MBE provides an ideal pathway to create BiFeO3 with unparalleled structural perfection. As a model system, we have synthesized epitaxial thin films (2-100 nm) of BiFeO3 on (110) TbScO3 substrates via oxide molecular-beam epitaxy. In this work, we explore how epitaxial strain affects the formation and orientation of the spin texture in BiFeO3 using nitrogen-vacancy magnetometry (NV magnetometry). NV magnetometry uses a nitrogen vacancy implanted at the tip of a diamond cantilever which acts as a singlespin magnetometer to sensitively map nanoscale surface stray fields. By correlating the spin texture to ferroelectric information measured using

piezoresponse force microscopy, we are able to explore coupling between the two order parameters. Electric-field-dependent measurements show that the spin cycloid can be manipulated by switching the ferroelectric polarization, which is of great interest in low dissipation magnonics and spintronics.

### 11:15am ST-WeM2-14 Adsorption-Controlled Growth of SrTiO<sub>3</sub> by Oxide MBE, Dylan Sotir, M. Barone, D. Schlom, Cornell University

Historically, the growth of stoichiometric SrTiO<sub>3</sub> by conventional molecularbeam epitaxy (MBE), utilizing elemental sources, has been challenging due to the precise calibration required. It would be useful if there were a thermodynamic regime where the composition of SrTiO<sub>3</sub> was selfregulating. One demonstrated way to achieve such automatic control of stoichiometry in SrTiO<sub>3</sub> is by supplying an excess of the volatile organometallic precursor titanium isopropoxide to grow SrTiO<sub>3</sub> by metalorganic MBE (MOMBE). Another means, and the one investigated here, is to stick with elemental sources but increase the substrate temperature. When SrTiO<sub>3</sub> is heated to high temperature, it does not evaporate congruently; rather, it loses more strontium than titanium. This difference in the vapor pressures of the volatile species suggests that adsorption-controlled growth of SrTiO<sub>3</sub> may be possible by conventional MBE. We report a thermodynamic window in which SrTiO3 can be grown via adsorption control bv

conventional MBE. A new high-temperature laser substrate heater installed in Cornell University's PARADIM Thin Film Facility allows substrate temperatures to reach up to 2000  $^{\circ}$ C.

We show that the growth window for our highest quality adsorptioncontrolled SrTiO<sub>3</sub> is from approximately 1450-1475°C, with a Sr:Ti ratio of 5:1. All films were grown on LaAlO3 (100) substrates. Figure 1 shows XRD data for samples grown at several temperatures including the adsorptioncontrol window.

## 11:30am ST-WeM2-15 MBE of Ba<sub>2</sub>BiTaO<sub>6</sub>,a Candidate *p*-type Oxide Semiconductor, Anna Park, Y. Birkhölzer, M. Barone, D. Schlom, Cornell University

Complementary metal-oxide-semiconductor (CMOS) technology is an important part of today's integrated circuit technology. CMOS replaced nchannel metal-oxide-semiconductor (NMOS) in the 1980s and with the 100 to 1000x power savings advantage it provides, enabled integrated circuits to grow from tens of thousands of NMOS transistors on a chip to tens of billions of CMOS transistors today. Today we stand at similar cross-roads for transistors made from oxide semiconductors. Only high-performance nchannel oxide transistors (and thus NMOS) exist for oxide transistors. If high performance p-channel oxide transistors could also be made. low-power CMOS would be possible in oxide systems and enable low-power transparent electronics. Although many p-type semiconducting oxides have been predicted based on their theorized electronic properties, few have been realized in experiment and those that have been achieved have much lower mobility than established n-type oxides like indium-gallium-zincoxide or In<sub>2</sub>O<sub>3</sub>. Unfortunately, the realization of *p*-type oxides is particularly difficult due to the localization of the oxygen 2p orbitals. One design criterion to realize p-type oxides is to create oxides with Sn<sup>2+</sup>, Pb<sup>2+</sup>, or Bi<sup>3+</sup> cations that have dispersive filled s-orbitals that will hybridize with oxygen 2p orbitals to delocalize the hole states and enhance mobility. We show that suboxide MBE can be used to grow Ba<sub>2</sub>BiTaO<sub>6</sub> a candidate transparent p-type oxide.

Suboxide MBE utilizes molecular beams of suboxides, where the incoming cation precursors are already in the desired oxidation states. Many oxides crystallize with a perovskite structure and offer flexibility in accommodating a variety of cations, resulting in a wide range of properties. Double perovskites, of which Ba<sub>2</sub>BiTaO<sub>6</sub> is an example, offer additional flexibility and another degree of freedom to explore in the interplay of structure and properties. For Ba<sub>2</sub>BiTaO<sub>6</sub>, we take advantage of suboxide MBE and avoid using an electron beam to evaporate Ta by supplying a molecular beam of TaO<sub>2</sub> from a Ta<sub>2</sub>O<sub>5</sub> source. Additionally, with the volatility of Bi at our growth temperatures, we grow in an adsorption-controlled regime and fine tune the flux of Ba and Ta cations to create the desired phase.

### Wednesday Afternoon, September 20, 2023

**Novel Materials** 

### Room Ballroom A - Session NM-WeA

Low Dimensional and Topological Materials Moderator: Prof. Dr. Joshua Zide, University of Delaware

12:15pm NM-WeA-1 Piezo- and Flexoelectricity Arising from Extreme Strain Gradients in Bent GaAs Nanowires, F. Marin, O. Brandt, Lutz Geelhaar, Paul-Drude-Institut für Festkörperelektronik Leibniz-Institut im Forschungsverbund Berlin, Germany

Strain induced by lattice mismatch can be employed to engineer the electronic properties of semiconductors and enhance device performance. Conventionally, only homogeneous strain is considered, but in nanostructures strain gradients can be relevant. In particular, our group has demonstrated for freestanding GaAs nanowires (NWs) controlled bending resulting in extreme strain gradients [1]. Here, we analyze the effect of such gradients on charge carrier recombination.

GaAs NWs are grown under Ga droplets by molecular beam epitaxy on prepatterned Si substrates under conditions that lead to an exceptionally low degree of polytypism so that their low-temperature photoluminescence spectra exhibit only two transitions characteristic for zincblende GaAs [2]. Spectra with multiple transitions as observed in typical GaAs NWs would prohibit the subsequent analysis. Bending is induced by the growth of a lattice-mismatched asymmetric (Al,In)As stressor shell on only one NW side.

Even moderate bending induces a substantial red-shift of the free-exciton transition that cannot be explained by the change in band gap due to strain alone but indicates the presence of internal electric fields. These fields arise from both piezo- and flexoelectricity. We emphasize that flexoelectricity is an effect occurring in all dielectric materials under strain gradients but whose influence on the electronic transitions of semiconductors has not been studied experimentally.

For further analysis, systematic variations in NW diameter and bending curvature are needed. The former is modified by symmetric (Al,Ga)As shell growth or thinning by thermal evaporation. The latter is affected by the diameter and stressor shell thickness. We can predict the resulting experimental curvature variations by analytical strain calculations following linear elasticity.

In bent NWs with the same diameter, the free-exciton transition red-shifts with increasing curvature. For NWs with similar curvature but different diameter, we systematically observe a stronger red-shift with larger diameter. Since, for a given curvature, the strength of the piezoelectric field depends on the diameter but the flexoelectric one does not, these experimental data allow to disentangle the piezo- and flexoelectric effect on the electronic transitions of inhomogeneously strained GaAs.

More generally, our study has the potential to elevate the concept of strain engineering in semiconductors to a new level, exploiting spatially inhomogeneous instead of homogeneous strain.

[1] Lewis et al., Nano Lett. 18, 2343 (2018).

[2] Oliva et al., arXiv:2211.17167 (2022).

12:30pm NM-WeA-2 InAs Quantum Dot Nucleation on Finite Surface for Scalable Quantum Light Sources, Chen Shang, Y. Pang, M. Kennedy, University of California Santa Barbara; G. Moody, J. Bowers, University of California at Santa Barbara

The development of quantum photonic technologies will fuel a paradigm shift in data processing and communication protocols. A controlled generation of non-classical states of light is a challenging task at the heart of such technologies. Epitaxially grown self-assembled semiconductor quantum dots (QDs) offer the advantages of deterministic generation of single photons and prospects of device integration. Even though self-assembled QDs on planar substrates have demonstrated superior single deterministic single photon emission, the random distribution of the QDs limits the scalability of the emitters. In this work, we realize site-controlled nucleation of InAs QDs in SiO<sub>2</sub> pockets patterned on unetched GaAs substrates. Pockets with various dimensions aligned to the <1 1 O> directions of the III-V crystal were fabricated with the anticipation that the QD nucleation process would depend on the dimensions of the finite nucleation surface.

The initial 1.5  $\mu$ m SiO<sub>2</sub> was deposited on GaAs with PECVD. To protect the surface of epi-ready GaAs substrate, the oxide pockets were formed by ICP dry etching for the first 1.3  $\mu$ m oxide followed by BHF dip for the remaining oxide. After calibrating the growth temperature offset induced by the oxide

patterns, InAs QDs were deposited in the pockets with both buried dots and exposed surface dots for photoluminescent (PL) and AFM characterizations, respectively. The reference QD structures grown on planar GaAs substrate have an areal density of 5×1010 cm-2, a typical dot height of 8 to 9 nm, and an emission wavelength of 1292 nm. However, a blue shift of the emission wavelength and a significant reduction of density have been observed for QDs nucleated in the pockets on finite surfaces. The smaller the pocket dimensions, the fewer and bluer the dots, with no observable dots in pockets smaller than certain dimensions. Compared to the dots grown in rectangular pockets, dots grown in square pockets with the same width are generally bluer and sparser. This is attributed to indium adatom diffusion onto the edge of the finite surfaces and adatom evaporation at the corners. Square pockets provide an additional diffusion path to the surface edge. At the same time, the surviving indium adatoms would diffuse onto the fewer nucleation centers in smaller, yet not too small, pockets resulting in larger dot sizes. Thus, in combination with the growth conditions, one could control the location and the structure of the dots by simply adjusting the pocket dimensions without etching the substrate. Such QD nanoparticles could potentially function as scalable onchip quantum light sources for on demand single photon streams.

#### 12:45pm NM-WeA-3 Epitaxial Cd<sub>3</sub>As<sub>2</sub> Heterostructures for Vertical Device Architectures, Anthony Rice, J. Nelson, A. Norman, K. Alberi, National Renewable Energy Laboratory

Dirac semimetals provide exciting opportunities in a number of applications, owing to their gapless band structures, high electron mobilities, broadband light absorption, and fast carrier dynamics.  $Cd_3As_2$  is a particularly promising example due to it's similarity to a number of existing III-V and II-VI technologies. A significant limiting factor to date has been its high vapor pressure, leading to growth temperatures as low as 115 °C, also significantly limiting the ability to grow epitaxial layers on top. While there have been successful reports of device structures incorporating, among others,  $Al_2O_3$ , pentacene, and metallic layers, more sophisticated devices taking advantage of band structure engineering would require higher quality, epitaxial layers.

Here, we present the first demonstration of a fully epitaxial vertical heterostructure containing  $\mathsf{Cd}_3\mathsf{As}_2$  layers. Starting from existing growth approaches to high quality Cd<sub>3</sub>As<sub>2-</sub>(112) growth on II-VI/GaAs(111)B structures [1], CdTe capping layers were nucleated at Cd<sub>3</sub>As<sub>2</sub> growth temperatures. While underlaying Cd<sub>3</sub>As<sub>2</sub> layers survived temperatures much higher than possible without a capping layer, the low energy Cd<sub>3</sub>As<sub>2</sub>(112) surface results in visible pinholes and makes conformal layers <50nm thick impossible. Combining previously mentioned growth approaches, and demonstration of growth on GaSb(001), Cd3As2(001) is grown on GaAs(001). By moving to a higher energy (001) surface, smooth CdTe capping layers as thin as 5nm are achieved, while preserving Cd<sub>3</sub>As<sub>2</sub> film quality, with electron mobilities >10,000 cm<sup>2</sup>/V-s. Subsequent deposition of a p-type Zn<sub>3</sub>As<sub>2</sub> layer creates in a p-b-n structure, as measured with C-V and I-V measurements [2]. Photodiode behavior is demonstrated up to 1000 nm. This work sets the foundation of future epitaxial devices containing Dirac semimetals.

This work was performed by the National Renewable Energy Laboratory, operated by Alliance for Sustainable Energy, LLC, for the U.S. Department of Energy (DOE) under Contract No. DE-AC36-08GO28308. Funding was provided by the U.S. Department of Energy, Office of Science, Basic Energy Sciences, Division of Materials Sciences and Engineering, Physical Behavior of Materials Program under the Disorder in Topological Semimetals project.

[1] A. D. Rice, K. Park, E. T. Hughes, K. Mukherjee, K. Alberi. *Phys. Rev. Mat.* **3**, 121201(R) (2019)

[2] A.D. Rice et al. Adv. Funct. Mater. 2022, 32, 2111470.

1:00pm NM-WeA-4 Closing Remarks and Thank Yous,

- A -

Abbas, A.: NM-MoP-21, 24; NM-TuM1-9, 34 Abdul Karim, M.: QME-SaP-6, 5 Acuna, W.: NM-TuM1-5, 33 Ahadi, K.: NM-TuA2-10, 38 Ahammad, J.: NM-MoA2-15, 17; NM-TuA2-11, 38 Akers, S.: ST-WeM1-5, 41 Alberi, K.: NM-WeA-3, 44; QME-SaM2-15, 2; QME-SaP-17, 8 Alhelais, R.: ST-MoP-2, 31 Al-Hosainy, N.: ST-MoP-2, 31 Altvater, M.: NM-MoP-11, 22 Andrews, A.: NM-MoP-20, 24; QME-SaP-12, 6 Appathurai, N.: NM-MoP-1, 20 Arnold, M.: NM-TuM2-13, 35; NM-TuM2-14, 35; NM-TuM2-15, 35; QME-SaP-10, 6 Arony, N.: GD-MoP-12, 19 Asel, T.: NM-MoM1-5, 10; NM-MoM1-6, 10; NM-MoM2-17, 12 Assaf, B.: QME-SaP-6, 5 Averett, K.: NM-MoM2-17, 12 Azizie, K.: NM-MoM1-4, 10; NM-MoM1-7, 10 — B — Baan, M.: ST-WeM1-6, 41 Bac, S.: QME-SaP-6, 5 Bailey, N.: NM-MoP-13, 22 Baker, L.: GD-TuA1-5, 37 Balakrishnan, G.: GD-MoA1-2, 14; GD-MoP-8, 19; NM-TuM1-8, 34 Bank, S.: GD-TuA1-1, 36; ST-WeM2-10, 41 Barone, M.: NM-TuA2-12, 38; NM-TuA2-13, 39; NM-TuA2-9, 37; QME-SaP-9, 5; ST-WeM2-14, 43; ST-WeM2-15, 43 Basnet, R.: NM-MoP-29, 26 Battles, S.: ST-WeM1-4, 40 Baugh, J.: GD-MoP-5, 18 Baumgartner, G.: NM-TuM1-7, 34 Bedford, R.: QME-SaP-1, 4 Benamara, M.: NM-MoP-27, 26 Bergeron, E.: GD-MoP-5, 18 Bergthold, M.: GD-TuA1-1, 36; ST-WeM2-10, 41 Bewley, W.: GD-MoA1-2, 14 Bhattacharya, A.: QME-SaM1-5, 1 Biermann, K.: NM-MoP-46, 30 Birchmier, R.: NM-MoP-11, 22 Birge, A.: GD-TuA1-3, 36; ST-MoP-3, 31 Birkhölzer, Y.: NM-MoM1-4, 10; NM-TuA2-9, 37; QME-SaP-9, 5; ST-WeM2-15, 43 Biswas, D.: QME-SuM1-5, 9 Blackston, A.: ST-WeM1-6, 41 Blaikie, T.: GD-MoP-5, 18; NM-MoP-1, 20 Boebinger, M.: NM-MoA2-15, 17; NM-MoP-3, 20; NM-TuA2-11, 38 Bork, J.: NM-TuM1-5, 33 Boschker, H.: NM-MoA2-12, 16; ST-WeM1-1, 40 Bowden, M.: NM-TuA2-14, 39 Bowers, C.: NM-MoM2-17, 12 Bowers, J.: GD-MoA1-3, 14; GD-MoA1-4, 15; NM-WeA-2, 44 Bracker, A.: NM-TuM1-7, 34 Brahlek, M.: NM-MoP-38, 28; NM-TuM1-4, 33; QME-SuM2-11, 9 Brandt, O.: NM-MoM2-14, 12; NM-WeA-1, 44 Braun, W.: NM-MoA2-12, 16; ST-WeM1-1, 40; ST-WeM1-3, 40 Brook, A.: GD-MoP-4, 18 Brützam, M.: NM-MoA2-14, 17 Buchholz, C.: NM-MoP-24, 25 Buehler-Paschen, S.: QME-SaP-12, 6

Bold page numbers indicate presenter Bugallo, D.: NM-TuM2-12, 34; QME-SaP-2, 4 Butera, V.: QME-SaP-12, 6 — C — Cai, J.: NM-MoA2-13, 17; QME-SaP-5, 4 Canedy, C.: GD-MoA1-2, 14 Caretta, L.: ST-WeM2-13, 42 Carr, M.: NM-MoP-13, 22 Carrasco, R.: GD-MoP-8, 19 Casallas Moreno, Y.: NM-MoP-16, 23 Cava, R.: NM-MoA2-14, 17 Chabak, K.: NM-MoM1-4, 10 Chakraverty, S.: NM-MoP-19, 24 Chambers, S.: NM-TuA2-14, 39; NM-TuM2-16, 35; QME-SaP-4, 4; QME-SuM1-5, 9 Chaney, A.: NM-MoM2-17, 12 Chang, C.: NM-TuA2-16, 39; QME-SaM2-13, 1 Chang, Y.: QME-SaP-13, 7 Chatterjee, S.: NM-MoP-17, 23 Chen, A.: NM-MoP-38, 28 Cheng, C.: NM-MoP-10, 22 Cheng, S.: NM-MoP-43, 29 Chiang, C.: NM-MoP-10, 22 Chlipala, M.: GD-MoP-7, 19 Choo, S.: NM-TuM2-16, 35; NM-TuM2-17, 35; QME-SaP-4, 4; QME-SaP-7, 5 Choquette, K.: ST-WeM2-11, 42 Choudhary, R.: NM-MoA2-13, 17; QME-SaP-5, **4** Christudasjustus, J.: ST-WeM1-5, 41 Chu, J.: NM-MoA2-13, 17; QME-SaP-5, 4 Chyi, J.: NM-MoP-10, 22 Cocoletzi, G.: NM-MoM2-11, 11; NM-MoP-15, 23 Comes, R.: NM-MoA2-10, 16; NM-MoA2-15, 17; NM-MoP-2, 20; NM-MoP-3, 20; NM-TuA2-11, 38; QME-SaP-11, 6; QME-SuM1-3, 9; ST-WeM1-4, 40 Corbett, J.: QME-SaP-1, 4 Coye, S.: NM-MoM1-4, 10 — D – D. Williams, M.: NM-MoM1-7, 10 da Silva, A.: NM-MoP-39, 28; NM-MoP-46, 30 Dai, X.: GD-MoP-11, 19 Das, D.: QME-SaP-1, 4 Das, S.: NM-MoP-26, 25; NM-MoP-27, 26 David, J.: GD-MoA1-1, 14; GD-MoP-3, 18; NM-MoP-13, 22 Dempsey, C.: NM-MoP-17, 23; QME-SaP-13, 7 Dereh, D.: ST-WeM1-1, 40 Detz, H.: QME-SaP-12, 6

Deutsch, D.: NM-MoP-24, 25 Dhas, J.: NM-MoP-22, 24 Diercks, D.: NM-MoP-8, 21 Ding, X.: ST-WeM2-12, 42 Dinh, D.: NM-MoM2-14, 12 Dobrowolska, M.: NM-MoP-41, 29 Dong, J.: GD-TuA1-4, 36; GD-TuA1-6, 37; NM-MoP-17, 23 Downey, B.: NM-MoM2-15, 12; NM-MoP-25, 25; NM-MoP-28, 26 Dryden, D.: NM-MoM1-4, 10 Du, D.: NM-TuM2-13, 35; NM-TuM2-15, 35; QME-SaP-10, 6 Du, Y.: NM-MoP-22, 24; NM-TuA2-14, 39 Dumen, M.: NM-MoP-19, 24 Durbin, S.: NM-MoP-9, 21 — E — Elbaroudy, A.: GD-MoP-5, 18 Engel, A.: GD-TuA1-4, 36; GD-TuA1-6, 37; NM-MoP-17, 23; QME-SaP-13, 7

Engel-Herbert, R.: NM-MoA2-9, 15; NM-MoP-39, 28; NM-TuA2-16, 39; QME-SuM2-15, **9** Engel-Herbet, R.: NM-TuA2-10, 38 England, J.: NM-MoP-13, 22 Eom, C.: QME-SaM1-7, 1 Erickson, T.: NM-MoP-15, 23; NM-MoP-34, 27 Evangelista, I.: NM-TuM2-12, 34; QME-SaP-2, Eyink, K.: QME-SaP-1, 4 — F — Fabian Jocobi, J.: NM-MoP-16, 23 Faeth, B.: NM-MoA2-12, 16; ST-WeM1-1, 40 Falson, J.: QME-SaP-15, 7; QME-SuM2-13, 9 Fang, X.: ST-MoP-3, **31** Farzaneh, S.: GD-TuA1-5, 37 Fedorov, A.: QME-SaP-13, 7 Feng, K.: GD-MoA1-3, 14; GD-MoA1-4, 15 Feygelson, T.: NM-MoM2-13, 11 Fischer, L.: QME-SaP-12, 6 Fleck, E.: NM-TuA2-12, 38 Folkes, P.: QME-SaP-13, 7 Fonseca Montenegro, A.: ST-WeM1-6, 41 Forrester, C.: NM-MoP-50, 31; ST-WeM2-12, 42 Frammolino, L.: NM-MoP-40, 28 Frost, M.: GD-MoA1-2, 14 Furdyna, J.: NM-MoP-41, 29 — G – Gajowski, N.: NM-MoP-49, 30 Gallardo Hernández, S.: NM-MoP-16, 23 Ganschow, S.: NM-TuA2-10, 38 Garcia, A.: ST-WeM2-10, 41 Geelhaar, L.: NM-WeA-1, 44 Gemperline, P.: NM-MoP-2, 20; ST-WeM1-4, 40 Giparakis, M.: NM-MoP-20, 24; QME-SaP-12, 6 Giri, M.: NM-MoP-2, 20 Gofryk, K.: NM-MoM2-11, 11; NM-MoP-30, 26; QME-SaP-18, 8 Gokhale, V.: NM-MoM2-15, 12; NM-MoP-25, 25; NM-MoP-28, 26 Golani, P.: NM-TuM2-12, 34; QME-SaP-2, 4 Gong, J.: NM-MoP-32, 27 Goodge, B.: NM-TuA2-12, 38 Goorsky, M.: QME-SaP-19, 8 Gopalan, V.: NM-TuA2-10, 38 Gorsak, C.: NM-MoM1-4, 10 Grant, P.: GD-MoP-8, 19; NM-MoP-49, 30 Grassman, T.: ST-WeM1-6, 41 Grein, C.: GD-MoA1-1, 14 Grim, J.: NM-TuM1-7, 34 Grossklaus, K.: NM-MoP-6, 21; ST-WeM2-9, 41 Growden, T.: NM-MoP-28, 26 Guguschev, C.: NM-MoA2-14, 17 Guina, M.: NM-MoP-46, 30 Gul, Y.: NM-MoP-17, 23 Gundlach, L.: NM-TuM1-5, 33 Guo, B.: GD-MoP-11, 19 Guo, S.: NM-MoA2-10, 16; QME-SaP-11, 6 Guo, Y.: NM-MoM2-16, 12; NM-MoP-5, 20 Gupta, J.: QME-SaA1-5, 3 Gutierrez-Ojeda, S.: NM-MoM2-11, 11 — Н — Hachtel, J.: NM-MoP-31, 27 Haegel, N.: NM-MoP-31, 27 Haessly, S.: NM-MoP-32, 27 Hains, C.: GD-MoP-8, 19 Hajdel, M.: GD-MoP-7, 19 Hall, H.: NM-MoP-15, 23; NM-MoP-34, 27 Hanke, M.: NM-MoP-39, 28

Harame, D.: GD-MoA1-3, 14 Harbola, V.: NM-MoA2-12, 16 Hardy, M.: NM-MoM2-15, 12; NM-MoP-25, 25; NM-MoP-28, 26 Harilal, A.: ST-WeM1-5, 41 Hashimoto, H.: NM-MoP-37, 28 Hashimoto, M.: QME-SaP-13, 7 Hatefipour, M.: GD-TuA1-5, 37 Haugan, H.: QME-SaP-1, 4 Hazra, S.: NM-TuA2-10, 38 Helfrecht, B.: ST-WeM1-5, 41 Helms, L.: GD-MoP-8, 19 Hennessey, P.: ST-WeM2-9, 41 Hensling, F.: NM-MoA2-12, 16; NM-MoM1-4, 10; ST-WeM1-1, 40; ST-WeM1-3, 40 Herfort, J.: NM-MoP-39, 28 Herman, J.: GD-MoA1-3, 14 Hilse, M.: NM-MoP-47, 30; NM-TuA2-16, 39 Hilska, J.: NM-MoP-46, 30 Hilton, D.: NM-MoP-2, 20 Hinkle, C.: NM-MoP-12, 22; NM-MoP-42, 29 Hool, R.: GD-TuA1-2, 36; ST-WeM2-11, 42 Houser, E.: NM-MoP-45, 29 Hsu, J.: NM-MoP-11, 22 Hsu, M.: NM-MoP-10, 22 Hu, J.: NM-MoP-29, 26 Hua, Z.: NM-MoP-30, 26 Huang, V.: GD-MoP-11, 19 Hung, L.: NM-MoP-10, 22 Hurley, D.: QME-SaP-18, 8 -1-I Mazur, Y.: NM-MoP-26, 25 I. Mazur, Y.: NM-MoP-27, 26 Inbar, H.: QME-SaP-13, 7 Ince, F.: GD-MoA1-2, 14; NM-TuM1-8, 34 Ingram, D.: NM-MoP-21, 24; NM-MoP-34, 27 Isceri, S.: NM-MoP-20, 24; QME-SaP-12, 6 Ishikawa, F.: NM-MoP-37, 28 Issokson, J.: GD-TuA1-5, 37 -1-J. Salamo, G.: NM-MoP-26, 25 Jalan, B.: NM-MoA2-10, 16; NM-MoA2-11, 16; NM-MoA2-13, 17; NM-MoM1-8, 11; NM-TuM2-12, 34; NM-TuM2-16, 35; NM-TuM2-17, 35; QME-SaP-11, 6; QME-SaP-14, 7; QME-SaP-2, 4; QME-SaP-3, 4; QME-SaP-4, 4; QME-SaP-5, 4; QME-SaP-7, 5 James, R.: NM-MoA2-11, 16; NM-TuM2-16, 35; QME-SaP-3, 4; QME-SaP-4, 4 Jamison, C.: ST-WeM2-9, 41 Janotti, A.: NM-MoA2-10, 16; NM-TuM2-12, 34; QME-SaP-11, 6; QME-SaP-2, 4 Jaramillo, R.: ST-WeM1-2, 40 Jena, D.: NM-MoM1-7, 10 Jin, E.: NM-MoM2-15, 12; NM-MoP-25, 25; NM-MoP-28, 26 Jin, W.: NM-MoA2-10, 16; QME-SaP-11, 6 Jin, X.: GD-MoA1-1, 14; GD-MoP-3, 18 Johnson, S.: GD-MoP-8, 19; NM-MoA2-10, 16; NM-MoP-23, 24; QME-SaP-11, 6 Jöns, K.: NM-MoP-24, 25 Ju, Z.: GD-MoA1-6, 15; NM-MoP-29, 26 Jung, H.: GD-MoA1-1, 14; GD-MoP-3, 18 Jung, T.: NM-TuM2-13, 35; NM-TuM2-14, 35; NM-TuM2-15, 35; QME-SaP-10, 6 Jungfleisch, M.: NM-TuM1-5, 33 — K — Kamath Manjeshwar, A.: NM-MoA2-11, 16; NM-TuM2-12, 34; QME-SaP-2, 4; QME-SaP-3, 4 Kaspar, T.: NM-TuA2-14, 39; ST-WeM1-5, 41 Kassa, A.: NM-MoP-39, 28 Katzer, D.: NM-MoP-28, 26 Katzer, S.: NM-MoM2-15, 12; NM-MoP-25, 25

Kawakami, R.: NM-MoP-4, 20; NM-MoP-43, 29 Kawasaki, J.: NM-TuM2-13, 35; NM-TuM2-14, 35; NM-TuM2-15, 35; QME-SaP-10, 6 Kaya, S.: NM-MoP-34, 27 Kennedy, M.: NM-WeA-2, 44 Khromets, B.: GD-MoP-5, 18 Kiefer, B.: NM-MoP-2, 20 Kim, B.: GD-MoA1-5, 15; GD-TuA1-3, 36; ST-MoP-3, 31 Kim, C.: GD-MoA1-2, 14; QME-SaP-8, 5 Kim, D.: ST-WeM1-1, 40 Kim, G.: NM-MoP-35, 27 Kim, H.: NM-MoP-40, 28 Kim, J.: QME-SaA1-7, 3; QME-SaP-8, 5 Kim, M.: GD-MoA1-2, 14; GD-MoA1-5, 15; GD-TuA1-3, 36 Kim, Y.: NM-MoM1-4, 10; NM-MoM1-5, 10; NM-MoM1-6.10 Koester, S.: NM-TuM2-12, 34; NM-TuM2-17, 35; QME-SaP-2, 4; QME-SaP-7, 5 Koirala, K.: NM-TuA2-14, 39 Kolibalova, E.: QME-SaP-12, 6 Koscica, R.: GD-MoA1-3, 14; GD-MoA1-4, 15 Koudriavysev, I.: NM-MoP-16, 23 Kourkoutis, L.: NM-TuA2-12, 38; NM-TuM2-12, 34; QME-SaP-2, 4 Krishna, S.: GD-MoA1-1, 14; GD-MoP-3, 18; NM-MoP-49, 30 Krusin-Elbaum, L.: ST-WeM2-12, 42 Kuchuk, A.: NM-MoP-7, 21 Kuznetsova, T.: NM-TuA2-10, 38 -L-LaDuca, Z.: NM-TuM2-13, 35; NM-TuM2-14, 35; NM-TuM2-15, 35; QME-SaP-10, 6 Lähnemann, J.: NM-MoM2-14, 12 Lang, A.: NM-MoM2-13, 11; NM-MoM2-15, 12; NM-MoP-25, 25; NM-MoP-28, 26 Lasek, K.: NM-MoP-44, 29 Laukkanen, P.: NM-MoP-46, 30 Law, S.: NM-MoP-47, 30; NM-TuM1-6, 33; QME-SaA1-3, 3; ST-MoP-1, 31 Lay, T.: GD-MoP-2, 18 Leahy, I.: QME-SaP-17, 8 Leake, G.: GD-MoA1-3, 14 Lee, D.: NM-MoA2-10, 16; QME-SaP-11, 6 Lee, K.: NM-MoP-41, 29; NM-TuA2-12, 38 Lee, M.: GD-MoA1-5, 15; GD-TuA1-2, 36; GD-TuA1-3, 36; ST-MoP-3, 31; ST-WeM2-11, 42 Lee, S.: GD-MoA1-1, 14; GD-MoA1-5, 15; GD-MoP-3, 18; NM-MoP-41, 29; NM-MoP-49, 30 Lee, T.: QME-SuM1-5, 9 Lee, W.: NM-MoP-40, 28 Lee, Z.: NM-MoP-10, 22 Lemire, A.: ST-WeM2-9, 41 Leonard, T.: GD-TuA1-1, 36 Levy, I.: GD-TuA1-5, 37; ST-WeM2-12, 42 Li. B.: GD-TuA1-2. 36 Li, C.: NM-MoP-26, 25; NM-MoP-27, 26 Li. M.: GD-MoA1-5. 15 Li, X.: NM-MoP-40, 28 Li, Y.: NM-MoA2-10, 16; NM-MoA2-14, 17; NM-MoP-12, 22; NM-MoP-32, 27; NM-MoP-40, 28; NM-TuA2-15, 39; QME-SaP-11, 6 Lin, K.: GD-MoP-2, 18 Liu, C.: NM-MoP-10, 22 Liu, F.: NM-MoM1-8, 11; NM-TuM2-12, 34; QME-SaP-14, 7; QME-SaP-2, 4 Liu, J.: NM-MoM2-16, 12; NM-MoP-5, 20; NM-TuA2-14, 39 Liu, M.: NM-MoP-40, 28 Liu, W.: NM-MoP-10, 22 Liu, X.: NM-MoP-41, 29; QME-SaP-6, 5 46

Liu, Y.: GD-MoA1-1, 14; NM-MoP-32, 27 Liu, Z.: NM-MoA2-13, 17; NM-MoM1-4, 10; QME-SaP-5, 4 Llanos, A.: QME-SaP-15, 7 Lo, T.: NM-MoP-10, 22 Logan, J.: GD-MoP-8, 19 Logvenov, G.: NM-MoP-35, 27 Lopes, J.: NM-MoP-39, 28 López López, M.: NM-MoP-16, 23 Lopez, G.: ST-WeM2-12, 42 Lu. D.: OME-SaP-13, 7 Lu, Q.: NM-MoP-38, 28 Luna, E.: NM-MoP-46, 30 Lv, H.: NM-MoP-39, 28 Lygo, A.: GD-MoP-11, 19 - M -M Eldose, N.: NM-MoP-26, 25; NM-MoP-27, 26 Ma, Z.: NM-MoP-32, 27 MacDonald, A.: NM-MoP-40, 28 Maddaka, R.: NM-MoM2-16, 12 Maddakka, R.: GD-MoP-6, 18 Madhavan, V.: NM-MoP-11, 22 Maestas, D.: GD-MoP-8, 19 Mahalingam, K.: NM-MoM2-17, 12; QME-SaP-1, 4 Mahatara, S.: NM-MoP-2, 20 Maia de Oliveira, F.: NM-MoP-27, 26 Mais de Oliveira, F.: ST-MoP-2, 31 Majer, L.: NM-MoA2-12, 16; ST-WeM1-1, 40; ST-WeM1-3.40 Majumdar, A.: GD-MoA1-5, 15 Makin, R.: NM-MoP-9, 21 Malholtra, Y.: NM-MoM2-16, 12; NM-MoP-5, 20 Man, O.: QME-SaP-12, 6 Mannhart, J.: NM-MoA2-12, 16; QME-SaM1-3, 1; ST-WeM1-1, 40; ST-WeM1-3, 40 Mantooth, H.: NM-MoP-7, 21 Manzo, S.: NM-TuM2-13, 35; NM-TuM2-14, 35; NM-TuM2-15, 35; QME-SaP-10, 6 Marin, F.: NM-WeA-1, 44 Markland, R.: ST-WeM1-4, 40 Martin, L.: NM-TuA2-12, 38 Martínez López, A.: NM-MoP-16, 23 Matara Kankanamge, I.: NM-MoM1-7, 10 Matthews, B.: NM-TuM2-16, 35; QME-SaP-4, Maximenko, S.: NM-MoM2-13, 11; NM-TuM1-2, 33 May, B.: NM-MoM2-11, 11; NM-MoP-30, 26; QME-SaP-18, 8 May, S.: NM-TuM2-12, 34; QME-SaP-2, 4 Mazur, Y.: ST-MoP-2, 31 McCabe, L.: GD-MoP-12, 19 McCandless, J.: NM-MoM1-7, 10 McCarthy, T.: NM-MoP-29, 26 McCartney, M.: GD-MoA1-2, 14; NM-MoP-50, 31; NM-TuM1-8, 34 McChesney, J.: NM-TuM2-13, 35 McElearney, J.: NM-MoP-6, 21 McFadden, A.: GD-TuA1-6, 37 McKenzie, W.: NM-TuM1-7, 34 McMinn, A.: GD-MoA1-6, 15; NM-MoP-29, 26 Meisenheimer, P.: ST-WeM2-13, 42 Menasuta, P.: NM-MoP-6, 21 Messecar, A.: NM-MoP-9, 21 Meyer, D.: NM-MoM2-13, 11; NM-MoP-28, 26 Meyer, J.: GD-MoA1-2, 14 Mi, Z.: NM-MoM2-12, 11; NM-MoM2-16, 12; NM-MoP-5, 20 Miao, W.: GD-MoP-11, 19 Mikalsen, M.: GD-TuA1-5, 37

Miller, M.: NM-MoP-31, 27; NM-MoP-8, 21 Milosavljevic, M.: GD-MoP-8, 19; NM-MoP-23, **24** Min, J.: NM-MoM2-16, 12 Minehisa, K.: NM-MoP-37, 28 Mkhoyan, K.: NM-MoA2-10, 16; QME-SaP-11, 6 Montealegre, D.: GD-MoA1-5, 15; GD-TuA1-3, 36; ST-MoP-3, 31 Moody, G.: NM-WeA-2, 44 Moore, R.: NM-MoP-38, 28 Morath, C.: GD-MoP-8, 19 Moreno, B.: NM-MoP-1, 20 Moreno, J.: NM-MoP-15, 23 Mou, S.: NM-MoM1-4, 10; NM-MoM1-5, 10; NM-MoM1-6, 10; NM-MoM2-17, 12 Mukherjee, S.: ST-MoP-1, 31 Muller, D.: NM-MoM1-4, 10; NM-MoM1-7, 10; NM-TuA2-10, 38 Muziol, G.: GD-MoP-7, 19 Myers, R.: ST-WeM1-6, 41 — N — Nair, S.: NM-MoA2-10, 16; NM-MoA2-11, 16; QME-SaP-11, 6; QME-SaP-3, 4 Nakama, K.: NM-MoP-37, 28 Neal, A.: NM-MoM1-4, 10; NM-MoM1-5, 10; NM-MoM1-6.10 Nelson, J.: NM-WeA-3, 44 Nepal, N.: NM-MoM2-13, 11; NM-MoM2-15, 12; NM-MoP-25, 25; NM-MoP-28, 26 Newburger, M.: NM-MoP-4, 20 Newell, A.: GD-MoP-8, 19; NM-TuM1-8, 34 Ngai, J.: QME-SuM1-5, 9 Nishihaya, S.: QME-SaP-13, 7 Noesges, B.: NM-MoM1-5, 10; NM-MoM1-6, 10 Nolde, J.: NM-TuM1-2, 33 Norman, A.: NM-WeA-3, 44; QME-SaP-17, 8 North, W.: ST-WeM2-11, 42 -0 -Olguin Melo, D.: NM-MoP-16, 23 Olsen, G.: NM-TuA2-12, 38 Onuma, T.: NM-MoM1-7, 10 Opatosky, B.: NM-MoP-2, 20; NM-MoP-3, 20 Orloff, N.: NM-TuA2-12, 38 — P — Page, M.: NM-MoP-4, 20 Paik, H.: NM-MoP-2, 20; NM-TuA2-10, 38 Palmstrøm, C.: GD-TuA1-4, 36; GD-TuA1-6, 37; NM-MoP-17, 23; QME-SaM2-17, 2; OME-SaP-13.7 Pan, K.: QME-SaP-19, 8 Pandey, A.: NM-MoM2-16, 12; NM-MoP-5, 20 Pandey, S.: ST-WeM2-11, 42 Pang, Y.: NM-WeA-2, 44 Papac, M.: NM-TuA2-12, 38 Park, A.: NM-TuA2-9, 37; QME-SaP-9, 5; ST-WeM2-15. 43 Park, S.: NM-MoP-41, 29 Parker, N.: NM-MoM1-4, 10; NM-TuA2-13, 39 Parkin, S.: QME-SaM2-11, 1 Pate, B.: NM-MoM2-13, 11 Paudel, B.: NM-TuA2-14, 39 Paudel, R.: ST-WeM1-4, 40 Peiris, F.: NM-MoM2-14, 12; NM-MoP-45, 29 Pepper, M.: NM-MoP-17, 23 Pfeiffer, L.: NM-MoM1-2, 10 Pieczulewski, N.: NM-MoM1-4, 10; NM-MoM1-7, 10 Pope, J.: ST-WeM1-5, 41 Price, C.: QME-SaP-20, 8 Provence, S.: ST-WeM1-4, 40 Pryds, N.: QME-SuM1-1, 9

Punkkinen, M.: NM-MoP-46, 30 Puustinen, J.: NM-MoP-46, 30 Pyo, S.: ST-MoP-4, 32 - Q -Qi, X.: GD-MoA1-6, 15; NM-MoP-29, 26 — R — Raftery, E.: ST-WeM2-11, 42 Rahemtulla, A.: NM-MoP-1, 20 Rajapitamahuni, A.: NM-MoA2-11, 16; QME-SaP-3.4 Ramesh, M.: ST-WeM2-13, 42 Ramesh, R.: ST-WeM2-13, 42 Ramirez, E.: QME-SuM1-5, 9 Ram-Mohan, L.: QME-SaP-1, 4 Ramsteiner, M.: NM-MoP-39, 28 Ren, Y.: QME-SaP-19, 8 Reuter, D.: NM-MoP-24, 25 Rice, A.: NM-MoP-31, 27; NM-MoP-8, 21; NM-WeA-3, 44; QME-SaP-17, 8 Richardella, A.: NM-MoP-45, 29; NM-MoP-47, 30 Richards, R.: NM-MoP-13, 22 Ricks, A.: GD-TuA1-1, 36 Rimal, G.: NM-MoA2-15, 17; NM-MoP-2, 20; NM-MoP-3, 20; NM-TuA2-11, 38 Roberts, D.: NM-MoP-31, 27 Robinson, K.: NM-MoP-4, 20 Ronningen, T.: GD-MoA1-1, 14 Rotter, T.: GD-MoA1-2, 14; NM-TuM1-8, 34 — S — Saboor, A.: NM-MoA2-10, 16; QME-SaP-11, 6 Sadeghi, I.: ST-WeM1-2, 40 Sadeghi, S.: GD-MoP-5, 18 Sadowski, J.: NM-MoA2-10, 16; NM-MoA2-15, 17; NM-TuA2-11, 38; QME-SaP-11, 6 Salamo, G.: NM-MoP-27, 26; NM-MoP-7, 21 Salmani-Razaie, S.: NM-TuA2-10, 38 Samanta, T.: NM-TuM2-14, 35; NM-TuM2-15, 35; QME-SaP-10, 6 Samarth, N.: NM-MoP-45, 29; NM-TuA2-16, 39; QME-SuM2-9, 9 Saraswat, V.: NM-TuM2-13, 35 Sarollahi, M.: ST-MoP-2, 31 Sassi, M.: ST-WeM1-5, 41 Schlom, D.: NM-MoA2-14, 17; NM-MoM1-4, 10; NM-MoM1-7, 10; NM-TuA2-10, 38; NM-TuA2-12, 38; NM-TuA2-13, 39; NM-TuA2-15, 39; NM-TuA2-9, 37; QME-SaM1-1, 1; QME-SaP-8, 5; QME-SaP-9, 5; ST-WeM2-13, 42; ST-WeM2-14, 43; ST-WeM2-15, 43 Schrenk, W.: NM-MoP-20, 24; QME-SaP-12, 6 Schwaigert, T.: NM-TuA2-10, 38 Schwarz, B.: NM-MoP-20, 24 Senevirathna, M.: NM-MoM1-4, 10 Sfigakis, F.: GD-MoP-5, 18 Shabani, J.: GD-MoP-4, 18; GD-TuA1-5, 37 Shang, C.: GD-MoA1-3, 14; GD-MoA1-4, 15; NM-WeA-2, 44 Shang, S.: NM-MoM1-4, 10 Sharma, S.: NM-TuM2-16, 35; QME-SaP-4, 4 Sharpe, M.: NM-MoP-13, 22 Shetty, S.: NM-MoP-26, 25; NM-MoP-7, 21 Shi, Y.: GD-MoP-5, 18; NM-MoP-1, 20 Shih, C.: NM-MoP-40, 28 Shrestha, A.: NM-MoP-21, 24; NM-TuM1-9, 34 Sitaram, S.: ST-MoP-1, 31 Skierbiszewski, C.: GD-MoP-7, 19 Skipper, A.: GD-MoA1-3, 14; ST-WeM2-10, 41 Smart, T.: ST-WeM1-1, 40 Smeaton, M.: NM-TuM2-12, 34; QME-SaP-2, 4 Smink, S.: ST-WeM1-1, 40; ST-WeM1-3, 40

Smith, A.: NM-MoP-15, 23; NM-MoP-21, 24; NM-MoP-34, 27; NM-TuM1-9, 34 Smith, D.: GD-MoA1-2, 14; NM-MoP-50, 31; NM-TuM1-8, 34 Song, Y.: GD-MoA1-5, 15 Sotir, D.: ST-WeM2-14, 43 Spurgeon, S.: NM-TuM2-16, 35; QME-SaP-4, 4; ST-WeM1-5, 41 Srivastava, S.: QME-SaP-19, 8 Stanchu, H.: NM-MoP-26, 25; NM-MoP-27, 26; ST-MoP-2, 31 Stanley, M.: NM-MoP-45, 29 Steele, J.: NM-MoM1-4, 10; NM-MoM1-7, 10 Stemmer, S.: GD-MoP-11, 19 Sterbinsky, G.: NM-MoA2-15, 17; NM-MoP-3, 20; NM-TuA2-11, 38 Stevens, M.: NM-TuM1-7, 34 Stevenson, P.: ST-WeM2-13, 42 Storm, D.: NM-MoM2-13, 11; NM-MoP-28, 26 Strasser, G.: NM-MoP-20, 24; QME-SaP-12, 6 Strickland, W.: GD-TuA1-5, 37 Strohbeen, P.: GD-MoP-4, 18; GD-TuA1-5, 37 Stutzman, M.: GD-TuA1-4, 36 Su, K.: NM-TuM2-13, 35; NM-TuM2-14, 35; NM-TuM2-15, 35; QME-SaP-10, 6 Sun, K.: NM-MoM2-12, 11; NM-MoM2-16, 12; NM-MoP-15, 23 Sun, Y.: GD-TuA1-2, 36 Sushko, P.: NM-TuA2-14, 39; QME-SuM1-5, 9 Suyolcu, Y.: NM-MoP-35, 27; NM-TuA2-15, 39 Svagera, R.: QME-SaP-12, 6 - T -Tai, L.: QME-SaP-19, 8 Takeuchi, N.: NM-MoP-15, 23 Tam, A.: GD-MoP-5, 18 Tam, M.: NM-MoP-1, 20 Tamargo, M.: NM-MoP-50, 31; ST-WeM2-12, 42 Tamboli, A.: NM-MoP-8, 21 Tasnim, T.: NM-MoA2-15, 17; NM-MoP-2, 20; NM-MoP-3, 20 Taylor, P.: QME-SaP-13, 7; QME-SaP-16, 7 Tellekamp, B.: NM-MoP-8, 21 Tellekamp, M.: NM-MoP-31, 27 Tenne, D.: NM-TuA2-13, 39 Testelin, C.: ST-WeM2-12, 42 Thapa, S.: NM-MoP-2, 20; ST-WeM1-4, 40 Tian, Z.: NM-TuA2-12, 38 Tomasulo, S.: GD-MoA1-2, 14; NM-TuM1-2, 33 Trampert, A.: NM-MoP-39, 28; NM-MoP-46, 30 Trejo Hernández, R.: NM-MoP-16, 23 Trice, R.: NM-MoP-47, 30 Truttmann, T.: NM-MoM1-8, 11; NM-TuM2-12, 34; QME-SaP-14, 7; QME-SaP-2, 4 Tu. C.: NM-MoP-10. 22 Turski, H.: GD-MoP-7, 19 Twigg, M.: NM-TuM1-2, 33 — U – Upadhyay, S.: NM-MoP-15, 23; NM-MoP-34, 27 — v — Vallejo, K.: NM-MoM2-11, 11; NM-MoP-30, 26: QME-SaP-18.8 van Aken, P.: NM-MoP-35, 27 van Deurzen, L.: GD-MoP-7, 19 van Duin, A.: NM-MoA2-9, 15 Van Sambeek, J.: ST-WeM1-2, 40 van Schijndel, T.: GD-TuA1-6, 37 Vandervelde, T.: NM-MoP-6, 21; ST-WeM2-9, 41 Varshney, S.: NM-TuM2-17, 35; QME-SaP-7, 5

Vasudevan, R.: ST-WeM1-4, 40 Vazquez, V.: ST-WeM2-9, 41 Vincent, D.: NM-MoP-32, 27 Vogt, P.: NM-MoM1-4, 10 Voranthamrong, S.: NM-MoP-10, 22 Vukelich, R.: NM-MoP-2, 20 Vurgaftman, I.: GD-MoA1-2, 14; NM-TuM1-2, 33 – w -Waas, M.: QME-SaP-12, 6 Wang, D.: NM-MoM2-12, 11 Wang, J.: QME-SaP-6, 5 Wang, K.: QME-SaP-19, 8 Wang, L.: NM-TuA2-14, 39; ST-WeM1-5, 41 Wang, P.: NM-MoM2-12, 11 Wang, X.: ST-MoP-1, 31 Wang, Y.: ST-MoP-3, 31; ST-WeM2-11, 42 Ware, M.: ST-MoP-2, 31 Wasilewski, Z.: GD-MoP-5, 18; NM-MoP-1, 20 Wasserman, D.: GD-TuA1-1, 36 Webster, P.: GD-MoP-8, 19; NM-MoP-23, 24; NM-MoP-49, 30 Wen, J.: NM-TuM2-12, 34; NM-TuM2-17, 35; QME-SaP-2, 4; QME-SaP-7, 5

Wheeler, V.: NM-MoM2-15, 12; NM-MoP-25, 25; NM-MoP-28, 26 White, C.: GD-TuA1-1, 36 Wickramasinghe, K.: NM-MoP-50, 31; ST-WeM2-12, 42 Williams, M.: NM-MoM1-4, 10 Wu, W.: NM-TuM1-5, 33 Wu, Y.: NM-MoM2-12, 11; NM-MoM2-16, 12; NM-MoP-35, 27; NM-MoP-5, 20 — X — Xia, F.: GD-MoA1-5, 15 Xiao, Y.: GD-MoP-6, 18; NM-MoM2-12, 11; NM-MoM2-16, 12; NM-MoP-5, 20 Xie, Y.: QME-SaP-19, 8 Xing, G.: QME-SaA1-1, 3 Xing, H.: NM-MoM1-7, 10 Xu, C.: GD-MoP-2, 18 Xu, X.: NM-MoA2-13, 17; QME-SaP-5, 4 — Y — Yalcin, B.: NM-MoA2-9, 15 Yang, S.: NM-MoM2-16, 12; NM-MoP-5, 20 Yang, Z.: NM-MoA2-10, 16; NM-MoM1-8, 11; NM-TuA2-14, 39; NM-TuM2-17, 35; QME-SaP-11, 6; QME-SaP-14, 7; QME-SaP-7, 5 Ye, K.: ST-WeM1-2, 40

Yi, H.: NM-TuA2-16, 39 Yilmaz, D.: NM-MoA2-9, 15 Yoon, H.: NM-TuM2-16, 35; QME-SaP-4, 4 Yoshimura, K.: QME-SaP-6, 5 Younas, R.: NM-MoP-42, 29 Yu, M.: NM-MoP-47, 30; NM-TuM1-6, 33 Yu, S.: NM-MoP-26, 25; NM-MoP-29, 26; NM-MoP-32, 27 — z — Zak, M.: GD-MoP-7, 19 Zhang, F.: NM-MoP-40, 28 Zhang, Q.: NM-MoP-32, 27 Zhang, Y.: GD-MoA1-6, 15; NM-MoP-29, 26; NM-MoP-30, 26 Zheng, X.: NM-TuM2-13, 35 Zhou, G.: NM-MoP-12, 22; NM-MoP-42, 29 Zhou, H.: NM-MoA2-10, 16; QME-SaP-11, 6 Zhou, J.: NM-MoP-32, 27 Zhou, P.: NM-MoM2-12, 11 Zhou, S.: ST-WeM2-13, 42 Zhou, W.: NM-MoP-43, 29 Zide, J.: GD-MoP-12, 19; NM-TuM1-5, 33 Zolatanosha, V.: NM-MoP-24, 25 Zon, ..: NM-MoP-10, 22